Magnetic Measurements

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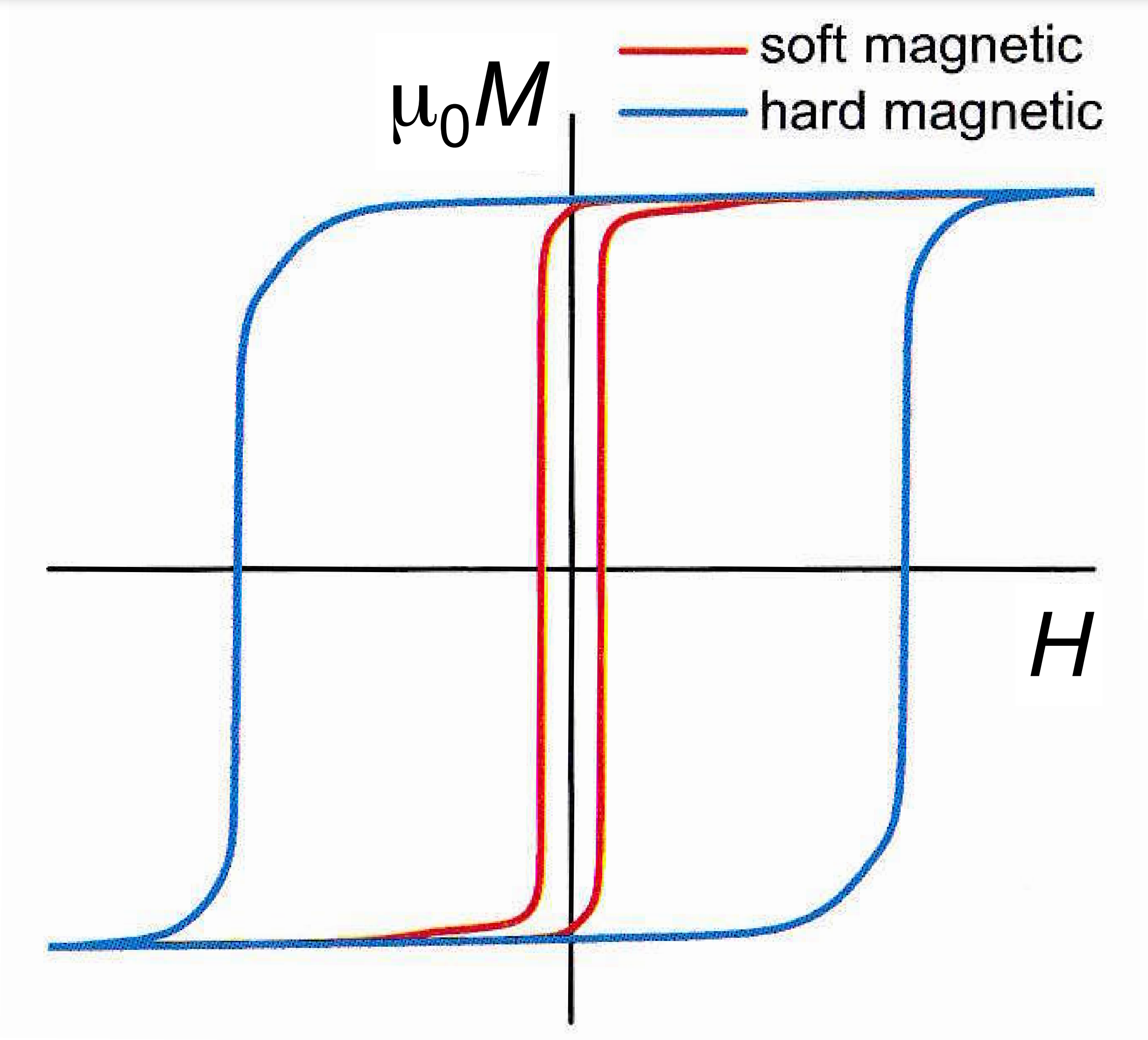
IEEE Magnetics Summer School 2014, Rio

Why Magnetic Measurements?

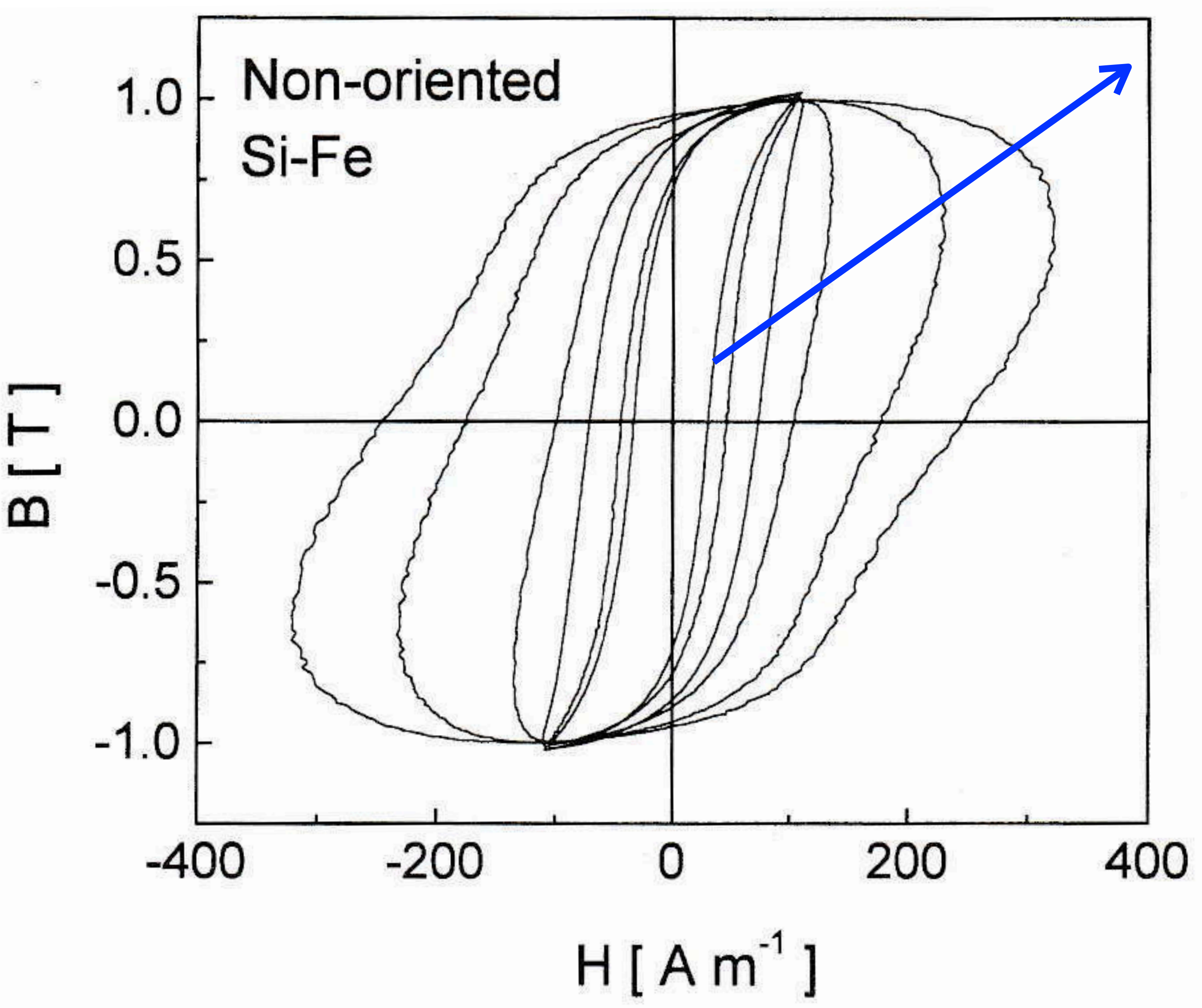
"Magnetism is an experimental science"
Mike Coey, 2010

" No clear understanding of magnetism can be attained without a sound knowledge of the way in which magnetic properties are measured."
B.D. Cullity & CD Graham:
Introduction to Magnetic Materials, 2009

What needs to be measured ?



freqency.: 1 – 1600 Hz

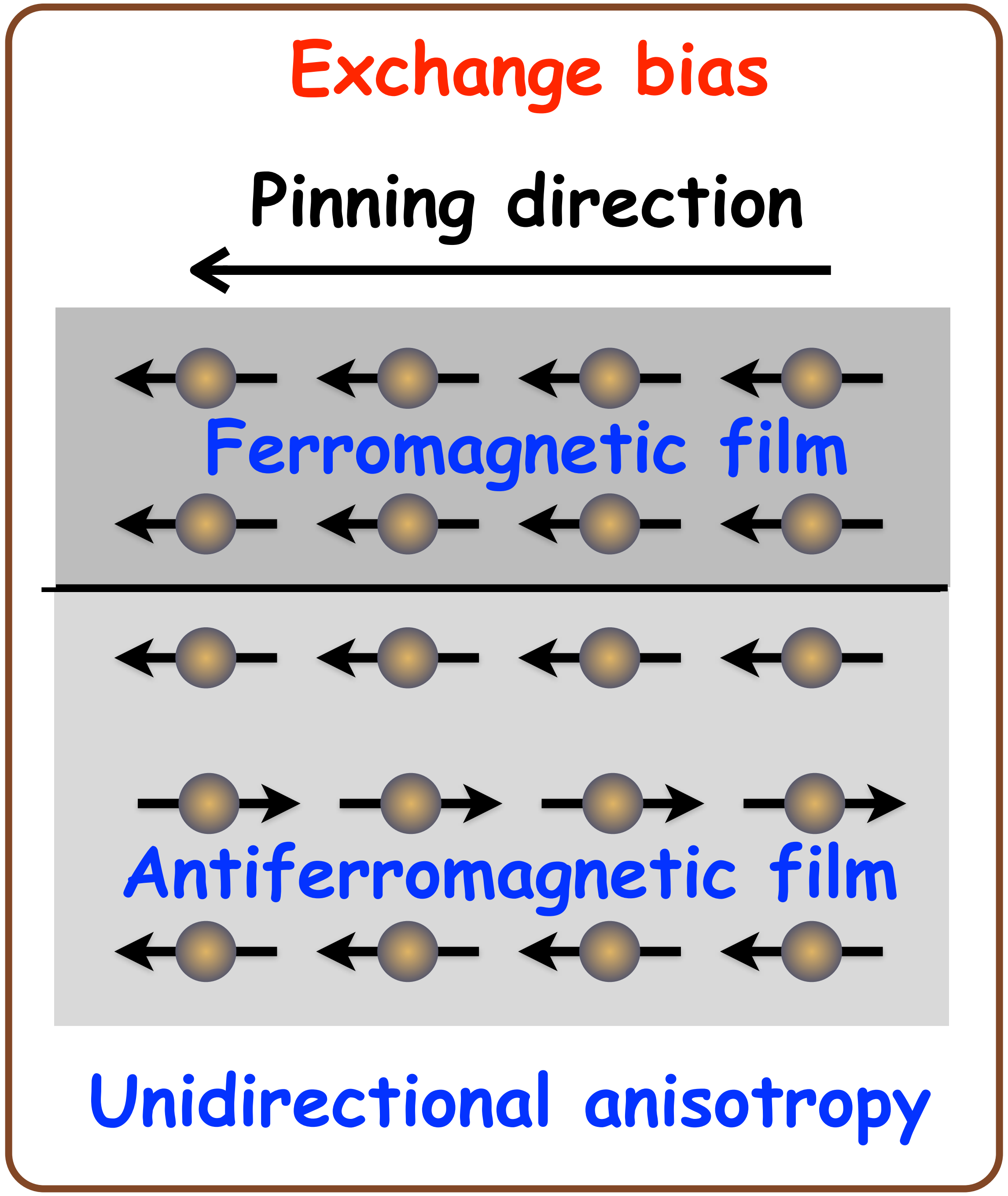


Hysteresis curve:

- coercivity, remanence
- permeabilities
- energy loss
- etc. …

M(H) loop and domains

Reversal of $Ni_{81}Fe_{19}$ (30 nm) / NiO (30 nm)



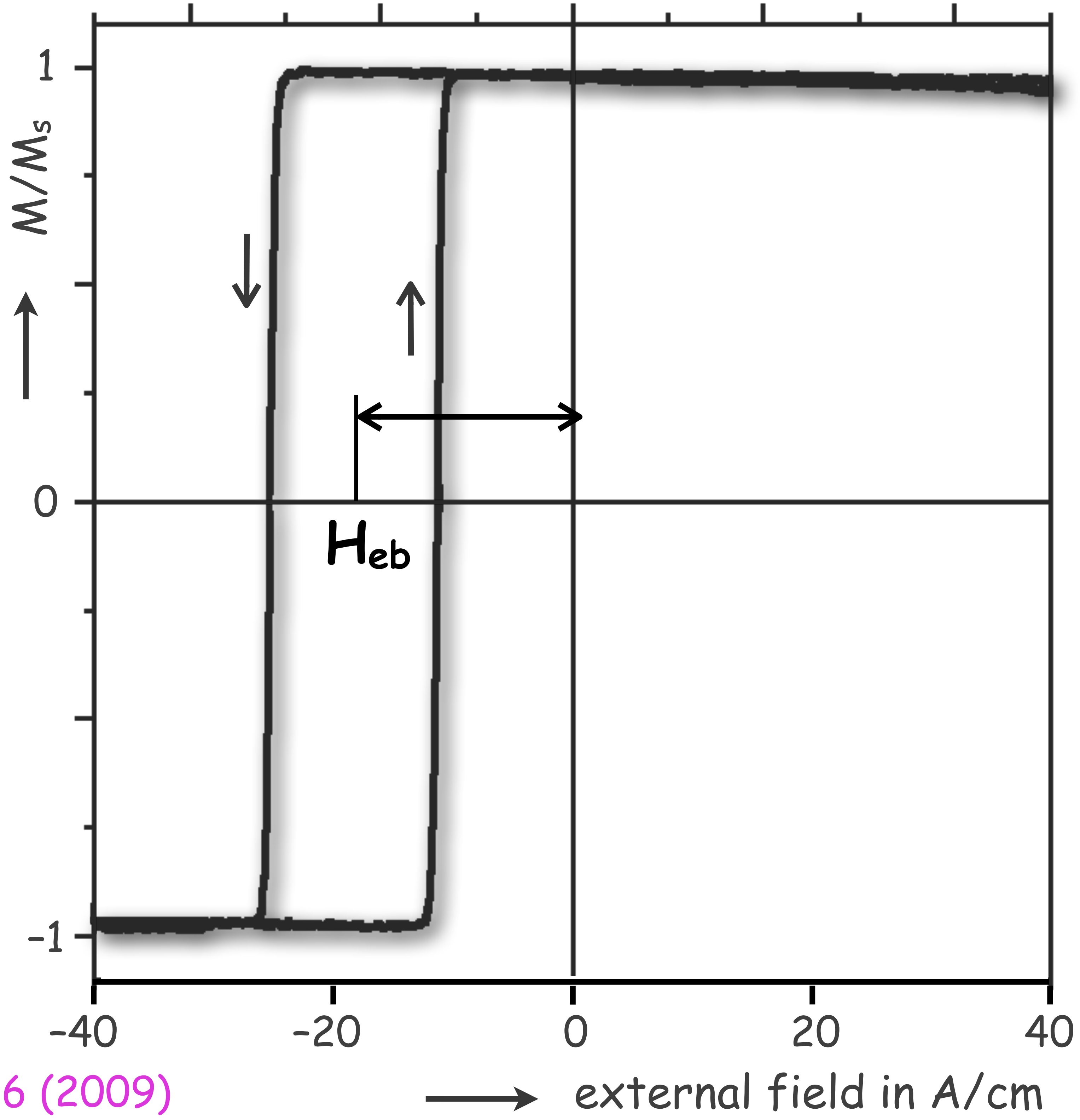
Exchange bias

Pinning direction

Ferromagnetic film

Antiferromagnetic film

Unidirectional anisotropy



J. McCord and R.S., New J. Phys. 11, 83016 (2009)

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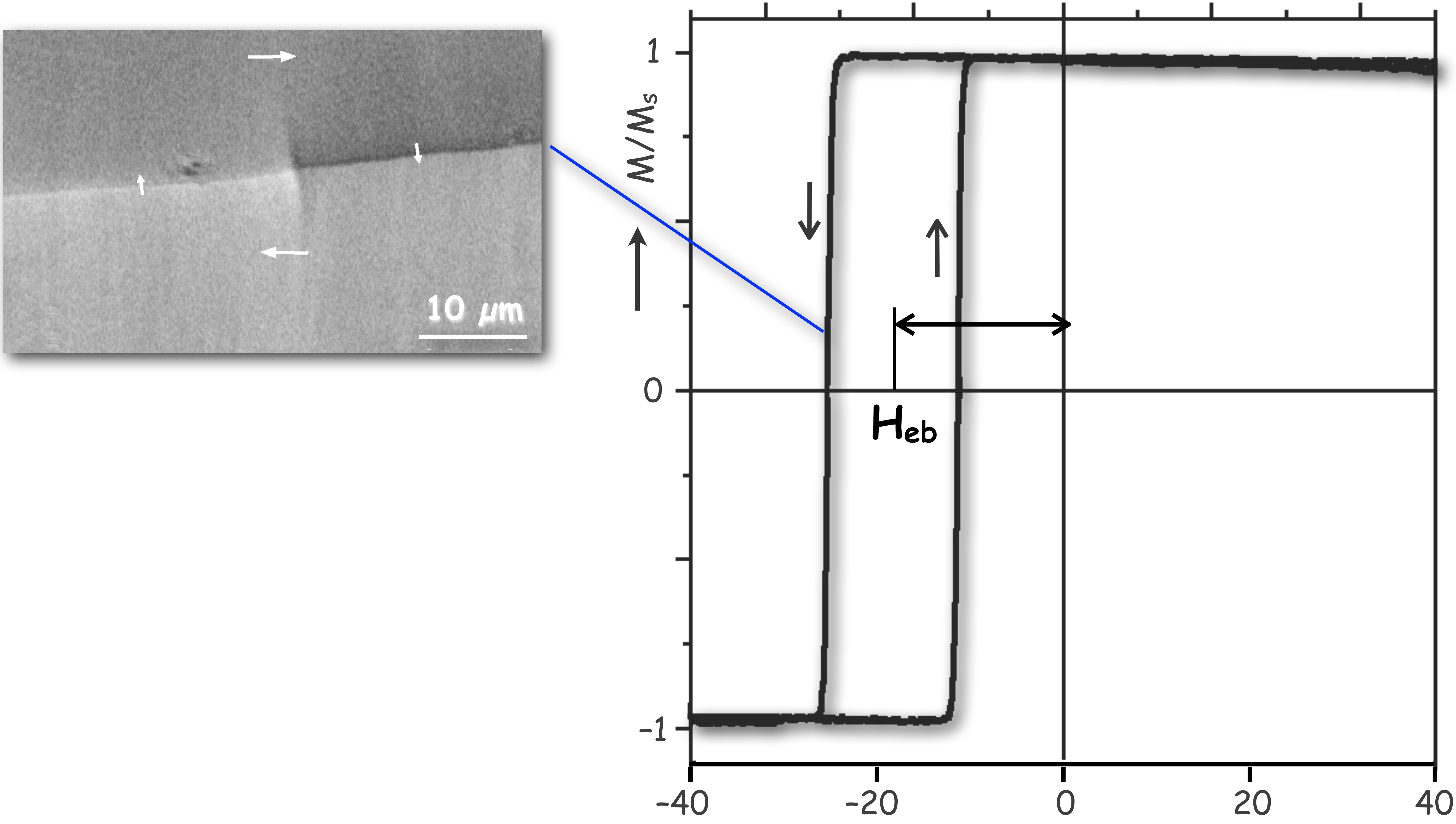

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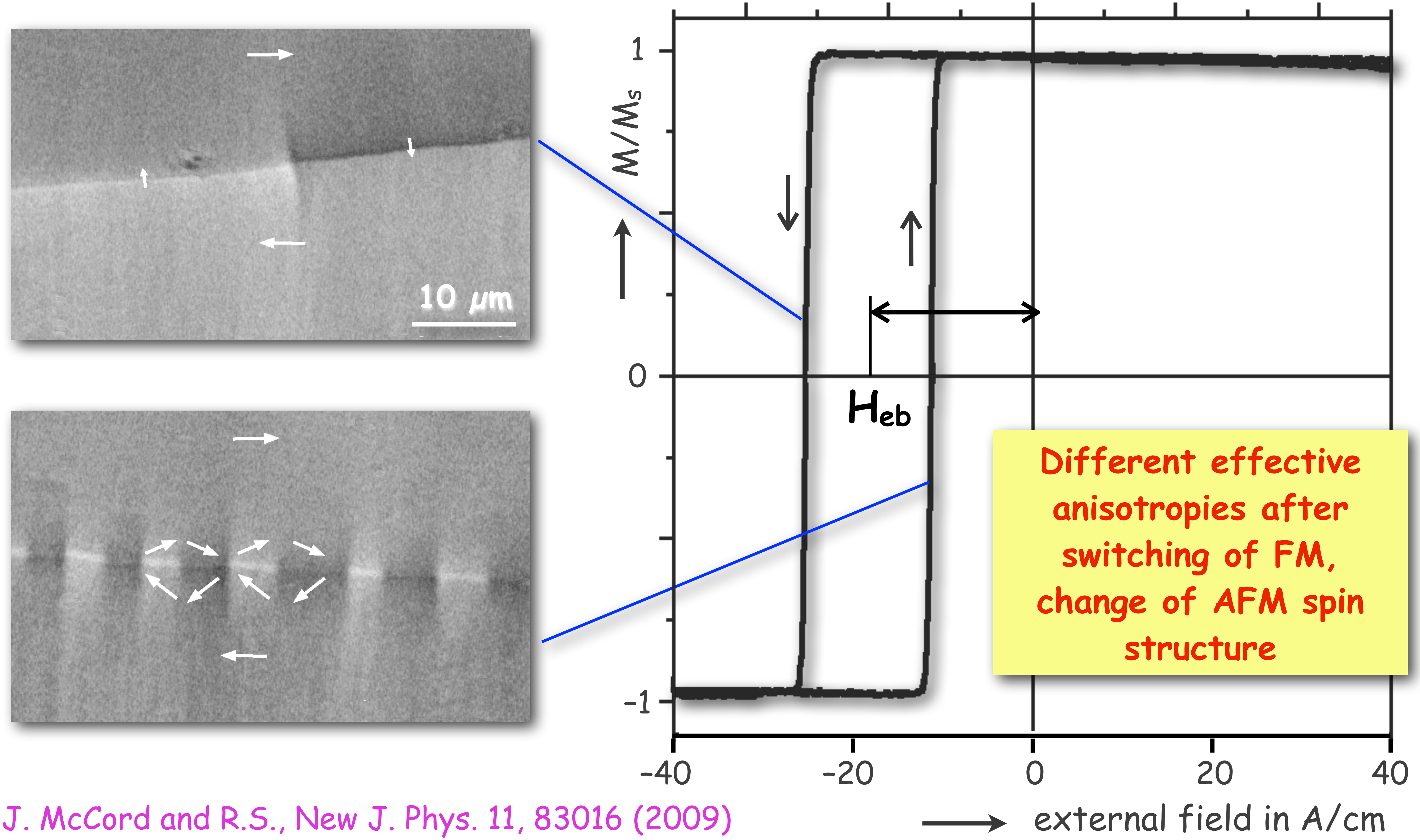


Different effective anisotropies after switching of FM, change of AFM spin structure

J. McCord and R.S., New J. Phys. 11, 83016 (2009)

What needs to be measured ?

Magnetization curve

Magnetoresistance

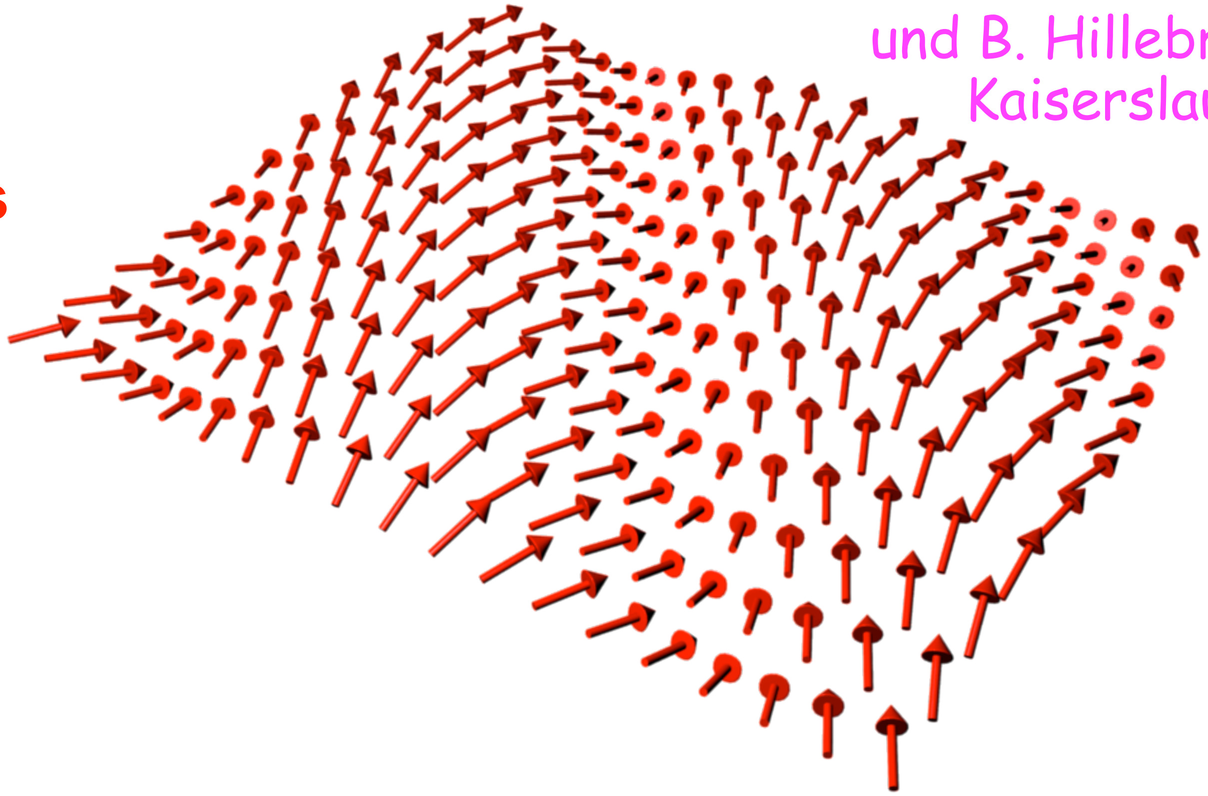
Domain scale masurements (Magnetic Imaging)

Damping constants α

Resonance frequency f_{res}

Saturation magnetization M_s

Animation: H. Schultheiß
und B. Hillebrands
Kaiserslautern



Crystal- or other anisotropy constants K

Exchange or stiffness constant A

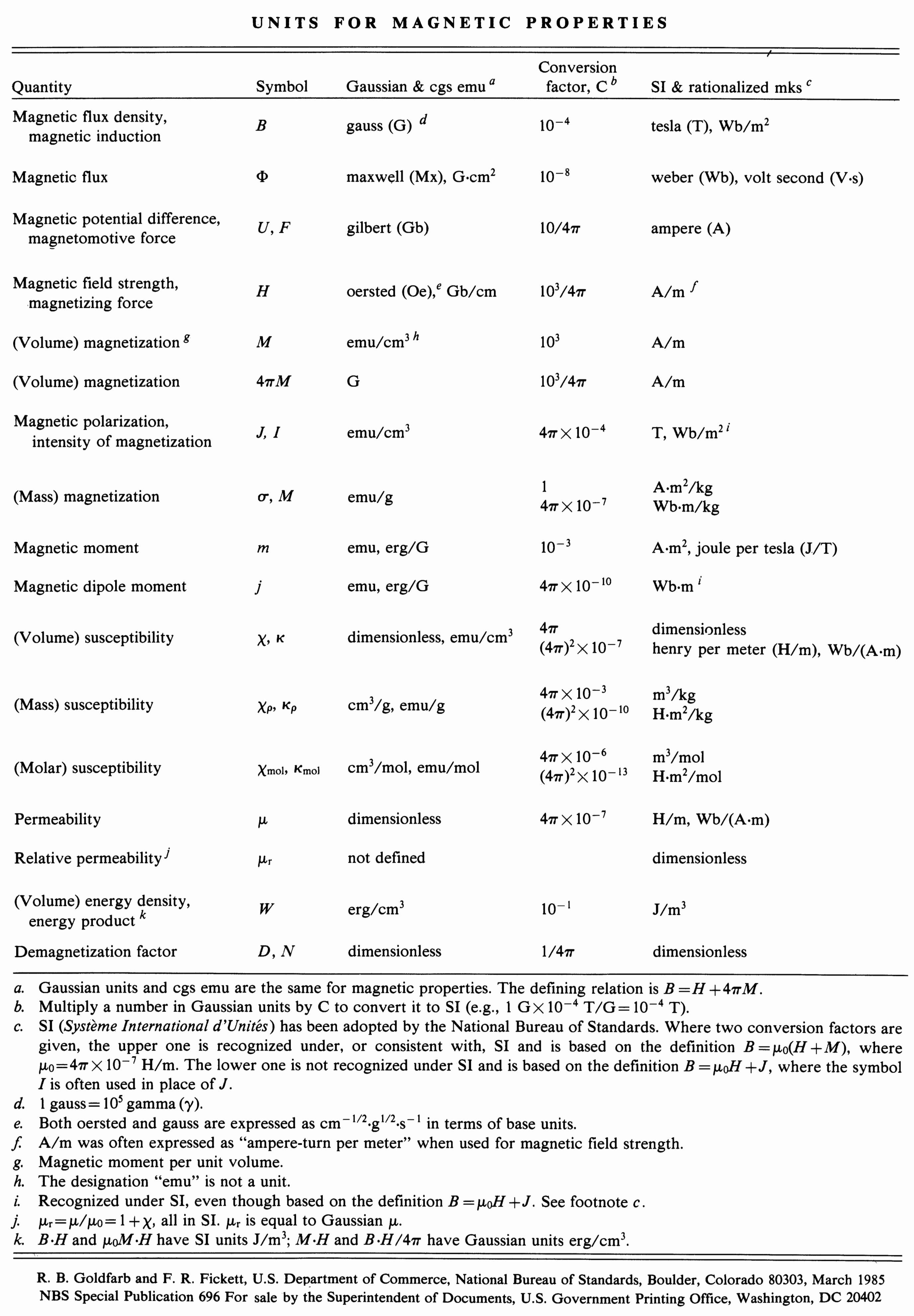
Magnetic order (neutrons)

Magnetostriction constants λ

and more…

Contents of lecture

Magnetic Units



UNITS FOR MAGNETIC PROPERTIES

Quantity	Symbol	Gaussian & cgs emu [a]	Conversion factor, C [b]	SI & rationalized mks [c]
Magnetic flux density, magnetic induction	B	gauss (G) [d]	10^{-4}	tesla (T), Wb/m^2
Magnetic flux	Φ	maxwell (Mx), G·cm^2	10^{-8}	weber (Wb), volt second (V·s)
Magnetic potential difference, magnetomotive force	U, F	gilbert (Gb)	$10/4\pi$	ampere (A)
Magnetic field strength, magnetizing force	H	oersted (Oe),[e] Gb/cm	$10^3/4\pi$	A/m [f]
(Volume) magnetization [g]	M	emu/cm^3 [h]	10^3	A/m
(Volume) magnetization	$4\pi M$	G	$10^3/4\pi$	A/m
Magnetic polarization, intensity of magnetization	J, I	emu/cm^3	$4\pi\times10^{-4}$	T, Wb/m^2 [i]
(Mass) magnetization	σ, M	emu/g	1 $4\pi\times10^{-7}$	A·m^2/kg Wb·m/kg
Magnetic moment	m	emu, erg/G	10^{-3}	A·m^2, joule per tesla (J/T)
Magnetic dipole moment	j	emu, erg/G	$4\pi\times10^{-10}$	Wb·m [i]
(Volume) susceptibility	χ, κ	dimensionless, emu/cm^3	4π $(4\pi)^2\times10^{-7}$	dimensionless henry per meter (H/m), Wb/(A·m)
(Mass) susceptibility	χ_ρ, κ_ρ	cm^3/g, emu/g	$4\pi\times10^{-3}$ $(4\pi)^2\times10^{-10}$	m^3/kg H·m^2/kg
(Molar) susceptibility	χ_{mol}, κ_{mol}	cm^3/mol, emu/mol	$4\pi\times10^{-6}$ $(4\pi)^2\times10^{-13}$	m^3/mol H·m^2/mol
Permeability	μ	dimensionless	$4\pi\times10^{-7}$	H/m, Wb/(A·m)
Relative permeability [j]	μ_r	not defined		dimensionless
(Volume) energy density, energy product [k]	W	erg/cm^3	10^{-1}	J/m^3
Demagnetization factor	D, N	dimensionless	$1/4\pi$	dimensionless

---

[a] Gaussian units and cgs emu are the same for magnetic properties. The defining relation is $B=H+4\pi M$.

[b] Multiply a number in Gaussian units by C to convert it to SI (e.g., 1 G$\times10^{-4}$ T/G$=10^{-4}$ T).

[c] SI (*Système International d'Unités*) has been adopted by the National Bureau of Standards. Where two conversion factors are given, the upper one is recognized under, or consistent with, SI and is based on the definition $B=\mu_0(H+M)$, where $\mu_0=4\pi\times10^{-7}$ H/m. The lower one is not recognized under SI and is based on the definition $B=\mu_0 H+J$, where the symbol I is often used in place of J.

[d] 1 gauss$=10^5$ gamma (γ).

[e] Both oersted and gauss are expressed as cm$^{-1/2}$·g$^{1/2}$·s^{-1} in terms of base units.

[f] A/m was often expressed as "ampere-turn per meter" when used for magnetic field strength.

[g] Magnetic moment per unit volume.

[h] The designation "emu" is not a unit.

[i] Recognized under SI, even though based on the definition $B=\mu_0 H+J$. See footnote *c*.

[j] $\mu_r=\mu/\mu_0=1+\chi$, all in SI. μ_r is equal to Gaussian μ.

[k] $B\cdot H$ and $\mu_0 M\cdot H$ have SI units J/m^3; $M\cdot H$ and $B\cdot H/4\pi$ have Gaussian units erg/cm^3.

R. B. Goldfarb and F. R. Fickett, U.S. Department of Commerce, National Bureau of Standards, Boulder, Colorado 80303, March 1985
NBS Special Publication 696 For sale by the Superintendent of Documents, U.S. Government Printing Office, Washington, DC 20402

In this presentation:

SI units

$$B = \mu_0(H + M) = \mu_0 H + J$$

$$B, J: \quad [\mathrm{T}] = [\frac{\mathrm{V\ sec}}{\mathrm{m}^2}]$$

$$H^*, M: \quad [\frac{\mathrm{A}}{\mathrm{m}}]$$

* Although SI unit of field is A/m, it is common to express field stregth in units of $\mu_0 H = B$ [Tesla]

from IEEE Magn. Soc. webpage

http://www.ieeemagnetics.org/images/stories/magnetic_units.pdf

For reading

B.D. Cullity and C.D. Graham: Introduction to Magnetic Materials.
IEEE Press and Wiley (2009)

S. Tumanski: Handbook of Magnetic Measurements.
CRC Press Taylor & Francis (2009)

F. Fiorillo: Measurement and Characterization of Magnetic Materials.
Elsevier Academic Press (2004)

R. Hilzinger and W. Rodewald: Magnetic Materials.
Edited by Vacuumschmelze GmbH, Publicis Publishing, Erlangen (2013)

D.C. Jiles: Introduction to Magnetism and Magnetic Materials.
Chapman & Hall, London (1995)

G. Bertotti: Hysteresis in Magnetism.
Academic Press, New York (1998)

M. Coey: Magnetism and Magnetic Materials.
Cambridge University Press (2010)

A. Hubert and R.S.: Magnetic Domains. Springer Verlag (1998)

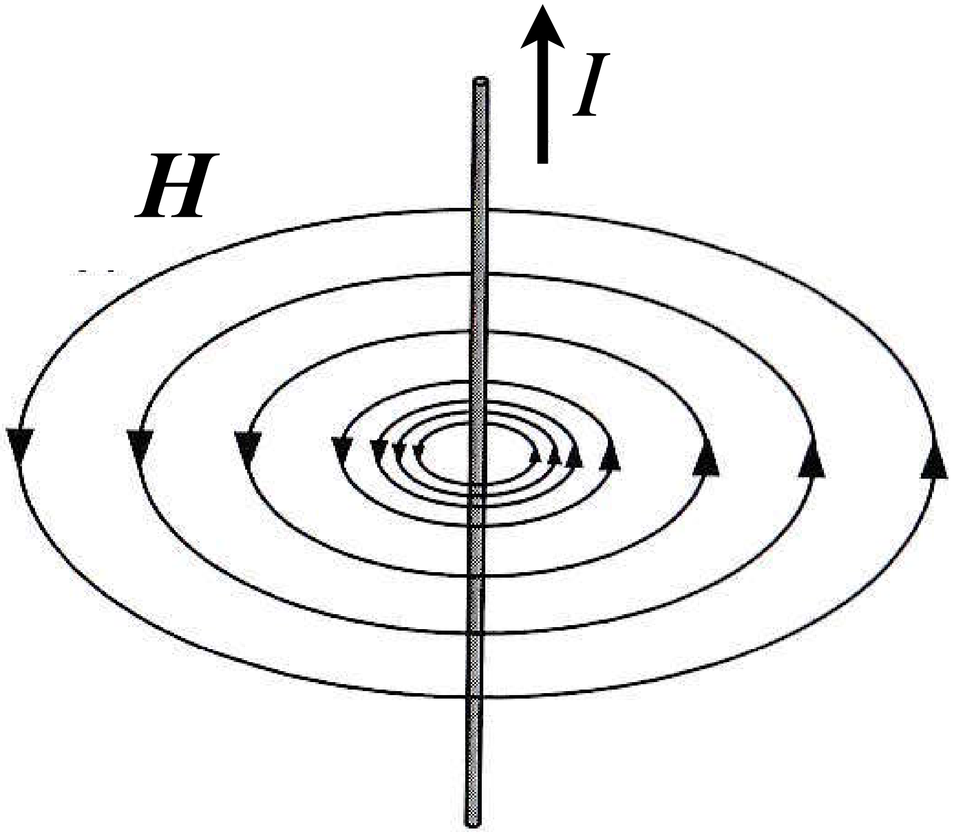
Many figures in this presentation are taken from these references – special thanks to the authors

1.

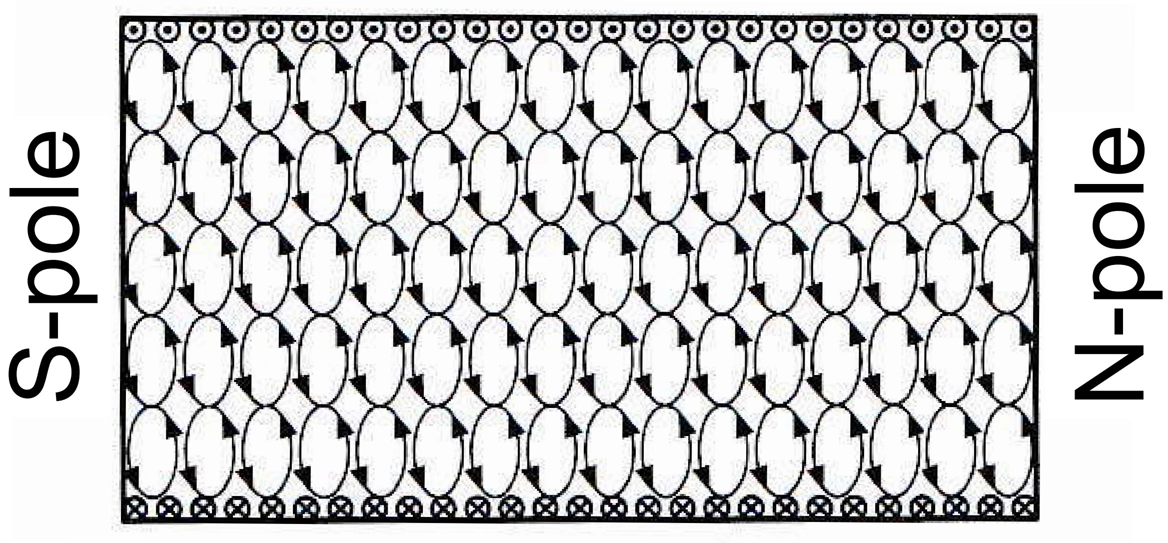
Production of Magnetic Field

1. Production of Magnetic Field

Two possibilities to generate magnetic field:



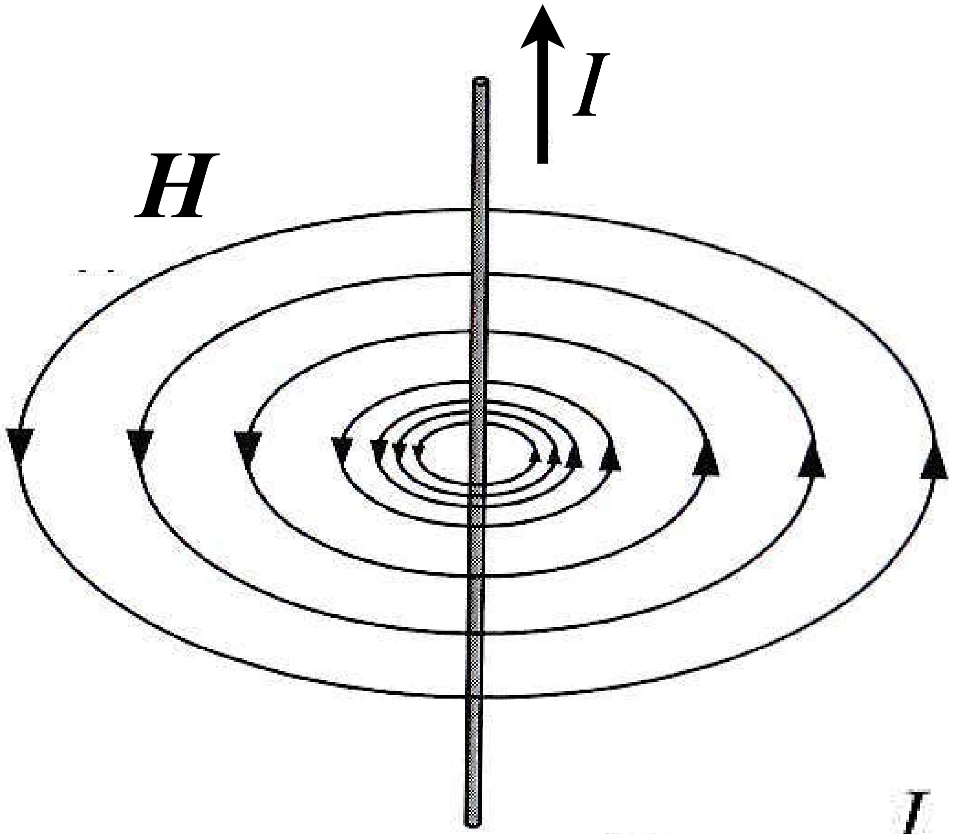
By electrical currents flowing in conductor
→ electromagnetic coils



By exploiting the ordered array of quantummechanical electronic currents circulating in a magnetic material (→ permanent magnets)

1. Production of Magnetic Field

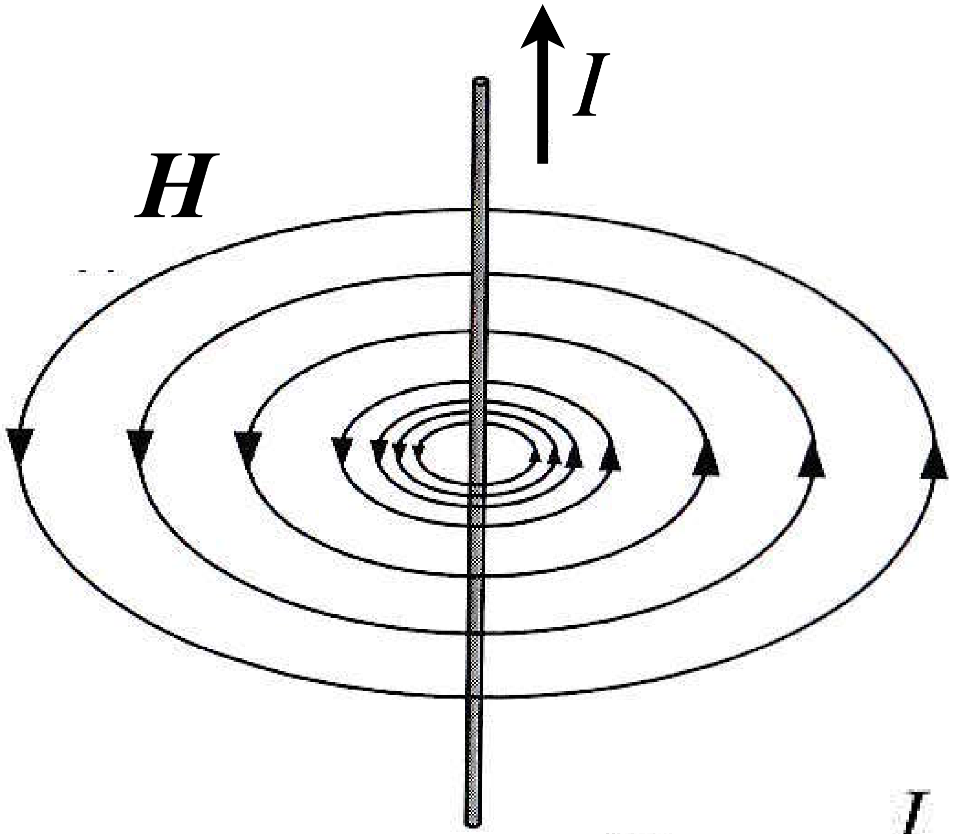
1.1 Electromagnetic coils

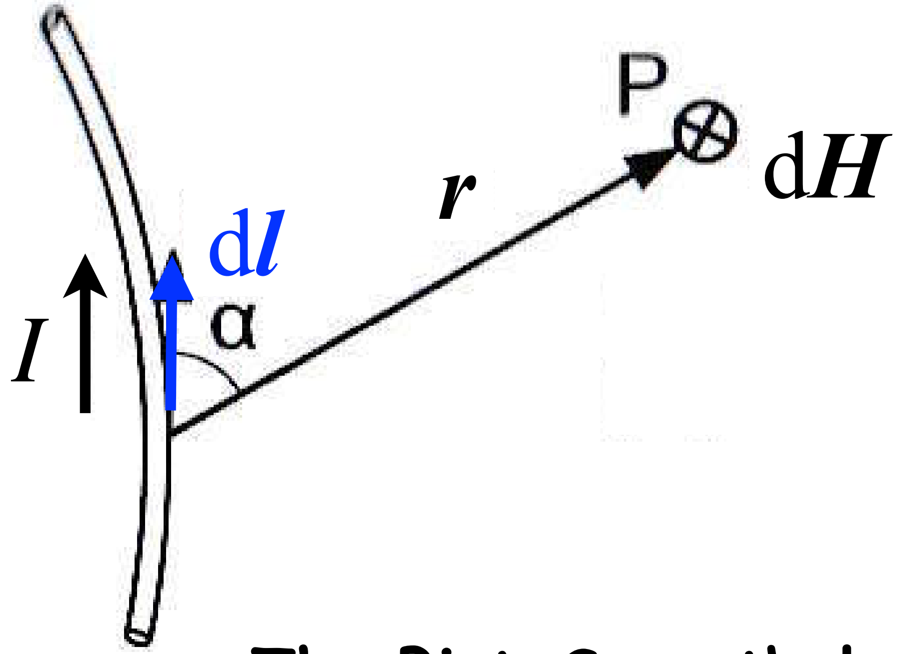


$$\mathrm{rot}\,\boldsymbol{H} = \boldsymbol{j} \longrightarrow H = \frac{I}{2\pi \cdot r}$$

1. Production of Magnetic Field

1.1 Electromagnetic coils





The Biot-Savart's law

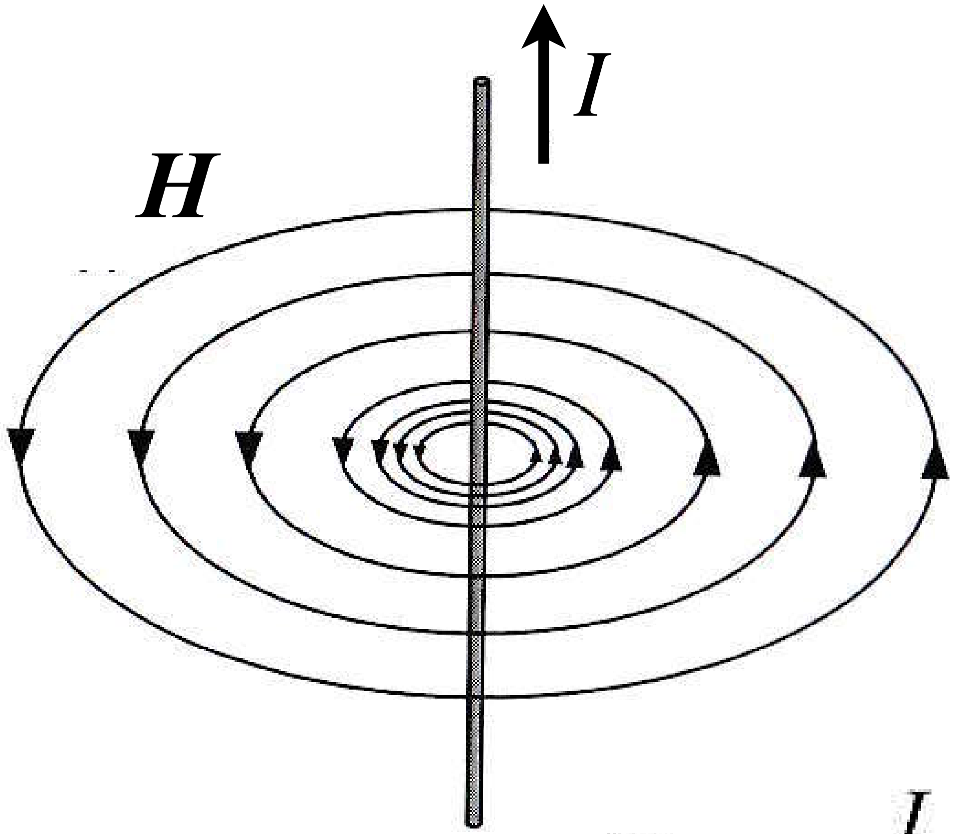
$$\mathrm{rot}\,\boldsymbol{H} = \boldsymbol{j} \longrightarrow H = \frac{I}{2\pi \cdot r}$$

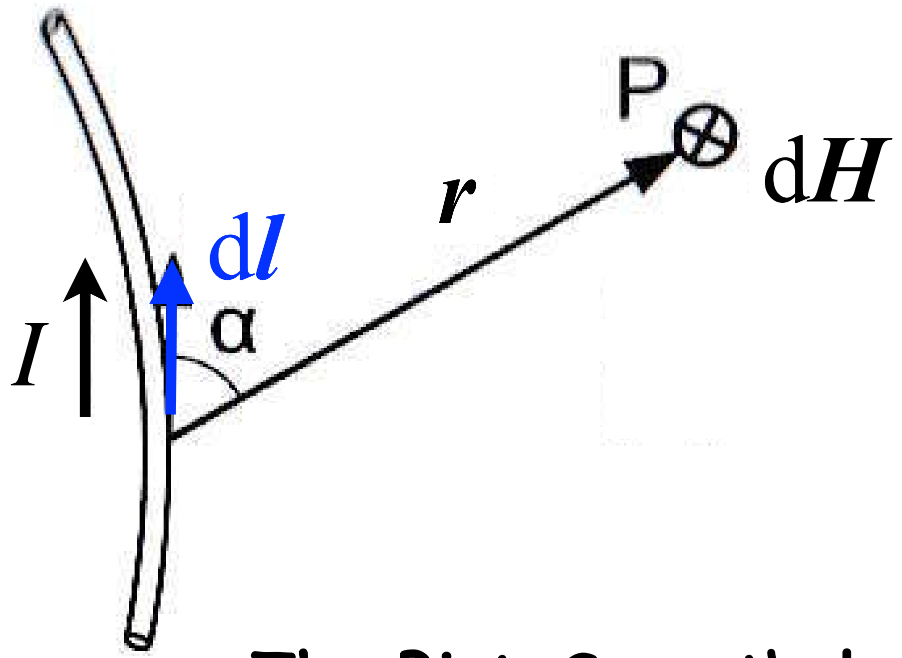
$$\mathrm{d}\boldsymbol{H} = \frac{1}{4\pi \cdot r^3} \cdot I \cdot \mathrm{d}\boldsymbol{l} \times \boldsymbol{r} \quad \text{or}$$

$$\mathrm{d}H = \frac{1}{4\pi \cdot r^2} \cdot I \cdot \mathrm{d}l \cdot \sin\alpha$$

1. Production of Magnetic Field

1.1 Electromagnetic coils





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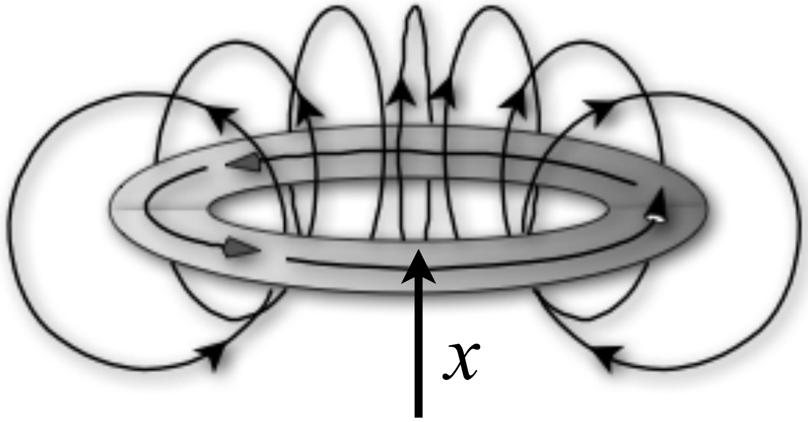
$$\Big\downarrow \int$$

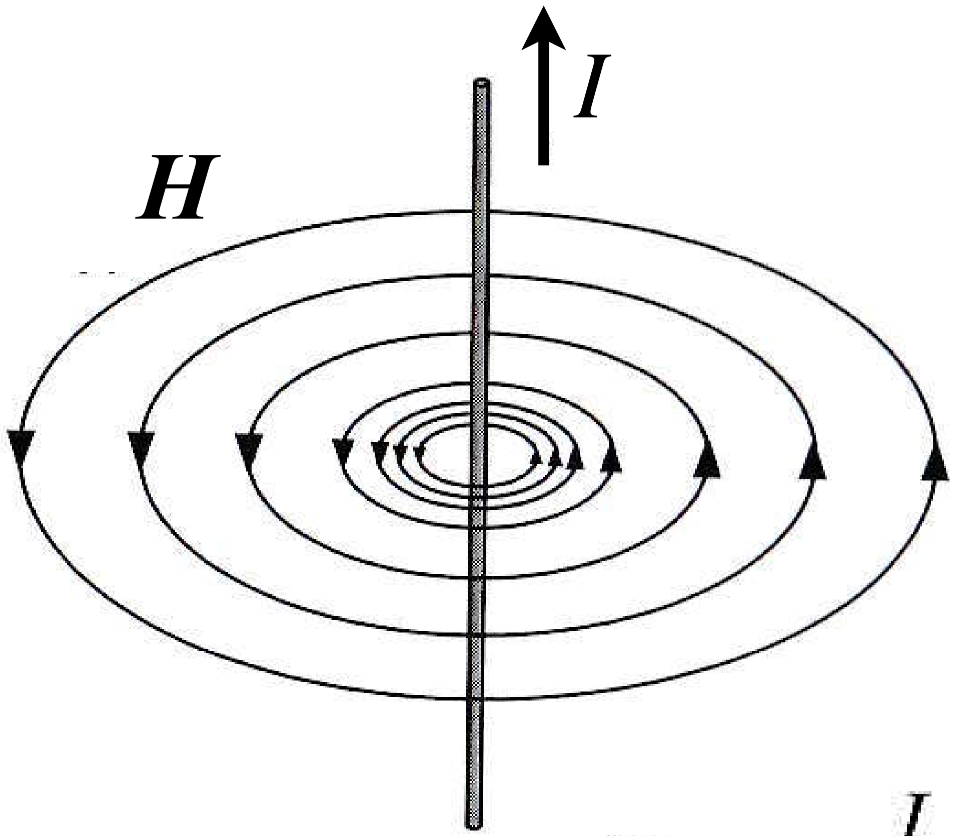
Current loop of radius r

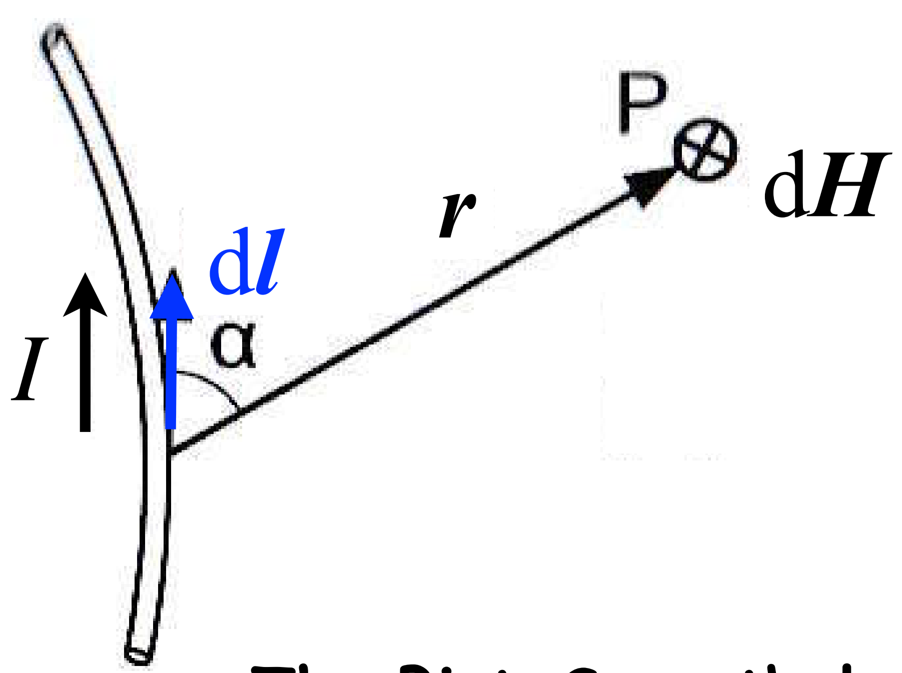


$$H_x(x) = \frac{I \cdot r^2}{2(r^2 + x^2)^{3/2}}$$

1. Production of Magnetic Field

1.1 Electromagnetic coils





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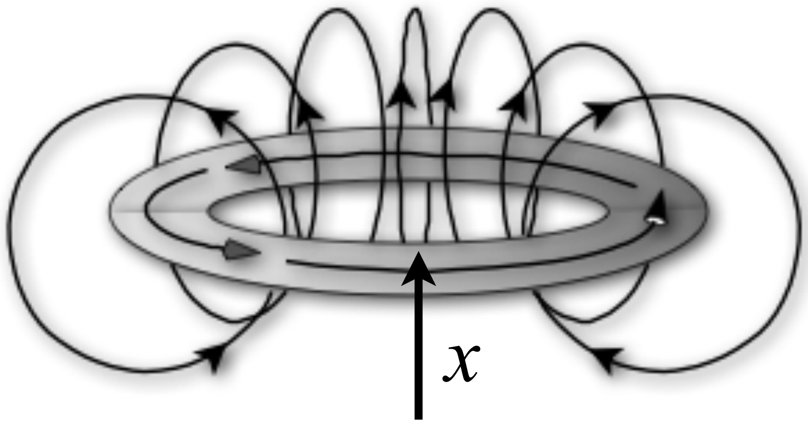
$$\Big\downarrow \int$$

$$H_x(x) = \frac{n \cdot I \cdot r^2}{2(r^2 + x^2)^{3/2}}$$

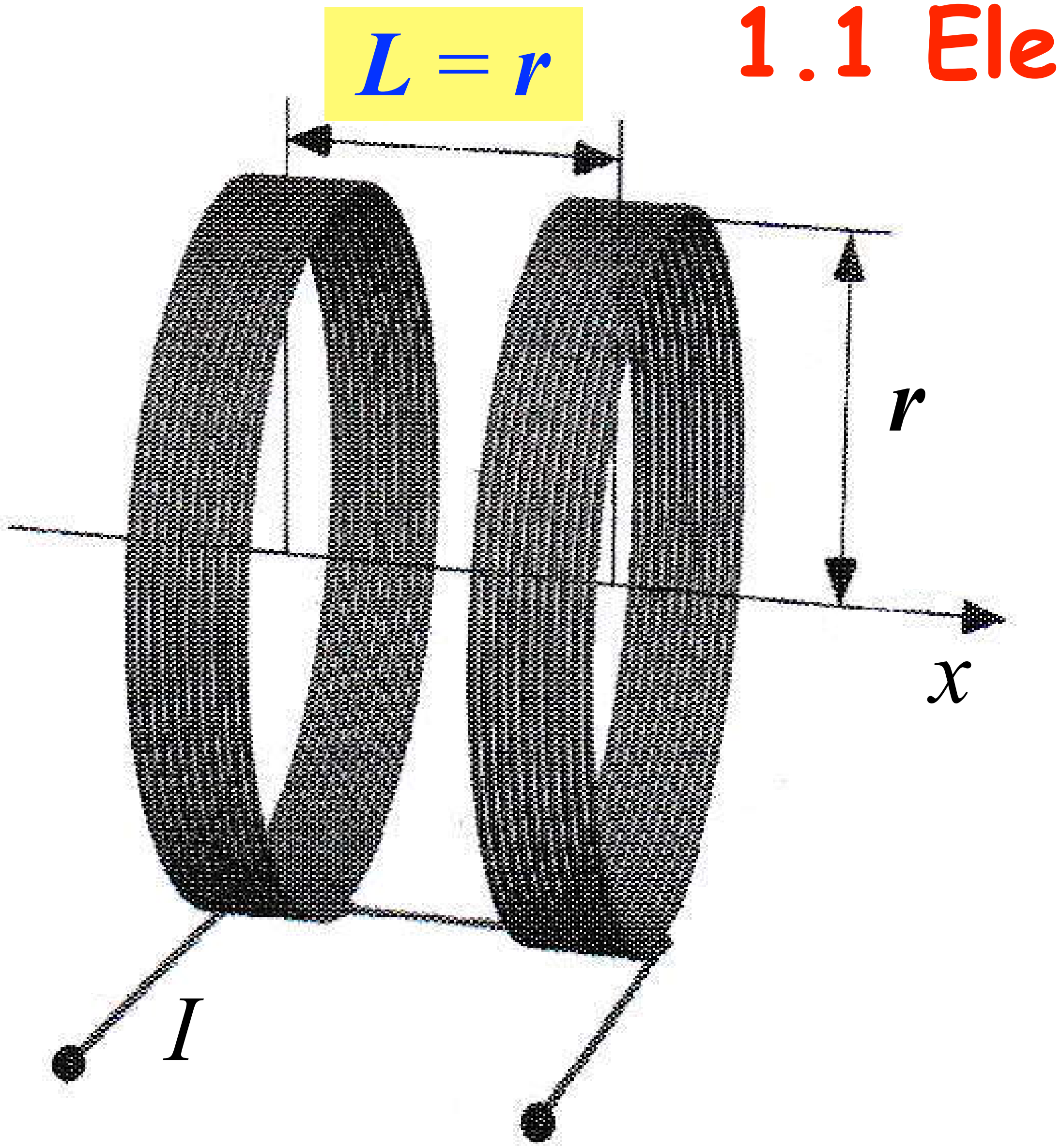
Field of coil with n windings

Current loop of radius r



1. Production of Magnetic Field

1.1 Electromagnetic coils



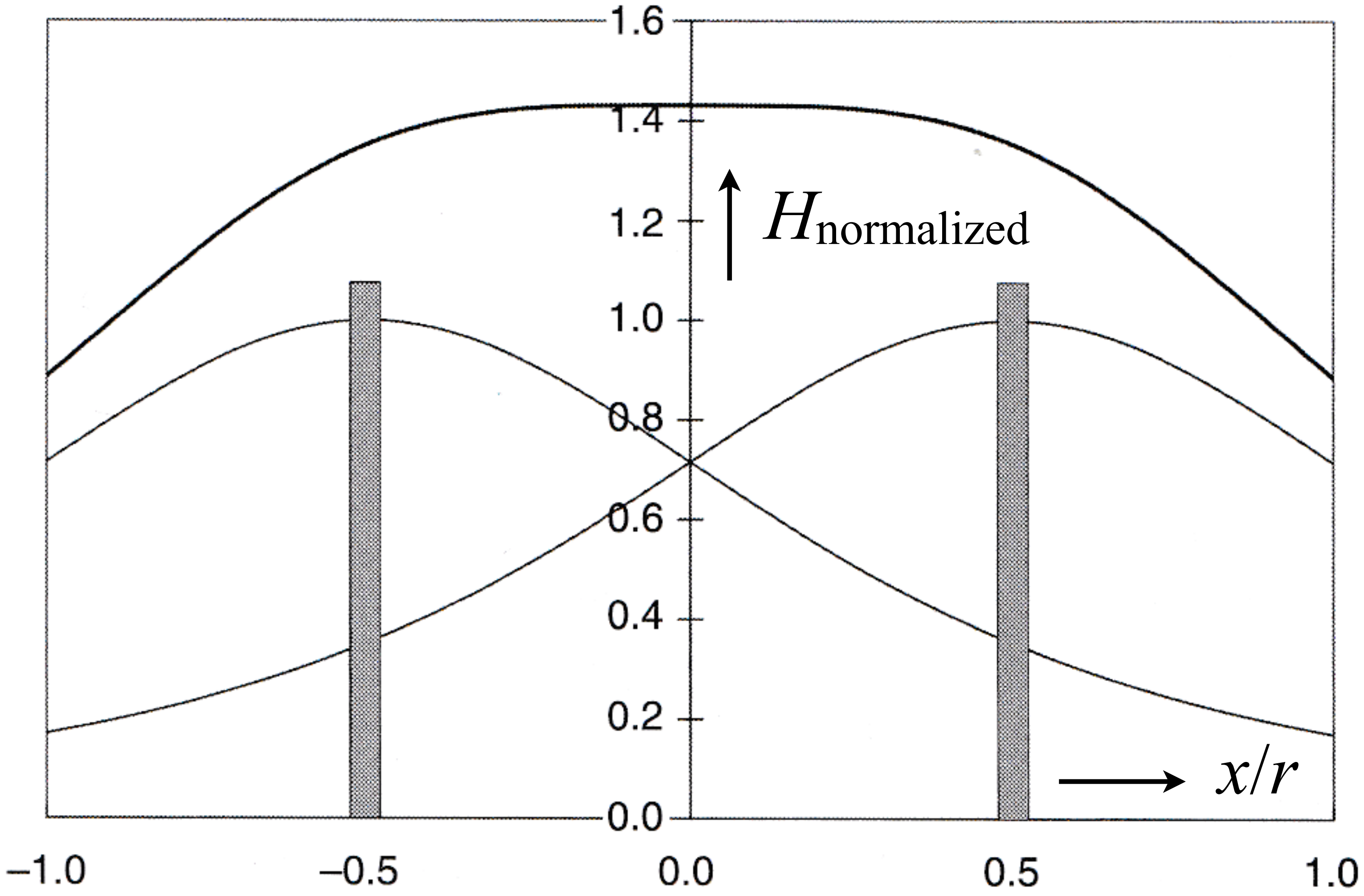
Axial field component at centre point:

$$H_x(x = r/2) = 0.7155 \cdot \frac{n \cdot I}{r}$$

Homogeneous (small) field in large volume

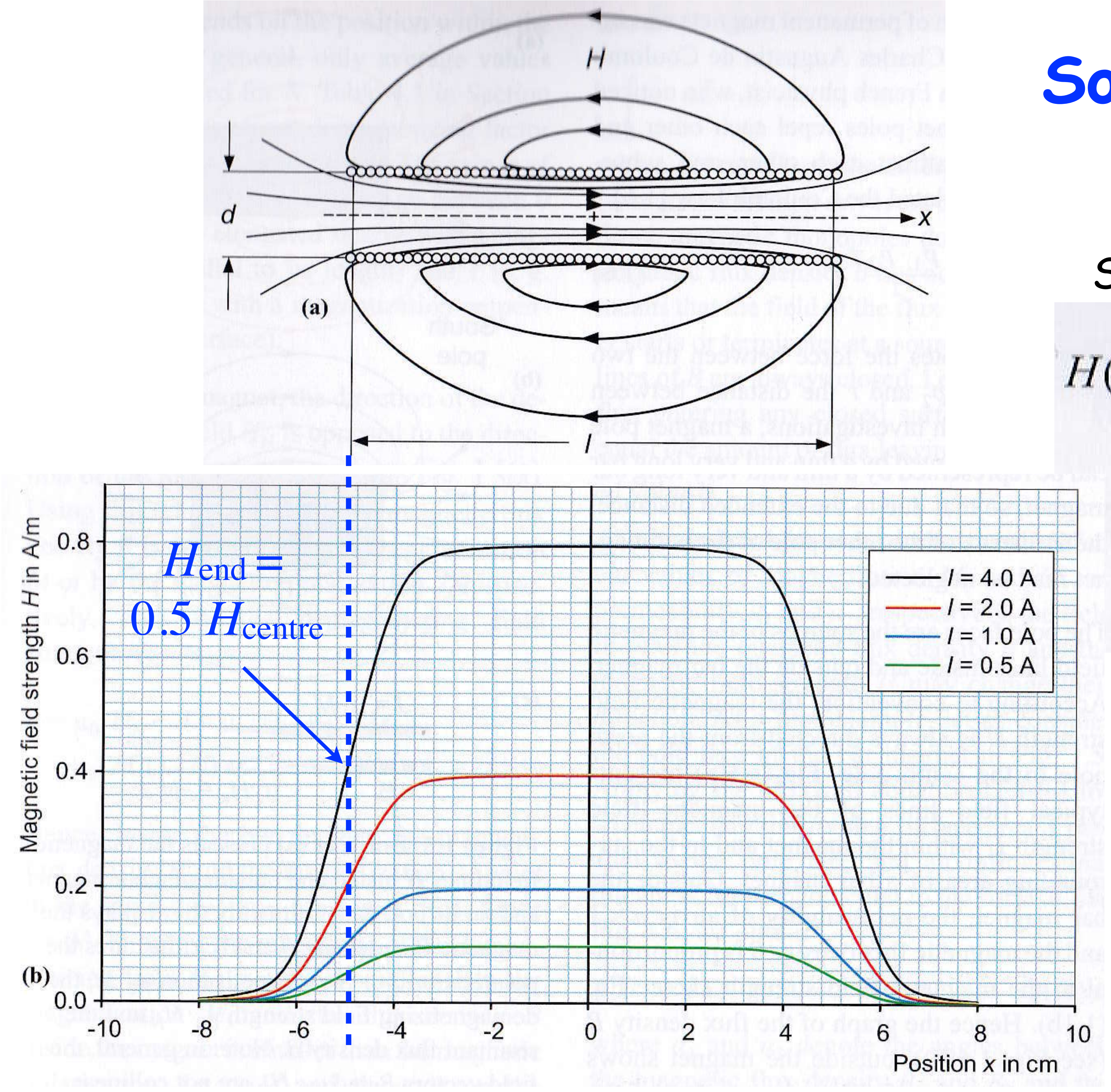
Helmholtz coils

$$H_x(x) = \frac{n \cdot I}{2r} \cdot \left[\left(1 + \frac{x^2}{r^2}\right)^{-\frac{3}{2}} + \left(1 + \frac{(r-x)^2}{r^2}\right)^{-\frac{3}{2}} \right]$$



1. Production of Magnetic Field

1.1 Electromagnetic coils



Solenoid

Solenoid with one layer of winding:

$$H(x) = \frac{n\cdot I}{l}\cdot\left\{\frac{(l+2x)}{2\sqrt{[d^2+(l+2x)^2]}} + \frac{(l-2x)}{2\sqrt{[d^2+(l-2x)^2]}}\right\}$$

Field at centre:

$$H(x=0) = \frac{n\cdot I}{l}\cdot\frac{l}{\sqrt{[d^2+l^2]}}$$

Long solenoid ($l \gg d$):

$$H(x=0) = \frac{n\cdot I}{l}$$

1. Production of Magnetic Field

1.1 Electromagnetic coils

Remarks:

- Higher field: better increase n/l by adding more layers of winding rather than increasing current

  $H \sim I$, but $heat \sim I^2R$

  Thus doubling number of winding layers and keeping current constant will double H, R, and amount of $heat$; whereas doubling of current will double H, but will quadruple $heat$

- Typical field: 0.1 T , higher field requires cooling

Solenoid

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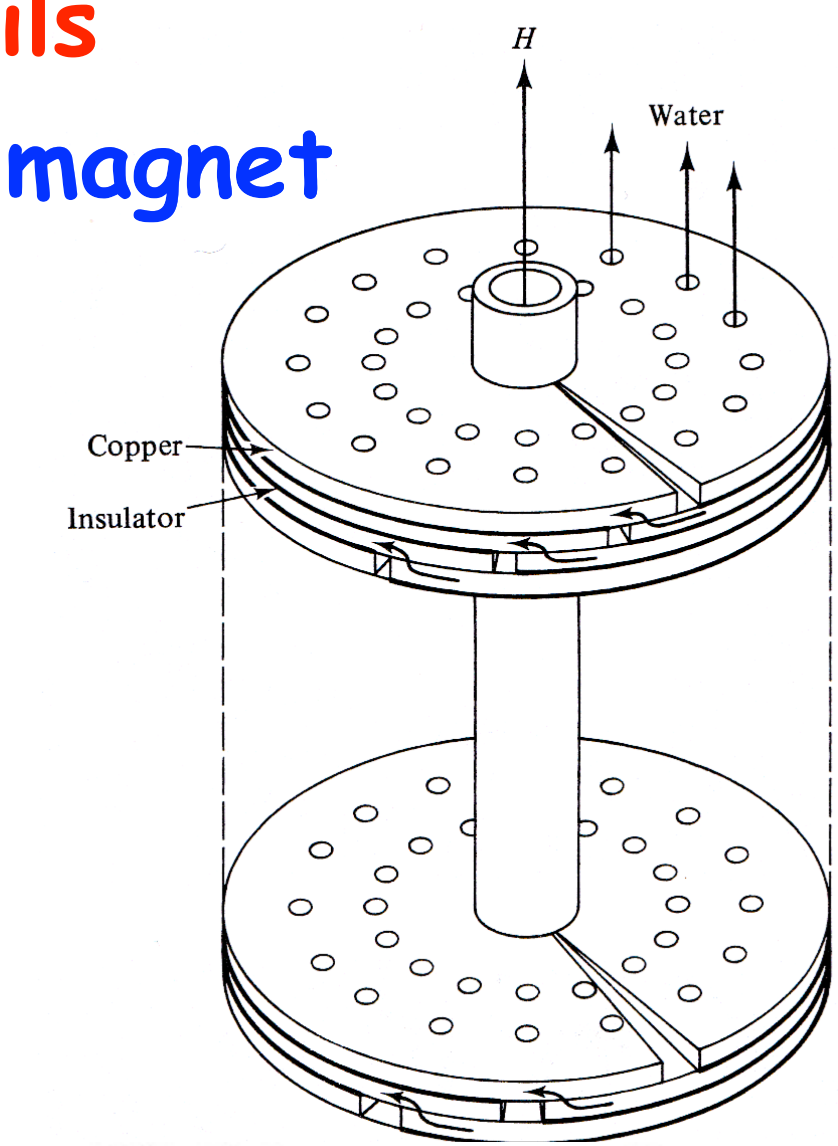
1. Production of Magnetic Field

1.1 Electromagnetic coils

High-Field Solenoid: Bitter magnet



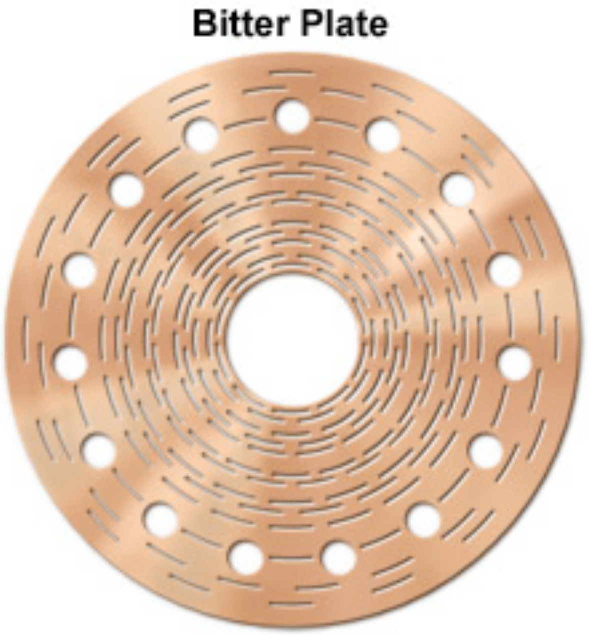
High field: requires large power input
⟶ two major design problems:

1) Large amount of heat (note: maintaining magnetic field by current is process of zero efficiency: all input power goes into heat)

2) Mechanical strength to resist large forces acting on current carriers has to be provided

Bitter magnet:

- Winding composed of Cu disks, ~30 cm diameter
- Insulated from each other, clamped together
- Rotated by 20°: overlap = conduction path
- Cooling water pumped through holes,
- Helical current path, acts like solenoid
- Typical field: 45 Tesla in 30 mm bore, requires current of 67.000 A, power input of 20 MW
- Requires large motor-generator sets

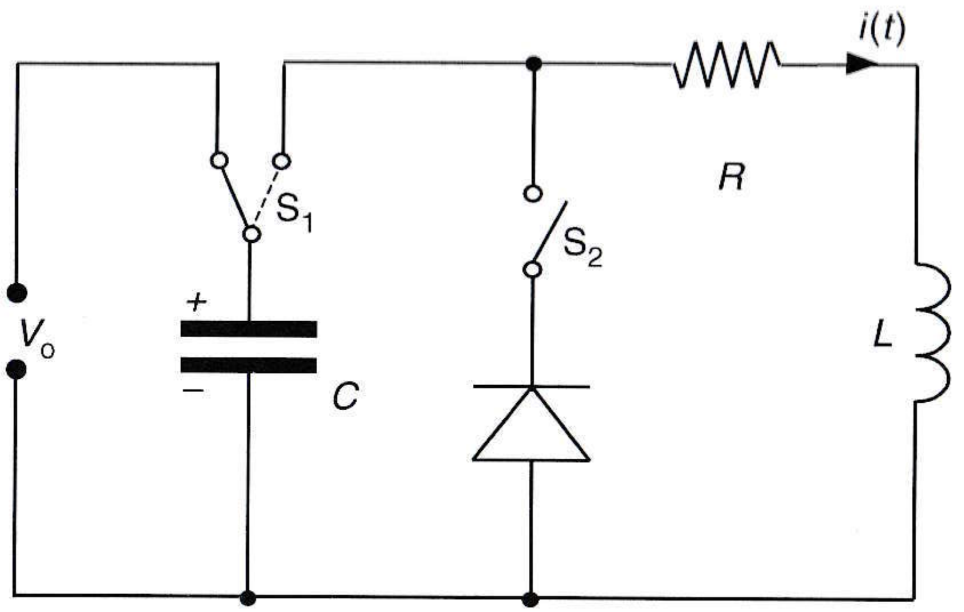


1. Production of Magnetic Field

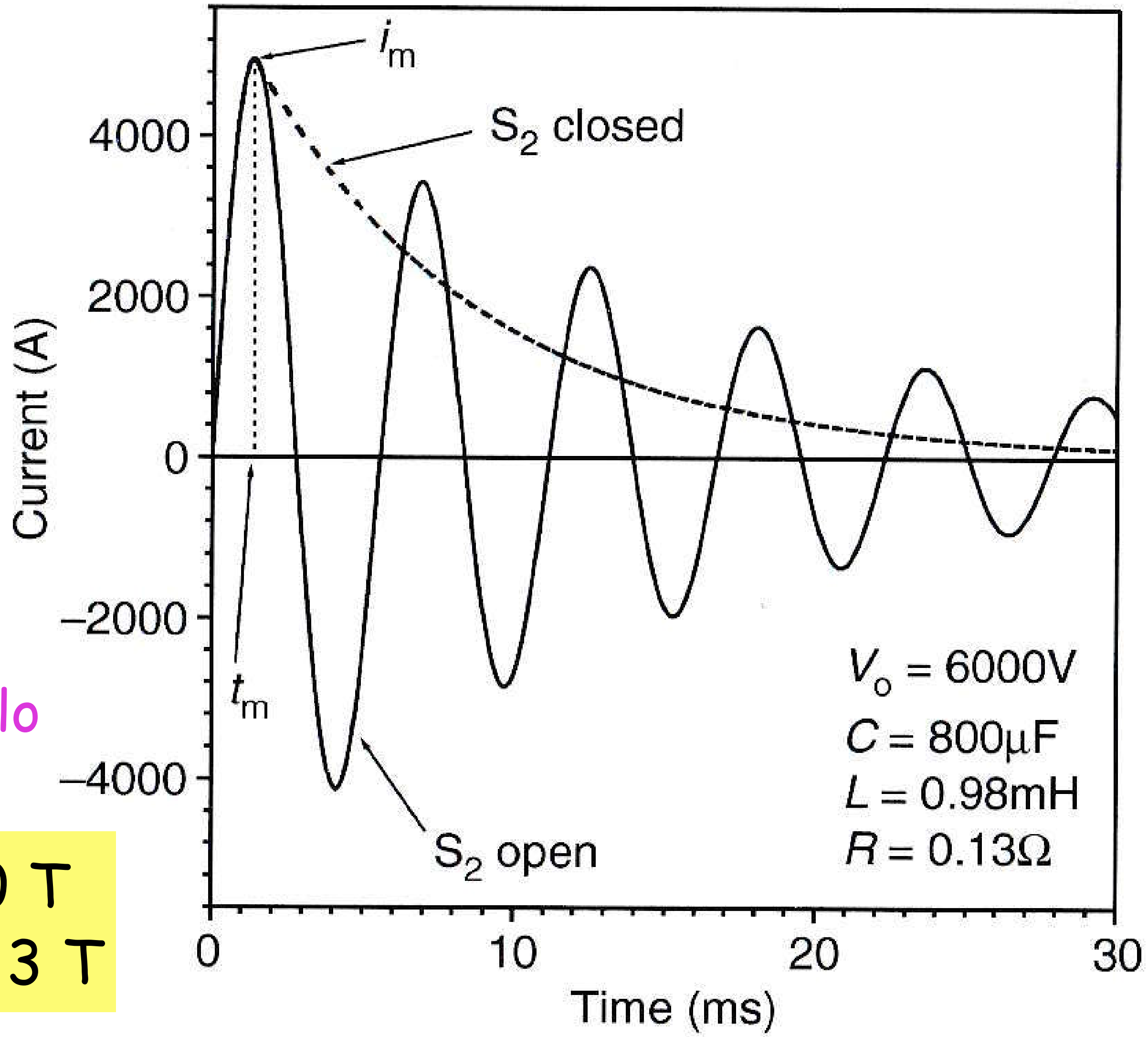
1.1 Electromagnetic coils

High-field Solenoid: Pulsed Fields

High pulsed fields by discharging capacitor bank through conventional solenoid



Courtesy F. Fiorillo



$$L\frac{\partial^2 i(t)}{\partial t^2} + R\frac{\partial i(t)}{\partial t} - \frac{1}{C}i(t) = 0,$$

$$i(t) = \frac{V_0}{L}\exp\left(-\frac{R}{2L}t\right)\frac{\sin \omega t}{\omega}$$

- Cooled: up to 20 T
- Uncooled: up to 3 T

In shown arrangement with given inductance L of solenoid, an oscillating damped discharge is obained (switch S_2 open). If S_2 is closed at maximum field, the diode prevents capacitor from discharging with reversed polarity and current decays from maximum value with time constant $\tau_1 = L/R$

1. Production of Magnetic Field

1.1 Electromagnetic coils

Superconducting Solenoids

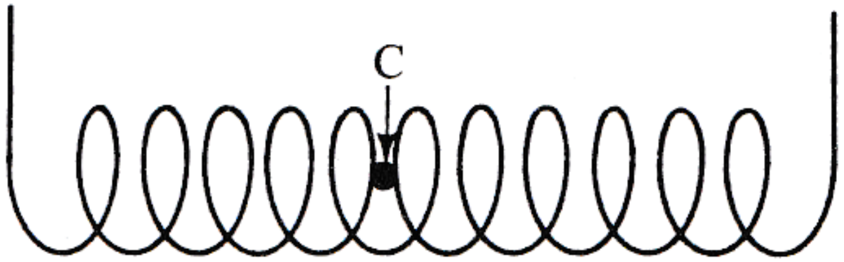
- DC fields up to about 20 T can be obtained by superconducting solenoids, commonly used for fields above 2 T
- Type II superconductor with high critical current and critical field: Nb-Ti or Nb_3Sn
- Cooling of coil by liquid helium (4.2 K), sample temperature up to room temperature
- Shortcut of superconducting coil: persistant mode, no power consumption over months
- Danger: quench by local heating

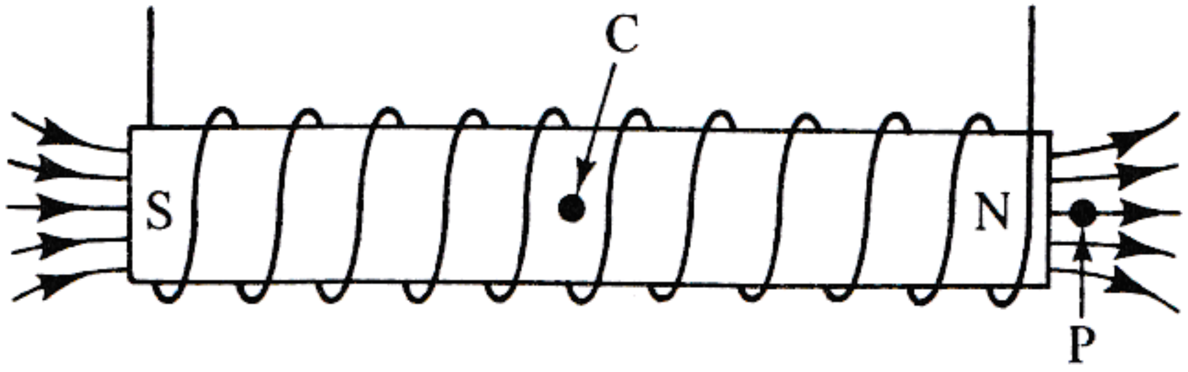
1. Production of Magnetic Field

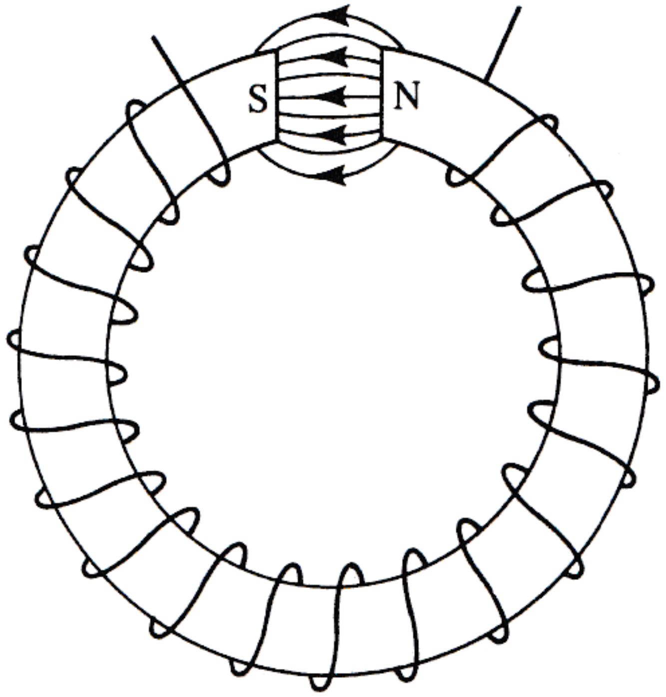
1.2 Electromagnets

DC fields up to 2 T, most commonly used magnetic field source in labs

Evolution of the electromagnet:







Solenoid, flux density at center C:

$$B = \mu_0 H = \mu_0 \frac{n \cdot I}{l}$$

Solenoid with iron rod, flux density at center C:

$$B = \mu_0(H + M) = \mu_0 \mu_r H = \mu_r \mu_0 \frac{n \cdot I}{l}$$

Iron has multiplied field due to the current by factor of μ_r ;
same field occurs just outside rod at P
⟶ large field obtained with low current
(e.g. H_{coil} = 1 mT, μ_r = 2000: H_{outside} = 2 T)
Problem: flux lines outside iron diverge and field decreases rapidly

Bended iron rod with gap:

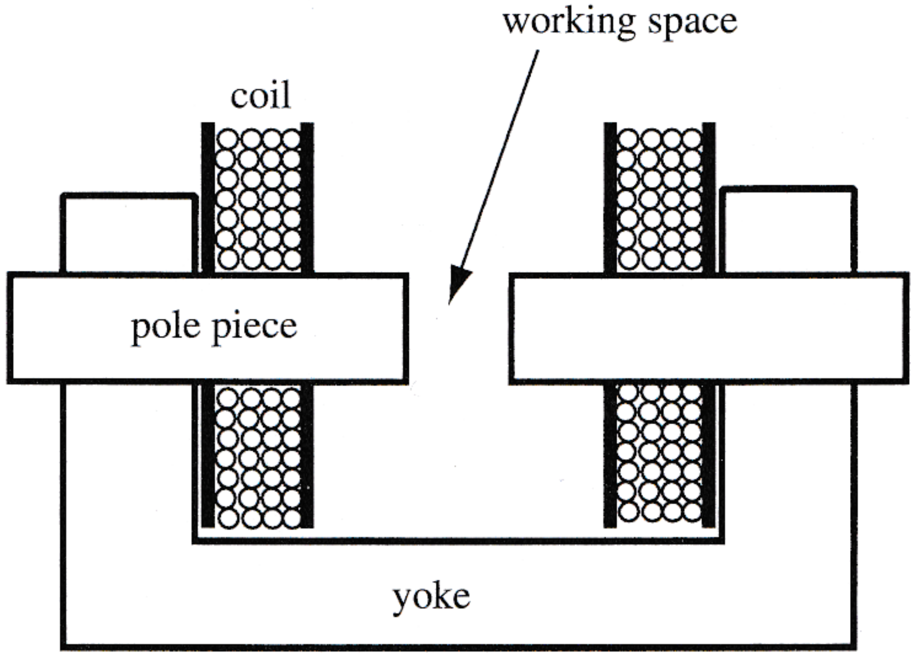
Flux travels directly from pole to pole across gap. Contribution of iron to gap field (if saturated) = 2.15 T

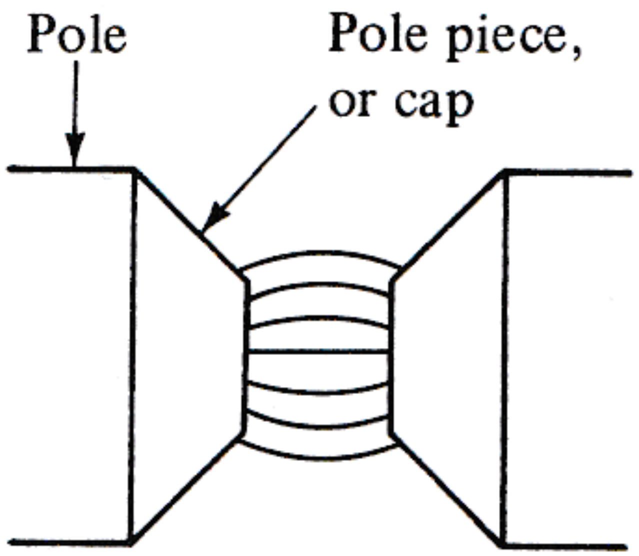
1. Production of Magnetic Field

1.2 Electromagnets

DC fields up to 2 T, most commonly used magnetic field source in labs

Evolution of the electromagnet:





Electromagnet:

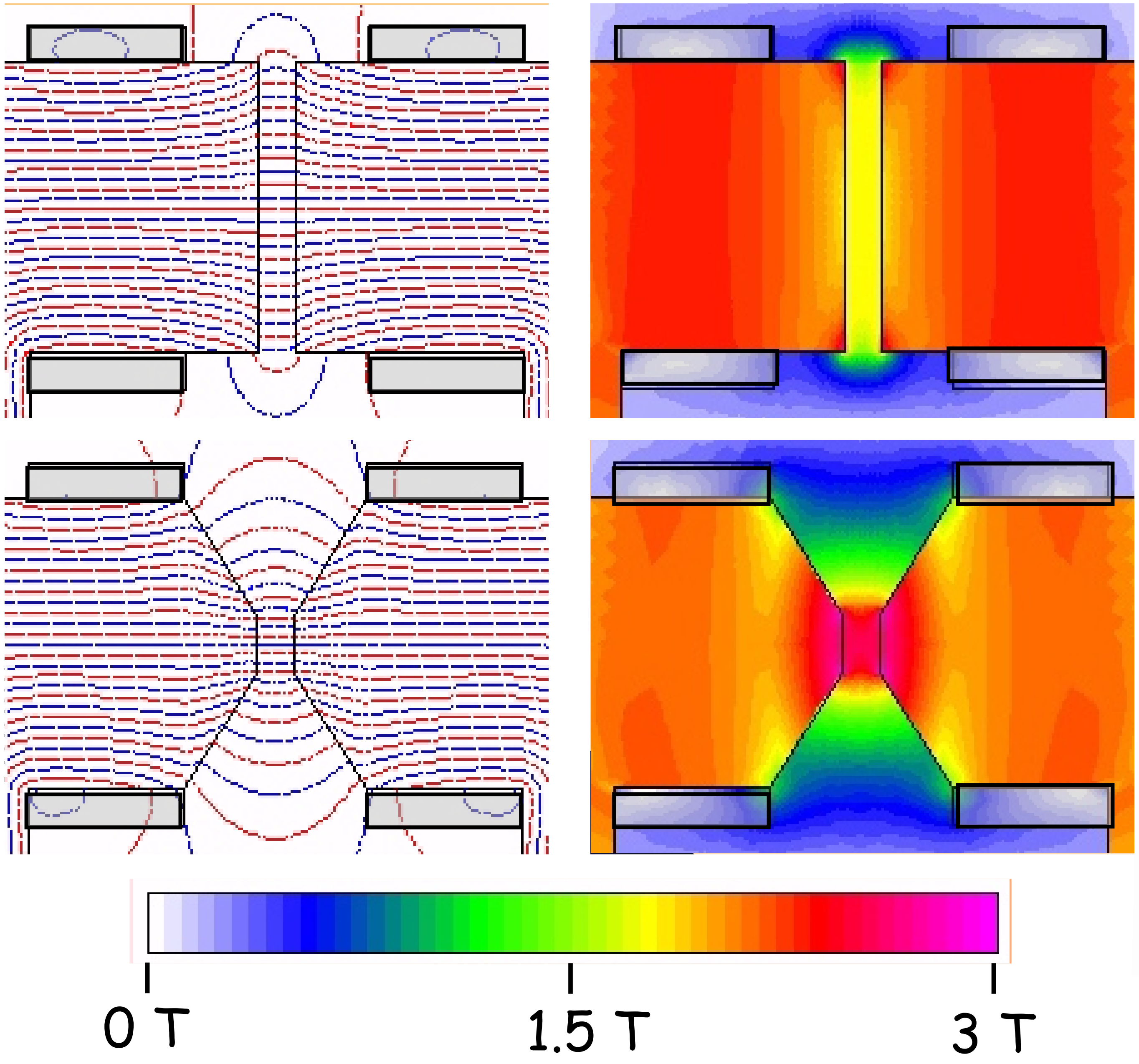
- Windings close to gap
- Core and yoke made of iron, annealed for high permeability
- Windings water-cooled
- Pole diameter: up to 30 cm
- Flat poles for uniform field
- Tapered poles: free poles formed on tappered surfaces contribute to field at gap center, can achieve fields higher than $\mu_0 M_s$ (> 3T for gap length of 5-10 mm). Optimum taper angle: 54,74°
- Pole pieces made of CoFe (M_s about 10% higher than for pure Fe)

1. Production of Magnetic Field

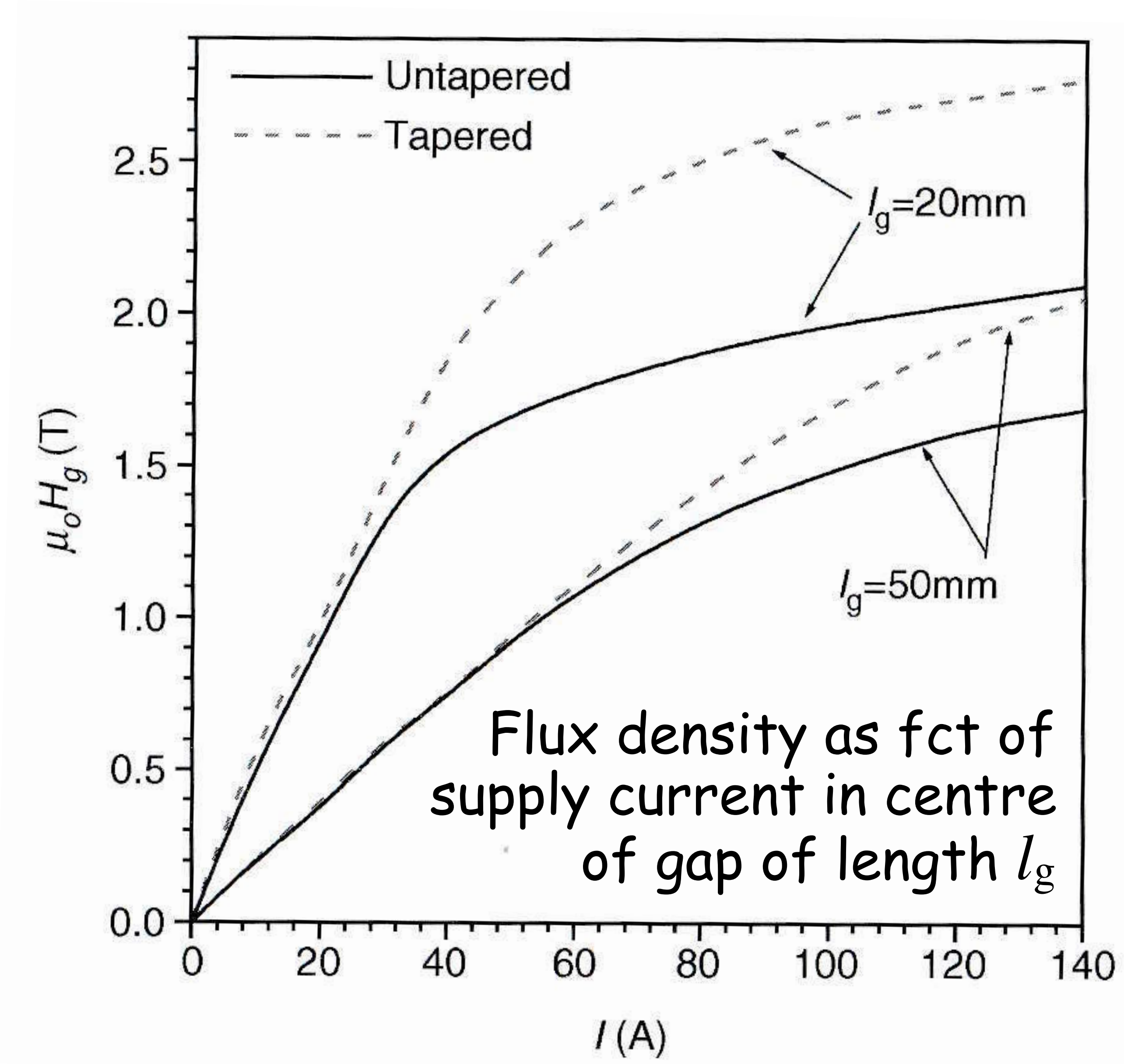
1.2 Electromagnets

DC fields up to 2 T, most commonly used magnetic field source in labs

Finite element simulations:
lines and contour maps of induction B



Courtesy F. Fiorillo



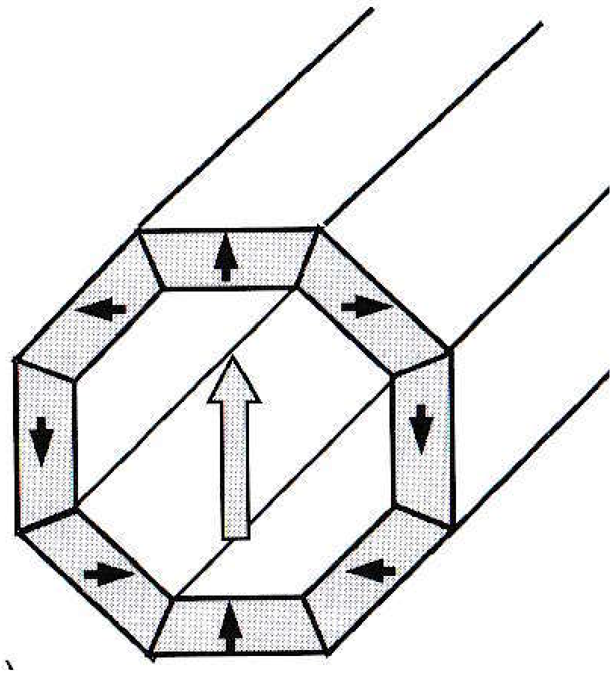
Flux density as fct of supply current in centre of gap of length l_g

The larger gap, the smaller field.

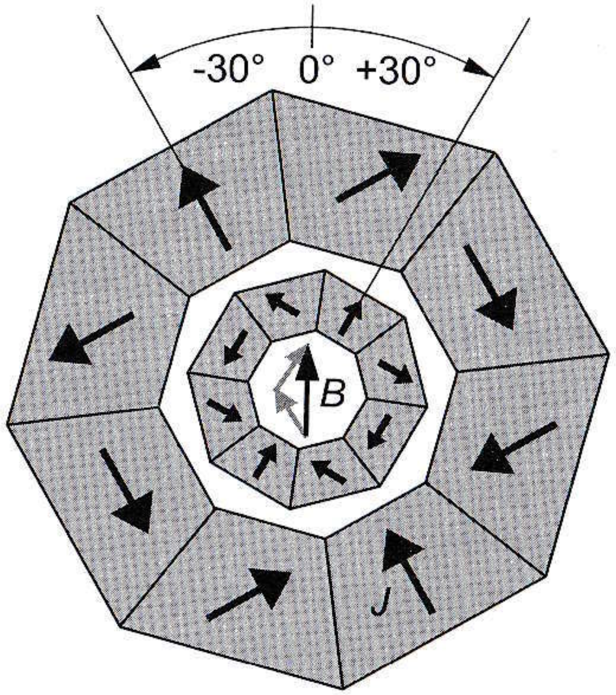
1. Production of Magnetic Field

1.3 Permanent Magnets

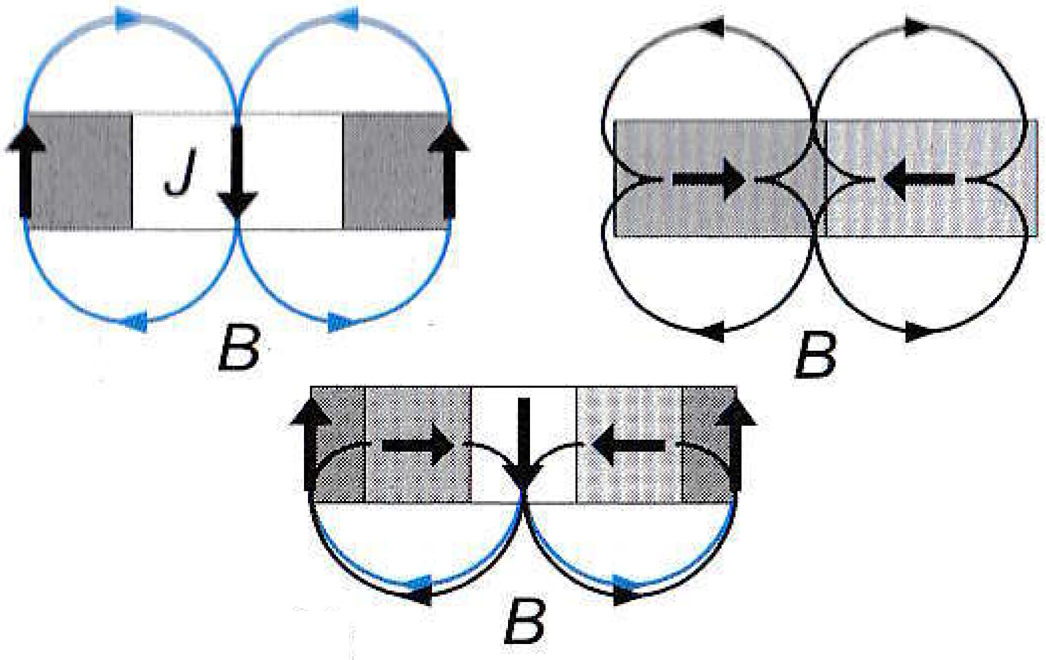
Fields up to ~2T by appropriate arrangement of permanent magnets



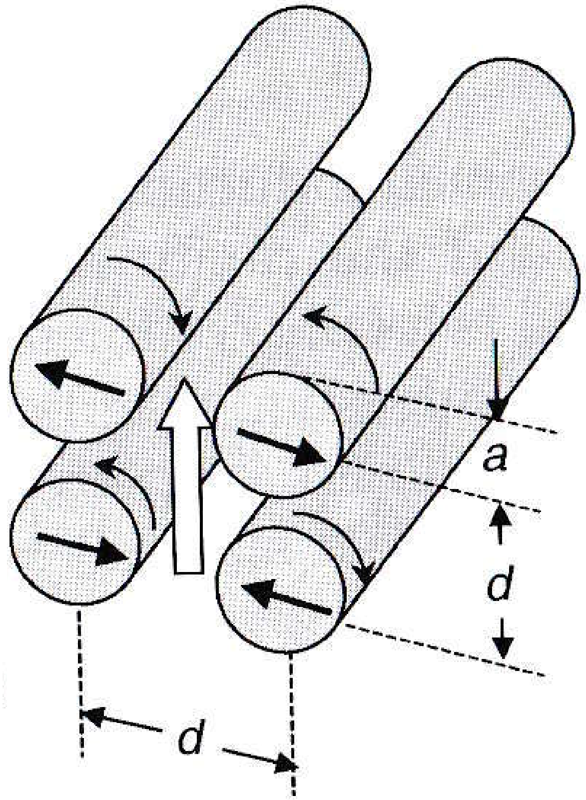
Technical design of Halbach cylinder: array of uniformly magnetized NdFeB magnets, uniform field accross diameter



Two concentric Halbach arrays: vectorial addition of flux density in centre. By synchronous but opposite rotation, the field strength can be continuously varied



One-sided field by superposition



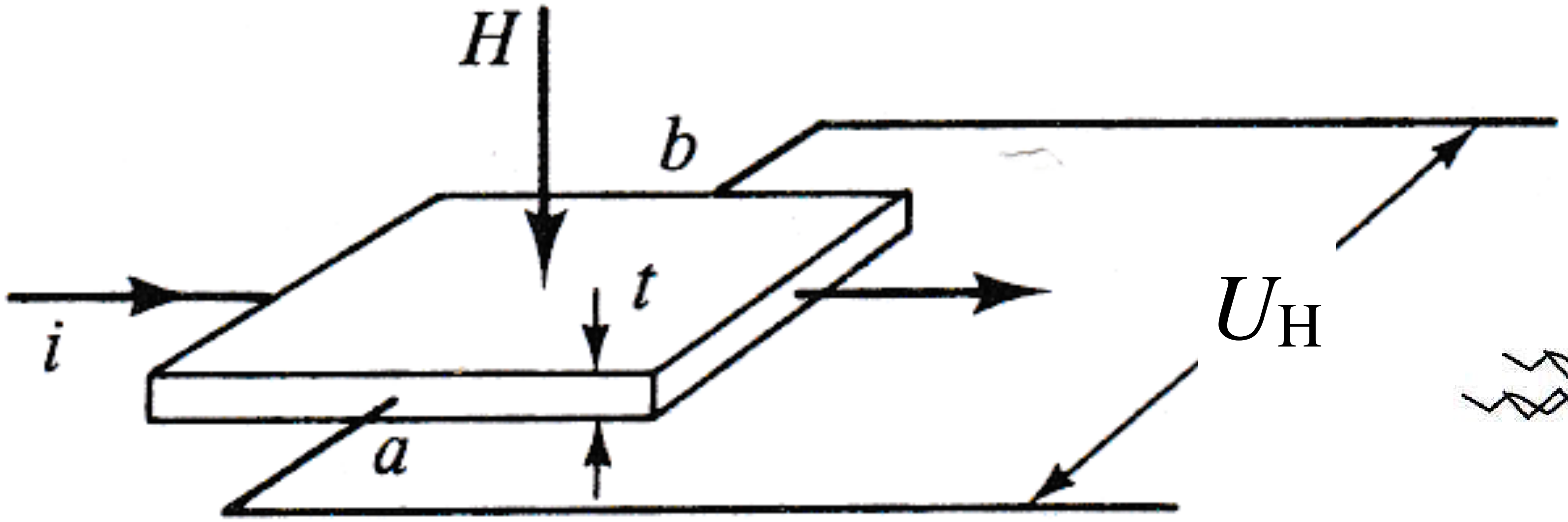
Simplification of above aray: continuously varying field by counter rotation of 4 transversly magnetized rods

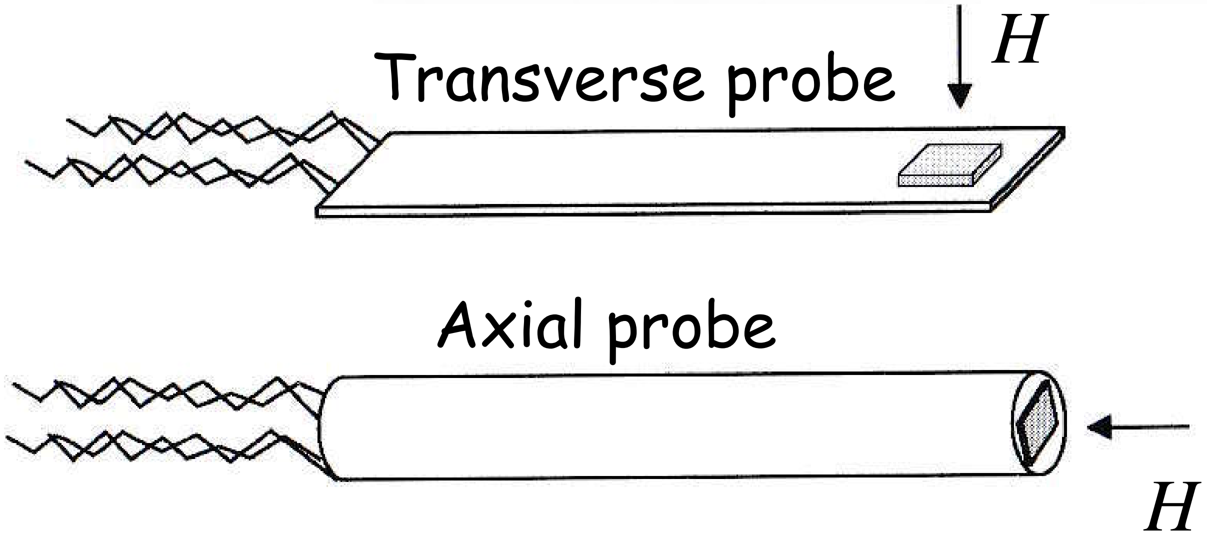
2.

Measurement of Magnetic Field Strength

2. Measurement of magnetic field strength

2.1 Hall Probe





$$U_H = R_H \frac{I \bullet H}{t}$$

R_H: Hall coefficient

- Hall probe = plate made of InSb or GaAs semiconductor
- H-field perpendicular to plate distorts current path and emf U_H is developed between a and b
- Multirange instruments, sensitive to field range from μT to 3T
- Uncertainty in field reading: 1-5% in hand-held Gaussmeters
- Alternating fields up to some 10 kHz can be measured
- Calibration by accuratly known fields required
- Low-field probes: zero must be set with probe in magnetically shielded cylinder to eliminate Earth's field

Gaussmeter

2. Measurement of magnetic field strength

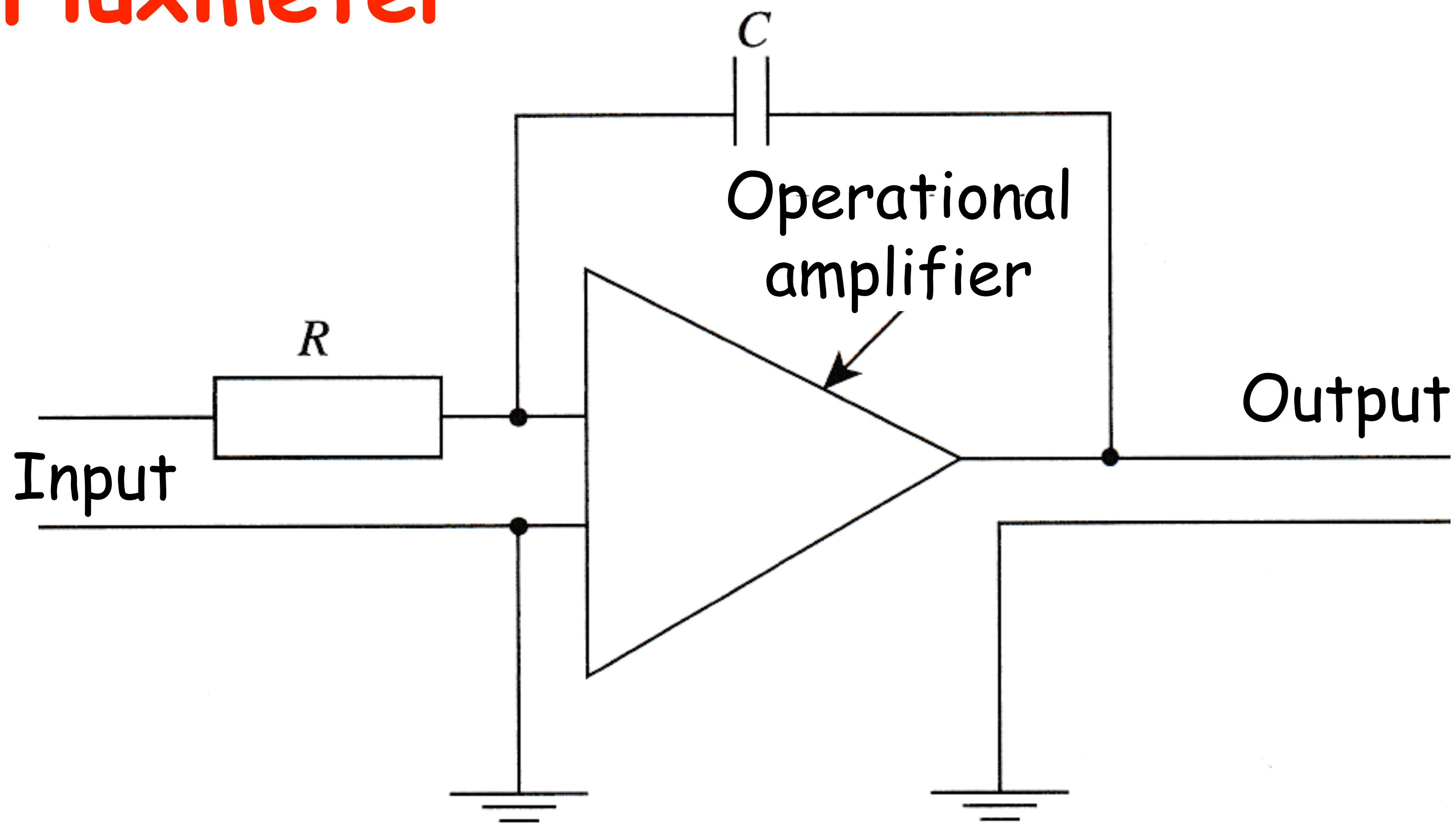
2.2 Fluxmeter

- Faraday's law: a changing magnetic flux φ through a coil of n turns generates a voltage in coil proportional to rate of change of flux:

$$U(t) = -n \frac{d\varphi}{dt} \quad [\text{Volt}]$$

$$U(t)\, dt = -n\, d\varphi$$

$$\int_0^t U(t)\, dt = -n \int_{\Phi 1}^{\Phi 2} d\varphi = -n\, \Delta\varphi$$



$$U_{\text{out}} = -1/RC \int U_{\text{in}}\, dt$$

- Instrument to integrate voltage from pick-up coil is called fluxmeter = electronic integrator (based on capacitive feedback around operational amplifier) that provides voltage output

- With $B = \varphi/A$ (flux density in pick-up coil of cross section A):

$$\int U(t)\, dt = -n\, A\, \Delta B \quad [\text{Vsec}]$$

Fluxmeter measures changes in flux density

2. Measurement of magnetic field strength

2.2 Fluxmeter

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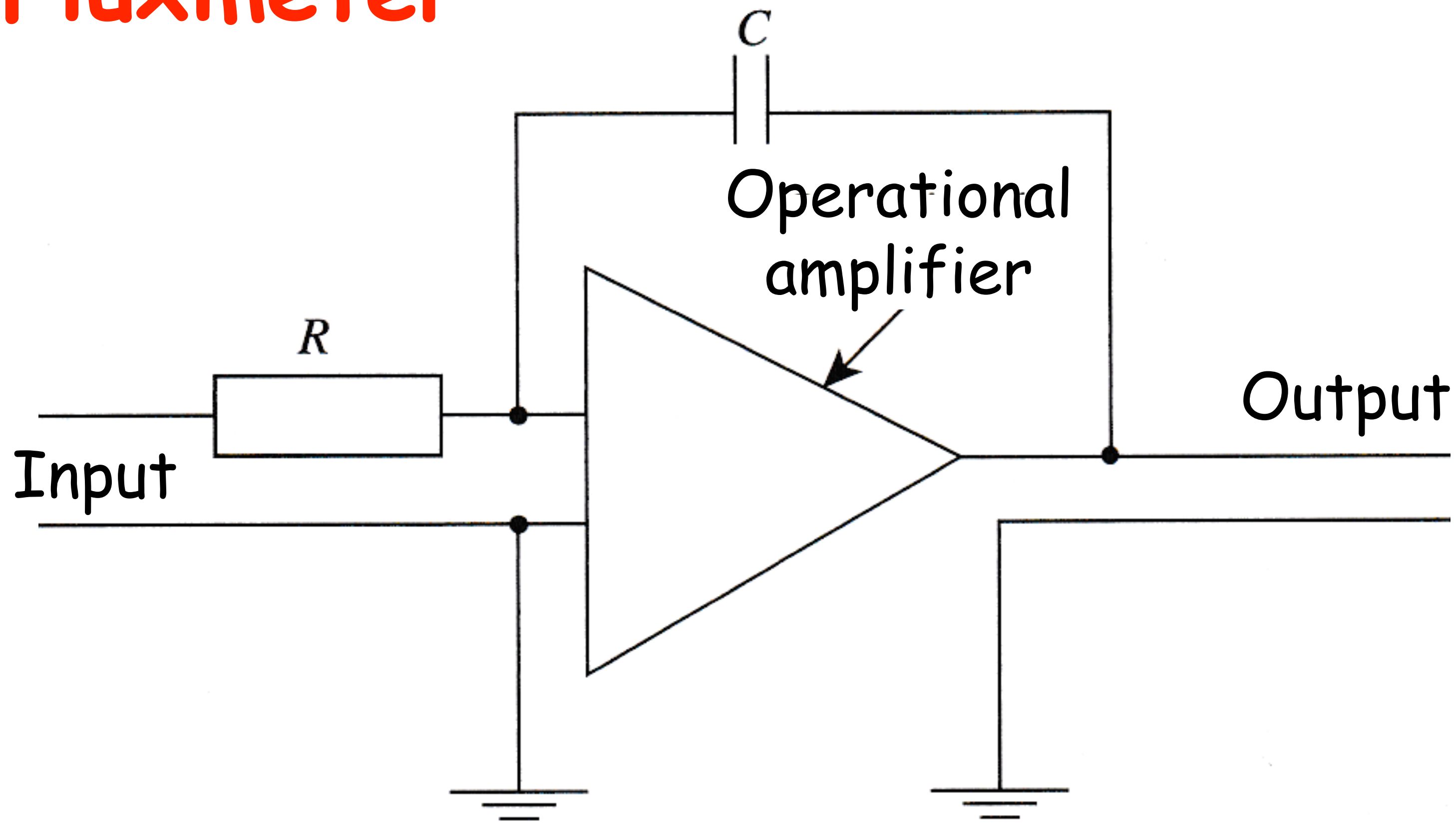

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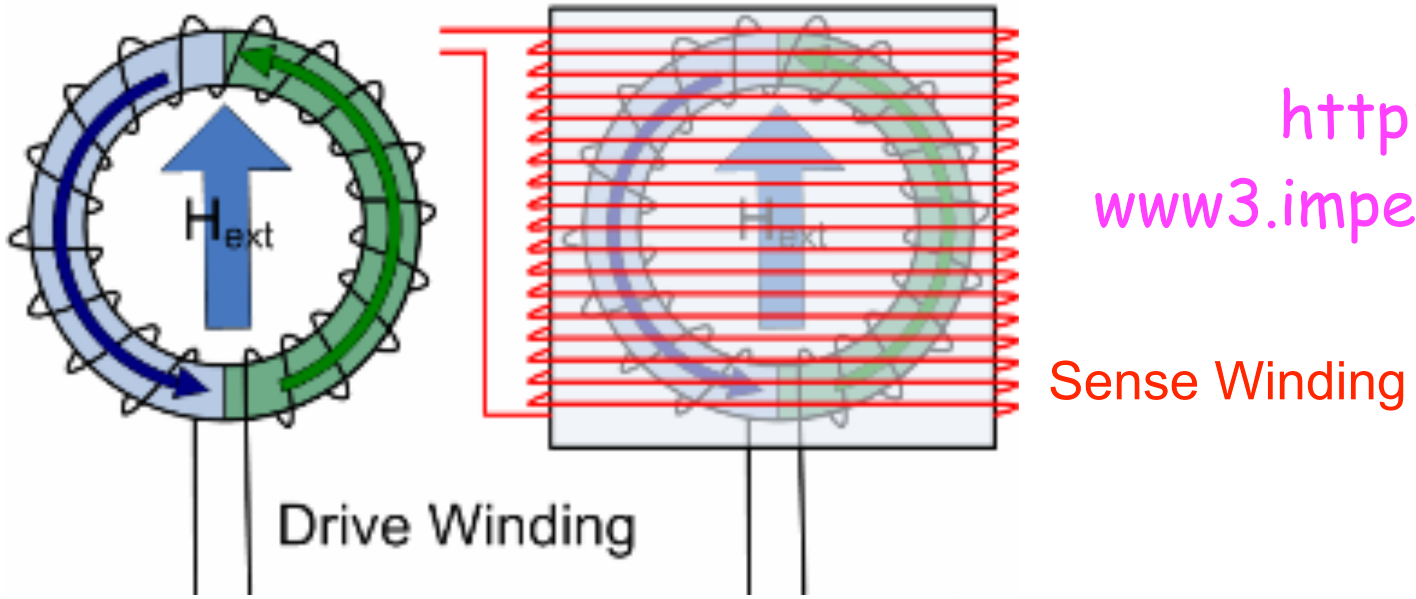
Fluxmeter measures changes in flux density

Measurement of constant field: search coil must be moved to zero-field region, or rotated through 180°

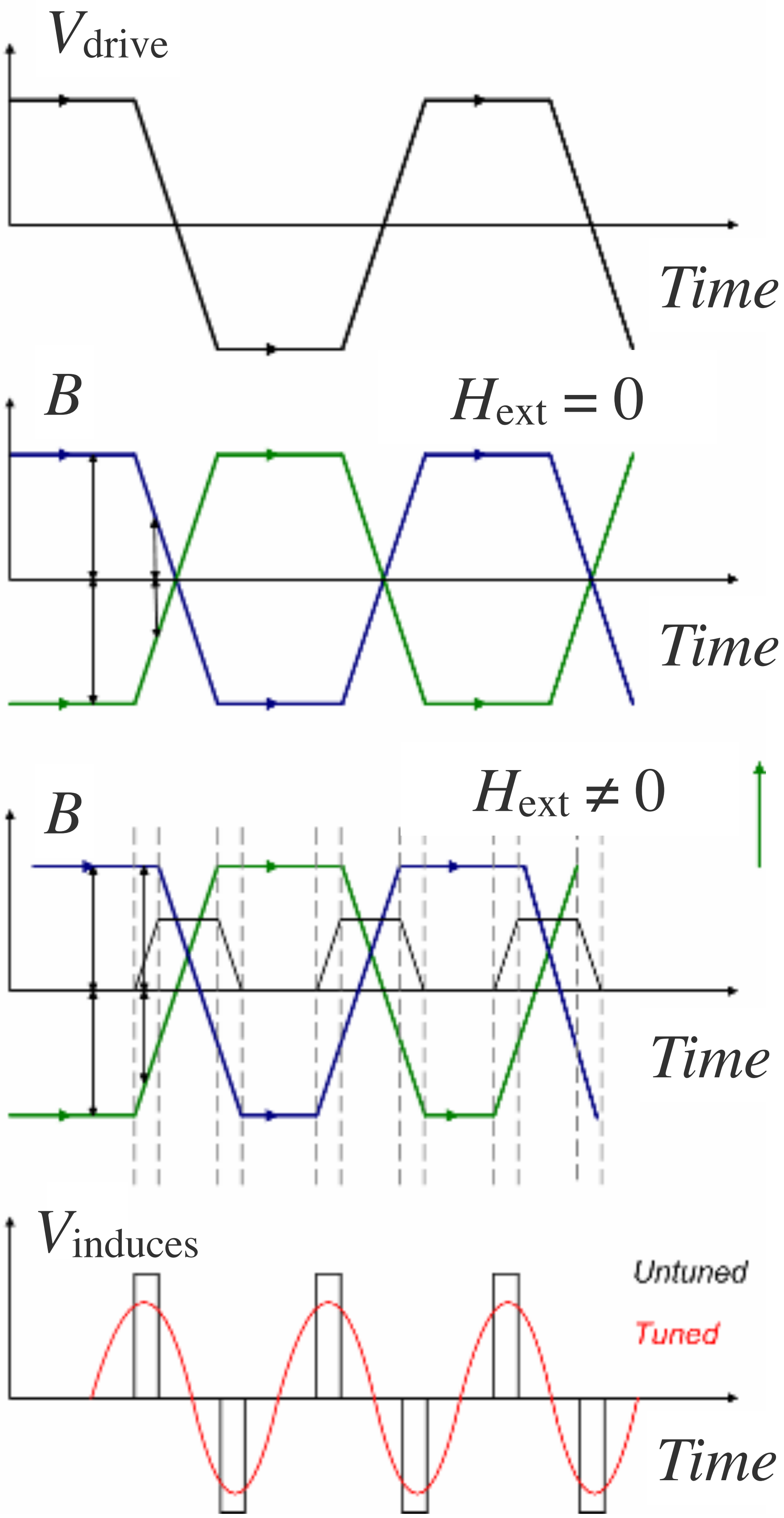
2. Measurement of magnetic field strength

2.3 Fluxgate Magnetometer



http://
www3.imperial.ac.uk

- Current through the drive: one half core generates B along H_{ext}, other half core in opposite direction
- $H_{\text{ext}} = 0$: two half cores go into and come out of saturation at same time → B-fields cancel → no net change of flux in sense winding → no voltage induced
- $H_{\text{ext}} \neq 0$: one half core comes out of saturation sooner, other half core later → net flux change → voltage → two spikes in voltage for each transition in drive
- Size and phase of induced spikes → magnitude and direction of H_{ext}
- Typical field range: 0.1 nT – 1 mT (used in e.g. geomagnetic and archeological surveys)



2. Measurement of magnetic field strength

2.4 Magnetic Potentiometer (Rogowski-Chattok coil)

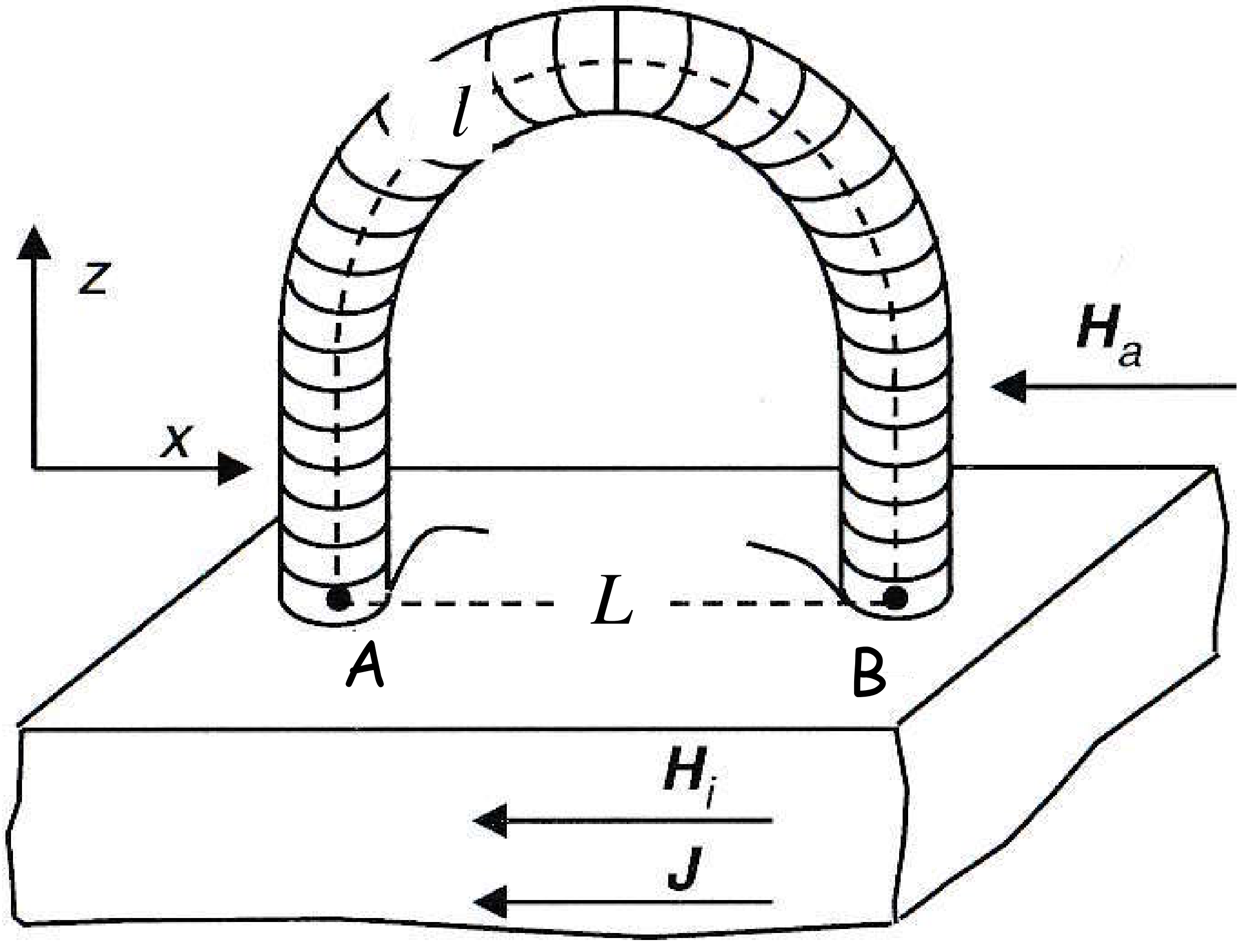
- Tightly wound coil whose ends lie in same plane, attached to magnetic sample
- Field at surface of magnetic sample is the same as internal field
- If no current flows in coil: line integral $\oint H dl$ around dotted path must be zero:

$$\int_A^B H\, dl + \int_B^A H\, dL = 0$$

- If field is uniform along L:

$$\int_B^A H\, dL = H\, \Delta L = -\int_A^B H\, dl$$

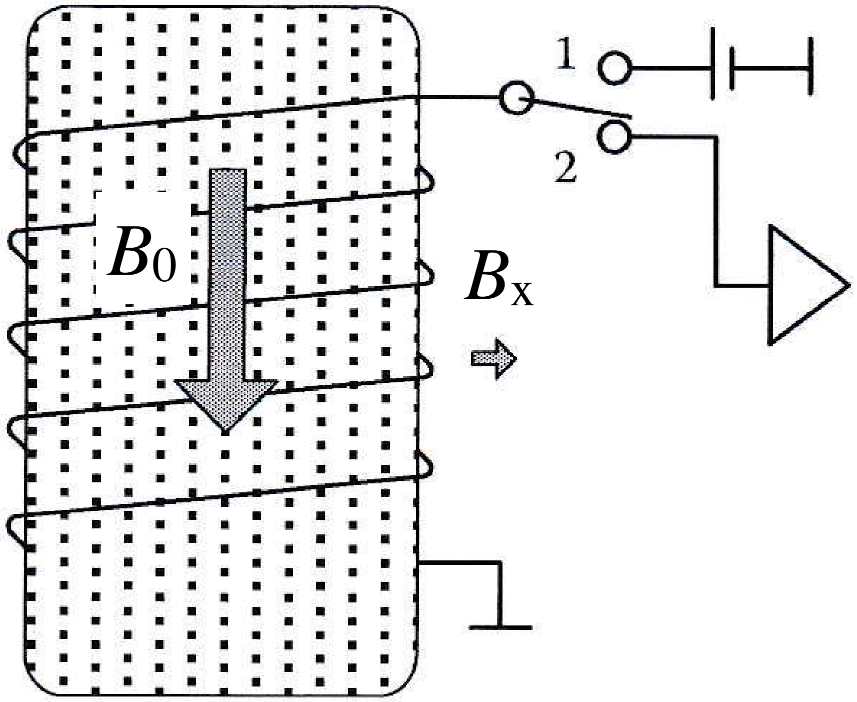
- Output of coil is connected to integrator
- Move Chattok coil to region of zero field: output of integrator proportional to (constant) H between A and B; $H = H_\mathrm{i}$
- Is used to measure magnetic field in flux-closing yokes (like singe sheet testers for electrical steel)

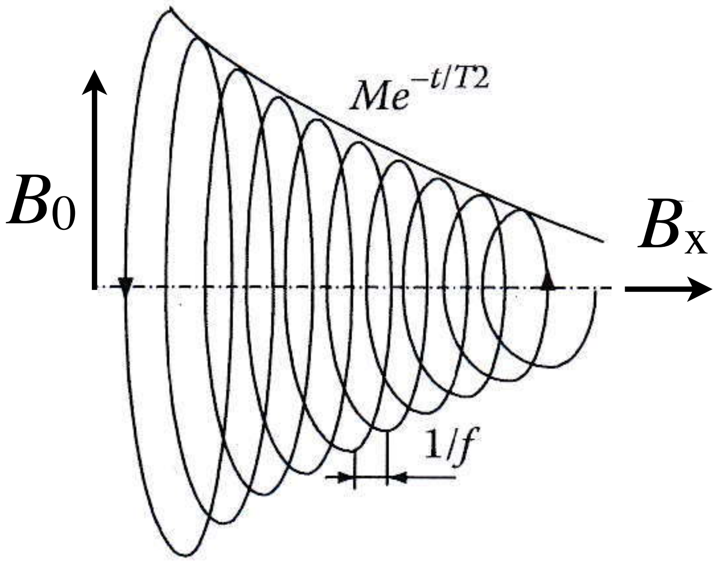


2. Measurement of magnetic field strength

2.5 Proton Precession Magnetometer

- High-precision measurement of weak magnetic field (like Earth's field, with uncertainty 1 ppm), also used for calibration
- Relies on Nuclear Magnetic Resonance (NMR): applyig magnetic field → magnetic moment of nucleus rotates in selected quantum directions with resonance frequency which strictly depends on value of field
- Principle:
  - (1) sample (e.g. 1 liter of water, kerosine…) is exposed to strong dc magnetic field B_0 (~ 10 mT) perpendicluar to the measured field B_x . B_0 aligns certain fraction of proton moments along coil direction
  - (2) B_0 switched-off → magnetic moments of protons align along B_x by decaying precession movement. Frequency of precession is measured by measuring frequency of induced voltage in coil. Frequency depends precisely on B_x .
  - Typical precessional frequency: a few kilohertz





2. Measurement of magnetic field strength

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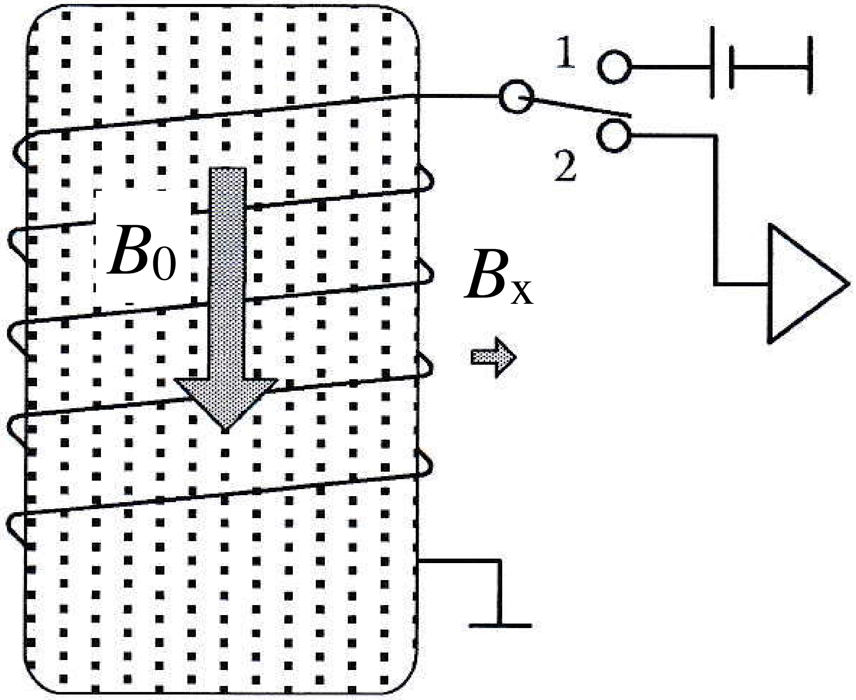
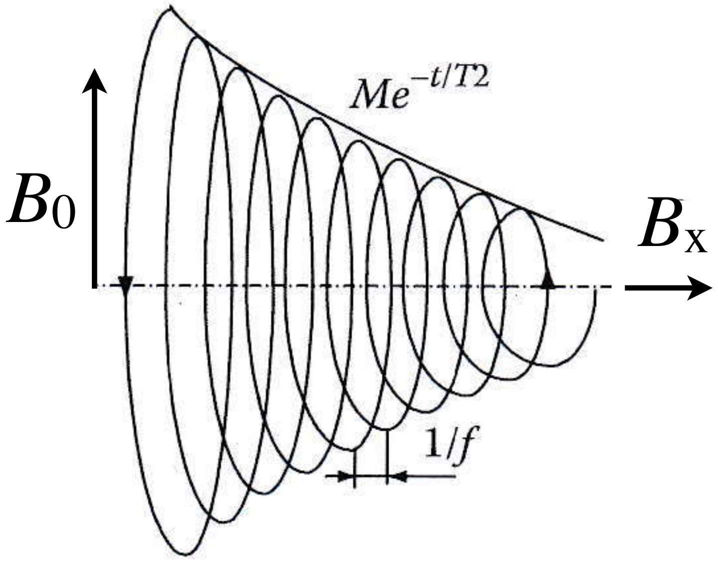
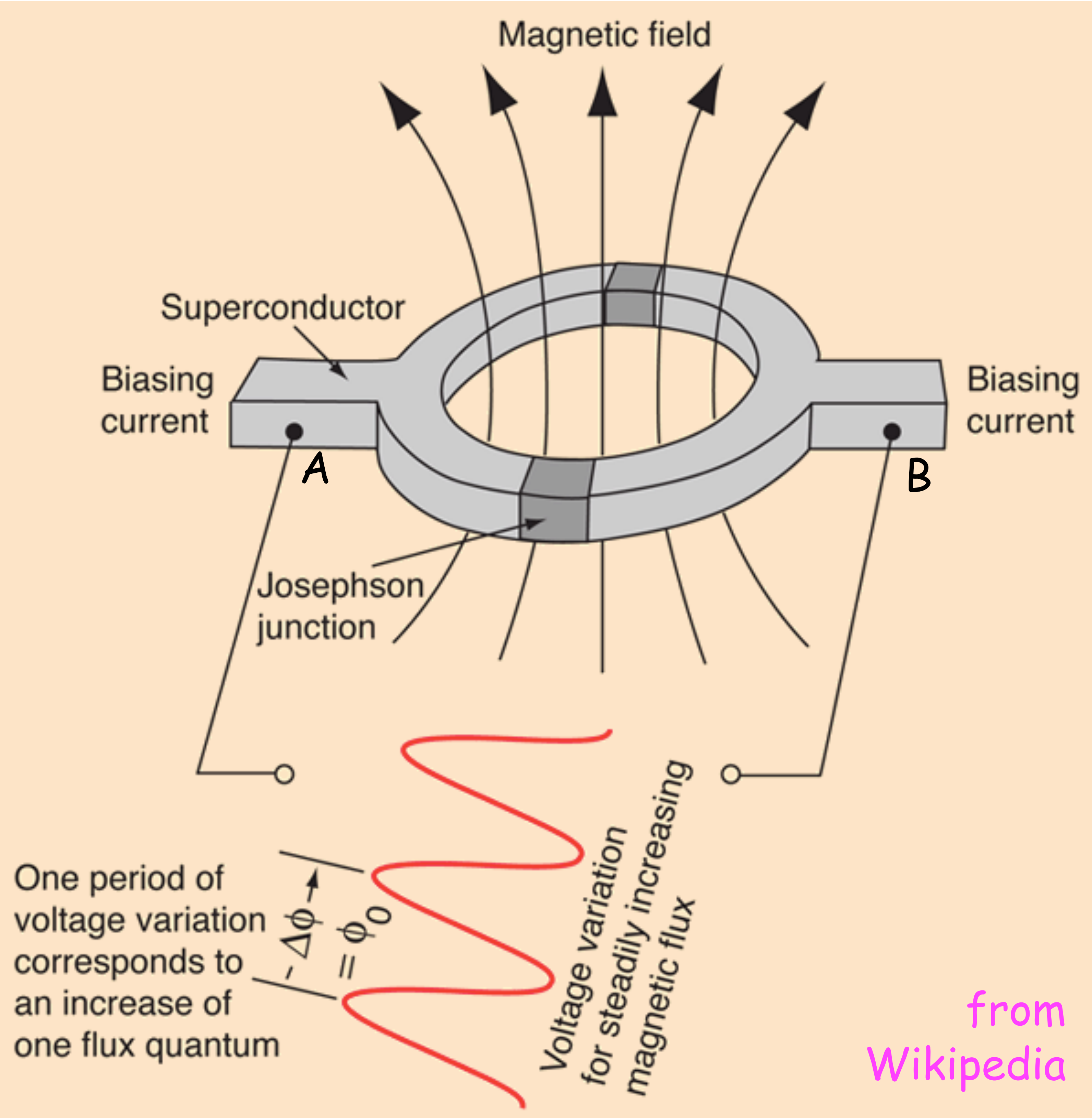




2. Measurement of magnetic field strength

2.6 SQUID Magnetometer

- SQUID: Superconducting Quantum Interference Device magnetometer
- Based on tunneling of superconducting electrons across narrow insulating gap, called Josephson junction
- Ring-shaped device, superconducting current from A to B, equal currents pass through each junction
- Changing magnetic flux through ring: induces „screening" current in ring (Faraday's law) which generates magnetic field that cancels external flux. Induced current adds to measuring current in one junction, subtracts in other

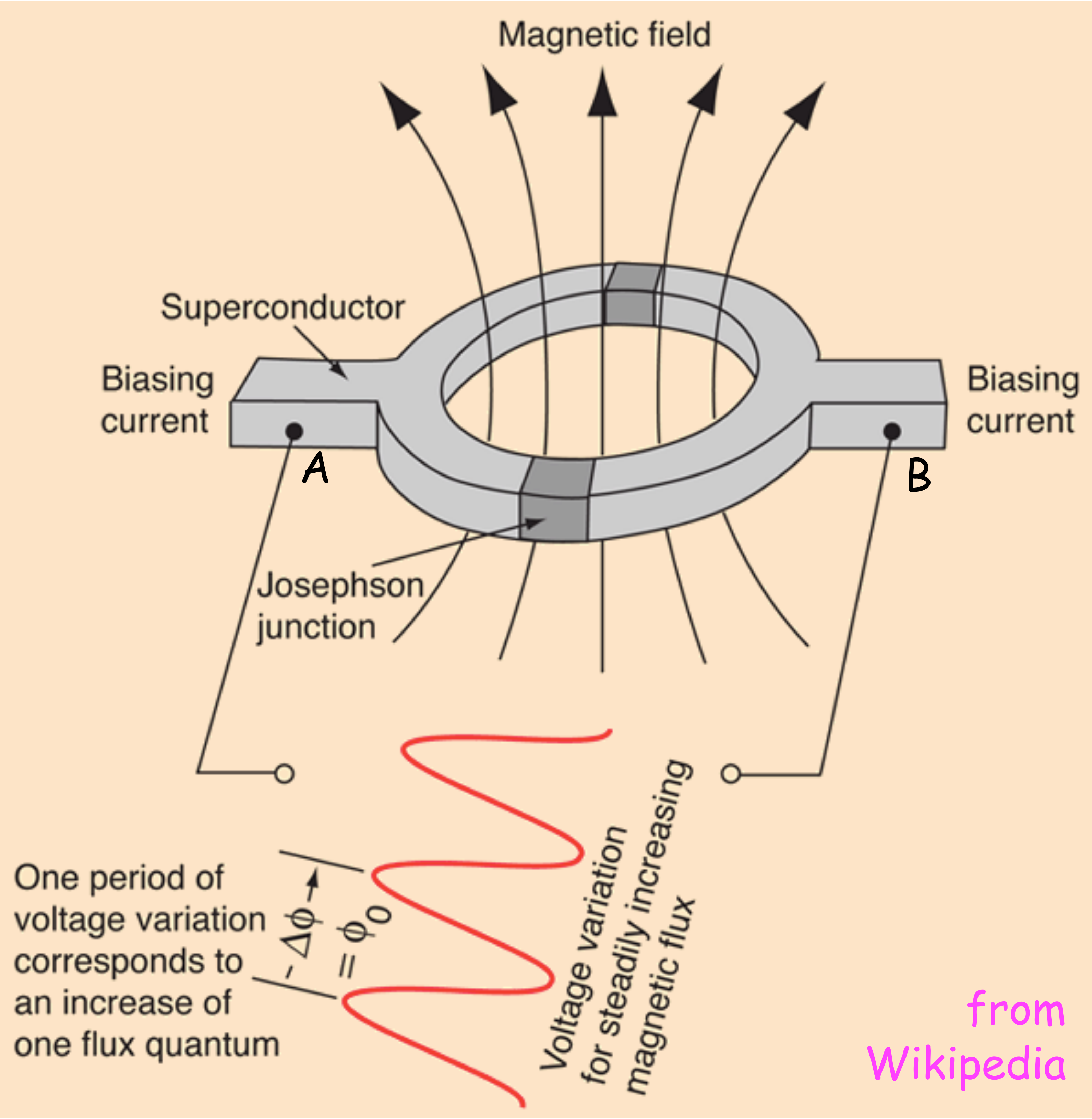


from Wikipedia

2. Measurement of magnetic field strength

2.6 SQUID Magnetometer

- In superconducting ring the magnetic flux is quantized, i.e because of wave nature of superconducting current (quantummechanics) the flux enclosed by the ring must be an integer number of the flux quantum $\Phi_0 = h/2e$



from Wikipedia

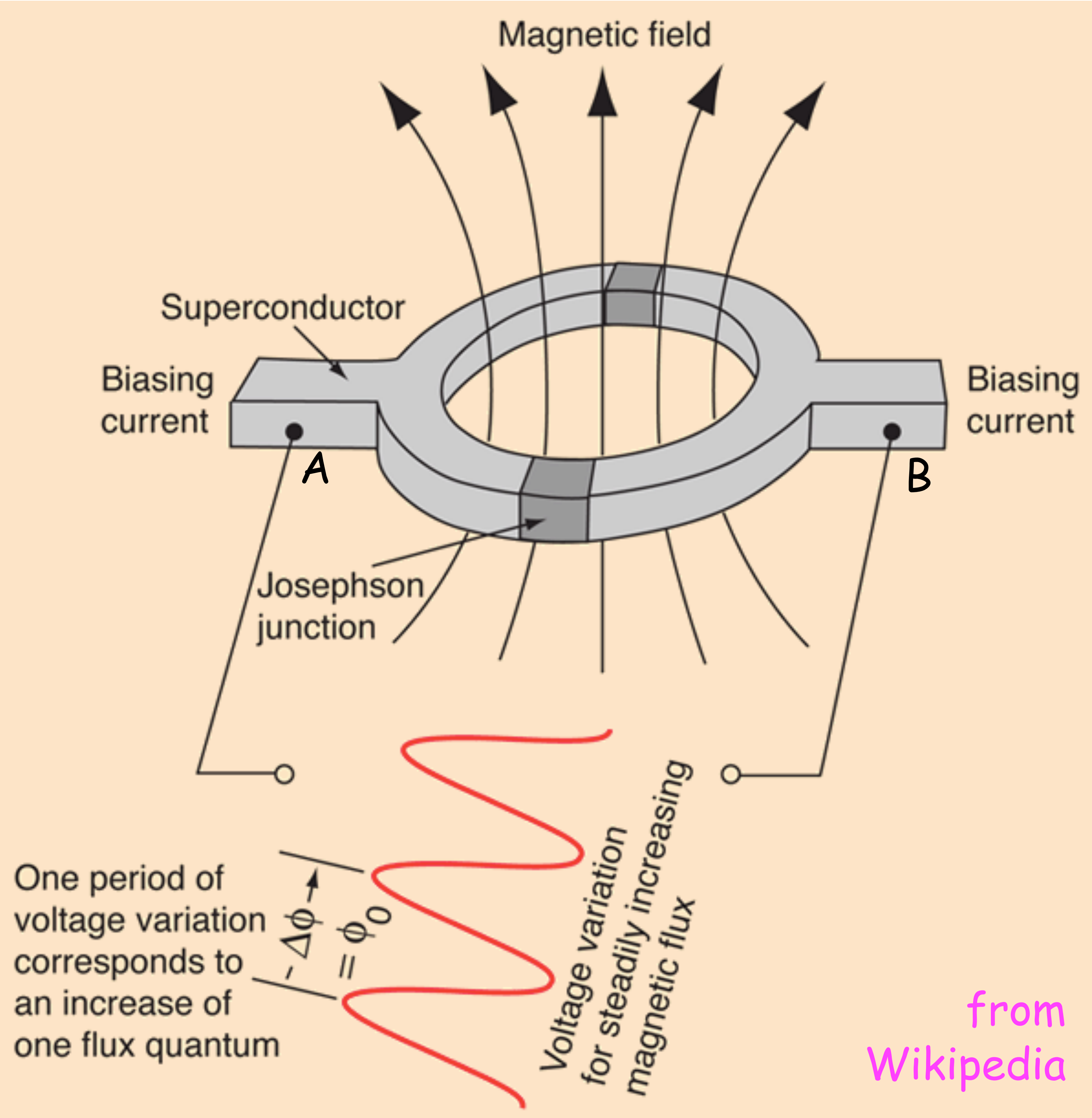
2. Measurement of magnetic field strength

2.6 SQUID Magnetometer

- Now suppose the external flux is further increased until it exceeds $\Phi_0 /2$. Since the flux enclosed by the loop must be integer number of flux quanta, instead of screening the flux the SQUID now energetically prefers to increase it to Φ_0. The screening current now flows in the opposite direction.
- → screening current changes direction every time the flux increases by half integer multiples of Φ_0

  → critical current oscillates as a function of the applied flux

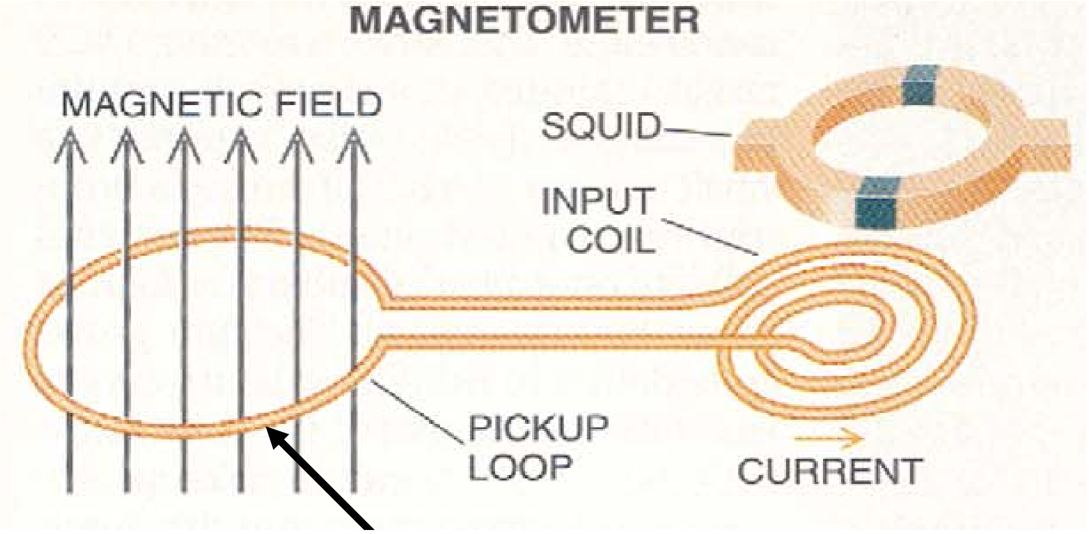
  → voltage between A and B is function of applied magnetic field and a period equal to Φ_0



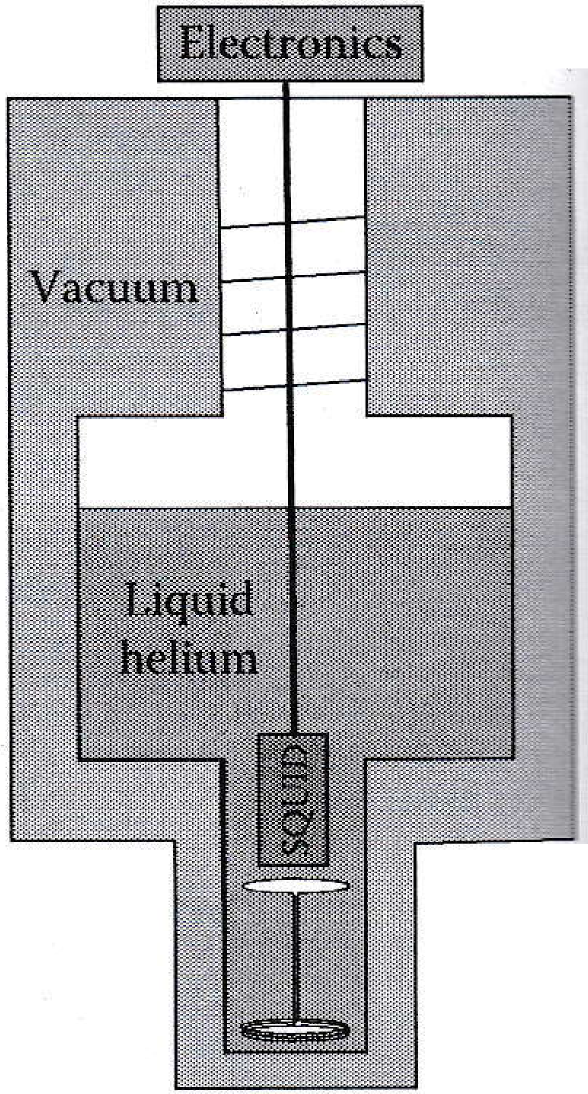
from Wikipedia

2. Measurement of magnetic field strength

2.6 SQUID Magnetometer



- In practise: SQUID is not directly contacted with magnetic source, device is rather linked to transformer coil to measure flux from small sample, i.e. sample magnetization
- SQUID magnetometer is high-sensitivity static fluxmeter
- Sensitivity: femto- to pico-Tesla
- Since SQUID requires low-T operation, it is usually used in conjunction with superconducting coil



3.

Measurements to determine magnetic material parameters & properties

3.

Measurements to determine magnetic material parameters & properties

General aspects

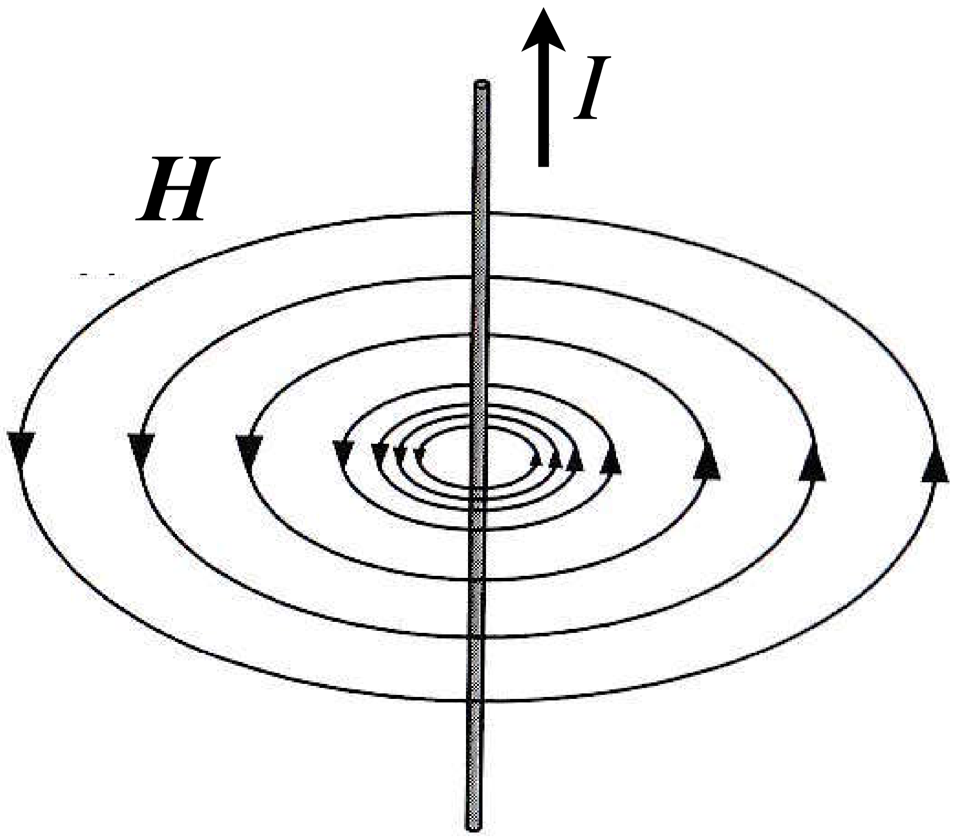
3. Magnetic Measurement …

General aspects

Magnetic field H

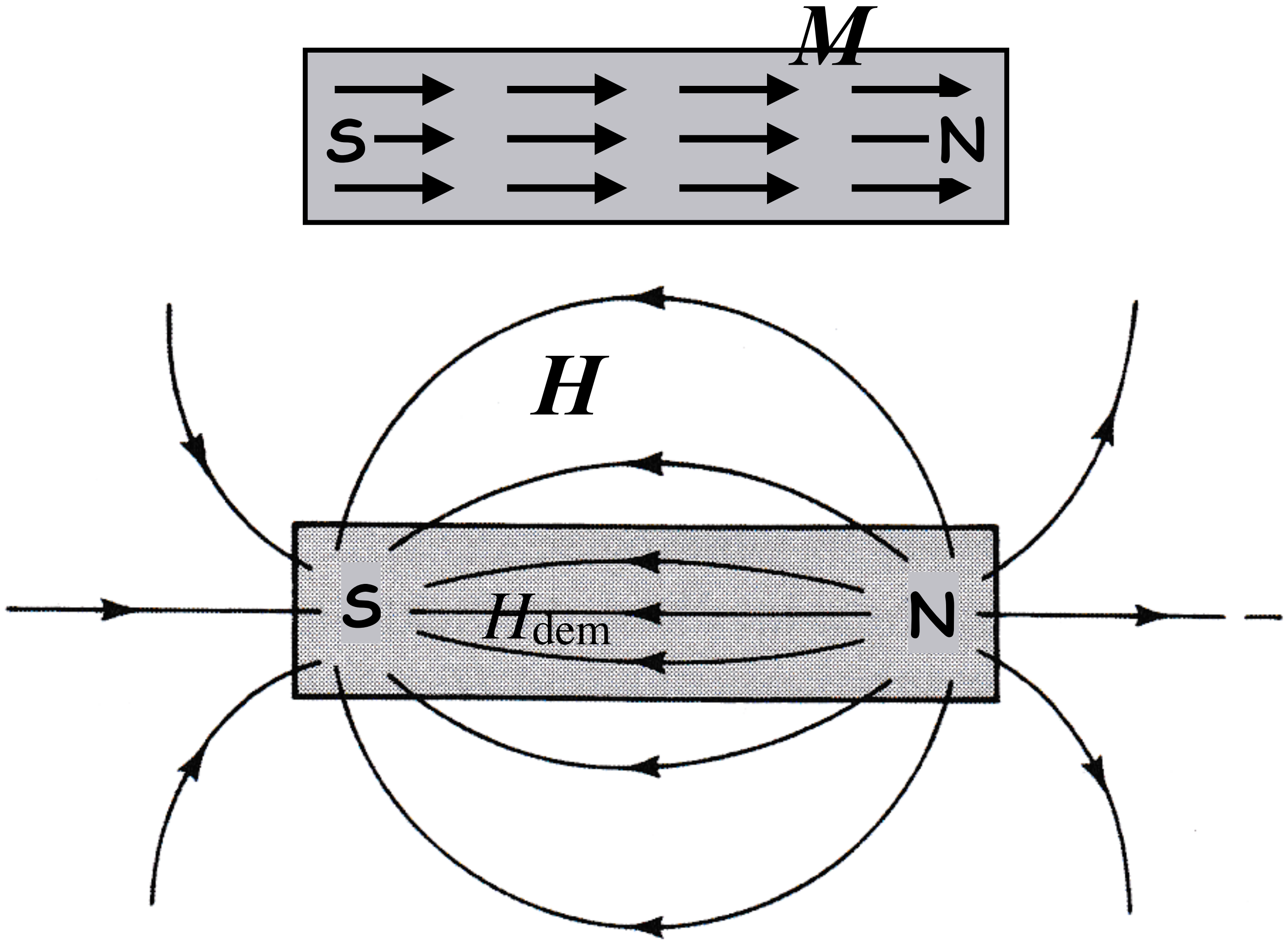
Field produced by currents:

Lines of H are continuous and form closed loops



Field produced by magnetic poles:

Lines of H begin at north poles and end at south poles (here: $H_{\mathrm{applied}} = 0$)



Demagnetizing field $\boldsymbol{H}_{\mathrm{dem}} = -N\overline{\boldsymbol{M}}$: acts opposite to magnetization M that creates it

3. Magnetic Measurement …

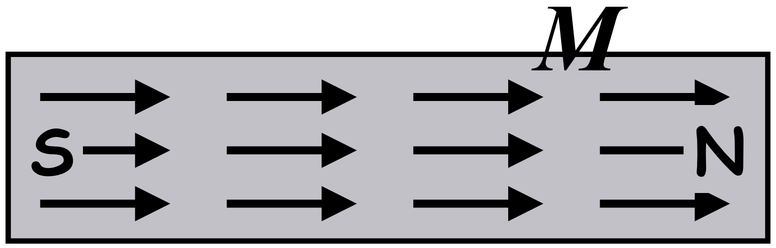
General aspects

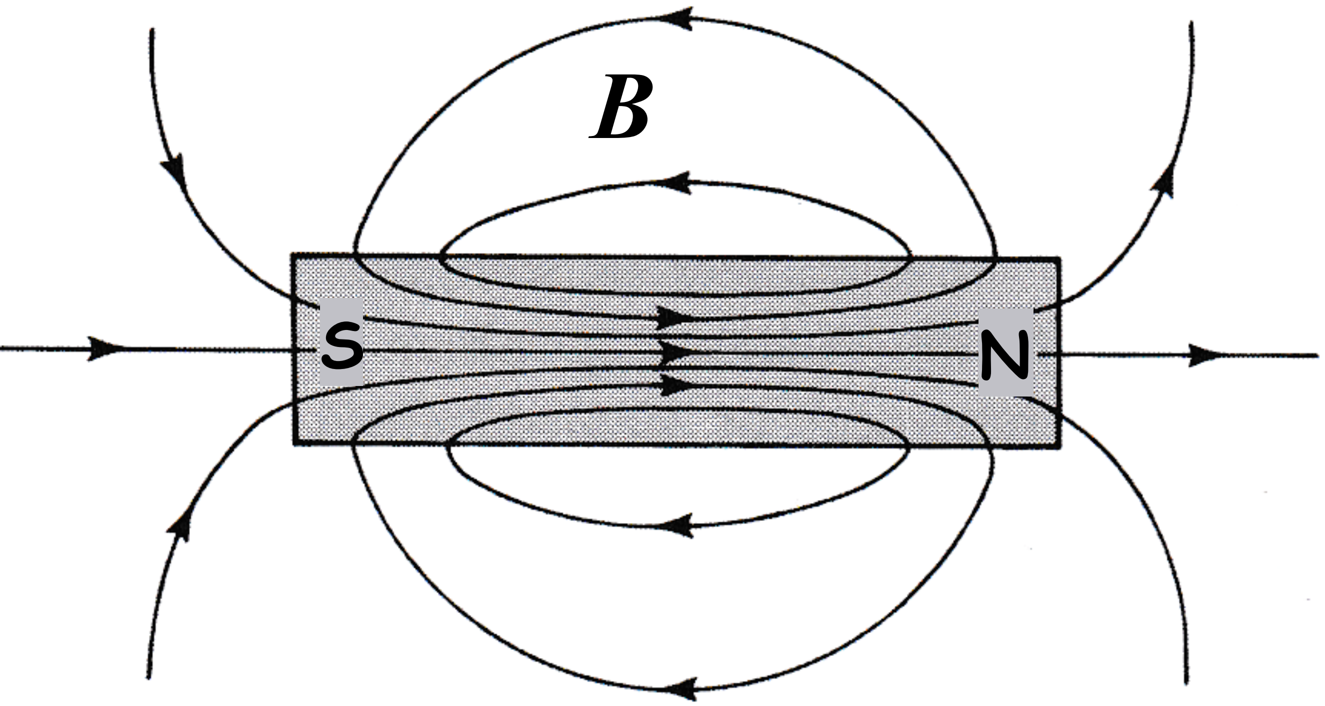
Magnetic field H and Induction B

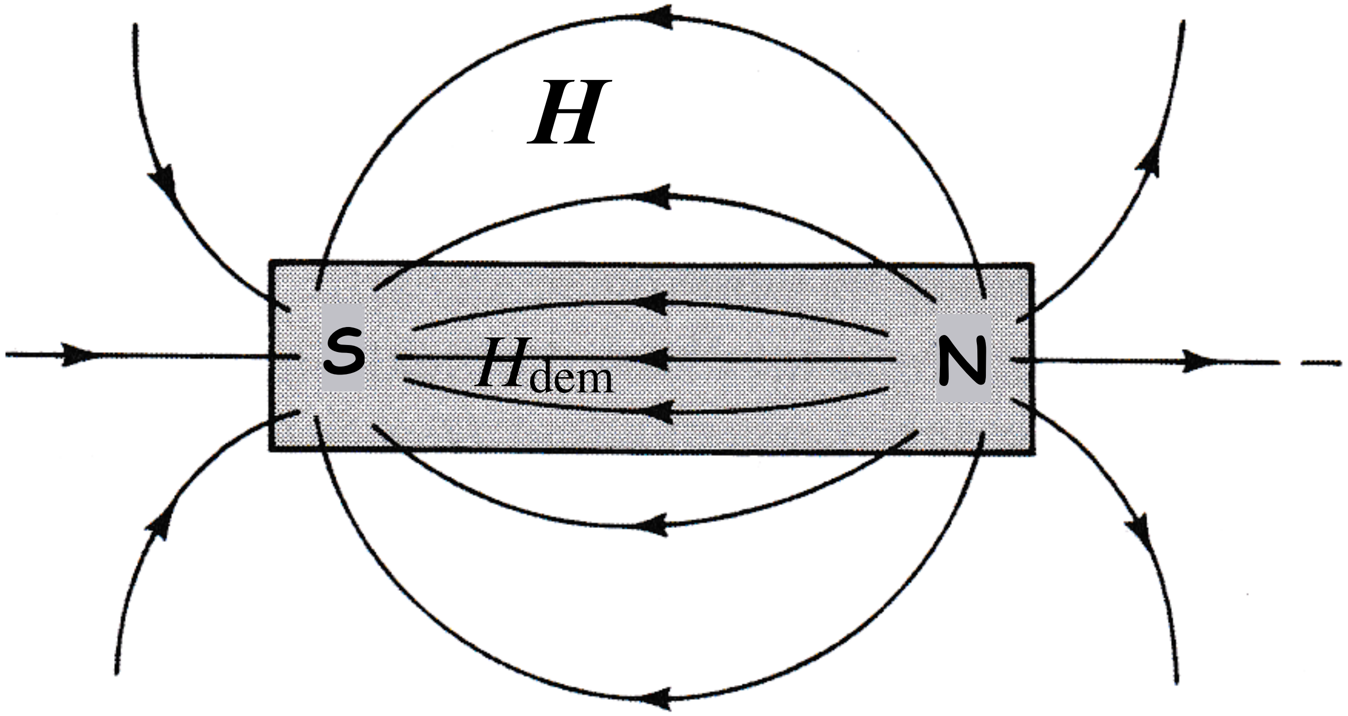
- $\boldsymbol{B} = \mu_0 (\boldsymbol{H} + \boldsymbol{M}) = \mu_0 (\boldsymbol{H}_{\text{applied}} - \boldsymbol{H}_{\text{dem}} + \boldsymbol{M})$
- If $\boldsymbol{H}_{\text{applied}} = 0$: $\boldsymbol{H}_{\text{dem}}$ is only field acting, and $\boldsymbol{B} = -\mu_0\boldsymbol{H}_{\text{dem}} + \mu_0\boldsymbol{M}$
- $\mu_0\boldsymbol{H}_{\text{dem}}$ can never exceed $\mu_0\boldsymbol{M}$, i.e. flux density $\boldsymbol{B}$ inside magnet is always smaller than $\mu_0\boldsymbol{M}$, but in same direction

Field produced by magnetic poles:

Lines of H begin at north poles and end at south poles (here: $H_{\text{applied}} = 0$)







- Lines of $\boldsymbol{B}$ are continuous, from S to N inside magnet, outside $\mu_0\boldsymbol{H} = \boldsymbol{B}$

Demagnetizing field $\boldsymbol{H}_{\text{dem}} = -N\overline{\boldsymbol{M}}$: acts opposite to magnetization M that creates it

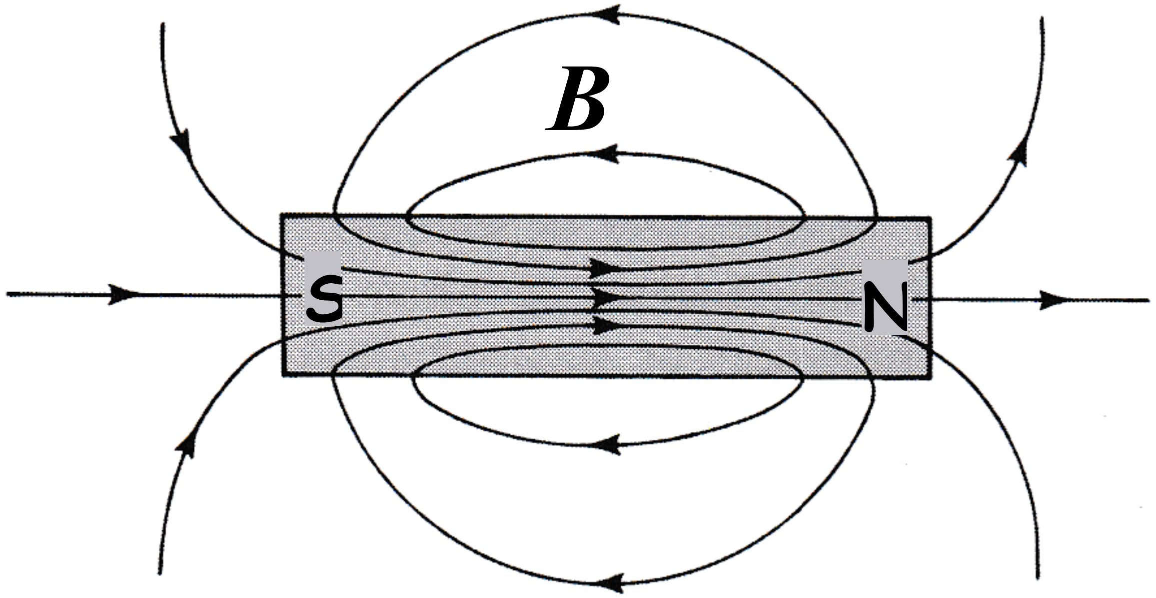
3. Magnetic Measurement …

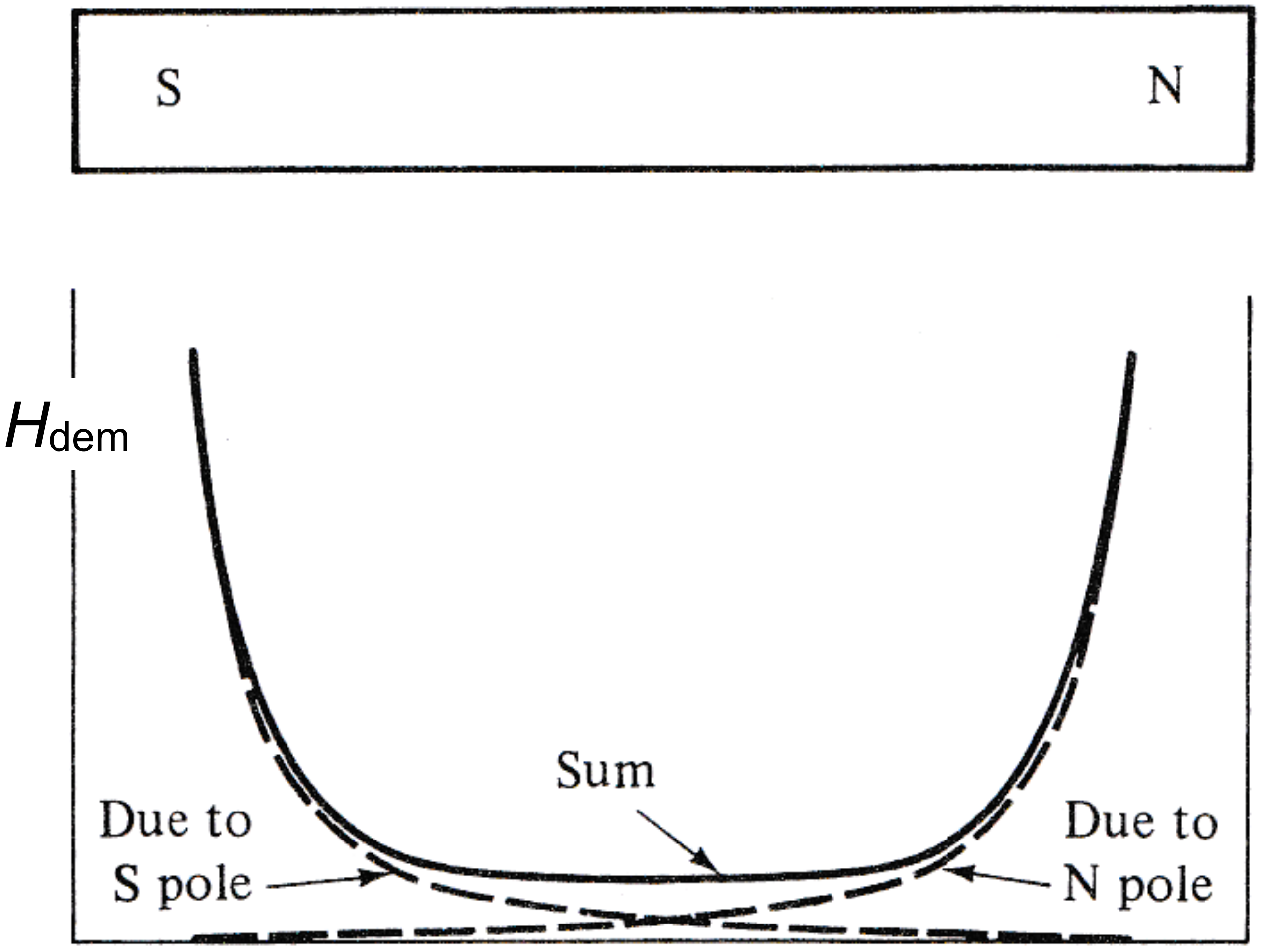
General aspects

Magnetic field *H* and Induction *B*

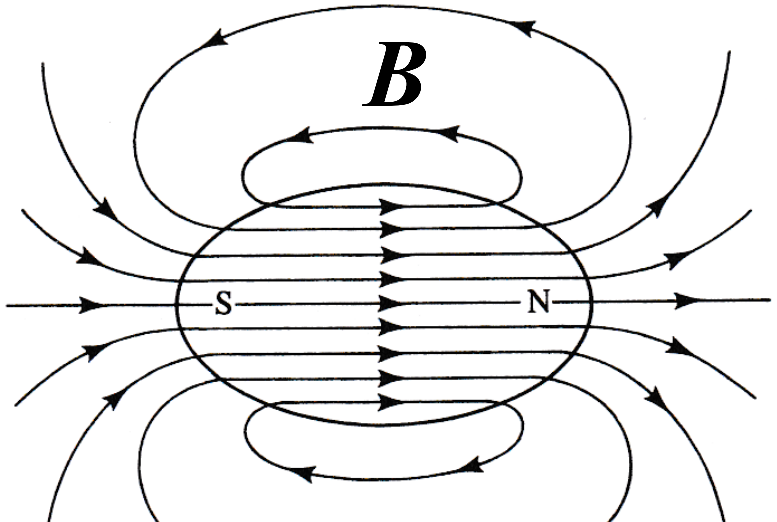
- Flux density of bar magnet is not uniform: *B*-lines diverge towards the ends → flux density is less than in center

- Reason: H_{dem} is stronger near the poles





- Exception: rotational ellipsoid:

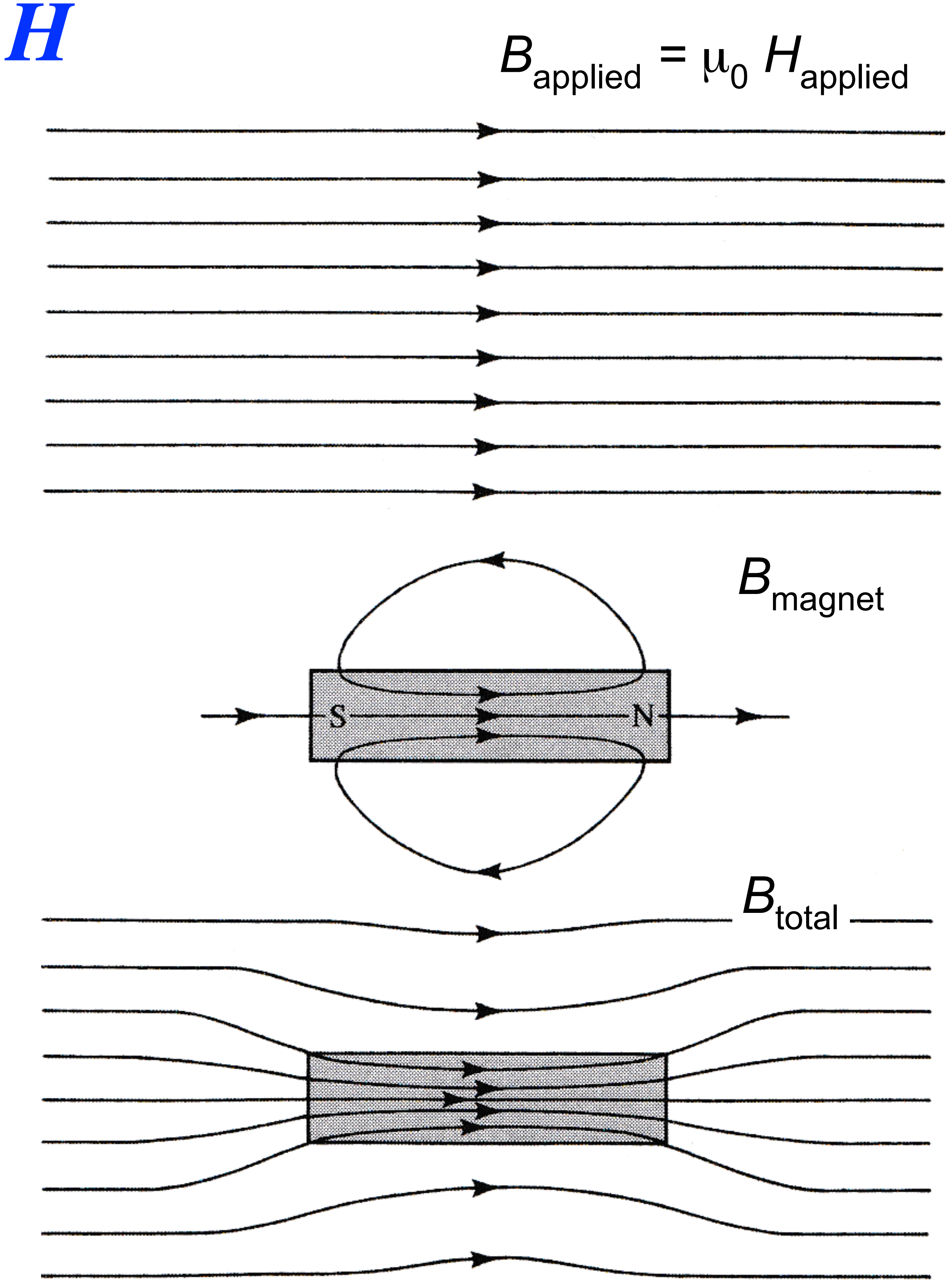


3. Magnetic Measurement …

General aspects

Finite sample in applied field H

- When soft magnetic body is placed in field, it alters shape of field.
- This is demonstrated by asuming a fully magnetized bar magnet. The total field B_{total} is the vector sum of applied field $B_{applied}$ and B-field of magnet
- The flux tends to crowd into the magnet, as though it were more permeable than surrounding air (→ term „permeability"). At points outside the magnet: field is reduced
- The same result is obtained if the body that is placed in the field is originally unmagnetized, because the field itself will produce magnetization (for material with μ >> 1)
- For strongly magnetic materials the disturbance of field is considerable



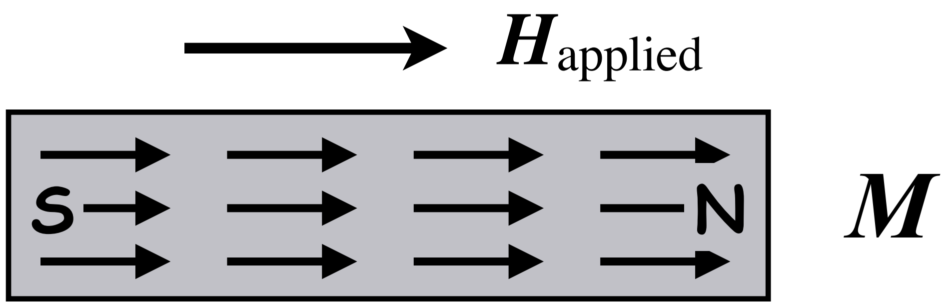
3. Magnetic Measurement …

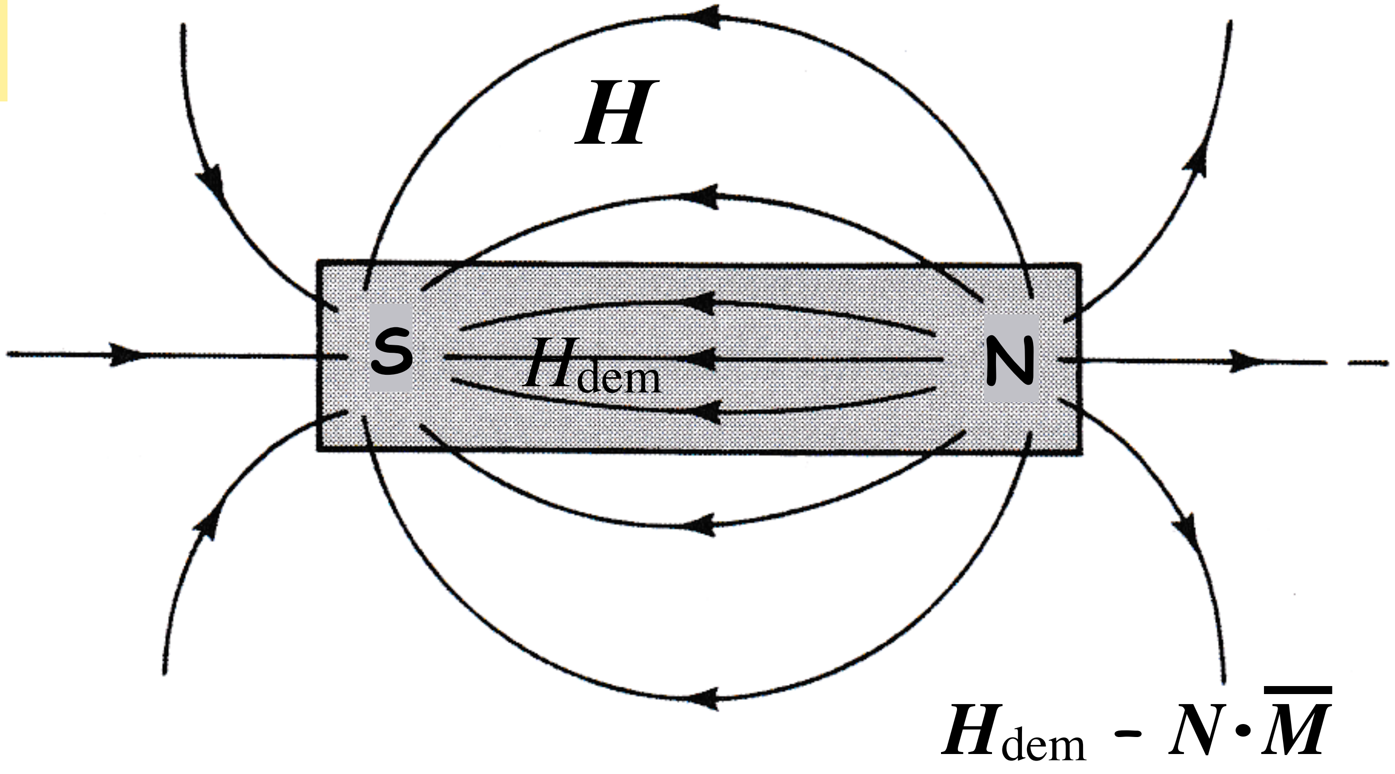
General aspects

Closed and open samples

- Open sample

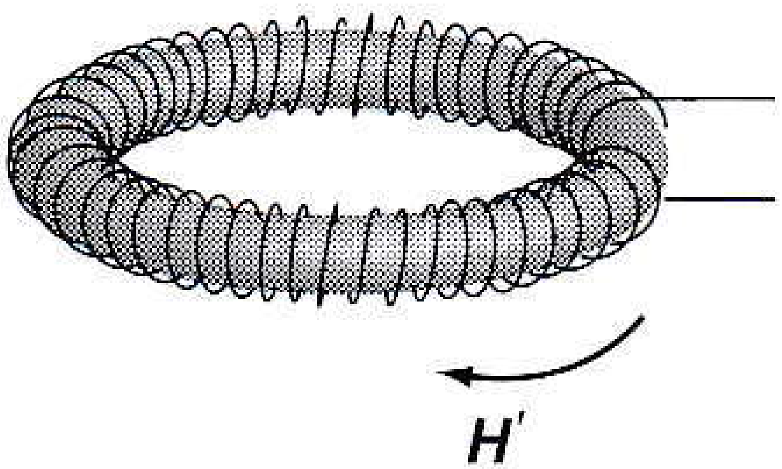
Internal field: $H_{in} = H_{applied} - N\cdot\overline{M}$





- Closed sample

Internal field: $H_{in} = H_{applied}$, $N = 0$

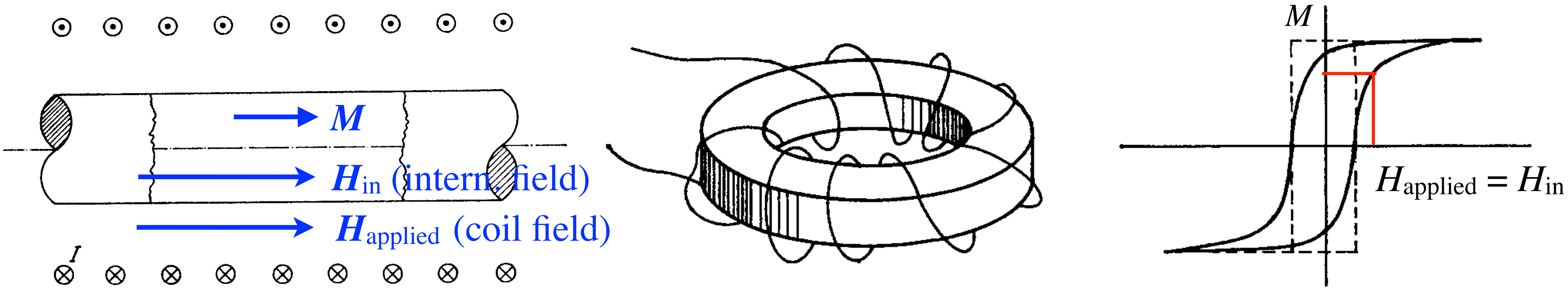


3. Magnetic Measurement …

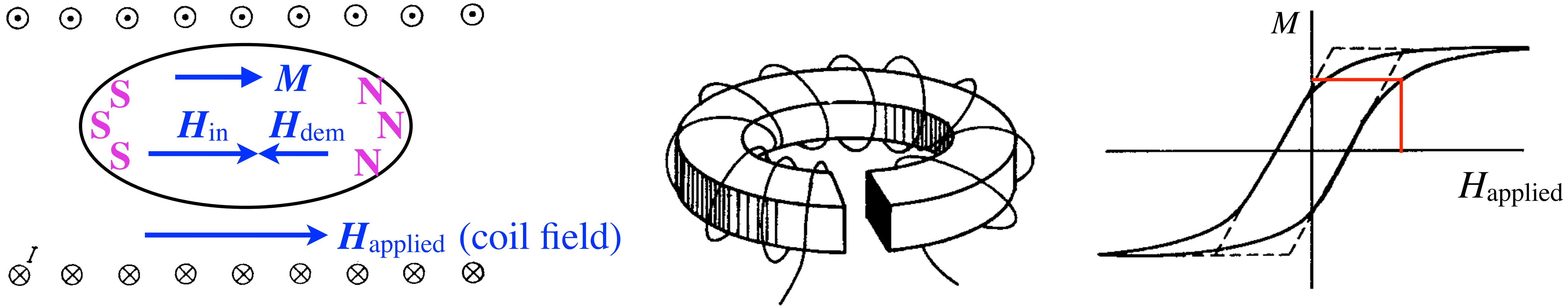
General aspects

Closed and open samples

Demagnetization effect
→ Shearing of magnetizaton curve



Infinite sample or closed ring: unsheared hysteresis curve: $N = 0$, i.e. $H_{in} = H_{ext}$



Finite sample or open core: sheared hysteresis curve due to demagnetization effect (a higher $H_{applied}$ is needed to achieve a given degree of M)

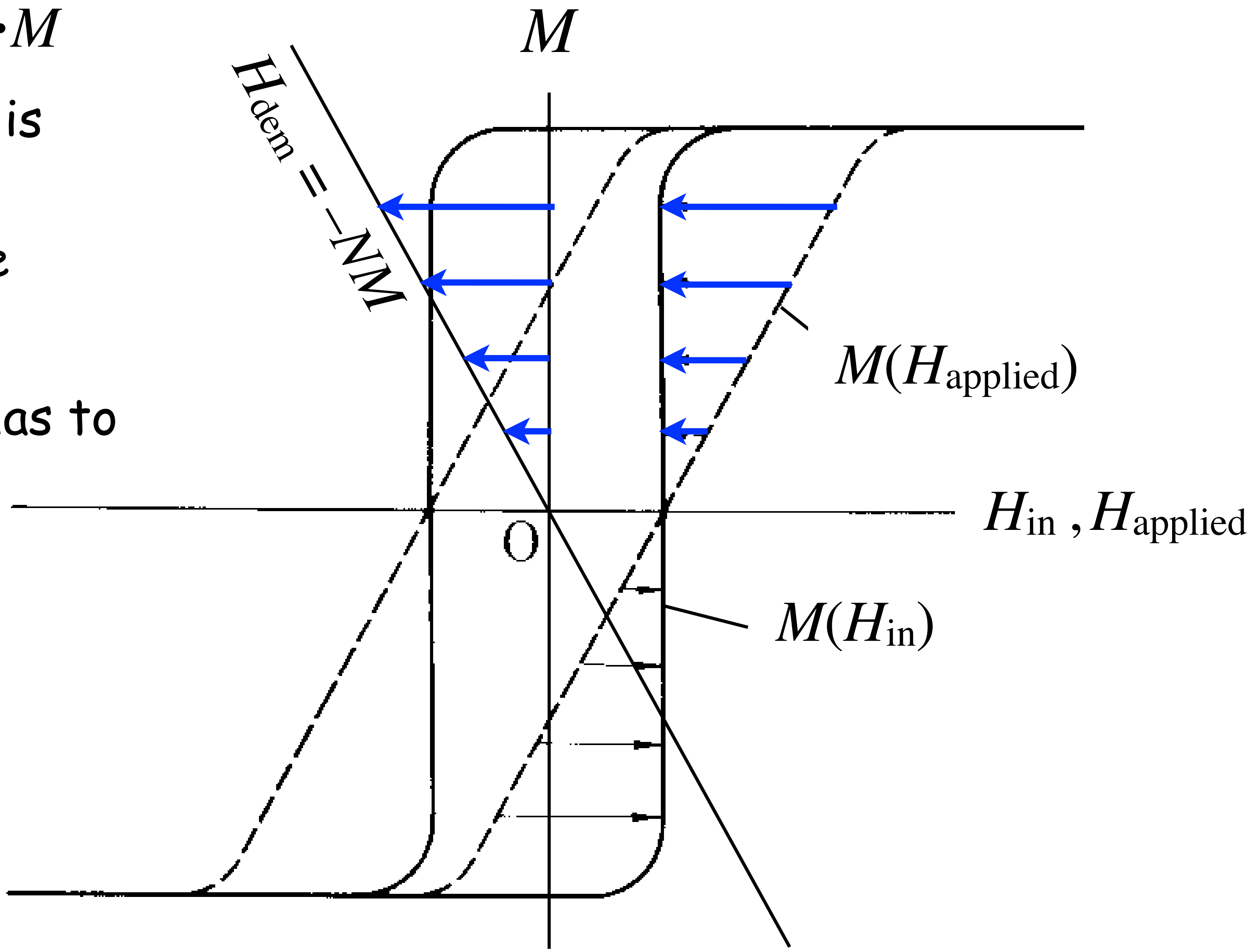
3. Magnetic Measurement …

General aspects

Closed and open samples

Demagnetization effect
→ Shearing of magnetizaton curve

- Internal field: $H_{\mathrm{in}} = H_{\mathrm{applied}} - N \cdot M$
- Relevant for magnetic materials is the $M(H_{\mathrm{in}})$-curve, as it is independent of the sample shape
- If a magnetization curve was measured on a finite sample, it has to be re-sheared.



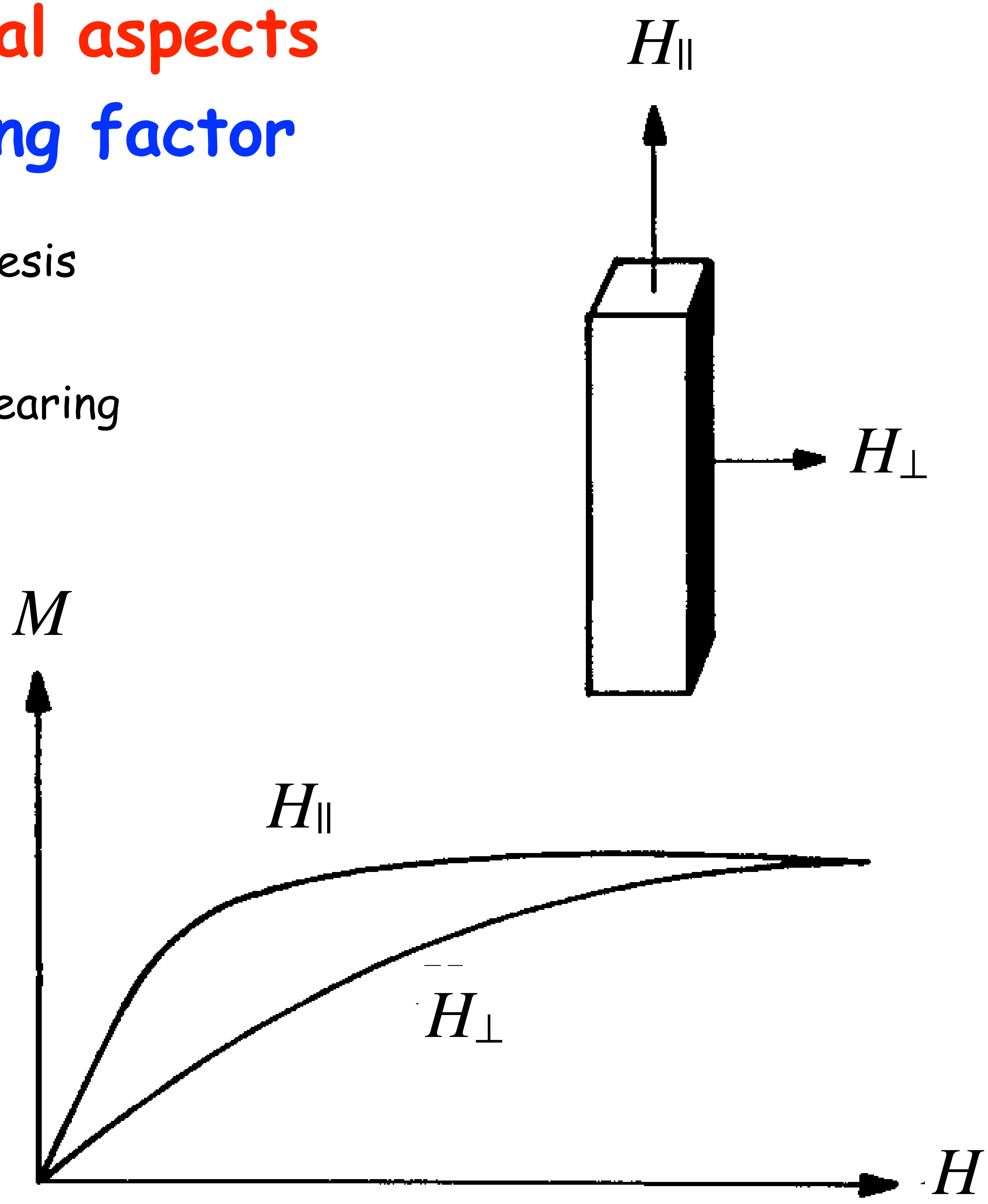
3. Magnetic Measurement …

General aspects

Influence of demagnetizing factor

- Influence of sample shape on hysteresis curve: Shape anisotropy
- Larger demag. factor → stronger shearing of magnetizaton curve

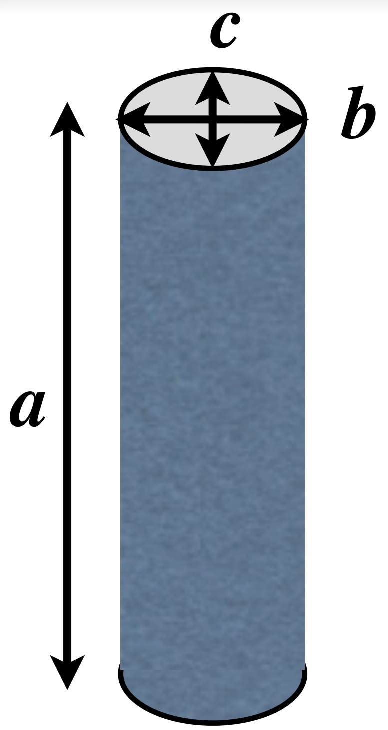
$N_{||} < N_{\perp}$



3. Magnetic Measurement …

General aspects

Demagnetizing factor



Long cylinder

$(b = c;\ a \longrightarrow \infty)$:

$N_\mathrm{a} \approx 0$;

$N_\mathrm{b} = N_\mathrm{c} \approx 1/2$

$\downarrow$

Cylinder prefers to be magnetized in longitudinal direction

Flat disk

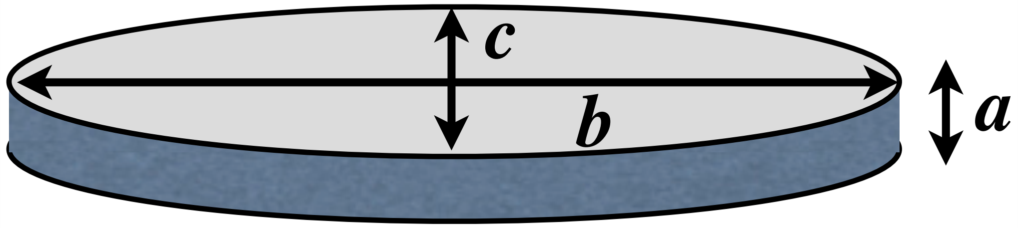
(thickness a; $b = c \longrightarrow \infty$):

$N_\mathrm{a} \approx 1$;

$N_\mathrm{b} = N_\mathrm{c} \approx 0$

$\downarrow$

Disk prefers to be magnetized in-plane



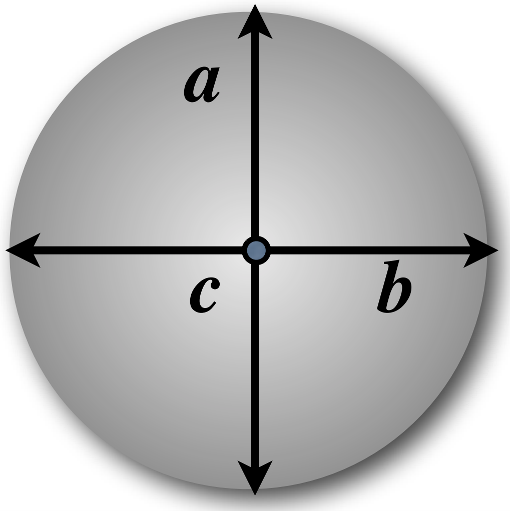
Sphere

$(a = b = c)$:

$N_\mathrm{a} = N_\mathrm{b} = N_\mathrm{c} = 1/3$

$\downarrow$

no prefered direction for magnetization



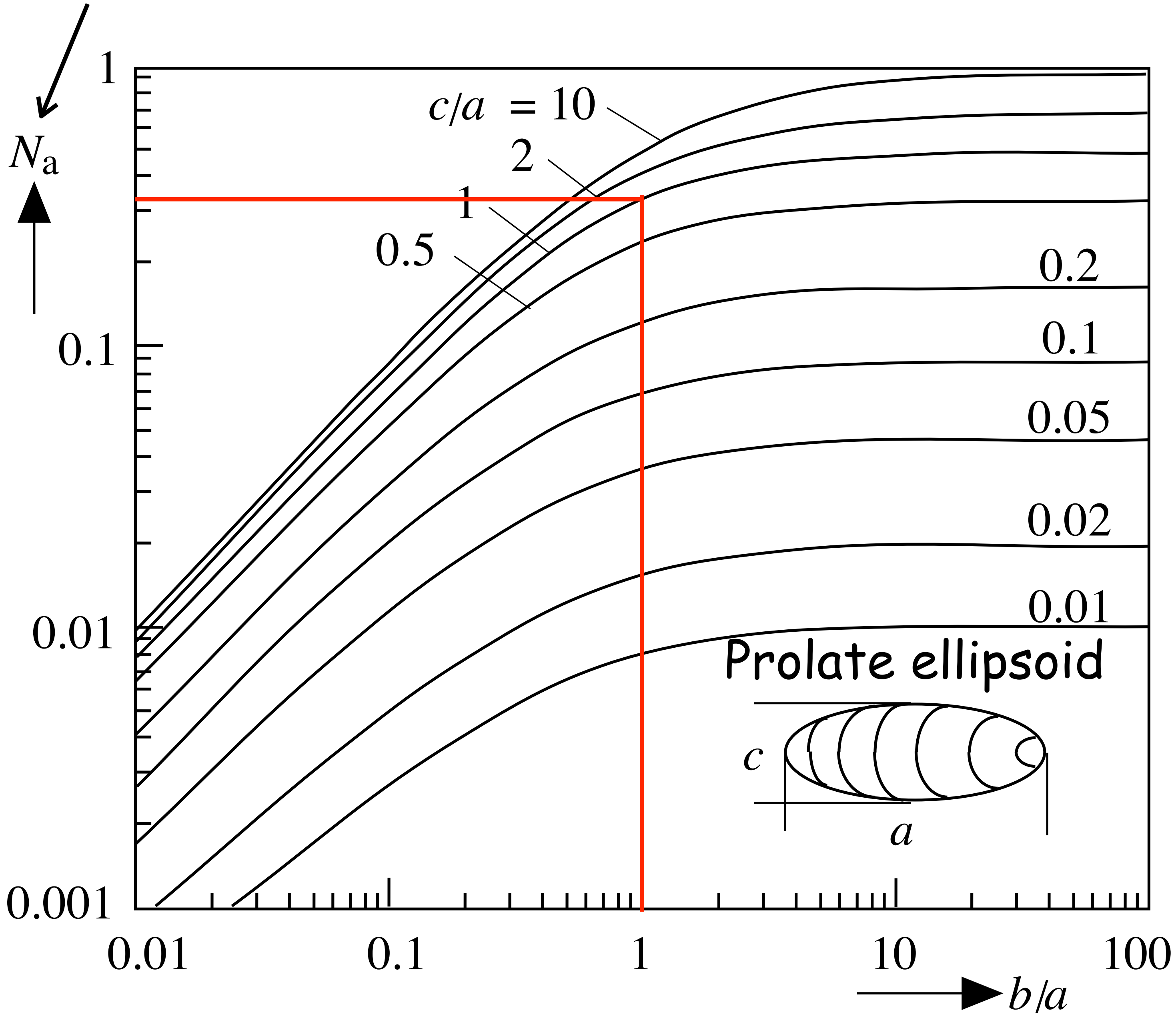
3. Magnetic Measurement …

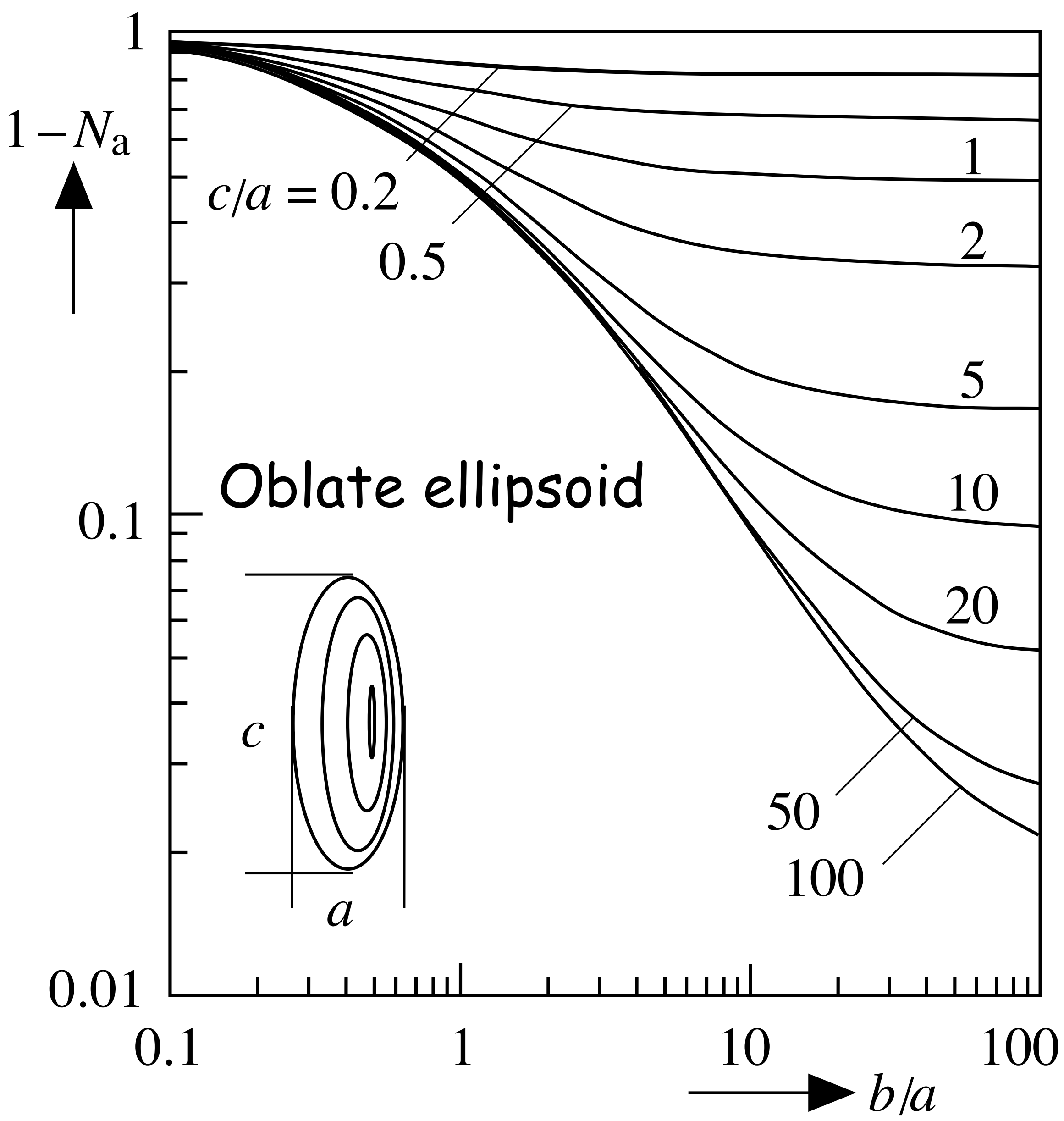
General aspects

Demagnetizing factor Can only be calculated exactly for rotational ellipsoid

N along a-axis

$N_a + N_b + N_c = 1$ (a, b, c: main axes of ellipsoid)





Sphere: $a = b = c \rightarrow N_a = 1/3 = N_b = N_c$

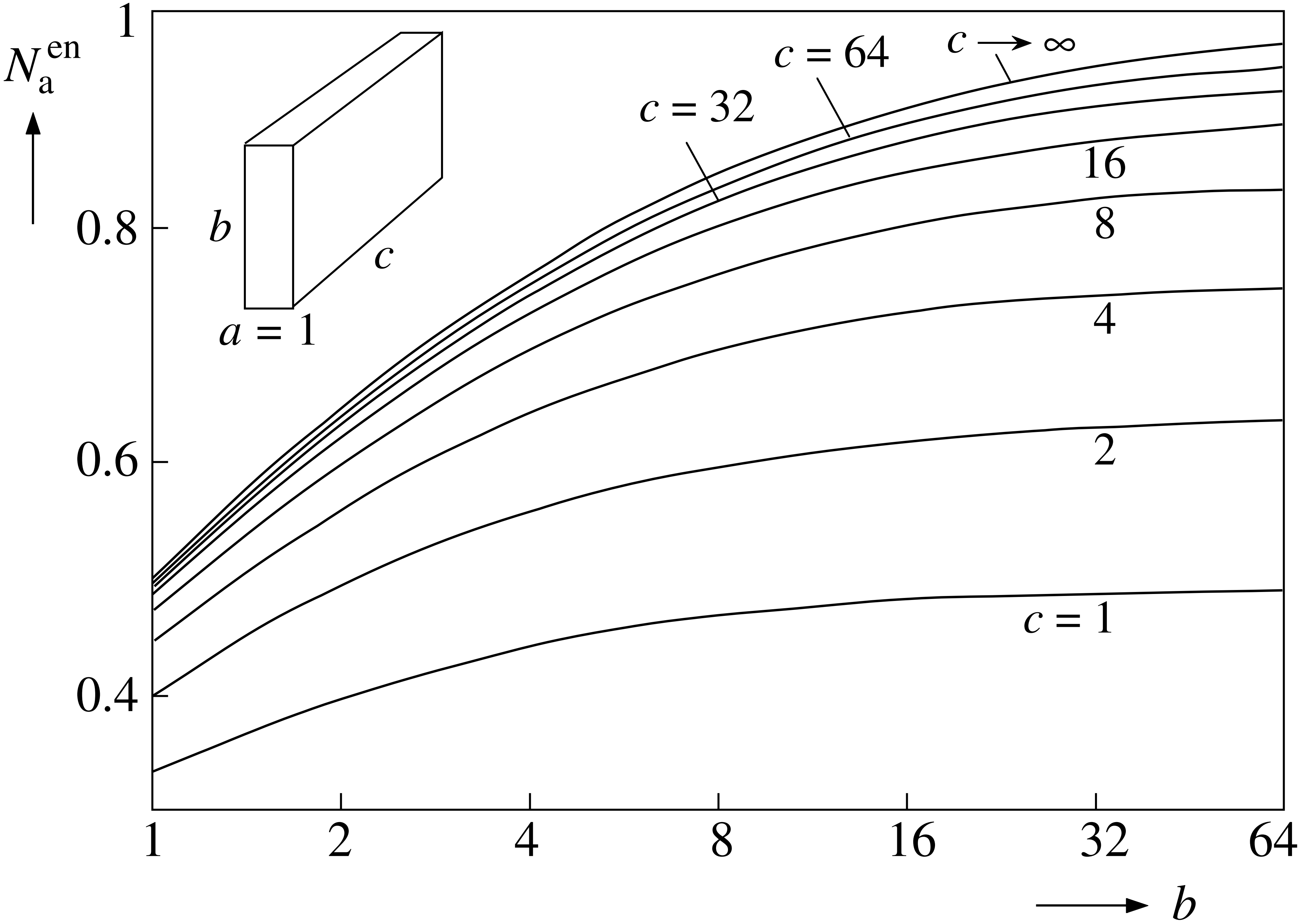
3. Magnetic Measurement …

General aspects

Demagnetizing factor

Numerical calculation for rectangular body

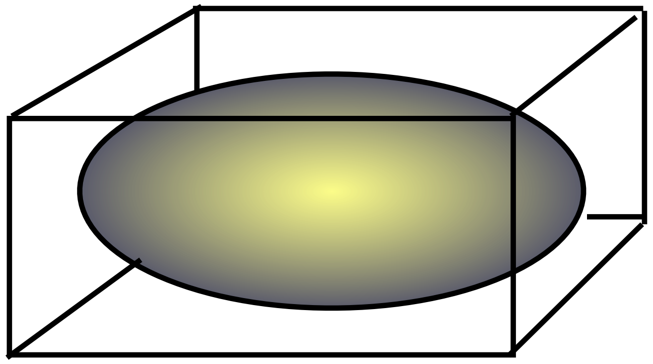
$N_a + N_b + N_c = 1$ applies



A. Hubert & RS,
Magnetic Domains

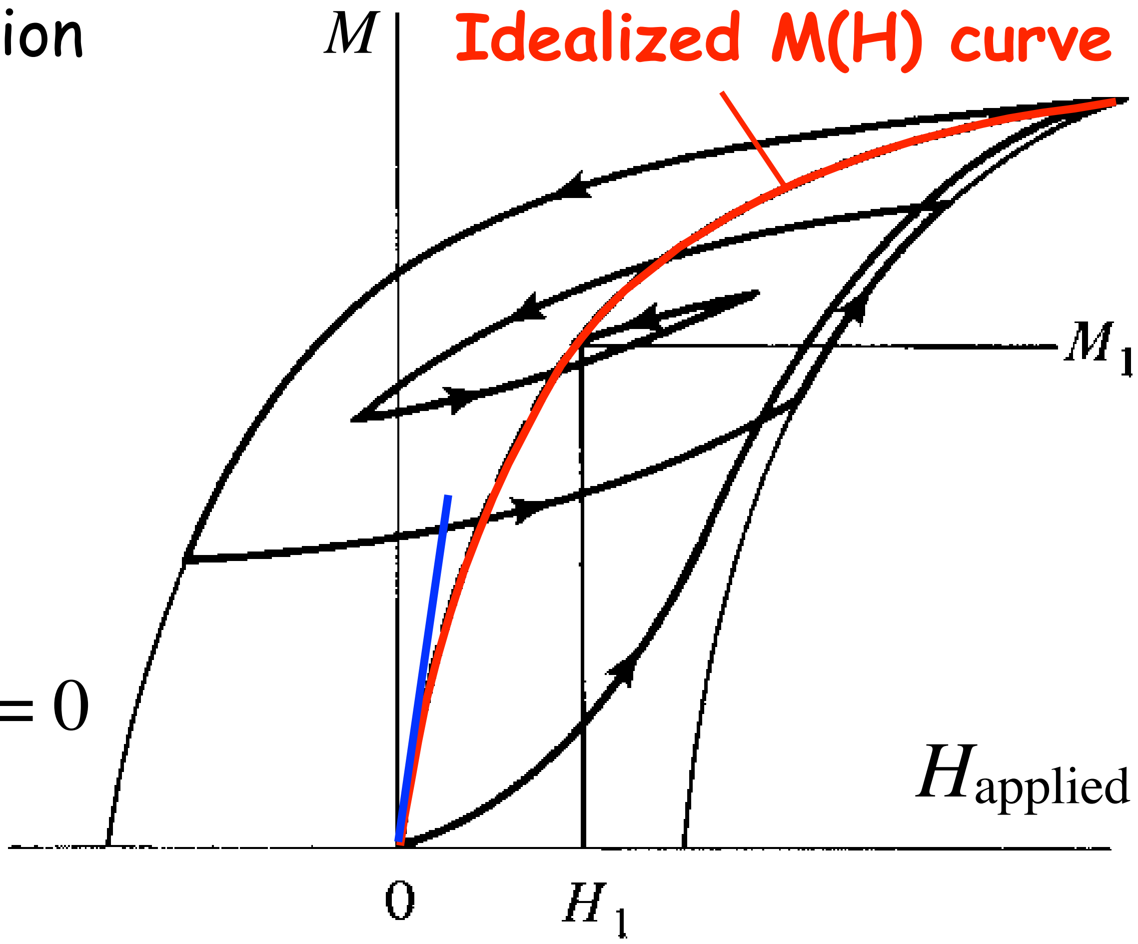
3. Magnetic Measurement …

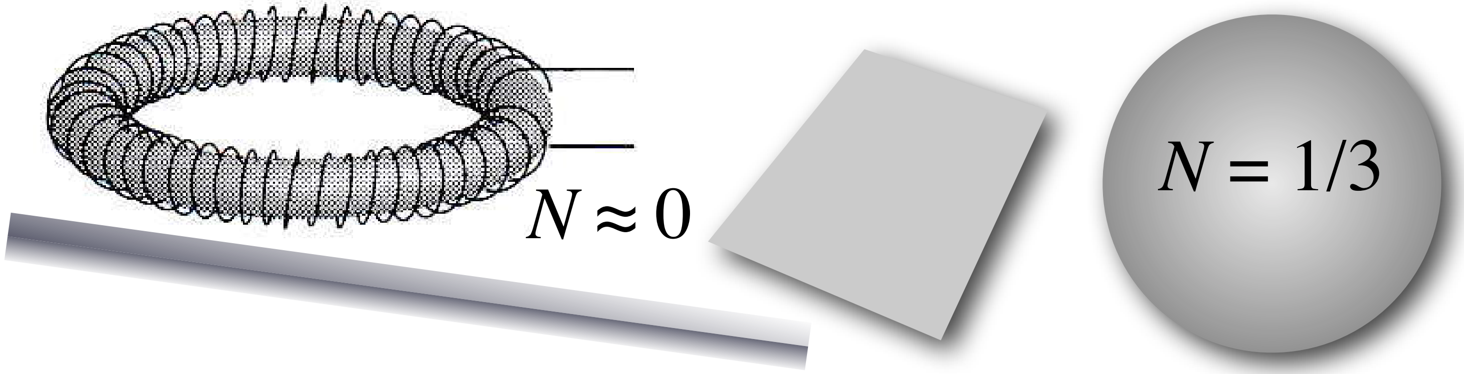
General aspects



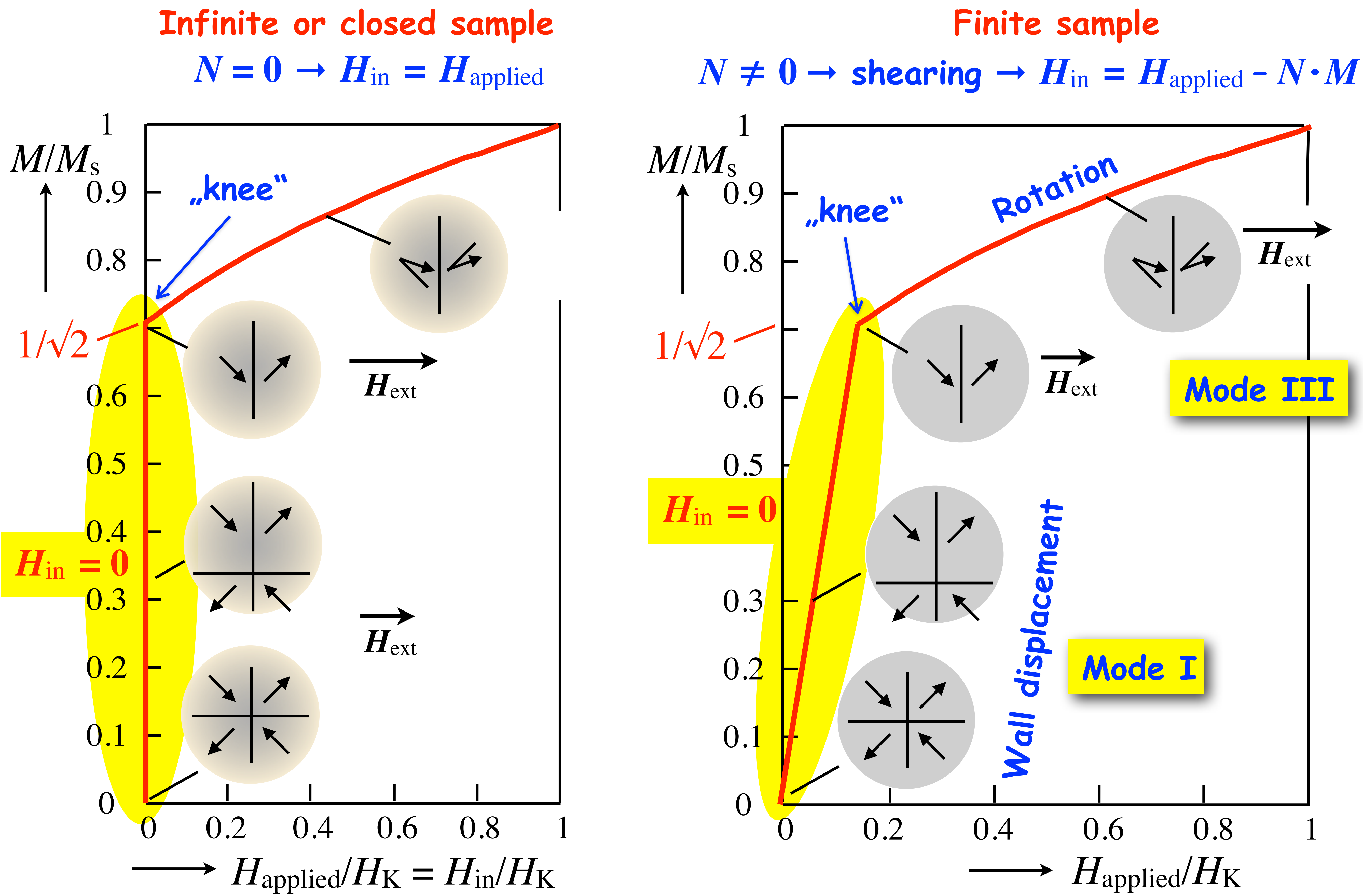
Demagnetizing factor

- Demagnetizing factor of compact bodies is often well approximated by that of inscribed ellipsoid
- Experimental determination of the demagnetization factor:
  - For every dc-field an ac-field of decreasing amplitude is superimposed (helps to over-come barriers in magnetization process)
  - Then approximately that magnetization is achieved, the demagnetizing field of which is equal to the applied field [$H_{\mathrm{in}} = H_{\mathrm{applied}} - N \bullet M = 0$ up to „knee“ → $M = (1/N) \bullet H_{\mathrm{applied}}$]
  - The initial slope of the magnetization curve is $\mathrm{d}M/\mathrm{d}H_{\mathrm{applied}} = 1/N$
- If possible: avoid demagnetization effect by chosing proper sample geometry for magnetic measurement





Phase Theory and M(H) curve



Infinite or closed sample

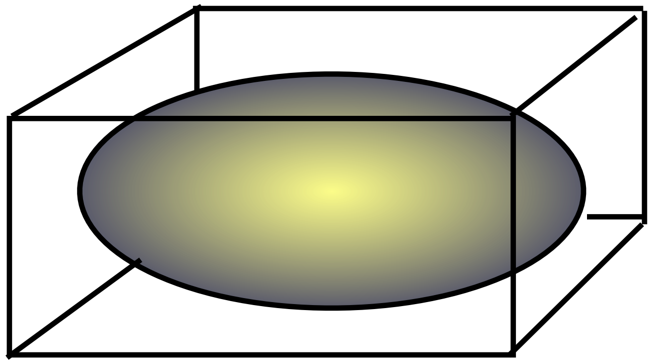
$N = 0 \rightarrow H_{\text{in}} = H_{\text{applied}}$

Finite sample

$N \neq 0 \rightarrow$ shearing $\rightarrow H_{\text{in}} = H_{\text{applied}} - N \cdot M$

3. Magnetic Measurement …

General aspects



Demagnetizing factor

- Demagnetizing factor of compact bodies is often well approximated by that of inscribed ellipsoid
- Experimental determination of the demagnetization factor:
  - For every dc-field an ac-field of decreasing amplitude is superimposed (helps to over-come barriers in magnetization process)
  - Then approximately that magnetization is achieved, the demagnetizing field of which is equal to the applied field [$H_{\text{in}} = H_{\text{applied}} - N \bullet M = 0$ up to „knee" → $M = (1/N) \bullet H_{\text{applied}}$]
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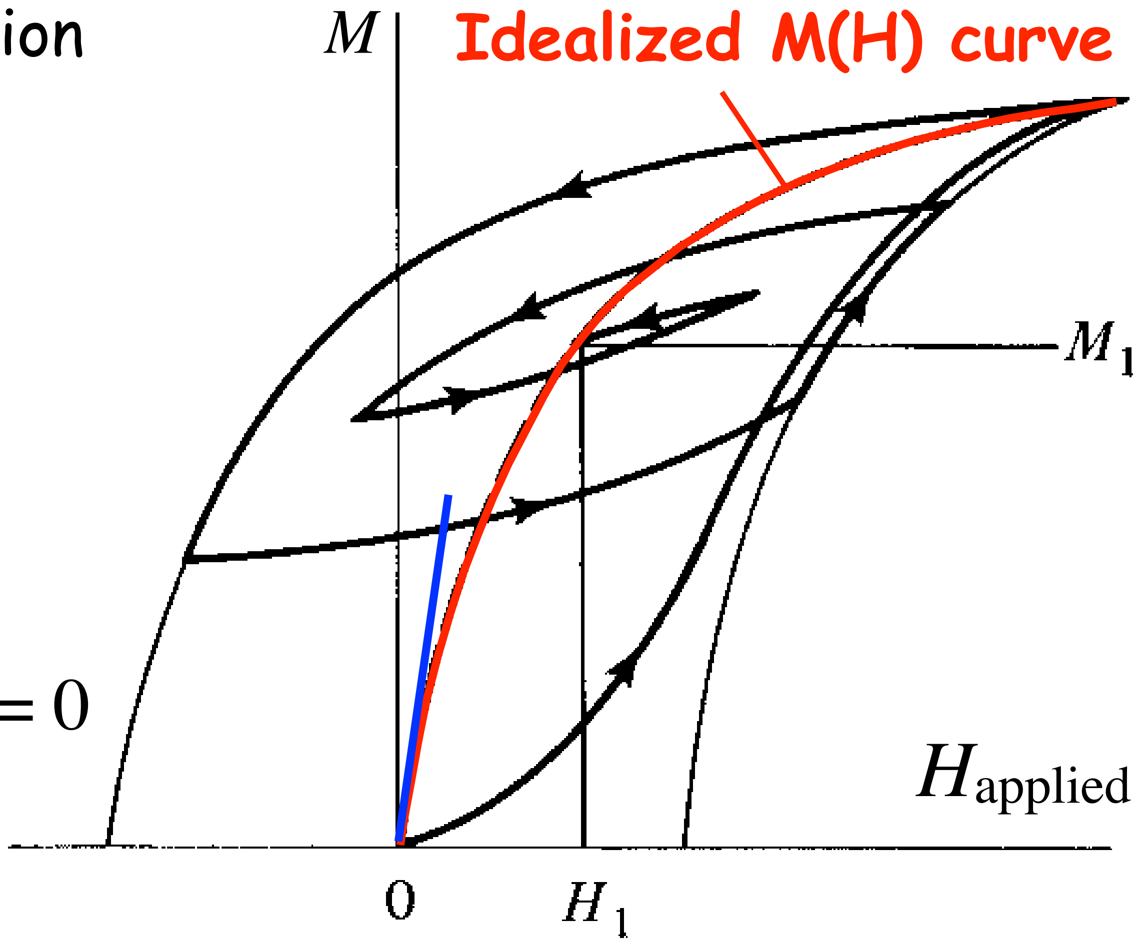
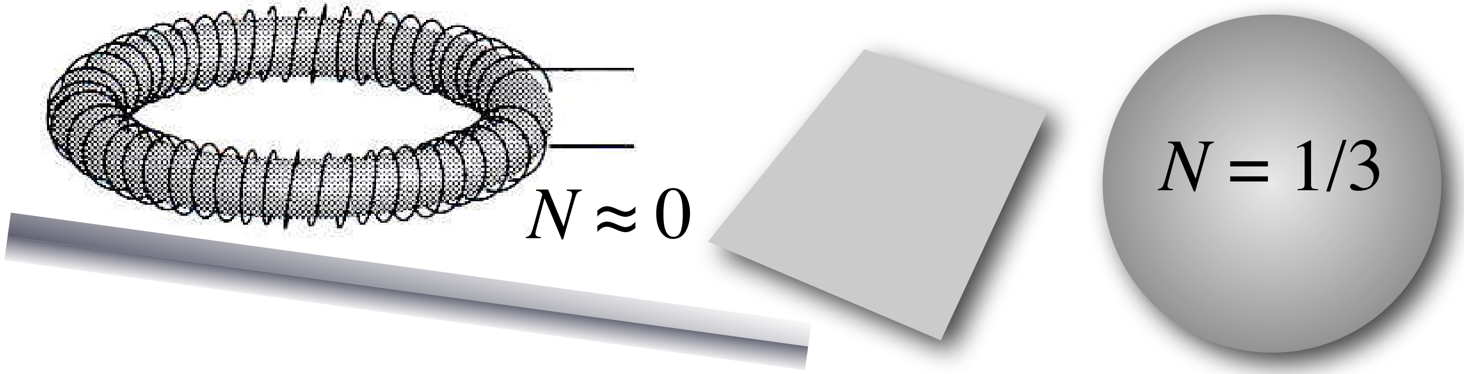




3. Magnetic Measurement …

General aspects

Conclusion (demagnetization problematics):

The long-range nature of the demagnetizing field makes the measured property of any test specimen geometry-dependent

Demagnetizing field should be avoided when characterizing **soft** magnetic materials

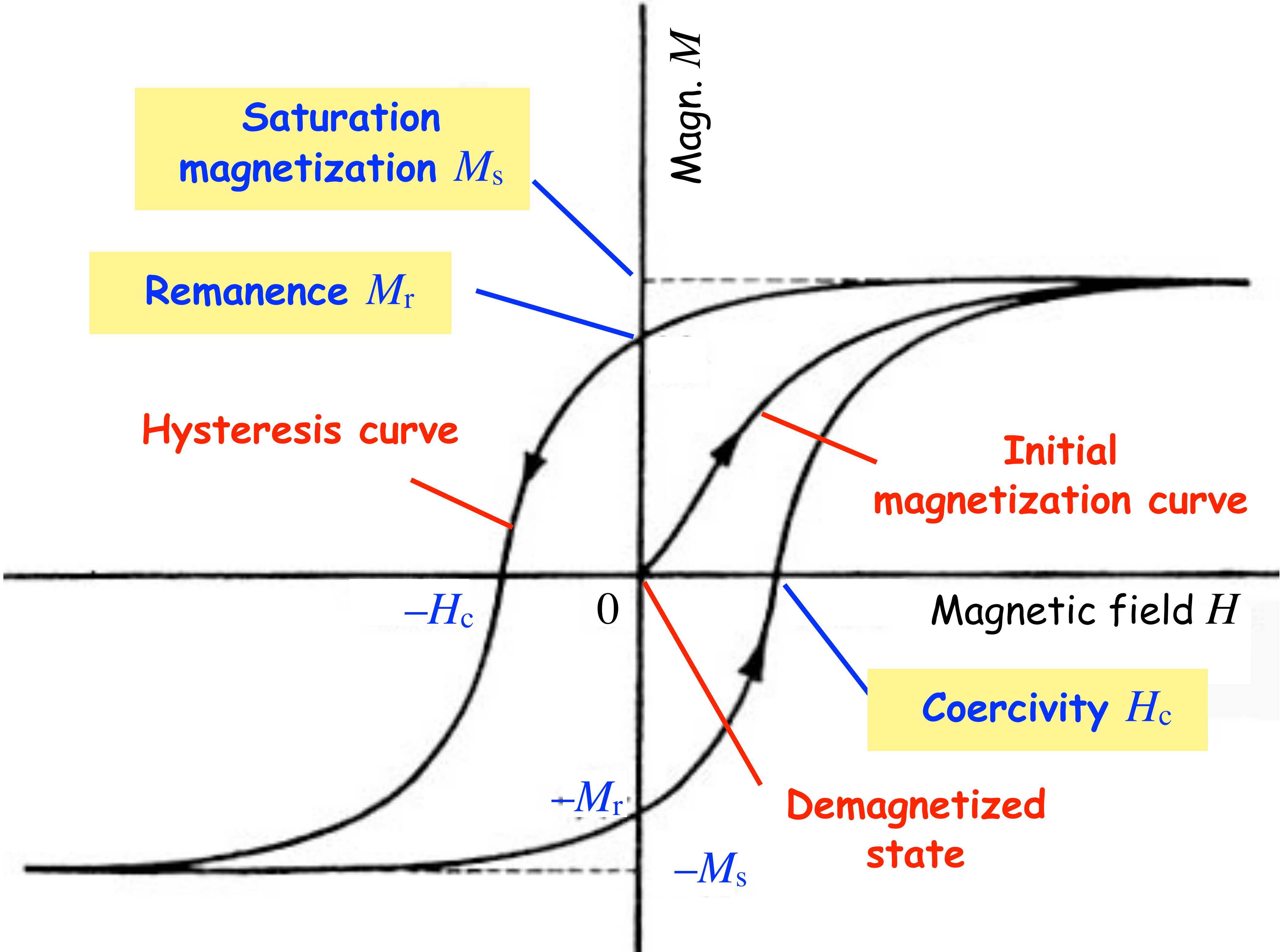
Demagnetizing field can be tolerated when characterizing **hard** magnetic materials, provided it is accurately known and is possibly uniform

For finite, non-ellipsoidal samples the magnetization is non-uniform due to demagnetizing effects → has to be considered when placing pick-up coil for inductive measurements

3. Magnetic Measurement …

General aspects

Hysteresis curve:



Saturation magnetization M_s

Remanence M_r

Hysteresis curve

Initial magnetization curve

Coercivity H_c

Demagnetized state

3. Magnetic Measurement …

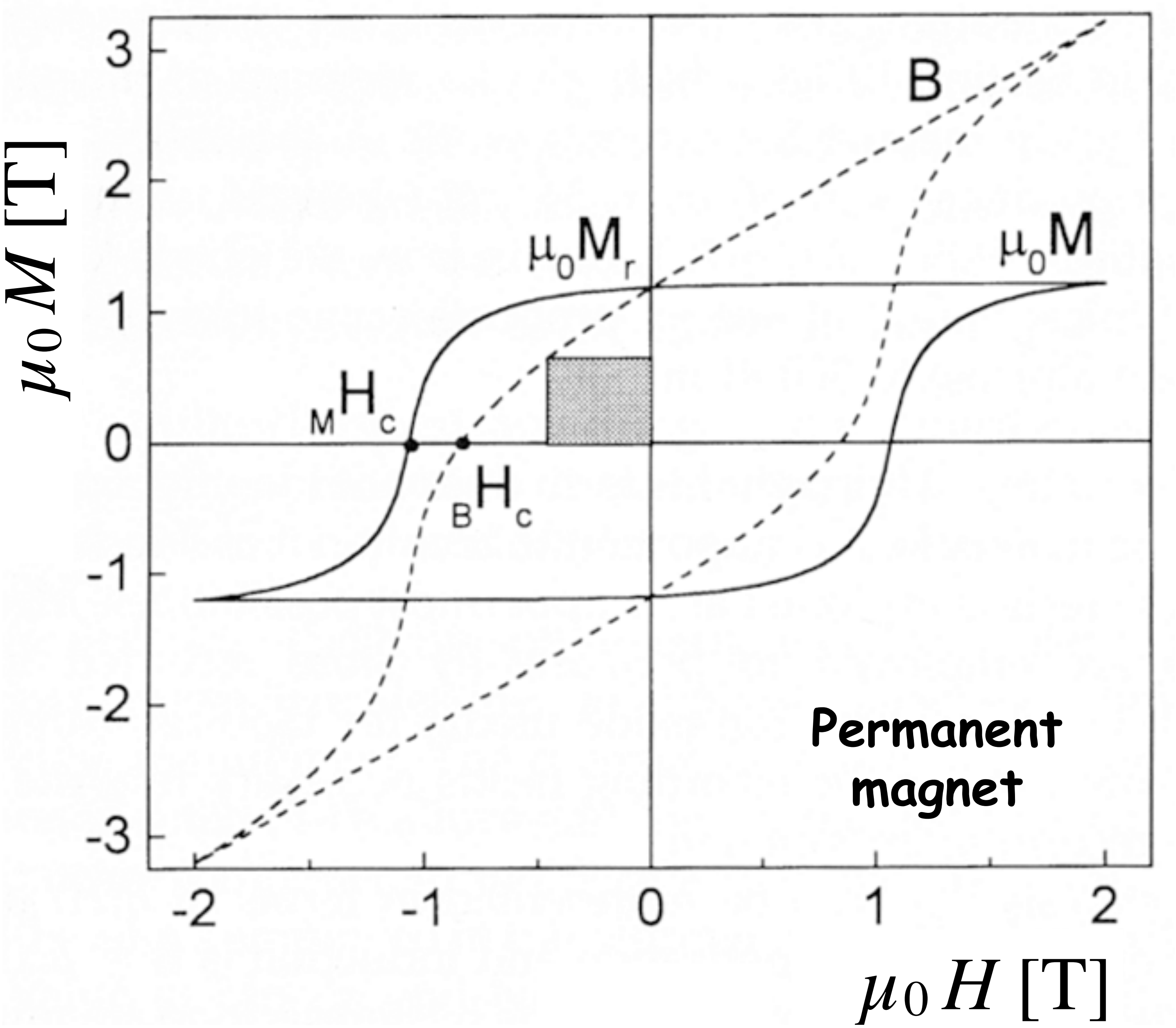
General aspects

Hysteresis curve: B and M

- Difference between $M(H)$ and $B(H)$

$$B(H) = \mu_0\,(H + M)$$

- Soft magnets:
  Fields involved in hysteresis loop are much smaller than corresponding magnetization values
  $\rightarrow B \cong \mu_0 M$
  $\rightarrow$ difference between $B(H)$ and $M(H)$ negligible
- Hard magnets:
  H and M have comparable orders $\rightarrow B(H)$ significantly different from $M(H)$



Permanent magnet

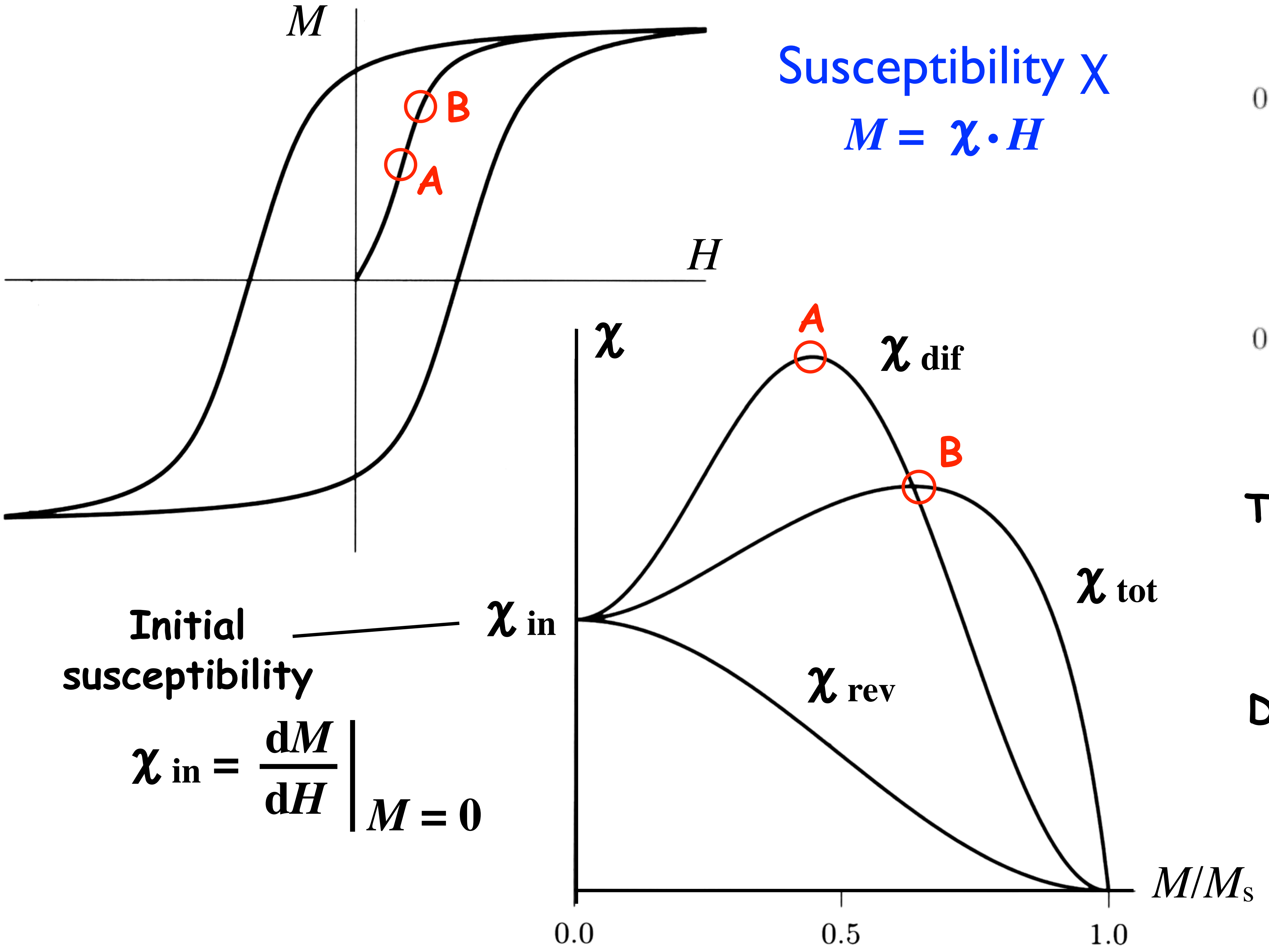
3. Magnetic Measurement …

General aspects

Hysteresis curve: Susceptibility

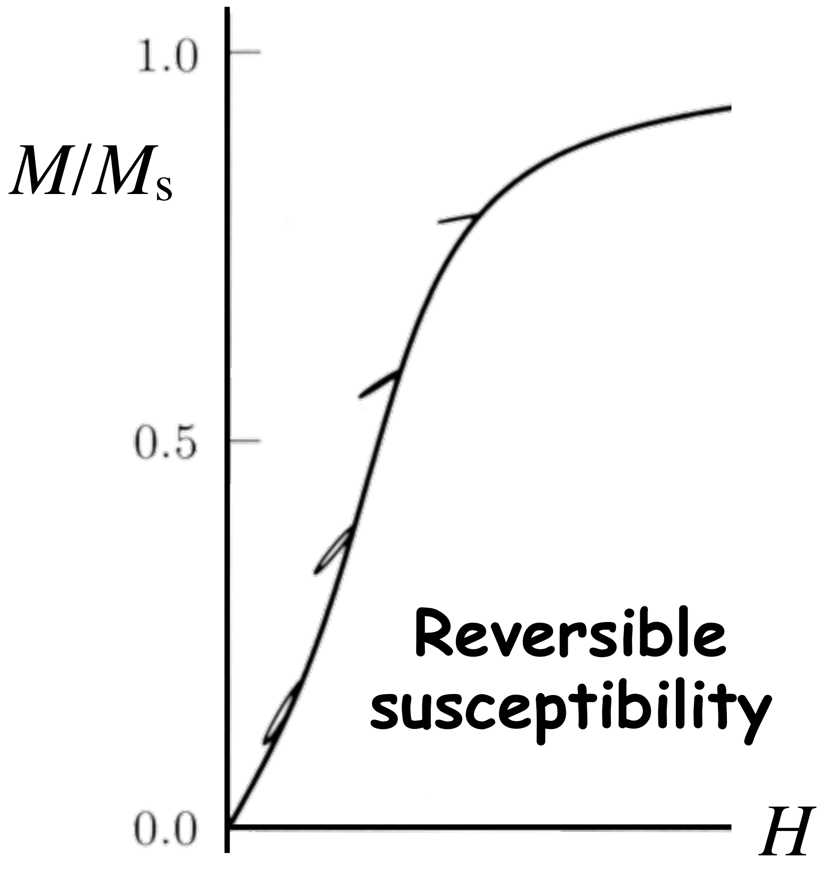
Susceptibility χ

$$M = \chi \cdot H$$



Initial susceptibility

$$\chi_{\text{in}} = \left.\frac{\mathrm{d}M}{\mathrm{d}H}\right|_{M=0}$$



Total susceptibility:

$$\chi_{\text{tot}} = \frac{M}{H}$$

Differential suscept.:

$$\chi_{\text{dif}} = \frac{\mathrm{d}M}{\mathrm{d}H}$$

3. Magnetic Measurement …

General aspects

Hysteresis curve: Anisotropy

- Magnetic anisotropy is defined as energy differences needed for saturation along different axes → Magnetic anisotropy can be determined from $M(H)$-curve for single crystals by comparing magnetization curves along hard- and easy directions
- Example: cubic magnetocrystalline anisotropy (case of iron)

$$e_{\mathrm{Kc}} = K_{\mathrm{c1}}\cdot(m_1^2m_2^2 + m_1^2m_3^2 + m_2^2m_3^2) + K_{\mathrm{c2}}\, m_1^2m_2^2m_3^2$$

m_{i} = Magnetization components along cubic axes (direction cosine)
K_{ci} = Anisotropy constants

$$\boldsymbol{m} = [100]: \quad e_{\mathrm{Kc}} = 0$$

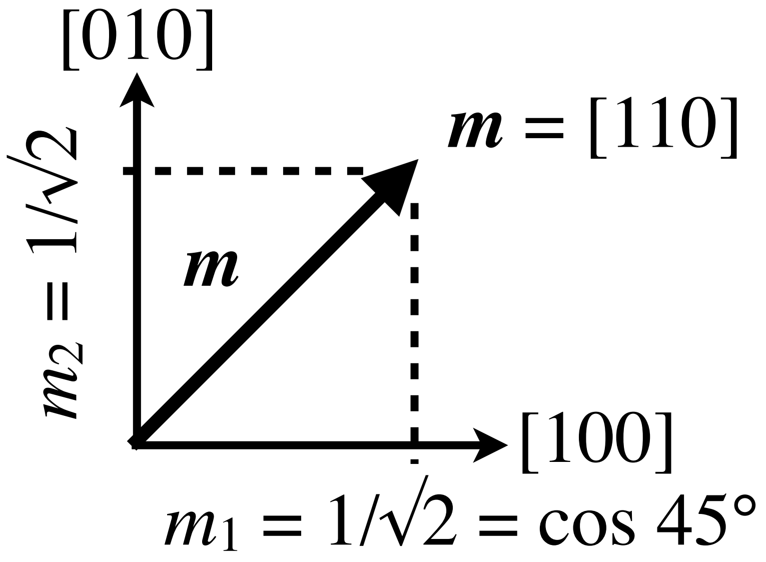
$$\boldsymbol{m} = [110]: \quad m_1 = m_2 = 1/\sqrt{2}$$

$$e_{\mathrm{Kc}} = K_{\mathrm{c1}}\ (1/2+0+0) = K_{\mathrm{c1}}/4$$

$$\boldsymbol{m} = [111]: \quad m_1 = m_2 = m_3 = 1/\sqrt{3}$$

$$e_{Kc} = K_{\mathrm{c1}}\ (1/9+1/9+1/9) + 1/27\, K_{\mathrm{c2}}$$

$$= 1/3\ K_{\mathrm{c1}} + 1/27\ K_{\mathrm{c2}}$$



3. Magne

G

Hysteresis curve: Ar

- Magnetic anisotropy is define
different axes → Magnetic ar
single crystals by comparing m
- Example: cubic magnetocrysta

$e_{Kc} = K_{c1}\cdot(m_1^2 m_2^2 + m_1^2 m_3^2$

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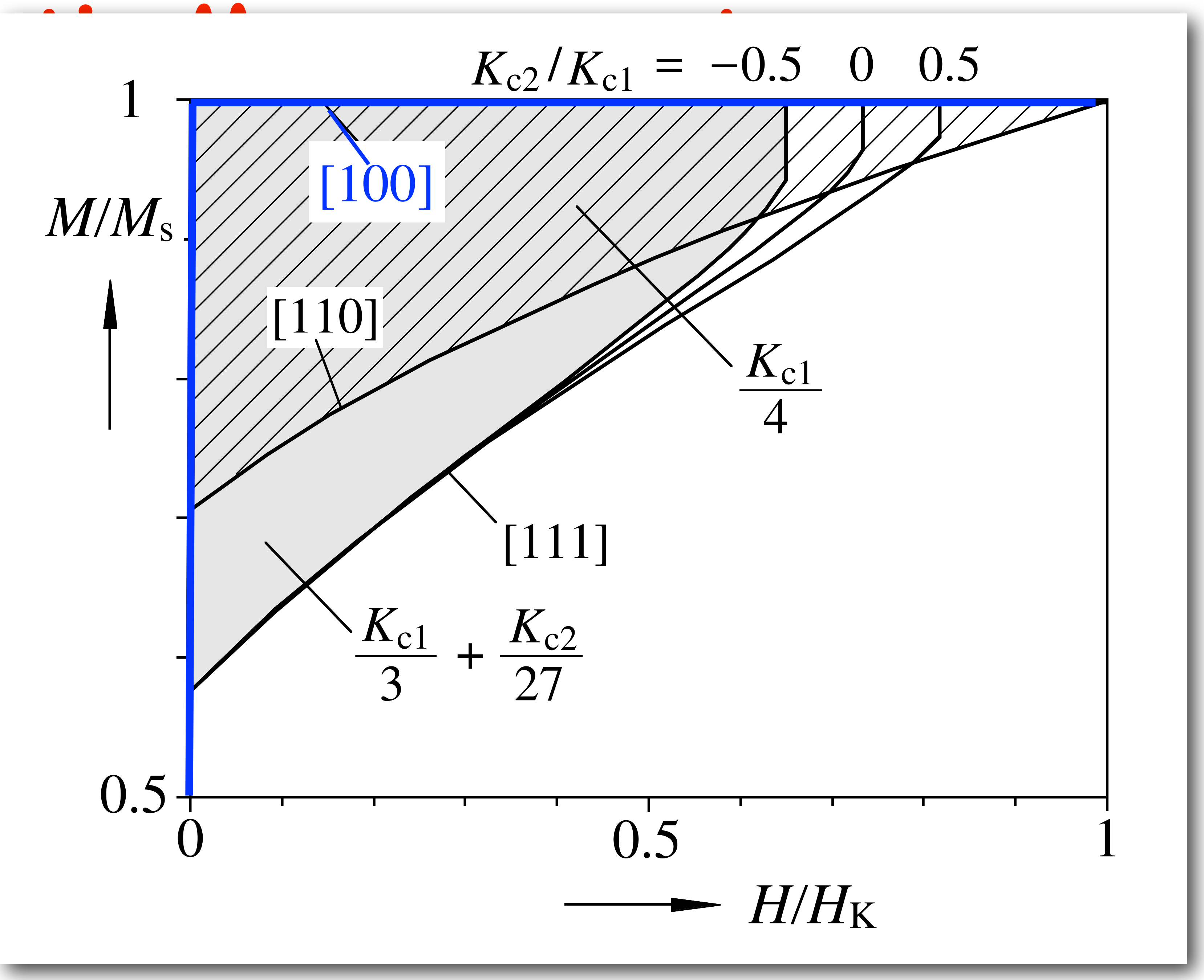
K_{ci} = Anisotropy consta



$\boldsymbol{m}$ = [100]: $e_{Kc} = 0$

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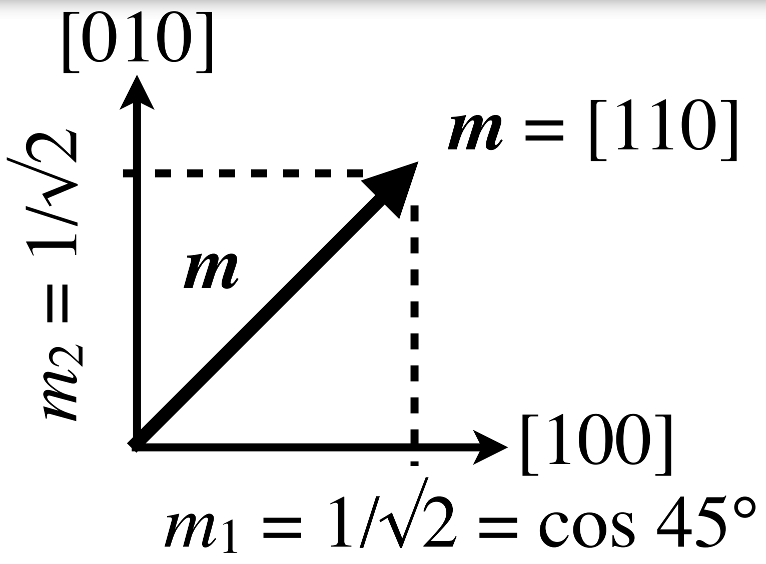
$e_{Kc} = K_{c1}\ (1/9+1/9+1/9) + 1/27\, K_{c2}$

$= 1/3\ K_{c1} + 1/27\ K_{c2}$



3. Magnetic Measurement …

General aspects

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m_i = Magnetization components along cubic axes (direction cosine)
K_ci = Anisotropy constants

- Limitations:
  - Hysteresis effects make the determination of the area ambiguous. Relying on the idealized magnetization curve offers fair solution to this problem
  - Non-ideal behaviour in approach to saturation. Internal stresses, inclusions and shape irregularities lead to a rounding of the magnetization curve. These effects depend on the magnetization direction because the magnetization deviations around such irregularities are influenced by anisotropy. The direct determination of the anisotropy from the magnetization curves is therefore often unreliable

3. Magnetic Measurement …

General aspects

Hysteresis curve: Anisotropy

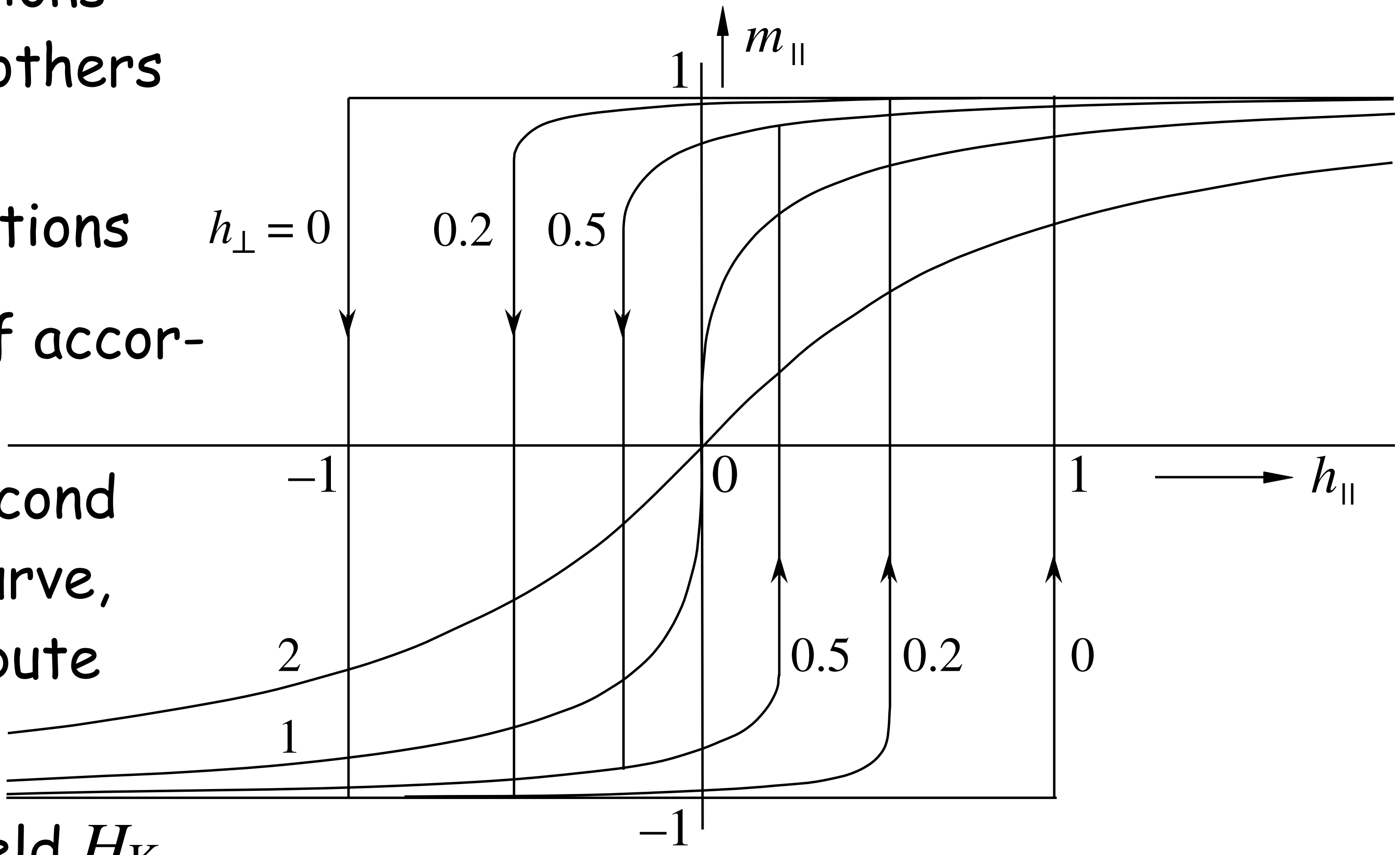
Singular Point Detection in polycrystals

- Detects singularities in magnetization curve of polycrystalline sample which are caused by singular contributions from certain grains

- Compare magnetization curves of uniaxial particles: There are field orientations for which curves are smooth, for others they show characteristic jumps
= first-order magnetization transitions

- In polycrystalline sample: jumps of accordingly oriented grains will show up as singularities (maxima) in the second derivative of the magnetization curve, while the other grains only contribute to a smooth background

→ Determination of anisotropy field H_K



3. Magnetic Measurement …

General aspects

Hysteresis curve: Anisotropy

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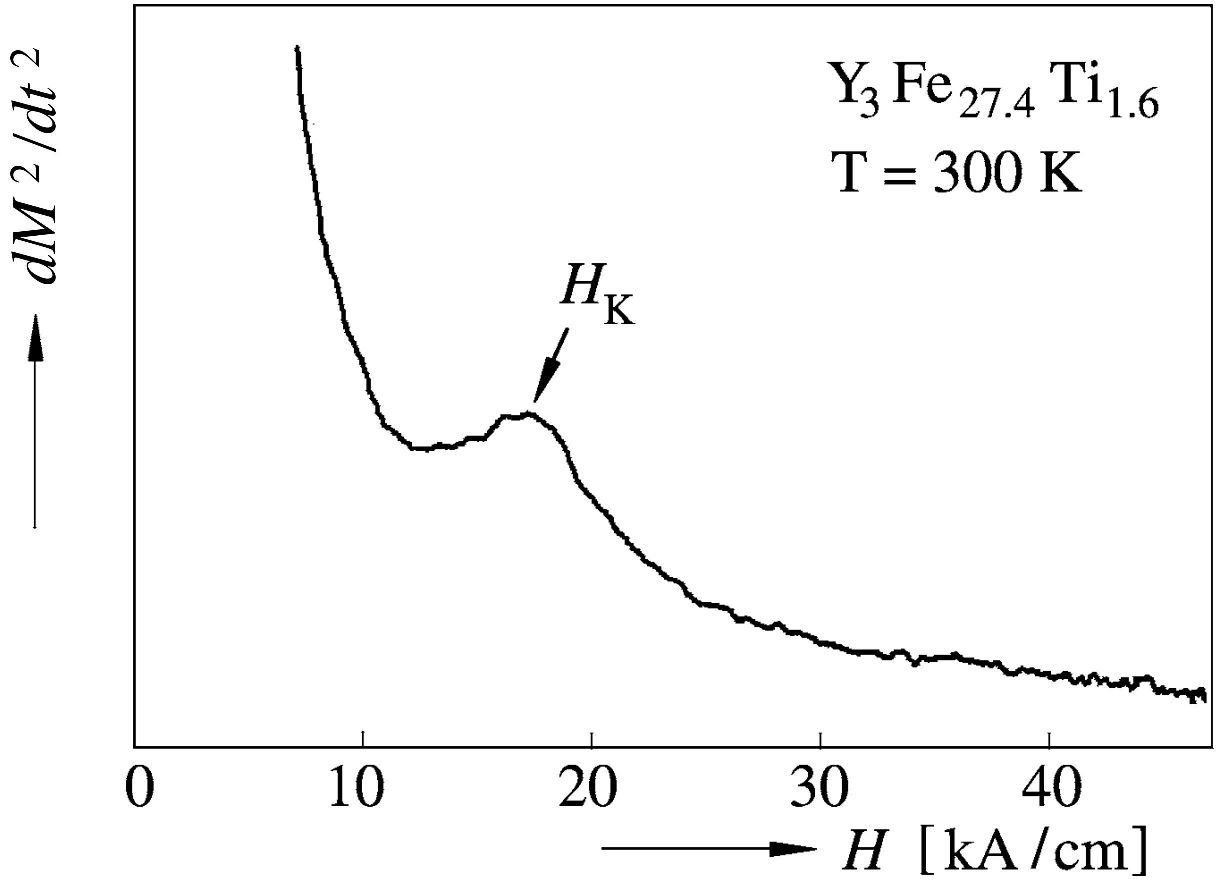


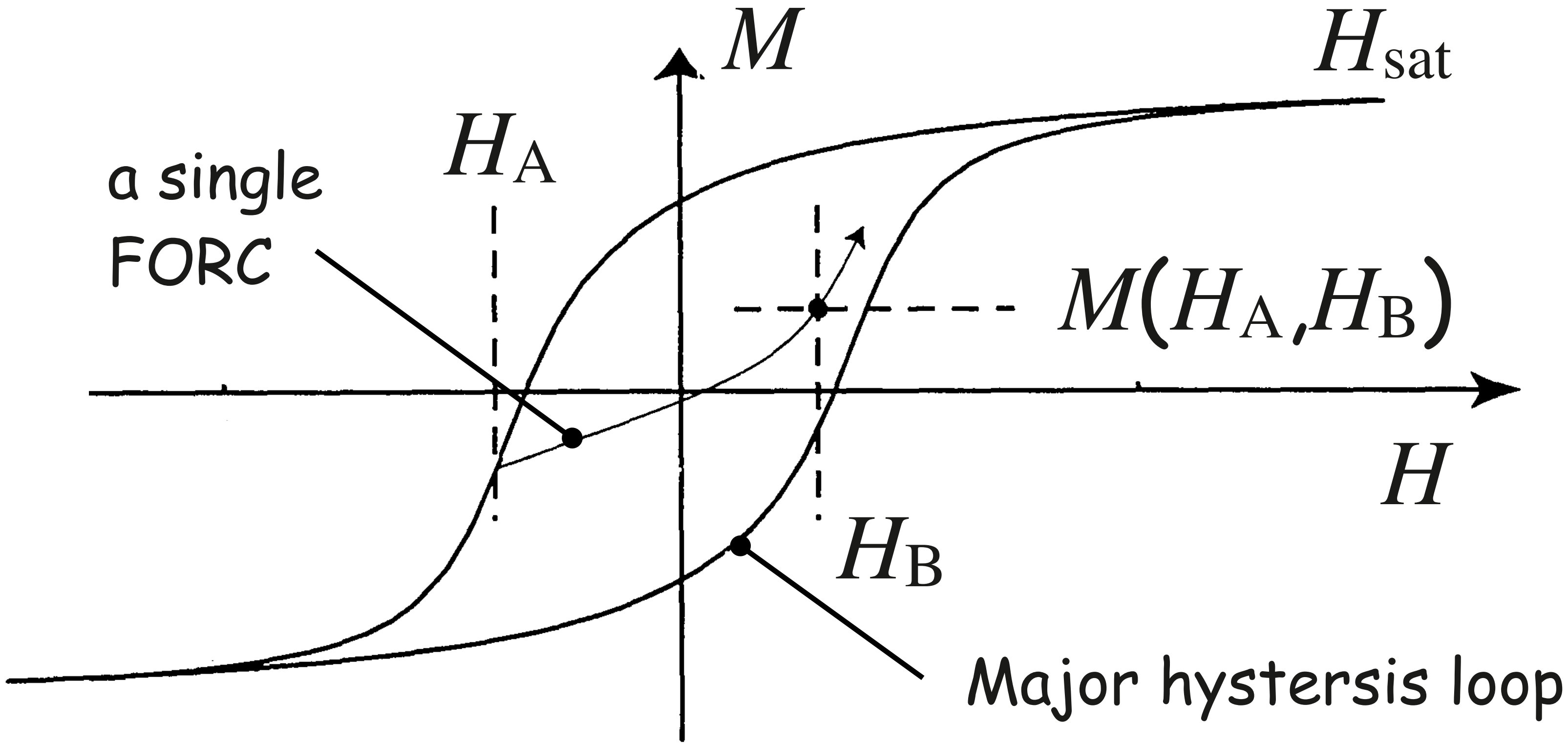
Courtesy R. Grössinger, Vienna

3. Magnetic Measurement ...

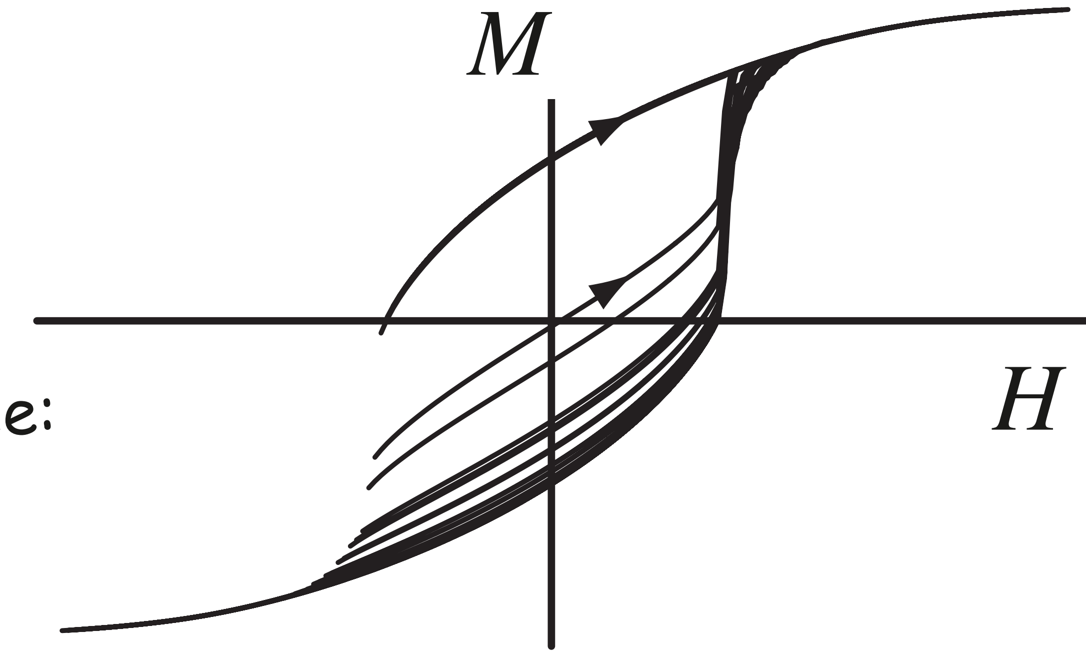
General aspects

FORC: First Order Reversal Curve

- A FORC is measured by saturating sample in field H_{sat}, decreasing the field to a reversal field H_{A}, then sweeping field back to H_{sat} in a series of **equal** field steps H_{B}. The magnetization curve between H_{A} and H_{B} is a FORC



- This process is repeated for many values of H_{A} yielding a series of FORCs, and the measured magnetization at each step as a function of H_{A} and H_{B} gives $M(H_{\mathrm{A}}, H_{\mathrm{B}})$ distribution
- The FORC distribution $\varrho(H_{\mathrm{A}}, H_{\mathrm{B}})$ is defined as the mixed second derivative of the $M(H_{\mathrm{A}}, H_{\mathrm{B}})$ - surface: $\varrho(H_{\mathrm{A}}, H_{\mathrm{B}}) = -\,\partial^2 M(H_{\mathrm{A}}, H_{\mathrm{B}})/\partial H_{\mathrm{A}}\,\partial H_{\mathrm{B}}$
- $\varrho(H_{\mathrm{C}}, H_{\mathrm{U}})$ is plotted as contour- or 3D-plot



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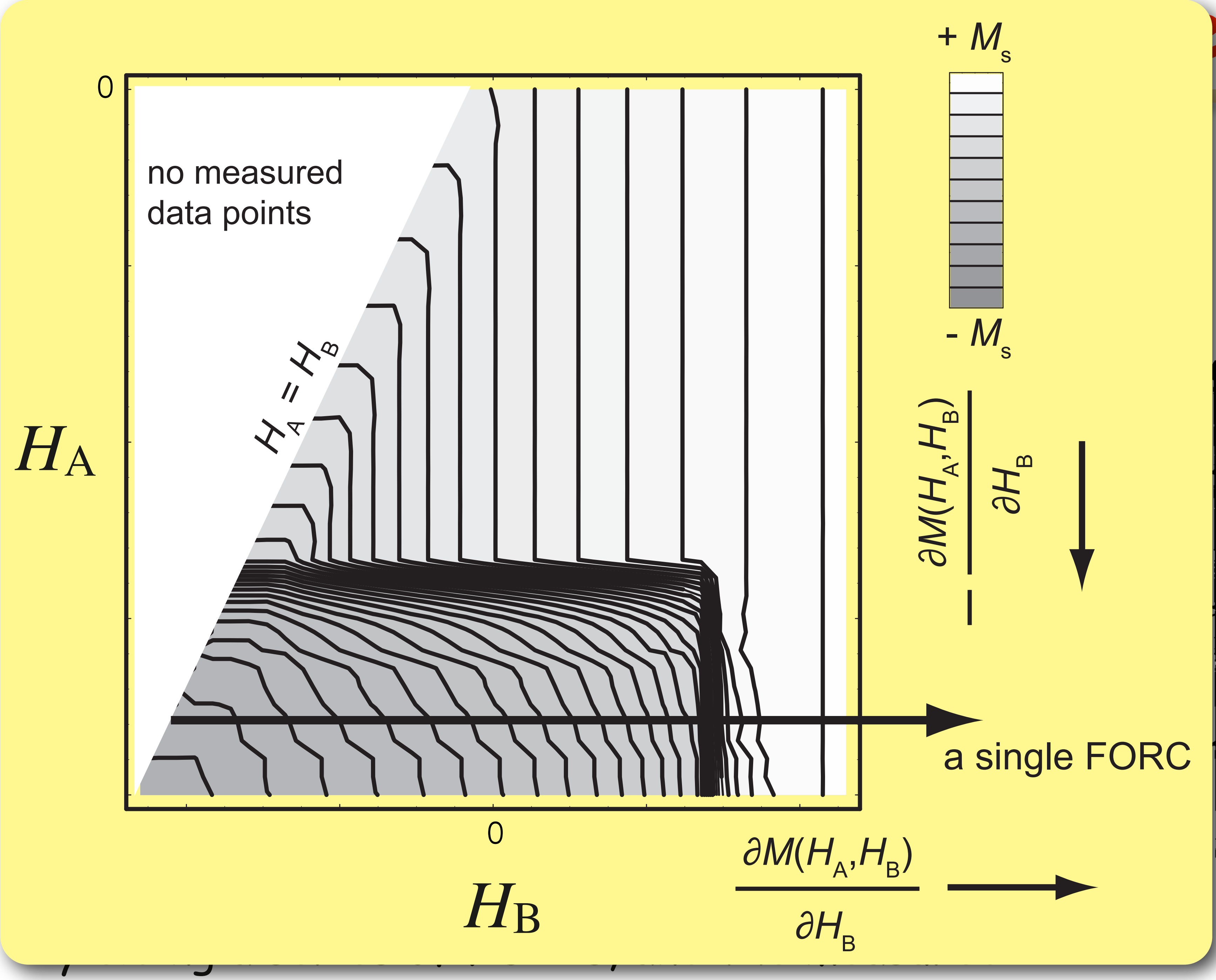

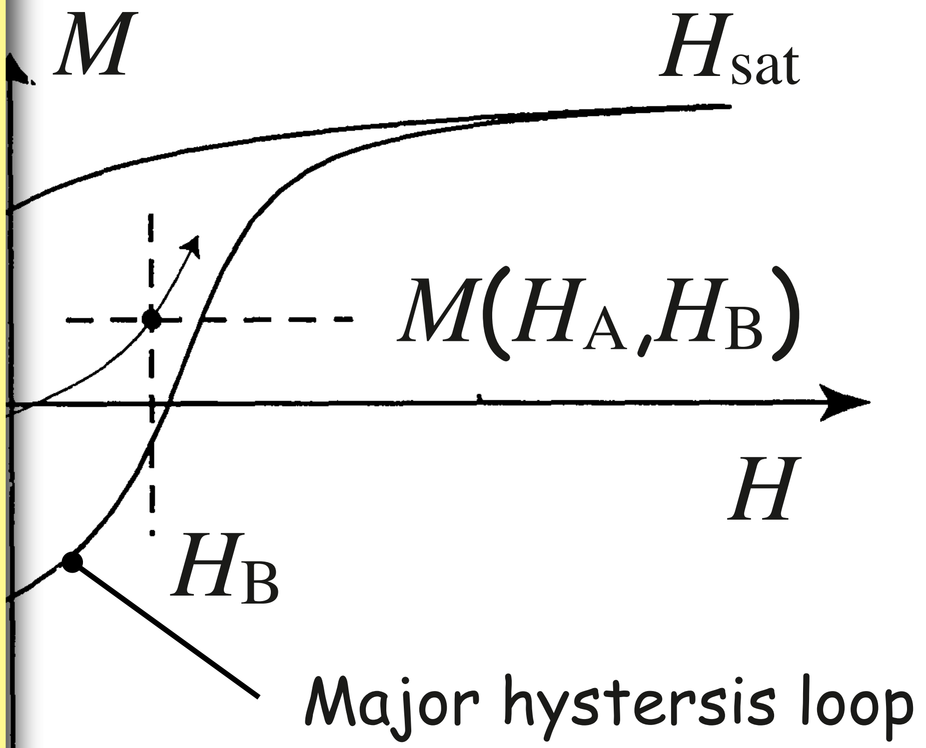

magnetization at each step as a function of H_A and H_B gives $M(H_A, H_B)$ distribution

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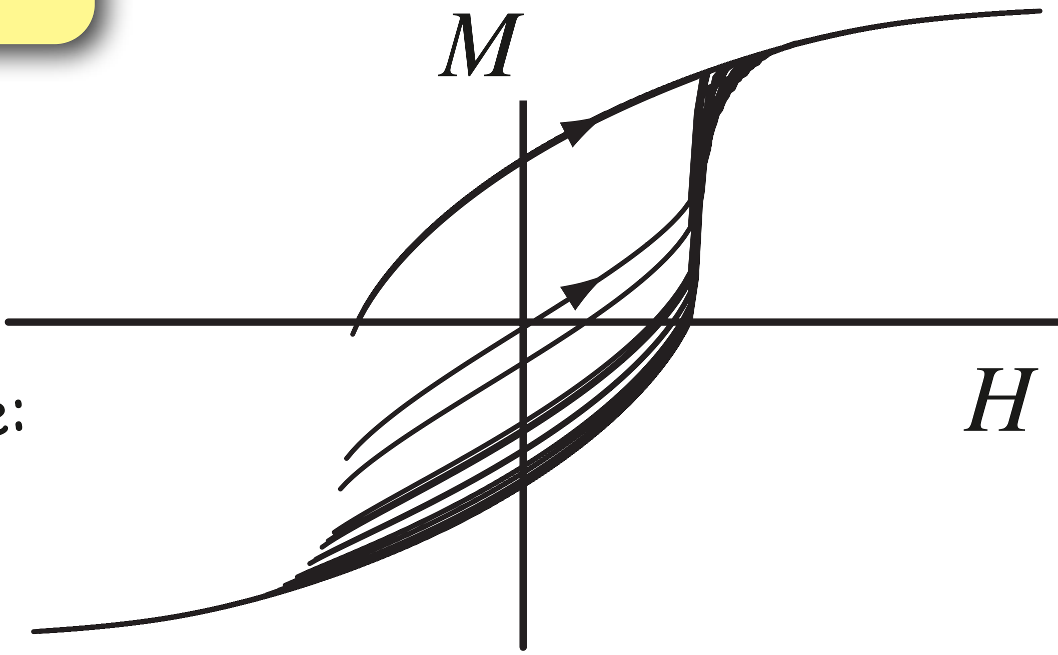
- $\varrho(H_C, H_U)$ is plotted as contour- or 3D-plot



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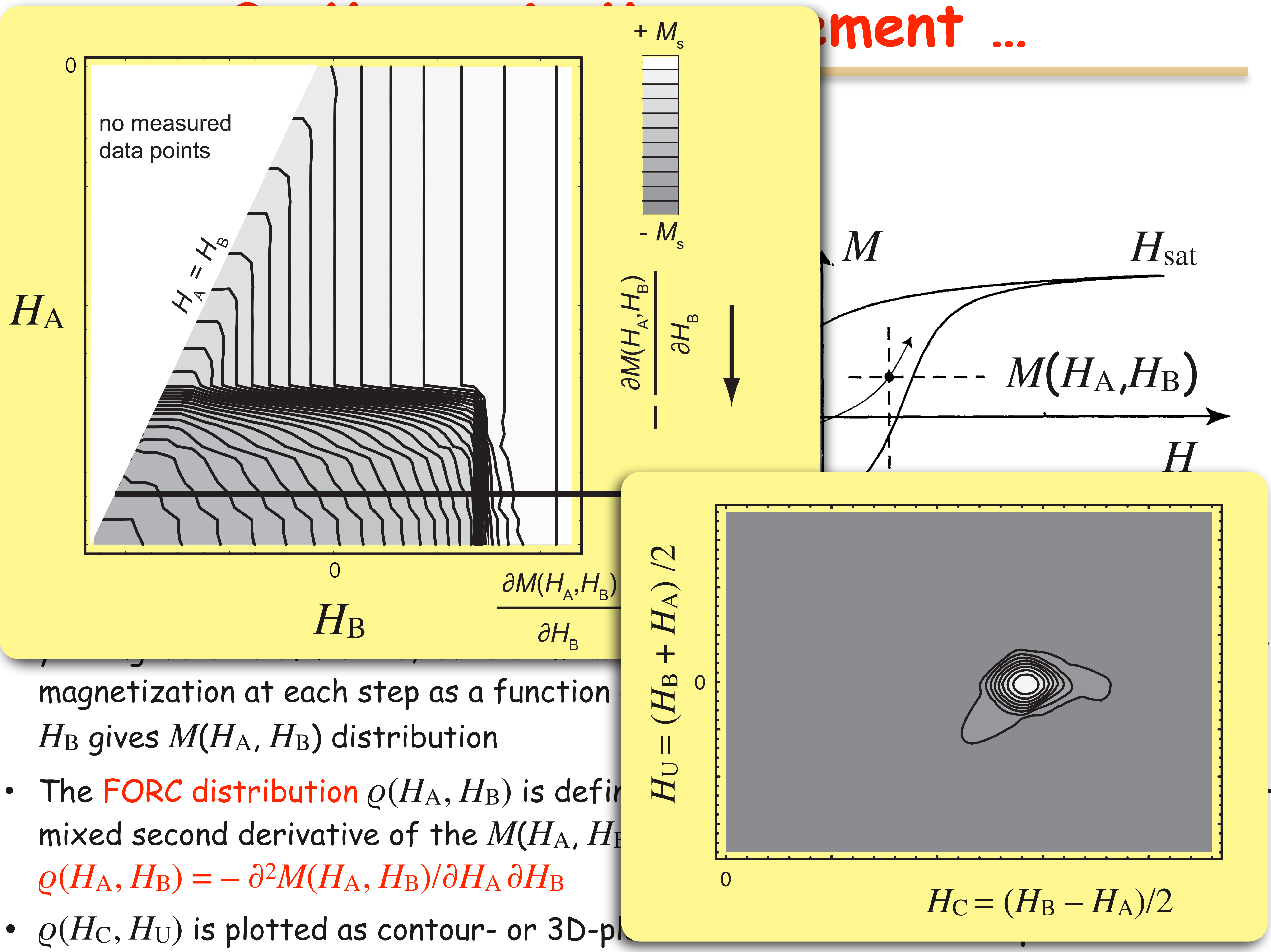


magnetization at each step as a function

H_B gives $M(H_A, H_B)$ distribution

- The FORC distribution $\varrho(H_A, H_B)$ is defi
  mixed second derivative of the $M(H_A, H$
  $\varrho(H_A, H_B) = -\,\partial^2 M(H_A, H_B)/\partial H_A\,\partial H_B$
- $\varrho(H_C, H_U)$ is plotted as contour- or 3D-pl

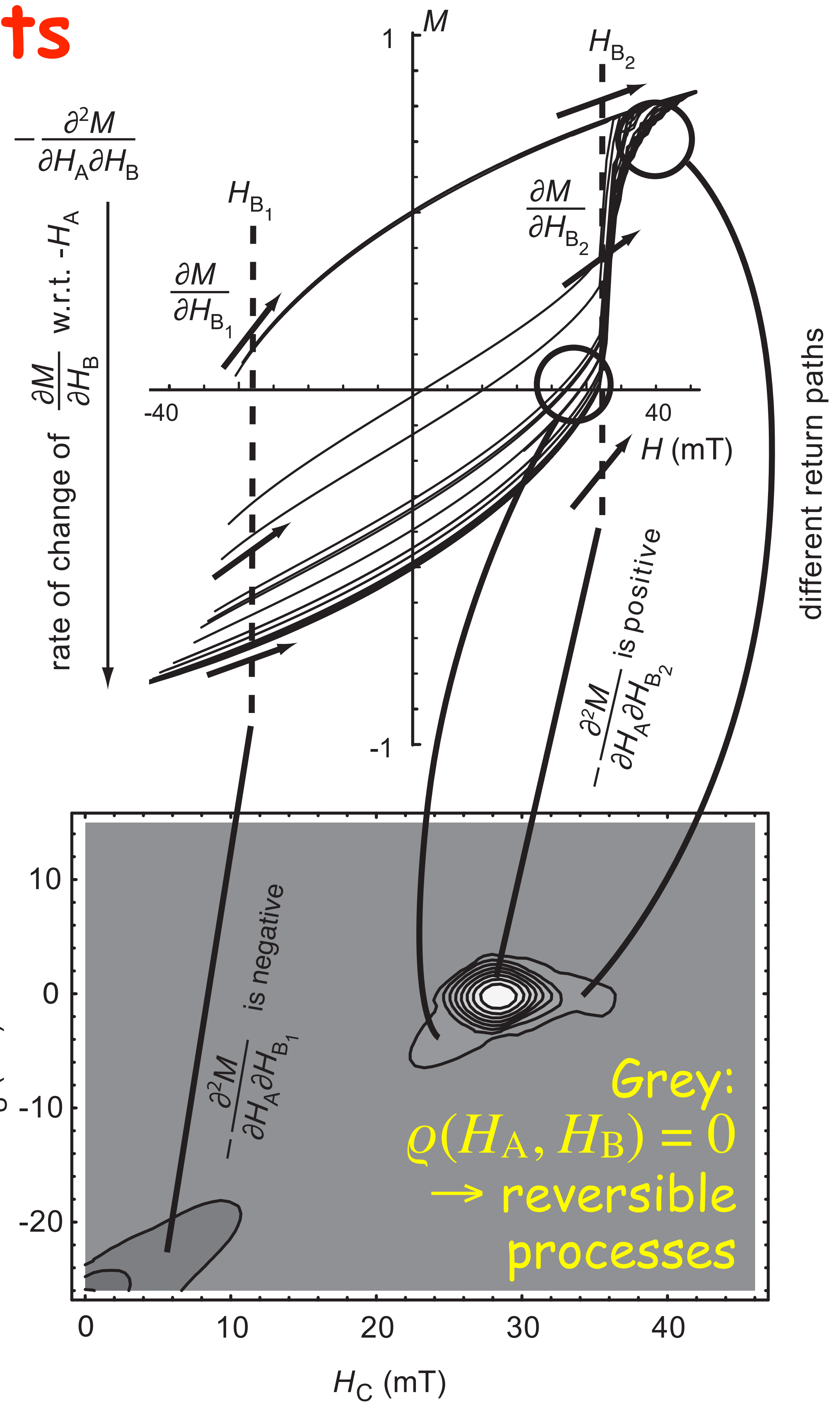
3. Magnetic Measurement ...

General aspects

FORC: First Order Reversal Curve

- Example: Assemble of randomly orientated, non-interacting, uniaxial Single-Domain grains
- Central peak; switching of the magnetization at H_{switch}
- Lower left-hand arm of the „boomerang“: related to FORCs near the relatively abrupt positive switching field
- The right-hand arm of boomerang: related to more subtle contours, which are due to the FORCs having different return paths
- Negative region: related to sections of the FORCs where $H_{\text{B}} < 0$

Muxworthy & Roberts, in: Encyclopedia of Geomagnetism and Paleomagnetism, Springer, 2007, Pages: 266 - 272



Grey: $\varrho(H_{\text{A}}, H_{\text{B}}) = 0$ → reversible processes

3. Magnetic Measurement …

General aspects

FORC: First Order Reversal Curve

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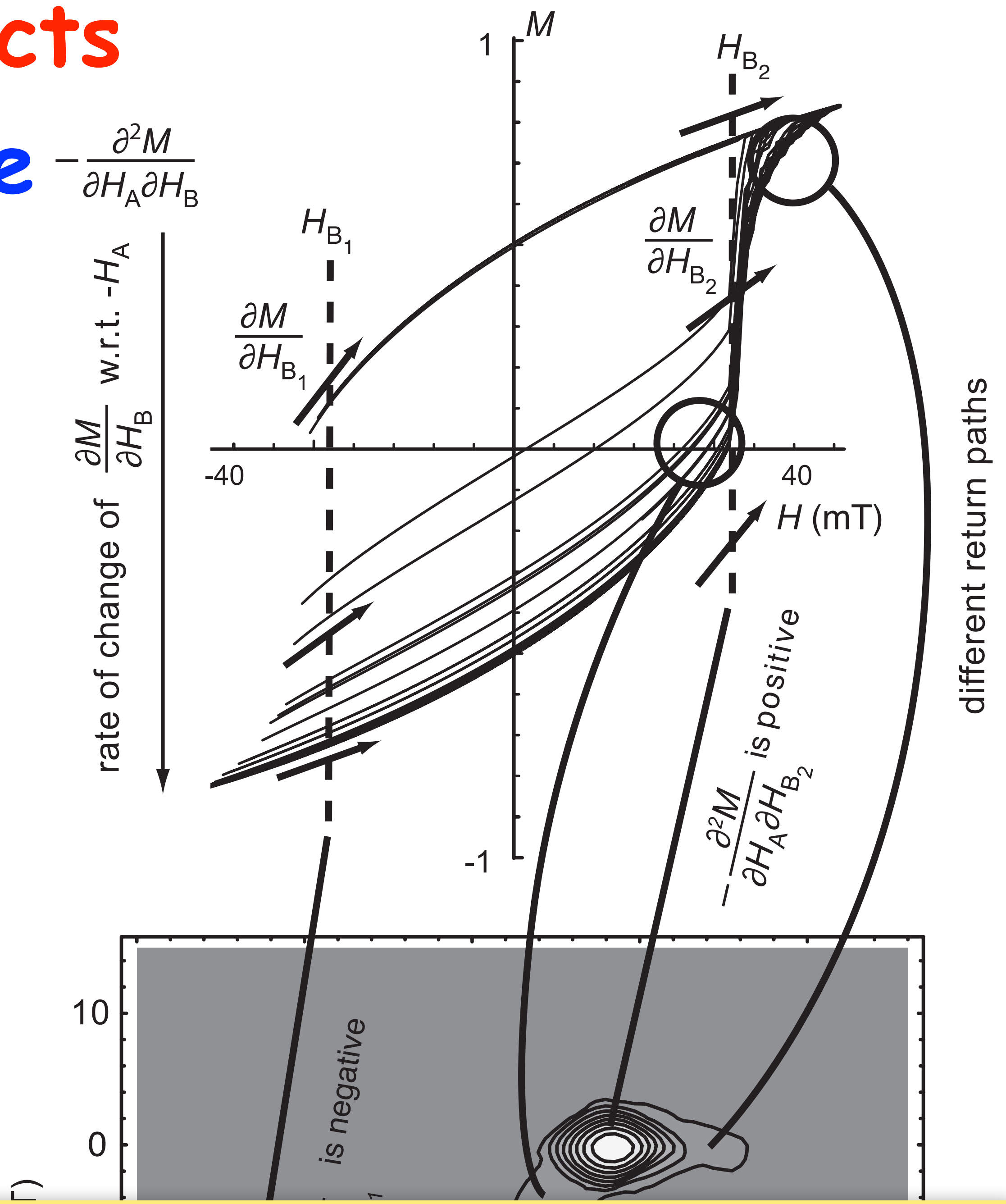


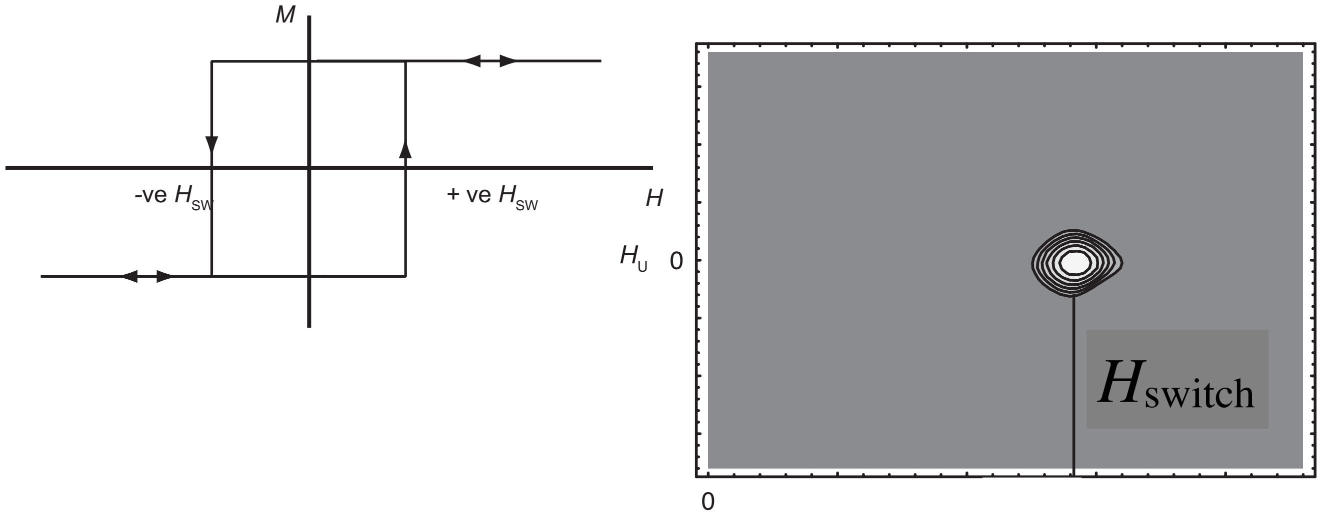
- FORC distribution eliminates purely reversible components of magnetization process. Thus any non-zero ϱ corresponds to irreversible switching processes
- FORC thus provide insight into relative proportions of reversible and irreversible components of the magnetization process

H_C (mT)

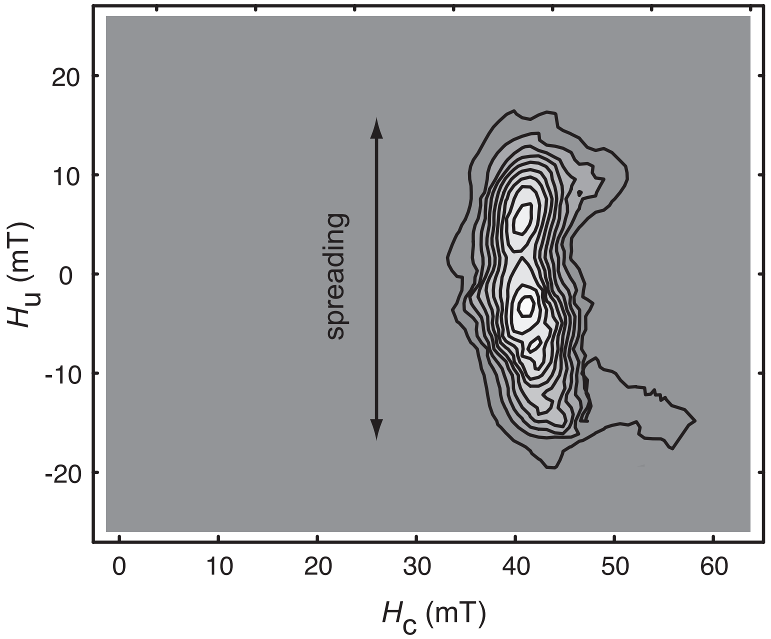
3. Magnetic Measurement …

General aspects

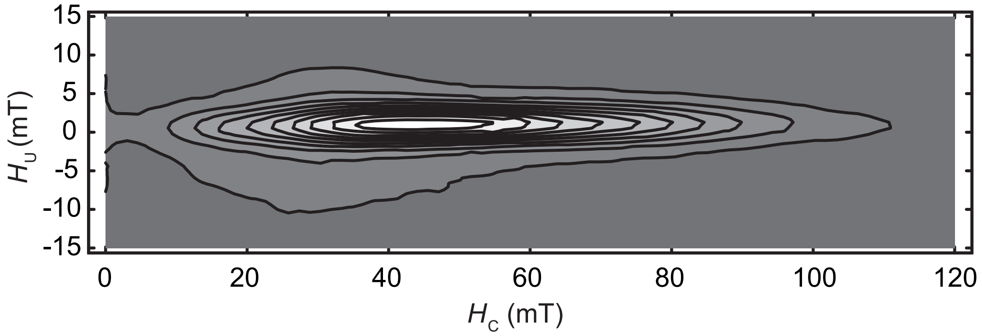
FORC: First Order Reversal Curve



Non-interacting SD grains, equally oriented: sharp peak, $H_C = H_{switch}$



Interacting SD grains: spreading in the HU direction



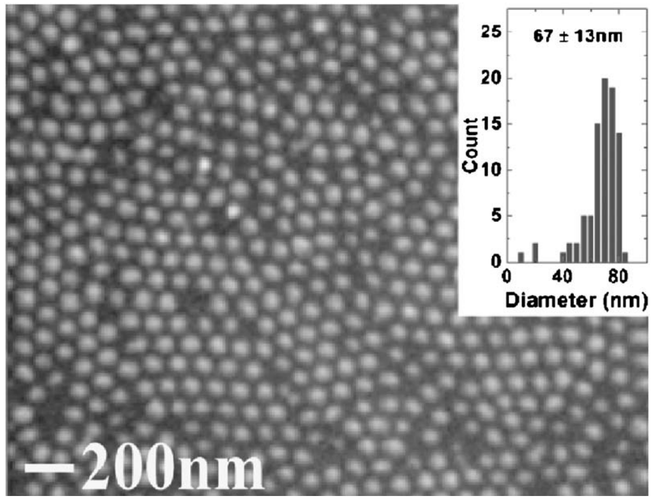
If the SD grains have a distribution of switching fields, this causes the FORC diagram to stretch out in H_C direction

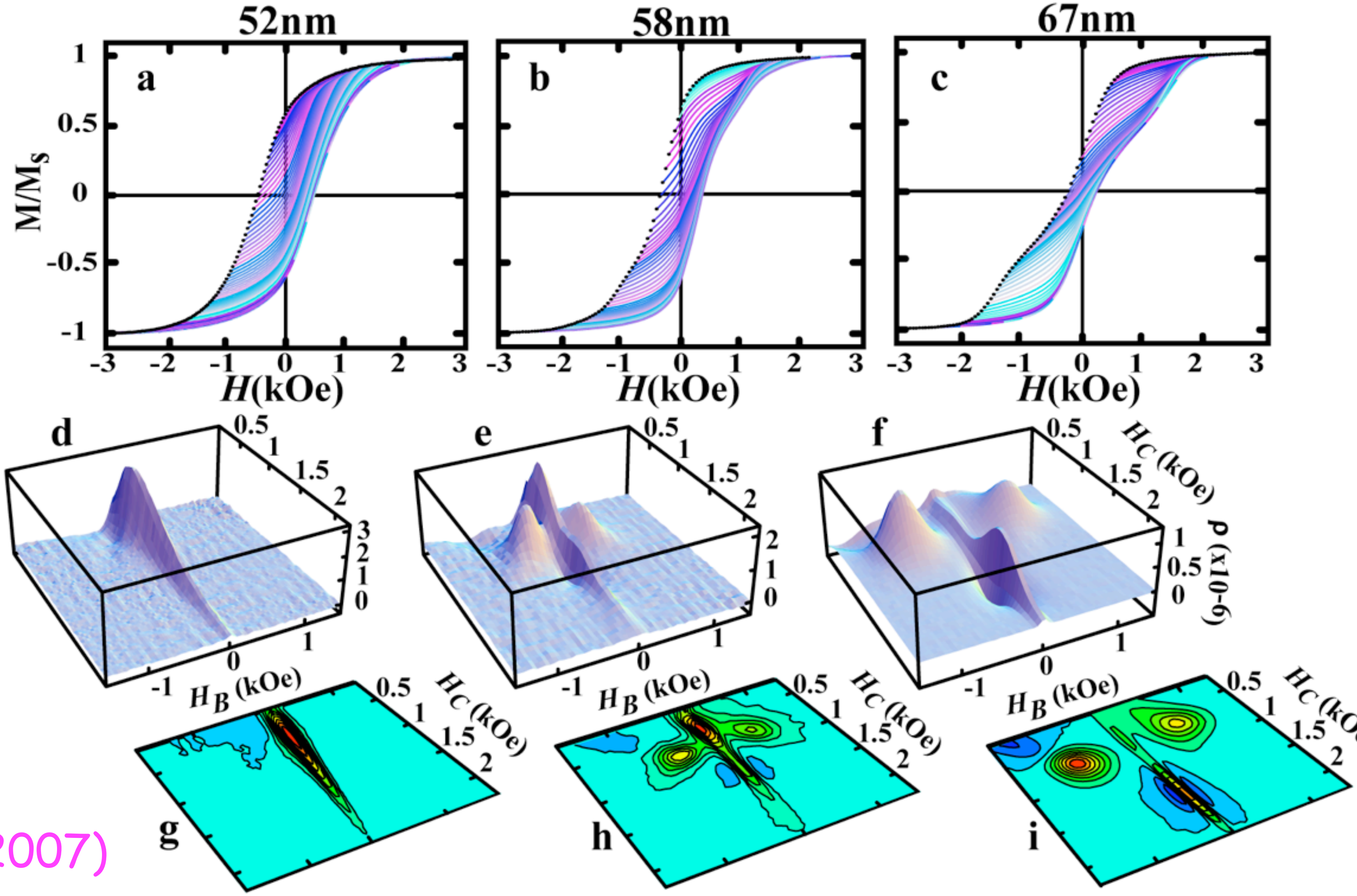
3. Magnetic Measurement …

General aspects

FORC: First Order Reversal Curve

- Example: nanodot array of different dot diameter. Distinctly different reversal mechanisms, despite only subtle differences in the major hysteresis loops.

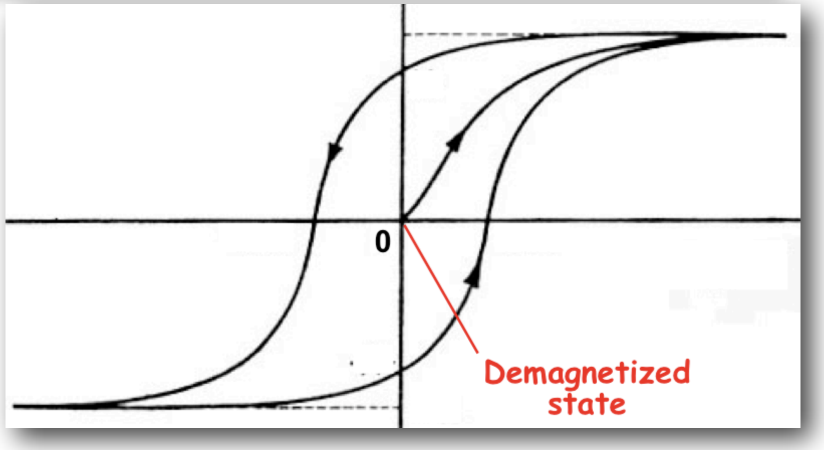




Courtesy Kai Liu, Davis
Phys. Ref. B 75, 134405 (2007)

3. Magnetic Measurement …

General aspects



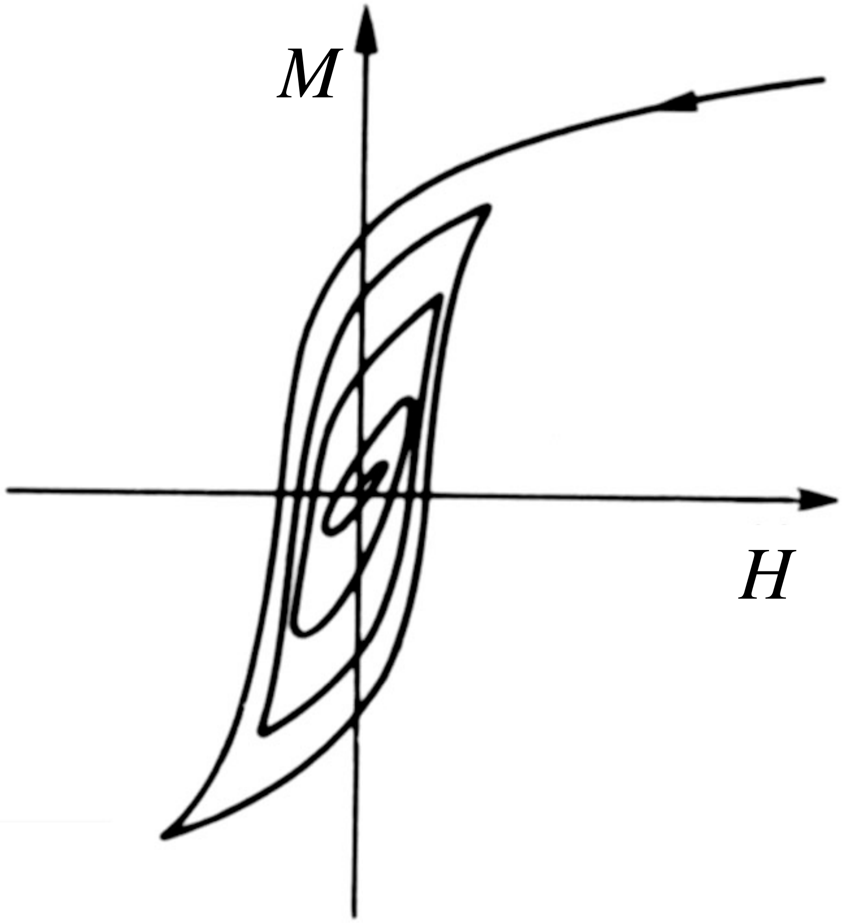
Demagnetiziation

3 possibilities, to „demagnetize" a magnet ($\overline{M} = 0$) :

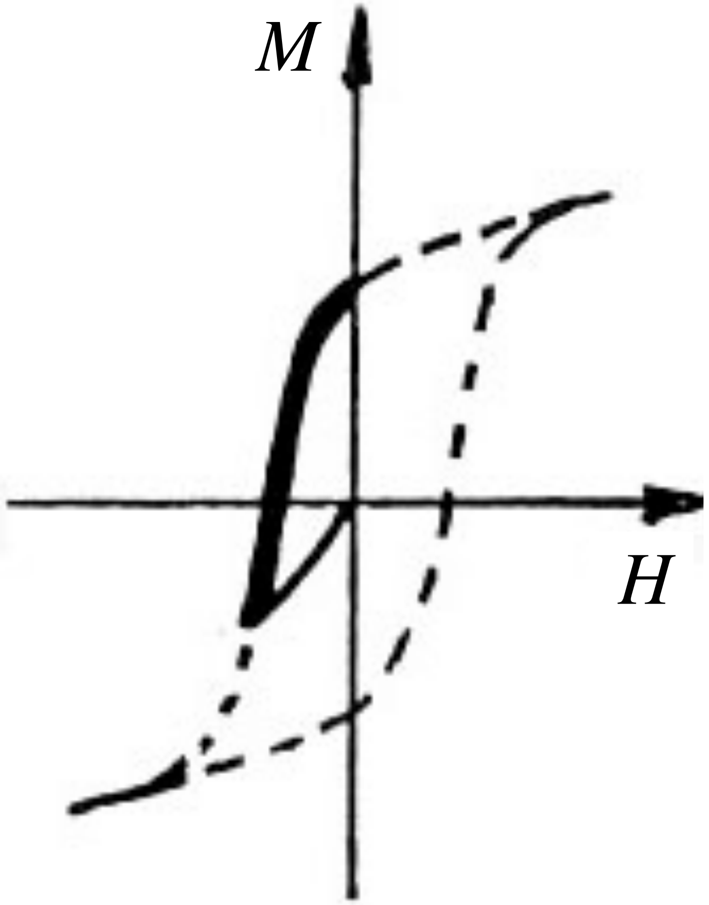
1. Thermal demagnetization

Heating above Curie temperature and cooling in absence of magnetic field

2. Cyclic (ac) demagnetization



2. dc-field demagnetization



In principle any point within or on the hysteresis loop can be obtained by choosing the right field history

3.

Measurements to determine magnetic material parameters & properties

3.1 Magnetic measurements

3.2 Mechanical measurements

3.3 Resonance techniques

3.4 Dilatometric measurements

3.5 Domain methods

3.

Measurements to determine magnetic material parameters & properties

3.1 Magnetic measurements

3.2 Mechanical measurements

3.3 Resonance techniques

3.4 Dilatometric measurements

3.5 Domain methods

3.1 Magnetic Measurements

a) Inductive methods

Sample surrounded by coil, in which voltage is induced when magnetization of sample is changed or when sample is moved. Voltage is integrated → signal proportional to magnetization

b) Magnetometric methods

For finite samples: demagnetizing field, which is proportional to mean magnetization ($H_{\text{dem}} = -N\overline{M}$), is measured

c) Optical magnetometry

Surface magnetization is measured by magneto-optic effect, useful for thin films where signal of inductive or magnetometric methods are too weak

3.1 Magnetic Measurements

a) Inductive methods

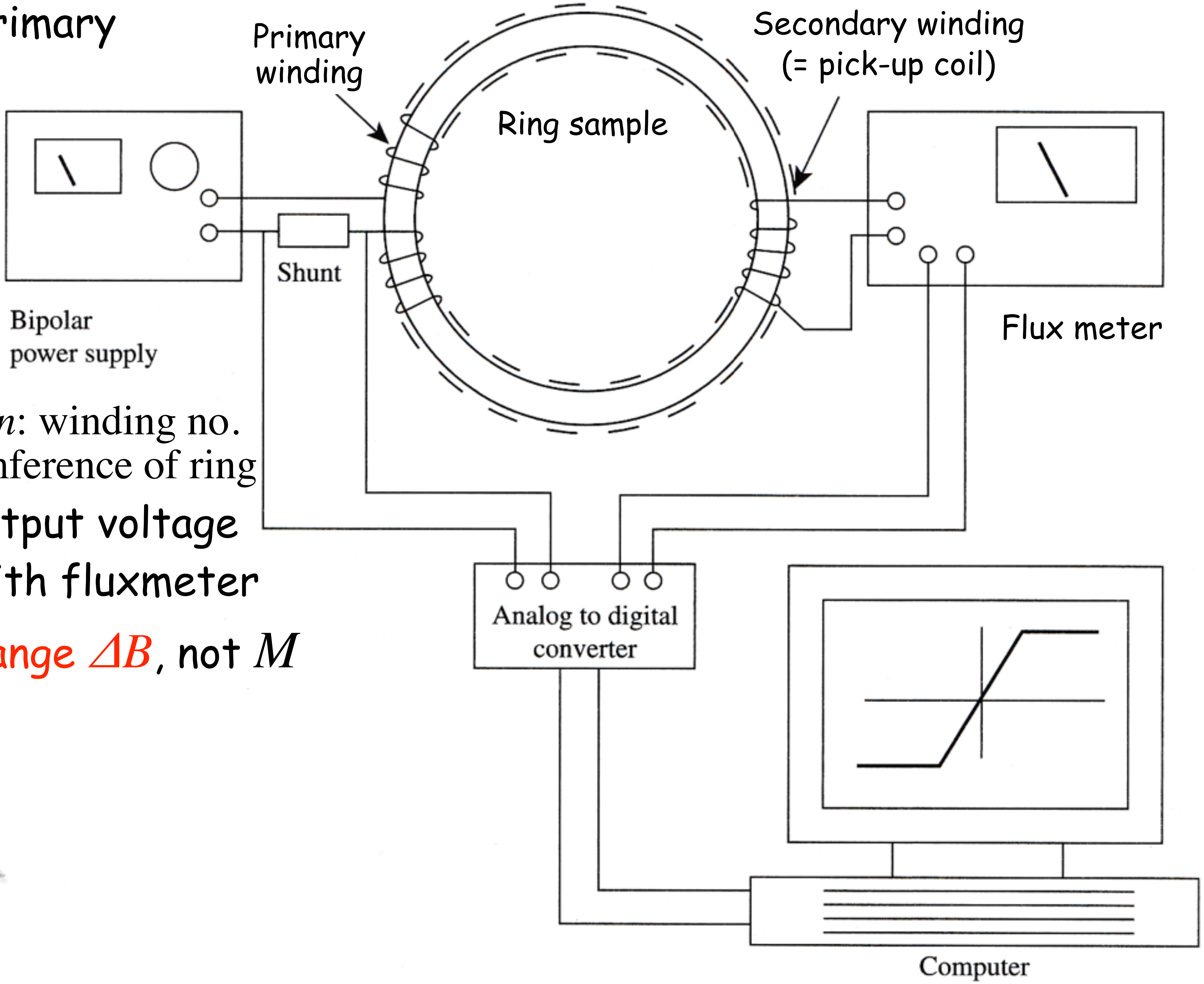
Measurement of closed circuit samples

- Conventional setup with primary and secondary winding
- Procedure: vary current through primary winding and measure its magnitude by voltage drop across shunt:

  $$H = nI/L_{Fe}$$

  I: current, n: winding no.
  L_{Fe}: circumference of ring

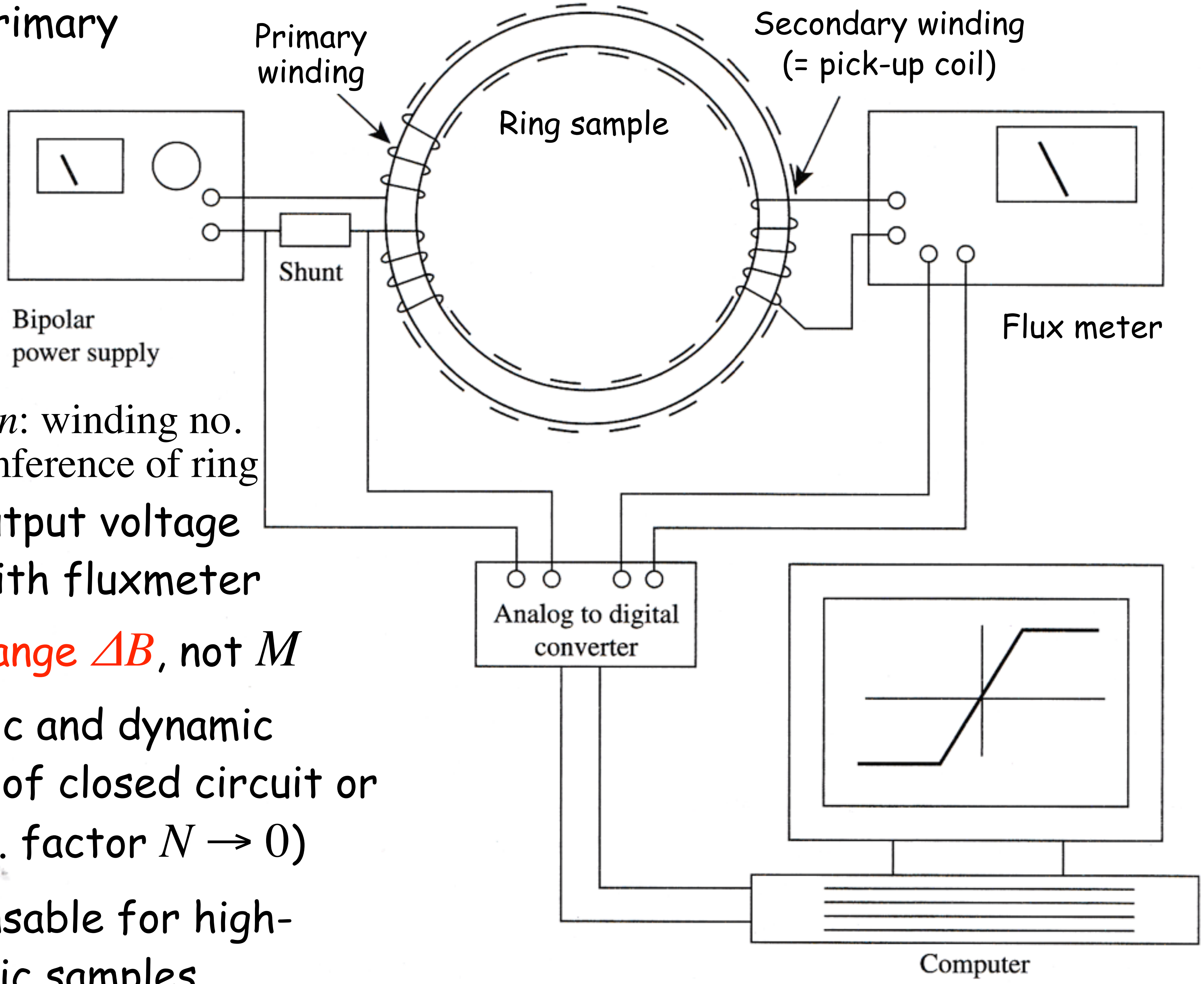
  simultanously integrate output voltage from secondary winding with fluxmeter
- Measures flux density change ΔB, not M



3.1 Magnetic Measurements

a) Inductive methods

Measurement of closed circuit samples



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  $$H = nI/L_{Fe}$$

  I: current, n: winding no.
  L_{Fe}: circumference of ring

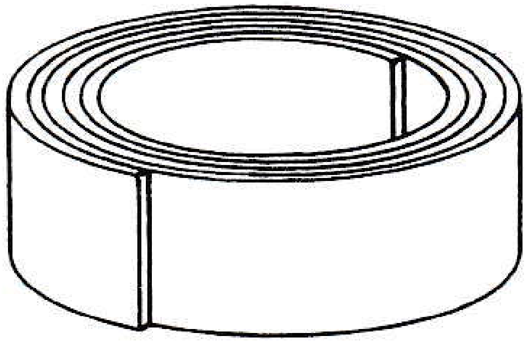
  simultanously integrate output voltage from secondary winding with fluxmeter
- Measures flux density change ΔB, not M
- Can be used for quasistatic and dynamic hysteresis measurements of closed circuit or elongated samples (demag. factor $N \rightarrow 0$)
- Such geometry is indispensable for high-permeability, soft magnetic samples

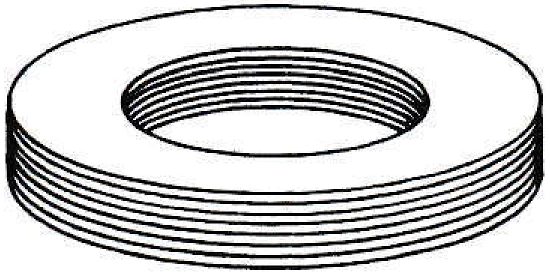
3.1 Magnetic Measurements

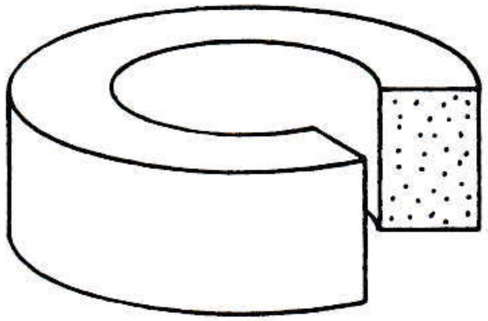
a) Inductive methods

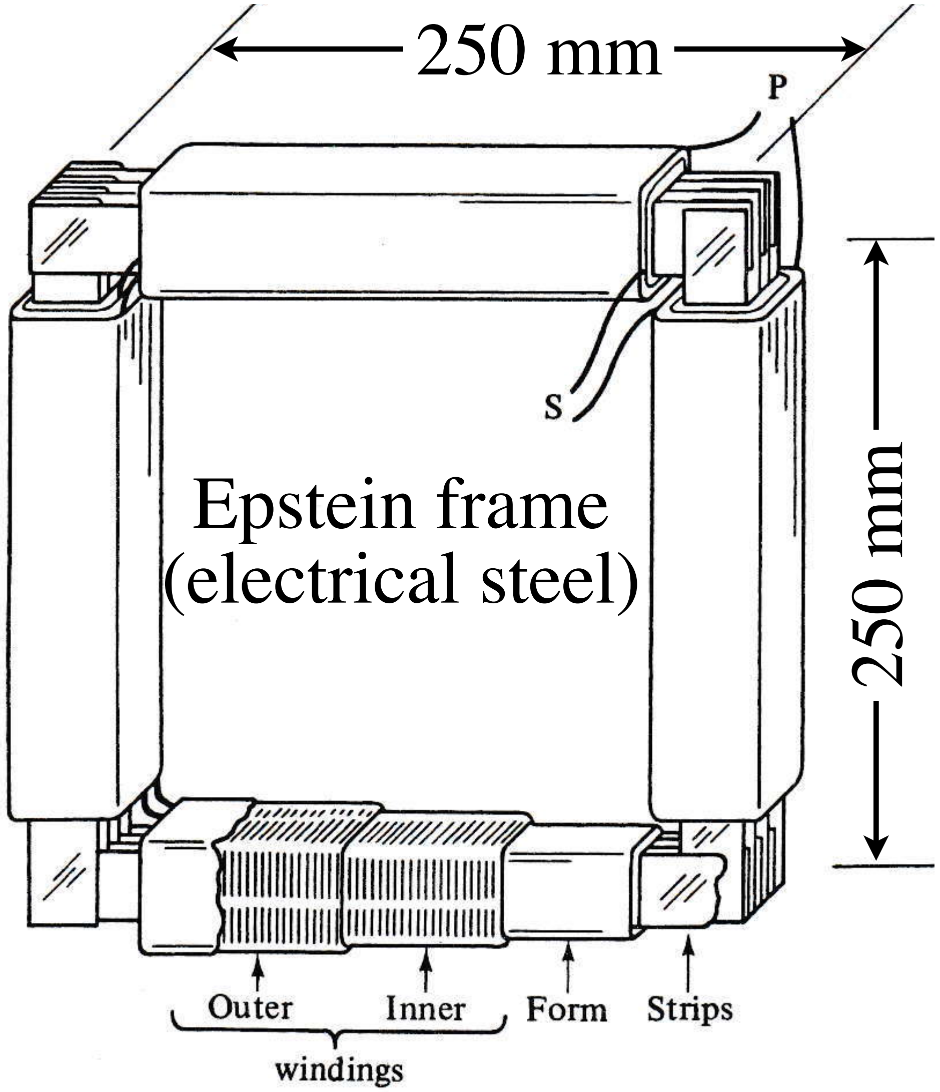
Measurement of closed circuit samples

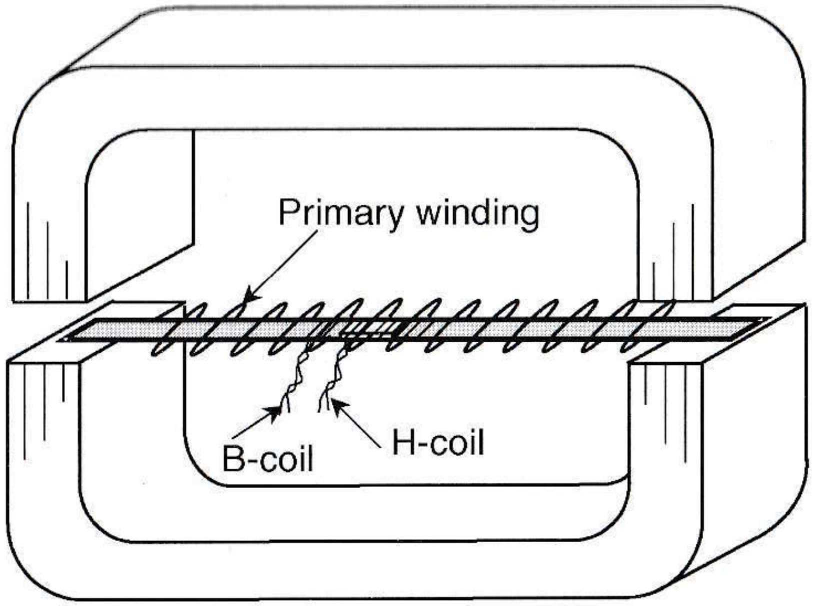
- Examples of closed samples

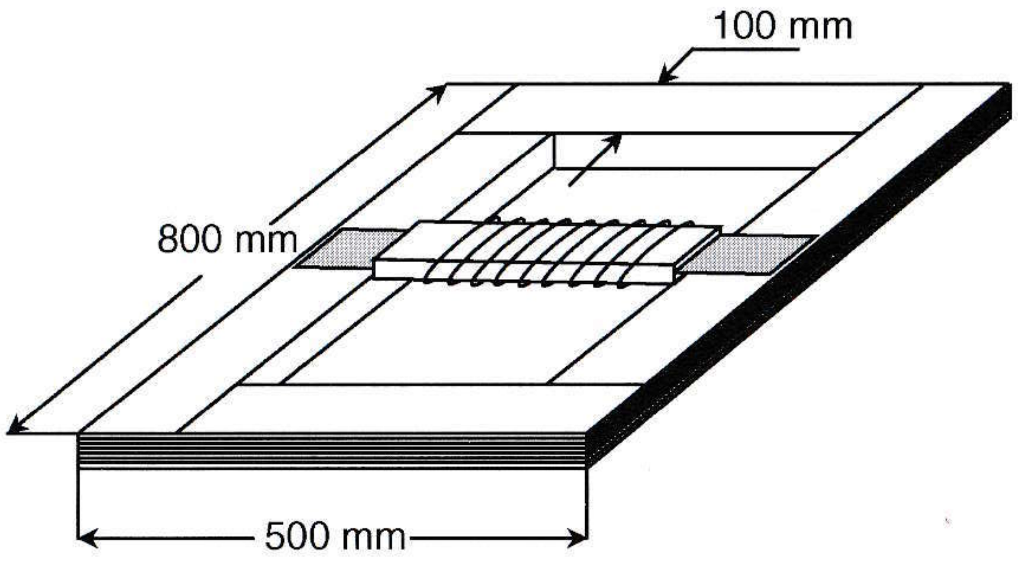

Tape-wound core


Stacked lamination


Powder core


Epstein frame (electrical steel)


Laminated yoke for sheet samples (permeameter)


Single sheet tester

3.1 Magnetic Measurements

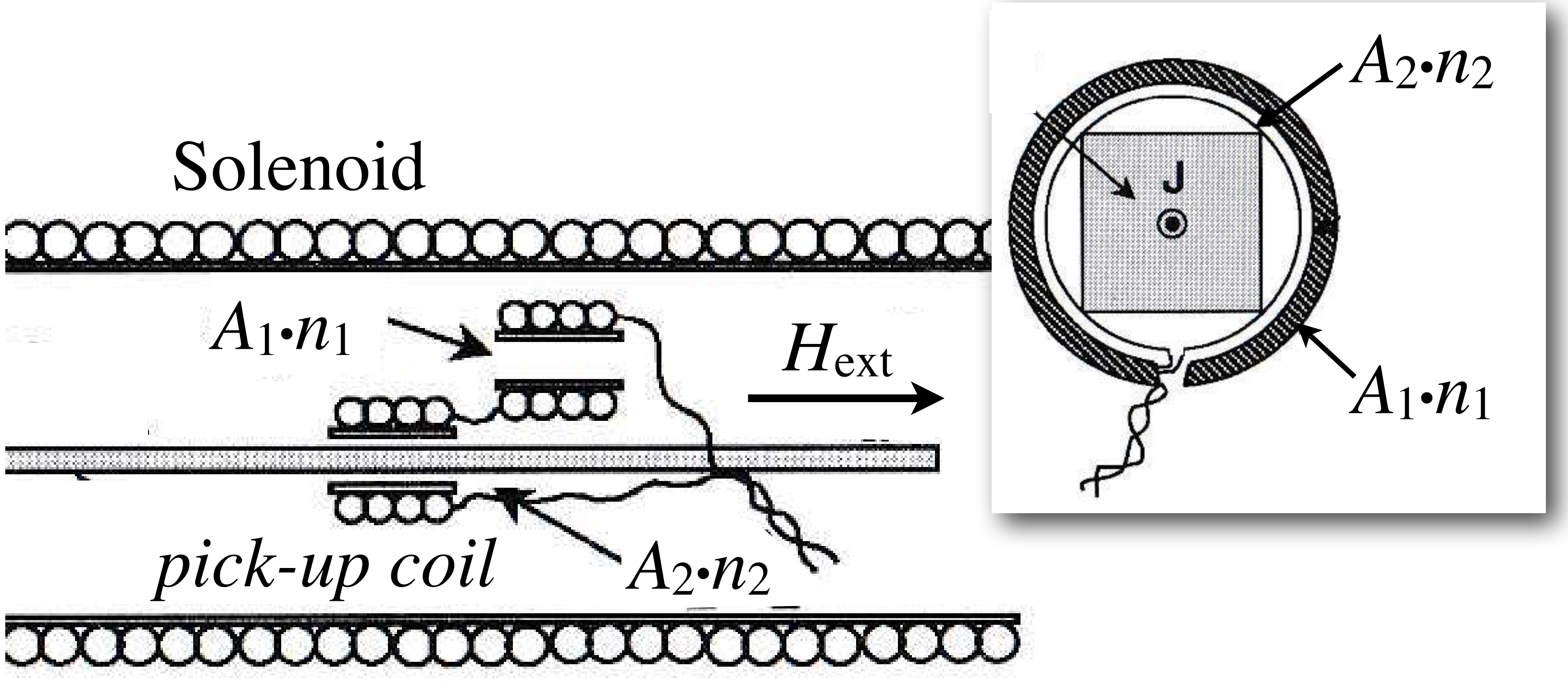
a) Inductive methods

Air flux compensation

- $B = \mu_0 H + J \longrightarrow$

  If $J(H)$ is to be measured instead of $B(H)$: Air flux compensation required to subtract effect of applied field

  - Compensation coil arrangement: 2 coils with equal winding areas $A_1 \cdot n_1 = A_2 \cdot n_2$ ($A_{1,2}$: cross sectional area of coils, $n_{1,2}$: number of turns) are connected electrically in opposition → difference signal is proportional to magnetization alone, i.e. without specimen in pick-up coil: no flux recorded by fluxmeter, with specimen in pick-up coil: fluxmeter reads ΔJ
  - Two possibilities: (i) two identical coils arranged side by side; (ii) external layers of the winding of a coil can be connected in such a way that their winding area is equal and opposite to the core winding area

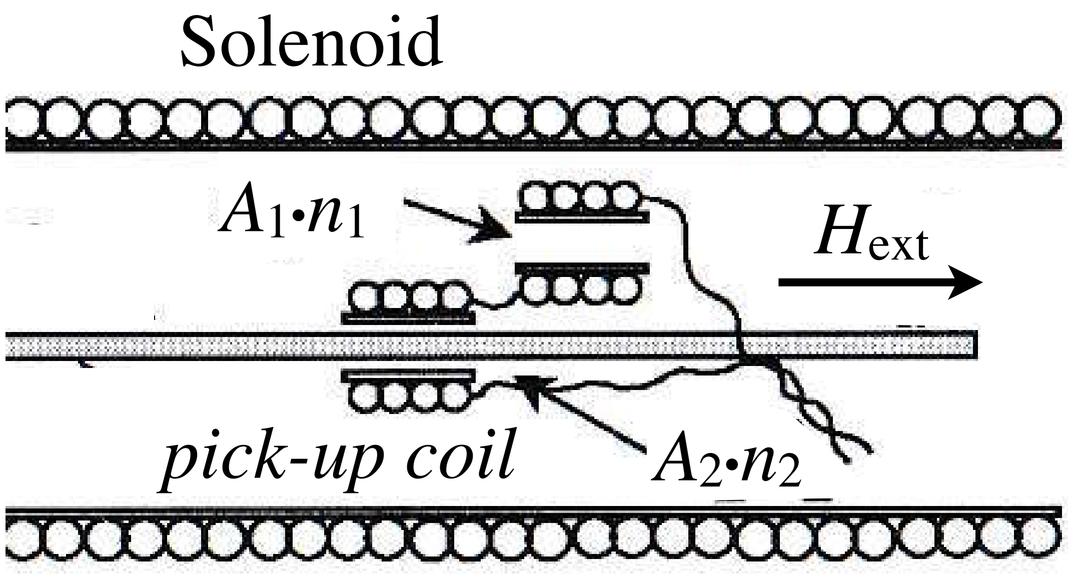


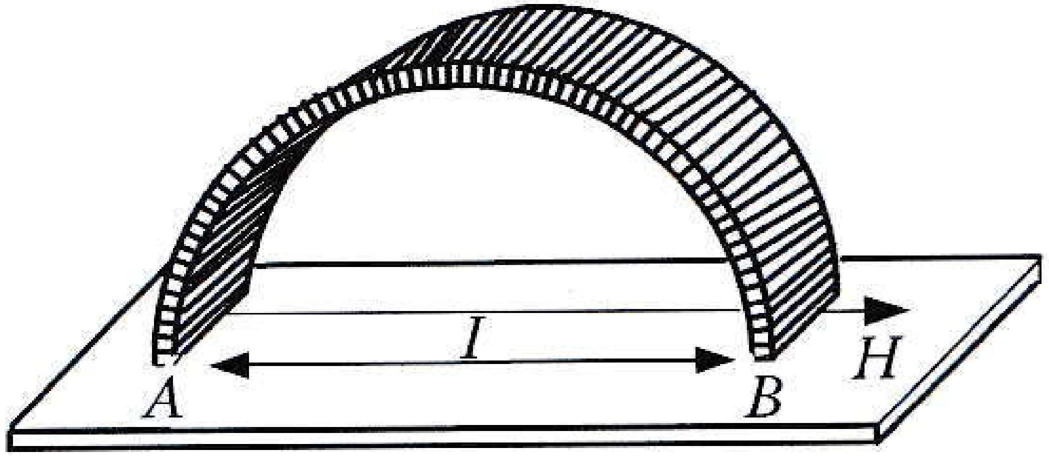
3.1 Magnetic Measurements

a) Inductive methods

Air flux compensation

- Measurement of field strength:
  - Signal induced in compensation coil can also be used to measure the effective internal field
  - Maxwell's equations: tangential component of magnetic field must be equal on both sides of sample surface (if no current is flowing in sample surface)
  - → A coil placed close to the surface may therefore measure the internal field
  - With this technique the unsheared magnetization curve can be measured even for short samples
  - Alternative: Rogowski-Chattok coil





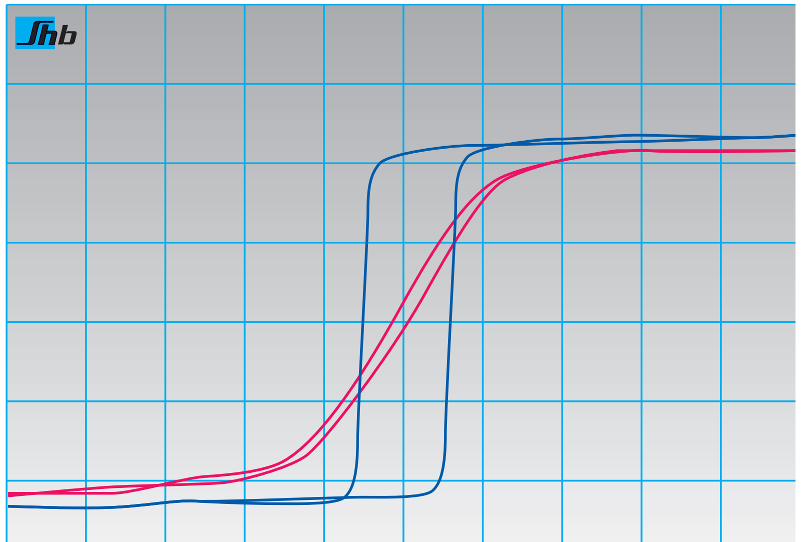
3.1 Magnetic Measurements

a) Inductive methods

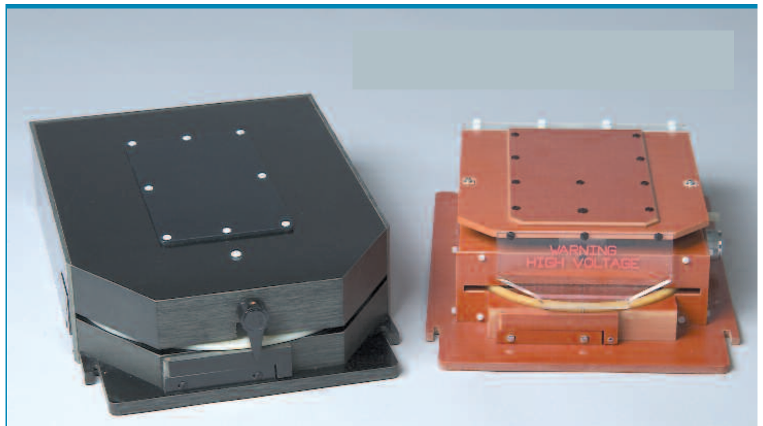
Loop tracer for magnetic films

- Highly sensitive commercial instrument to inductively measure M(H) loops in soft magnetic films
- Helmholtz coils: field up to 100 mT
- Frequency 1 - 10 Hz
- Alternative to Vibrating Sample Magnetometer

MESA
http://www.shbinstruments.com



Soft magnetic film, 0.3 nm thick, diameter 18 mm



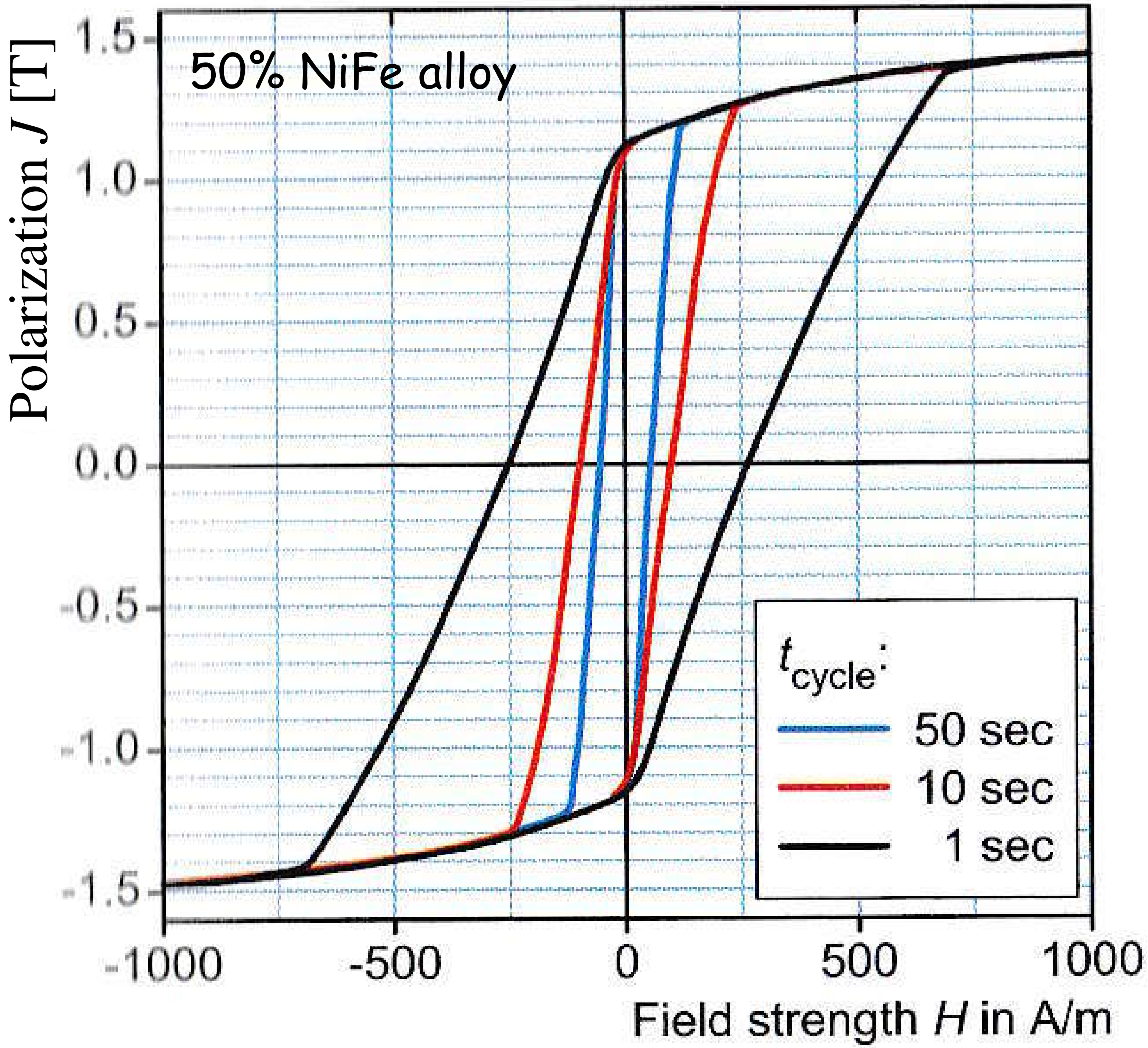
Pick-up assemblies

3.1 Magnetic Measurements

a) Inductive methods

(Quasi-)static magnetization

- Quasistatic $M(H)$ measurement: rate of magnetization $\mathrm{d}B/\mathrm{d}t$ and time for complete hysteresis cycle have to be low enough to eliminate all dynamic effects (like eddy currents, relaxaton processes etc.)
- Quasistatic loop is narrower than AC loop;
- Quasistatic coercivity is always lower than dynamic coercivity

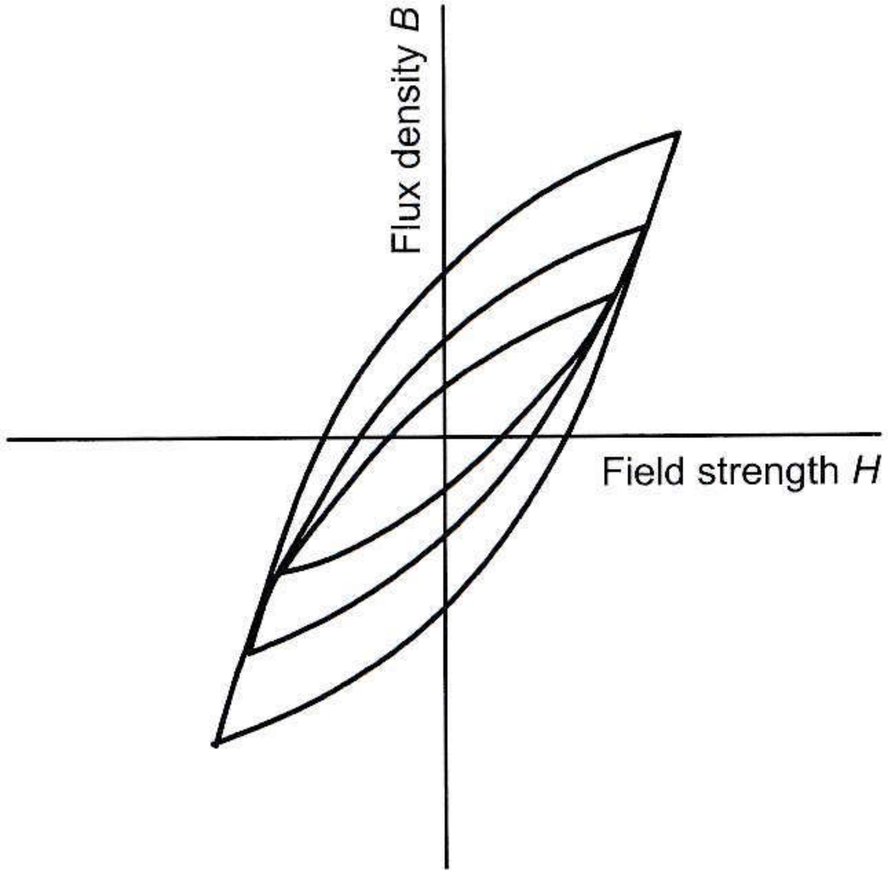


3.1 Magnetic Measurements

a) Inductive methods

Dynamic magnetization

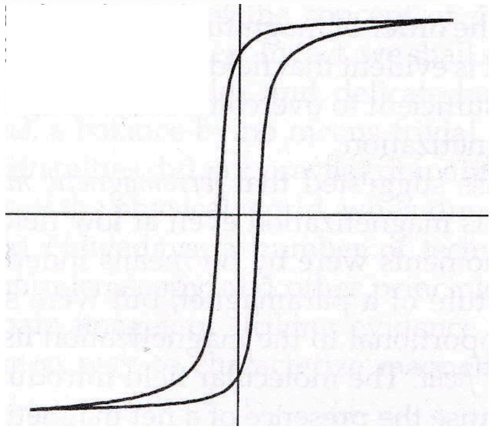
- **AC magnetization at low excitation level:**
  - Field amplitude below coercive field (Rayleigh region)
  - Relationship between AC flux density and AC field strength can be represented by ratio factor (permeability) and phase angle (magnetic loss angle due to eddy currents)
  - Can be described in terms of classical eddy current theory



3.1 Magnetic Measurements

a) Inductive methods

Dynamic magnetization



- **AC magnetization at high excitation level:**
  - AC excitation into region of maximum permeability and up to saturation
  - Severe distortions of magnetization due to non-linear behaviour of ferromagnetic material (S-like or rectangular hysteresis loops). Reason:
    - At high excitation level: material is subjected to rapid changes of $H(t)$ or $B(t)$
    - Local magn. flux density cannot follow changes due to eddy curents
    - Consequence: Magnetization curve depends on frequency and mode of excitation
  - 2 modes of dynamic magnetization:
    - **Voltage-controlled** magnetization: induced voltage is controlled to be sinusoidal (by feedback loop) → sinusoidal flux density $B(t)$ → magnetizing current $I(t)$ becomes dependent. Recommended as IEC standard
    - **Current-controlled** magnetization: magnetizing current is controlled to be sinusoidal (by high-impedence power source) → sinusoidal field strength $H(t)$ → $U(t)$ and $B(t)$ become dependent.

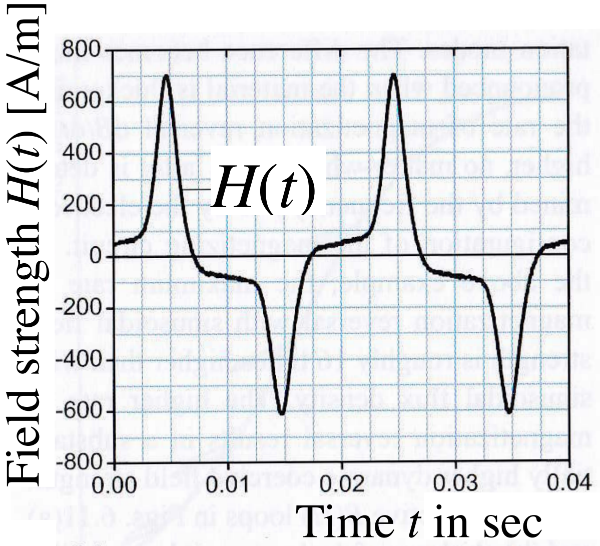
↓ cont.

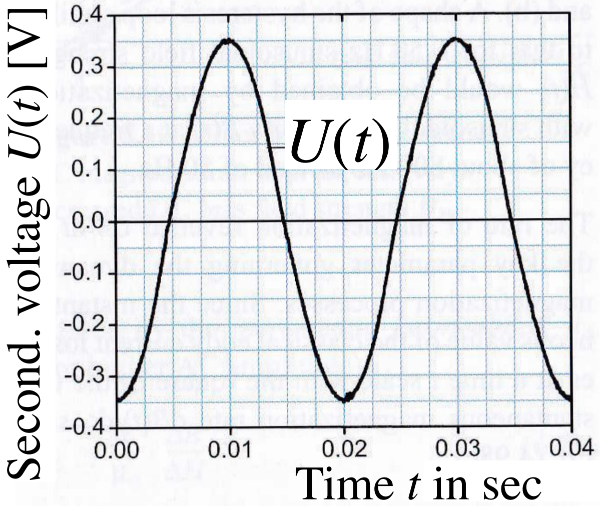
3.1 Magnetic Measurements

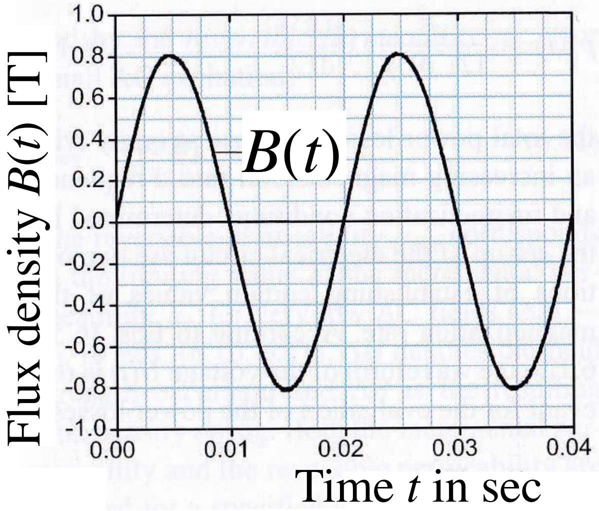
a) Inductive methods

high excitation level

Dynamic magnetization





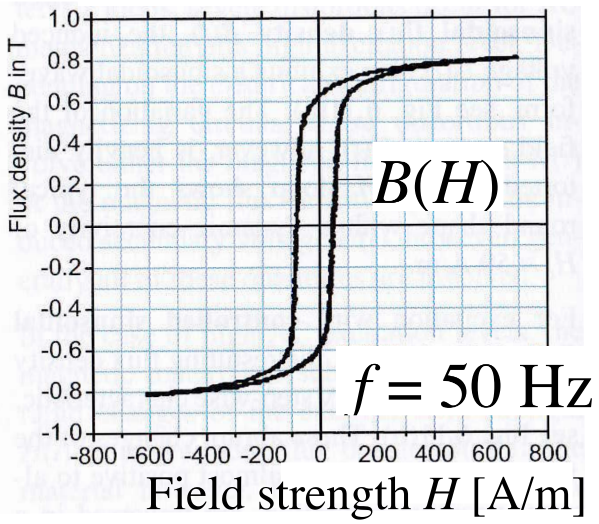


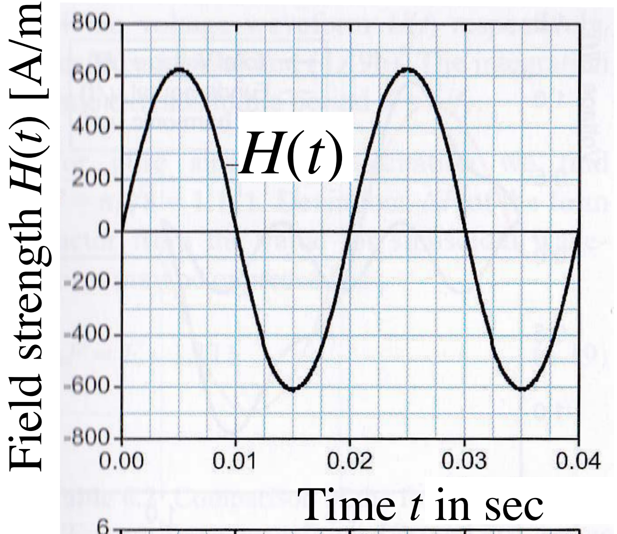
Controlled sinusoidal flux density $B(t)$

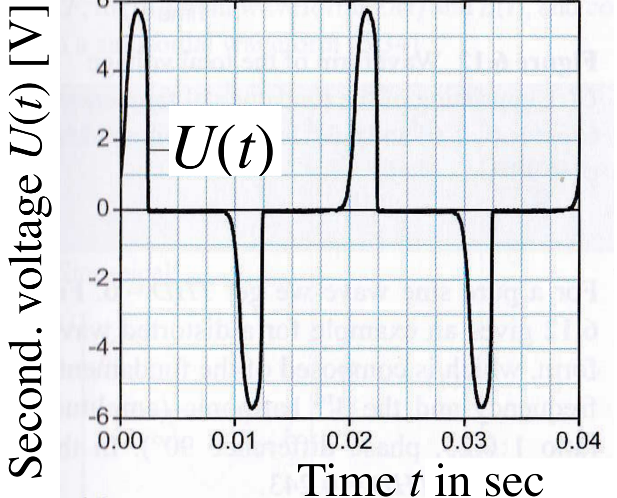
$$U(t) = (-)\, n_2 \cdot A_{\mathrm{Fe}} \frac{\mathrm{d}B}{\mathrm{d}t}$$

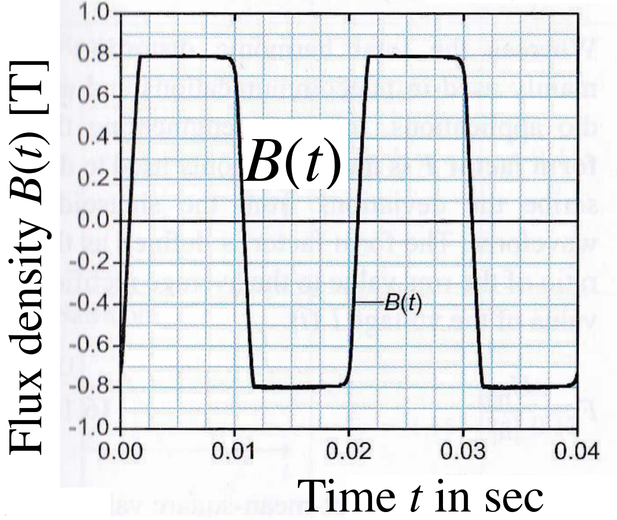
→ Induced voltage $U(t)$ also exhibits sinusoidal waveform

→ $H(t)$ heavily distorted







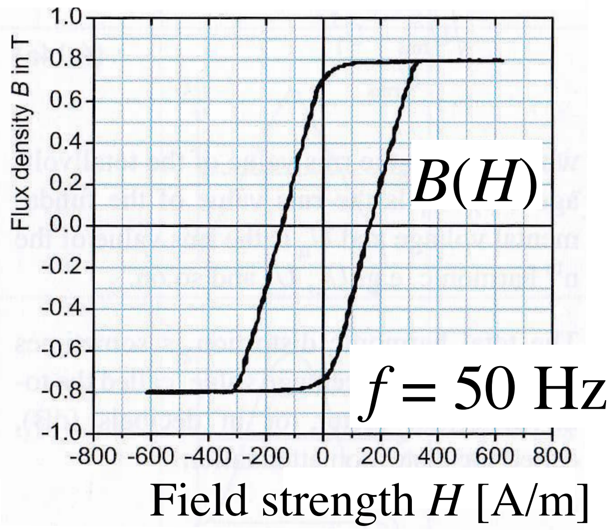


Controlled sinusoidal field strength $H(t)$

→ Step-wise characteristics of $B(t)$. Abrupt changes of $B(t)$ are traversed in short time interval

→ Spikes in induced voltage $U(t)$

→ Large eddy currents, sheared loop

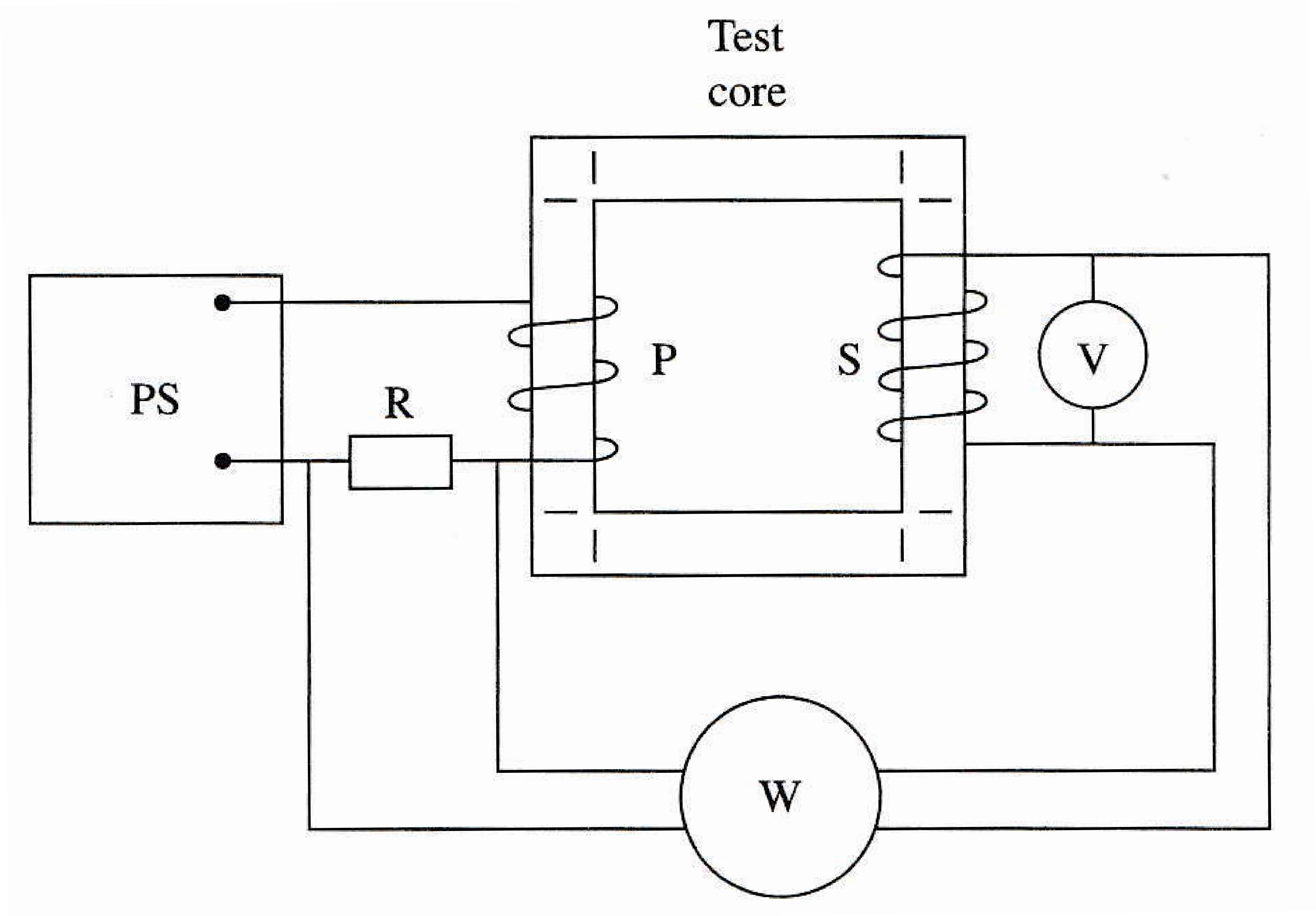


3.1 Magnetic Measurements

a) Inductive methods

Loss measurement

- Wattmeter method (e.g.)
  - Measures core loss (not copper loss)
  - Losses are measured at certain maximum flux density and given frequency
  - Total weight of sample is recorded, and losses are reported in W/kg

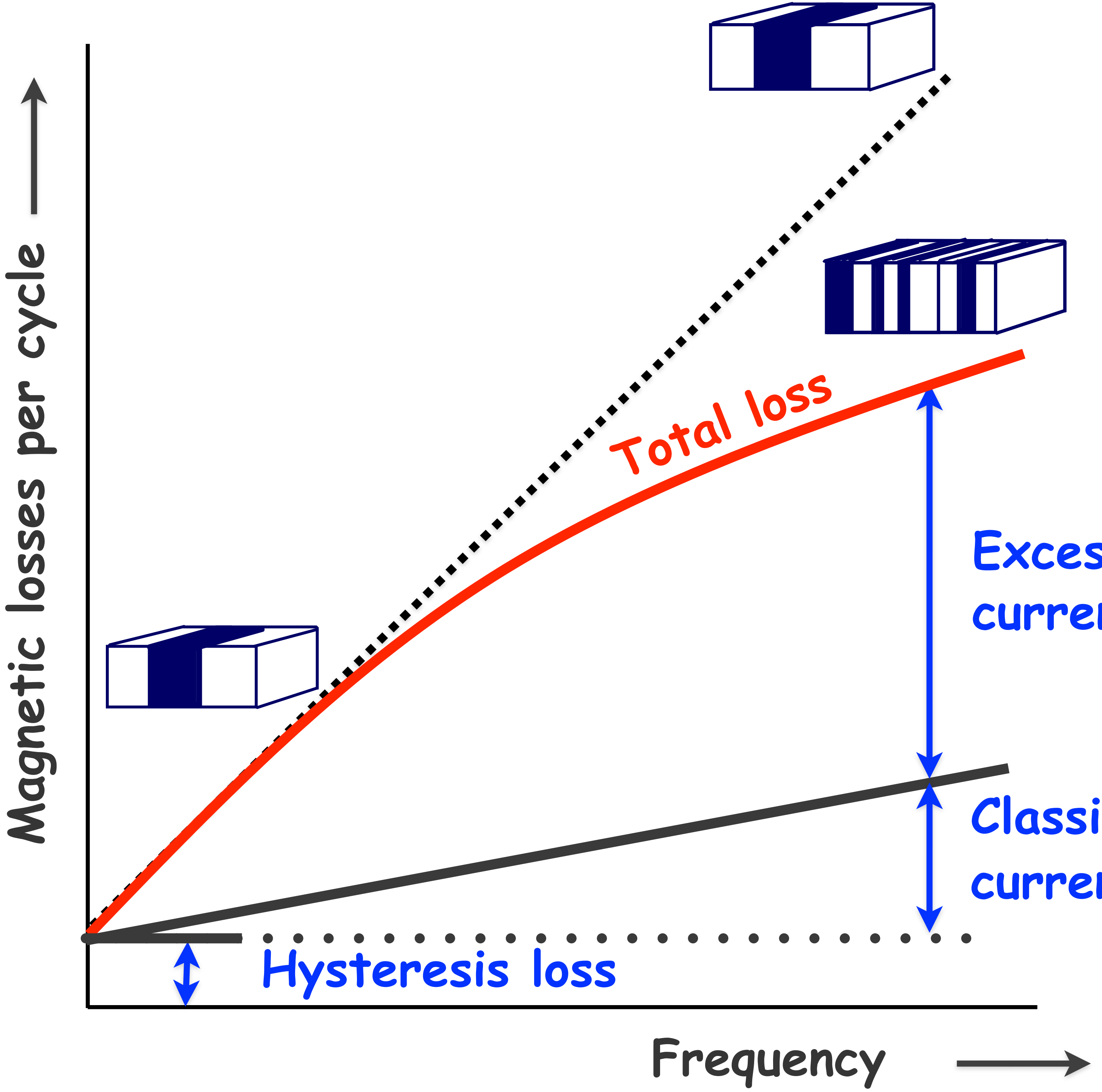


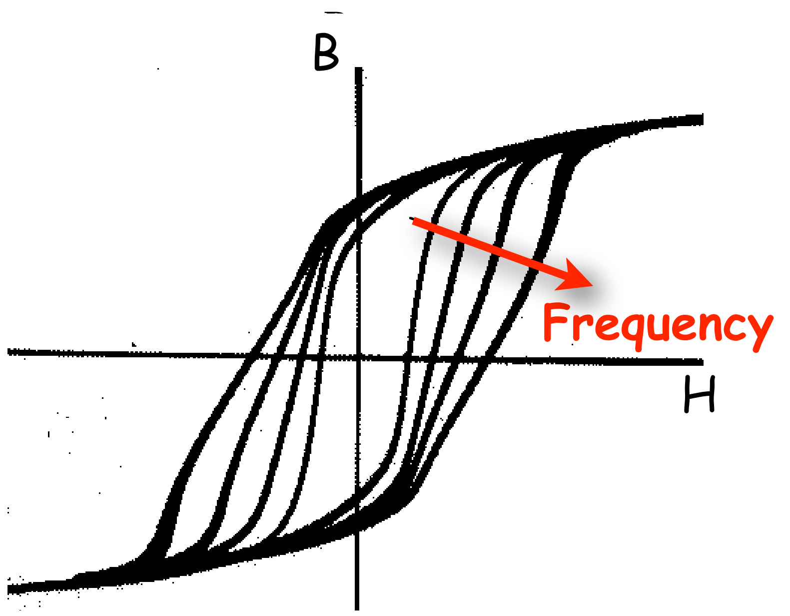
3.1 Magnetic Measurements

a) Inductive methods

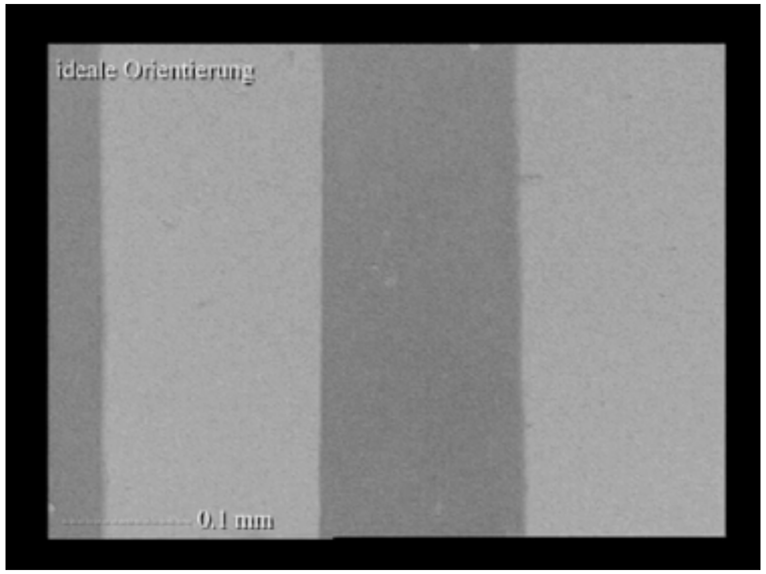
Loss measurement

- Energy loss





Hysteresis loops of typical material, measured at rising frequency



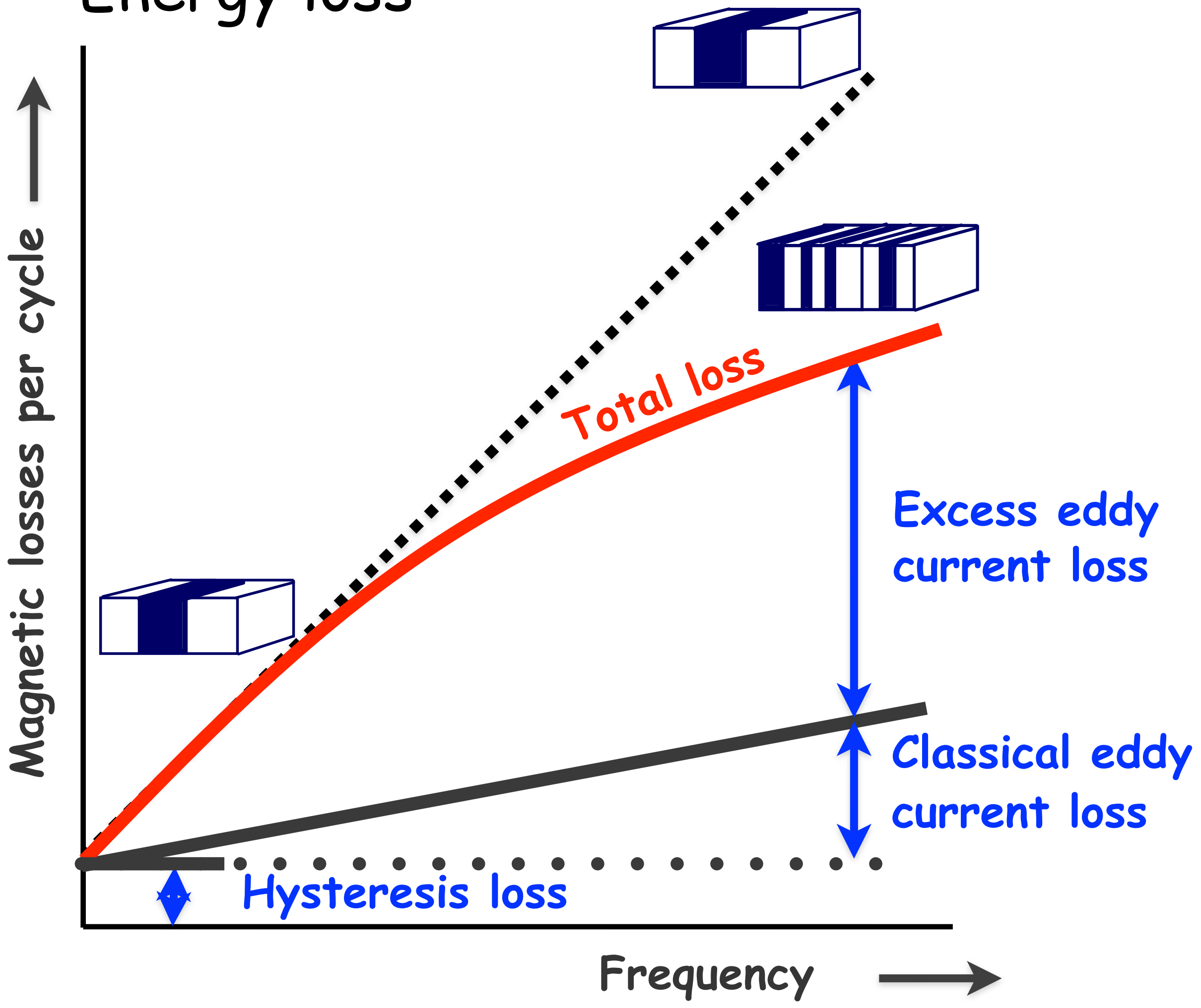
Transformer sheet

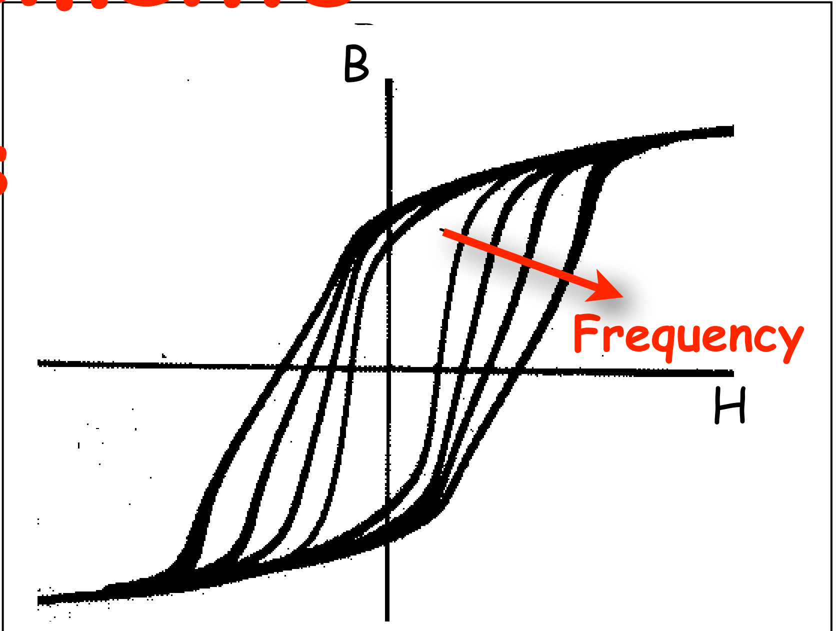
3.1 Magnetic Measurements

a) Inductive methods

Loss measurement

- Energy loss





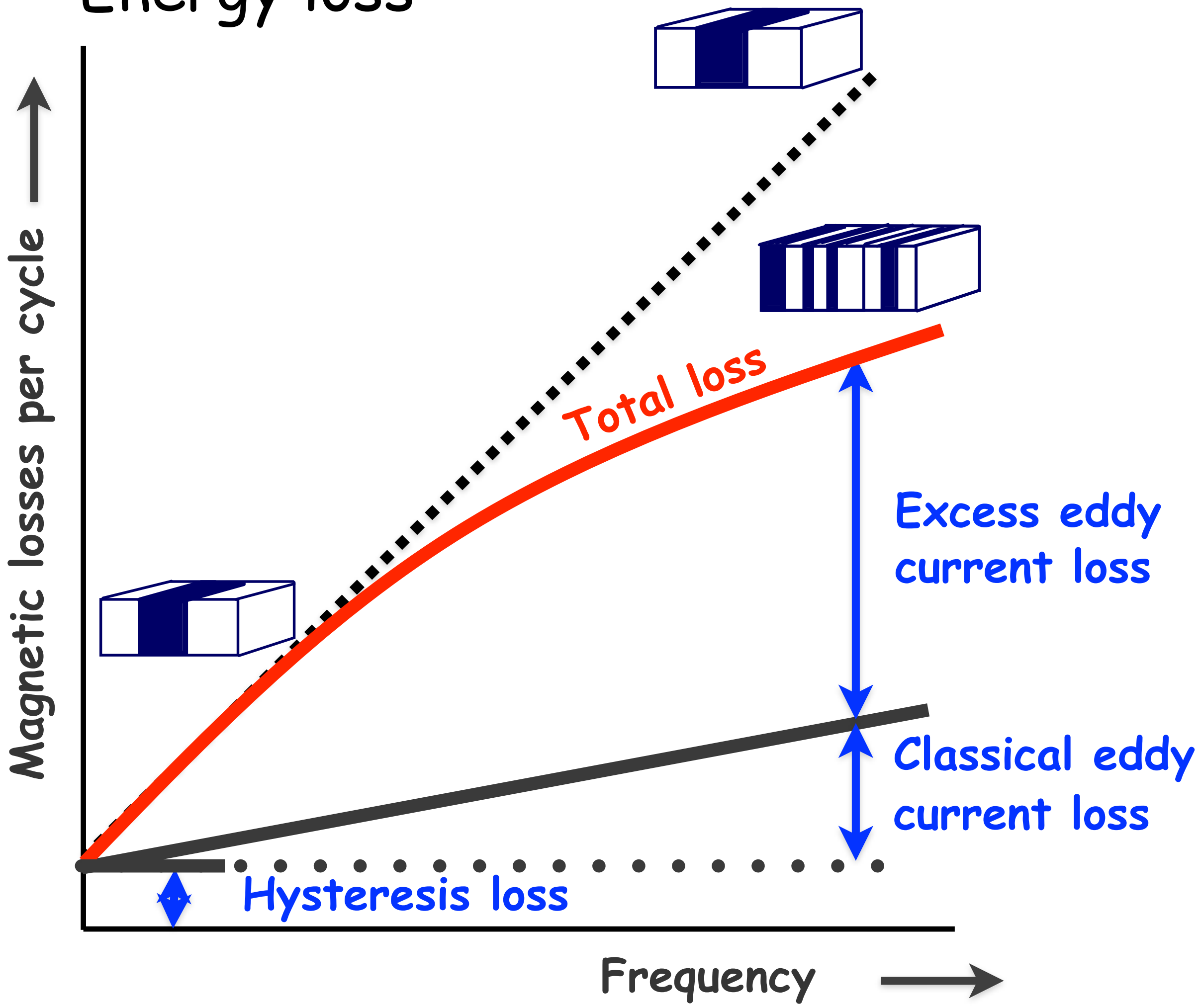
Hysteresis loops of typical material, measured at rising frequency

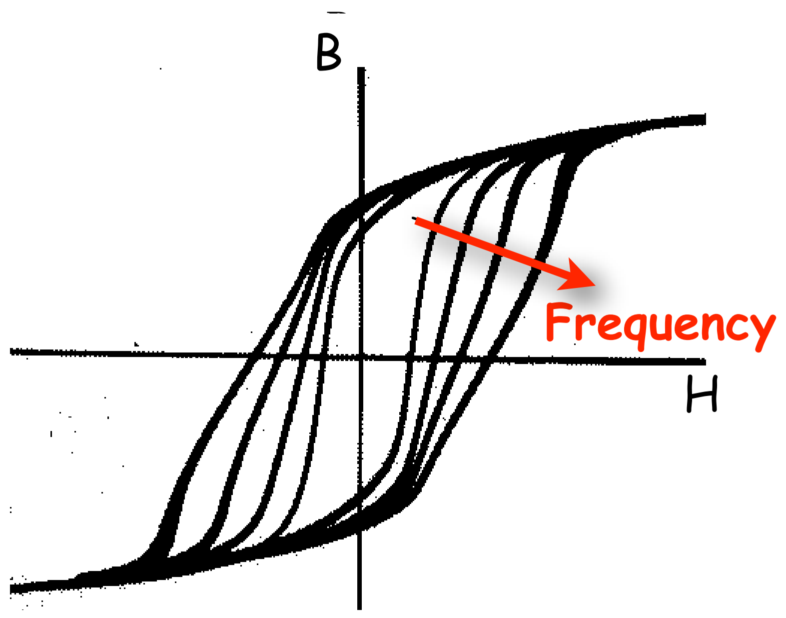
3.1 Magnetic Measurements

a) Inductive methods

Loss measurement

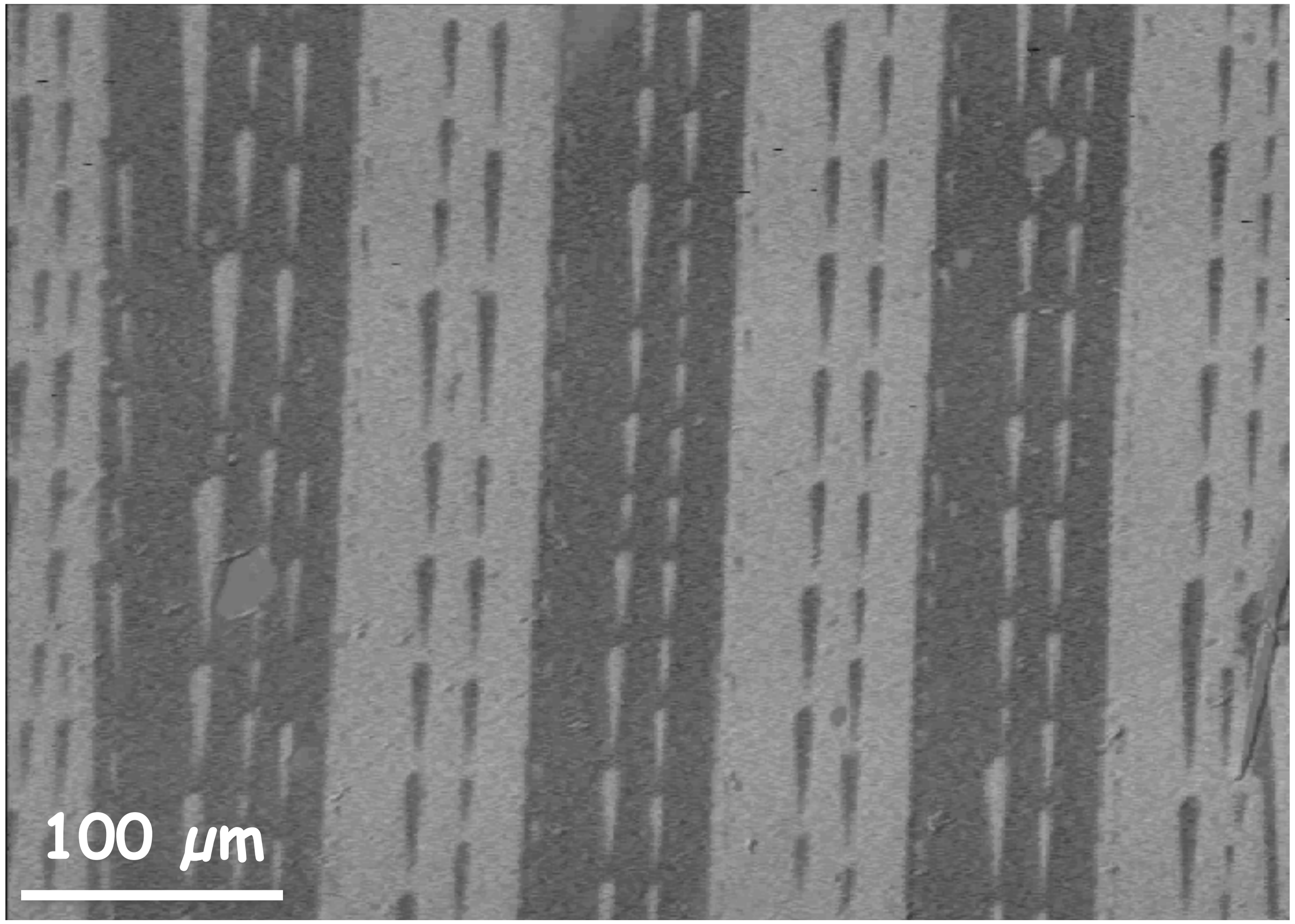
- Energy loss





Hysteresis loops of typical material, measured at rising frequency

Frequency increases 10 ⟶ 100 Hz



3.1 Magnetic Measurements

a) Inductive methods

Extraction Method

- Based on flux change in pick-up coil when sample is extracted from coil, or when specimen and pick-up coil together are extracted from field

  Total flux through pick-up coil:

  $$\Phi_1 = BA = \mu_0\,(H + M)\,A = \mu_0\,(H_{\text{applied}} - NM + M)\,A$$

  If sample is removed from pick-up coil, the flux through the coil becomes:

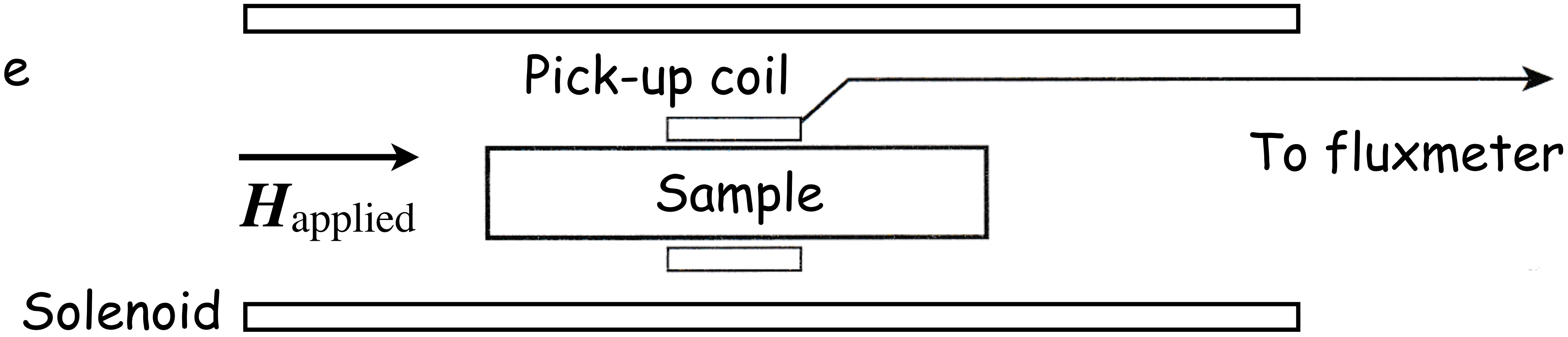
  $$\Phi_2 \;= \mu_0\,H_{\text{applied}}\,A \qquad A\text{: specimen or pick-up coil area}$$

  Fluxmeter will record a value proportional to flux change:

  $$\Phi_1 - \Phi_2 = \mu_0\,(1 - N)MA$$

- Extraction method measures M directly, rather than B

- Vibrating sample magnetometer: may be regarded as partial extraction method

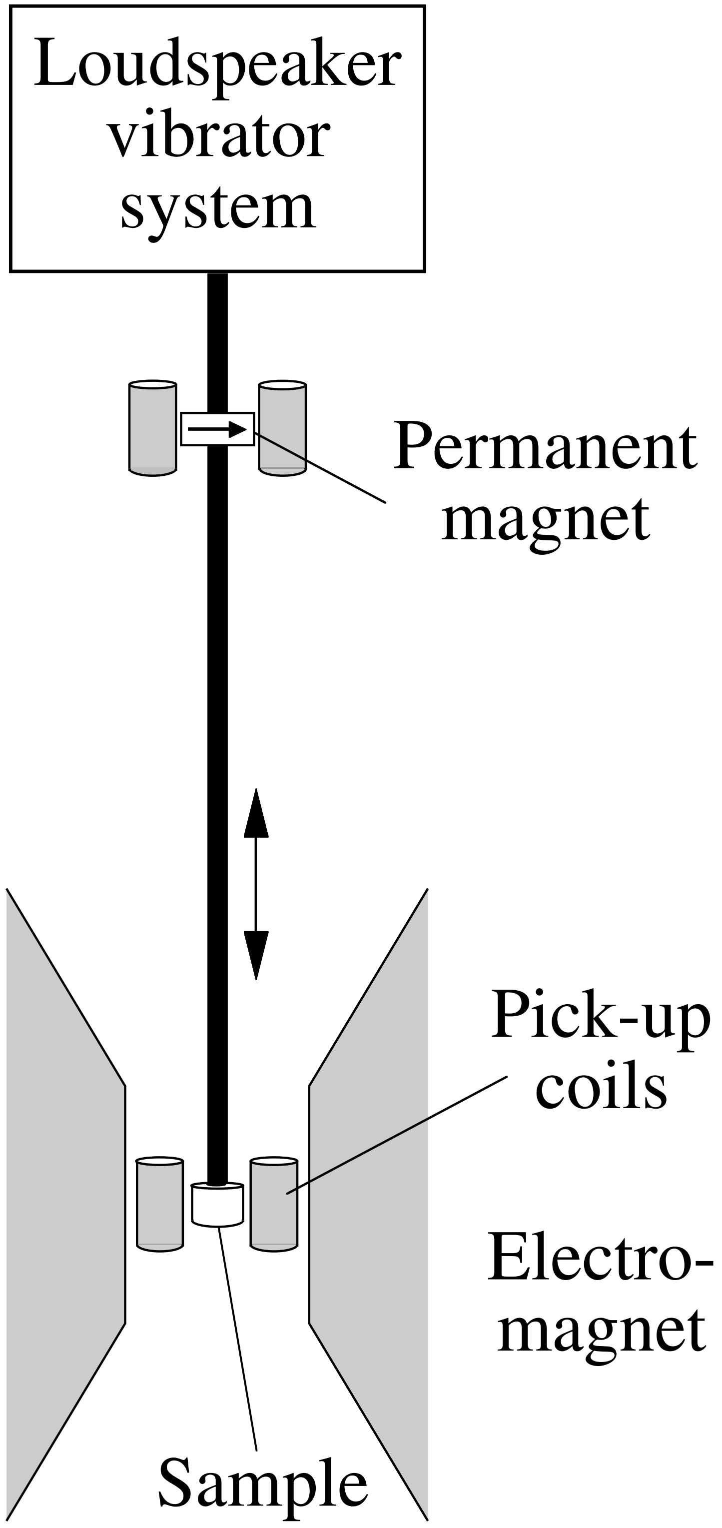


3.1 Magnetic Measurements

a) Inductive methods

Vibrating Sample Magnetometer (VSM)

- Sample placed inside magnet and vibrated perpendicular to field direction (frequ. ~ 100 Hz, vibration ampl. ~ 0.1 mm)
- Oscillating magnetization of moving sample → induces alternating voltage in pick-up coil, whose magnitude is proportional to M_{sample}: $U_{\mathrm{ind}} = const \cdot V \cdot M_{\mathrm{sample}}$ (V: sample volume)
- The pick-up signal is amplified with lock-in amplifier and compared with the signal induced in a pair of reference coils by a permanent magnet or by some variable capacitor setup (only sensitive to vibration frequency)
- Strong external fields can be applied (superconducting magnets for hard magnetic materials)
- Since sample is well-separated from the pick-up coils, it can be surrounded by cooling or heating devices
- The sample magnetization is static in the VSM, so that no eddy current effects have to be considered



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Loudspeaker

Courtesy Lakeshore

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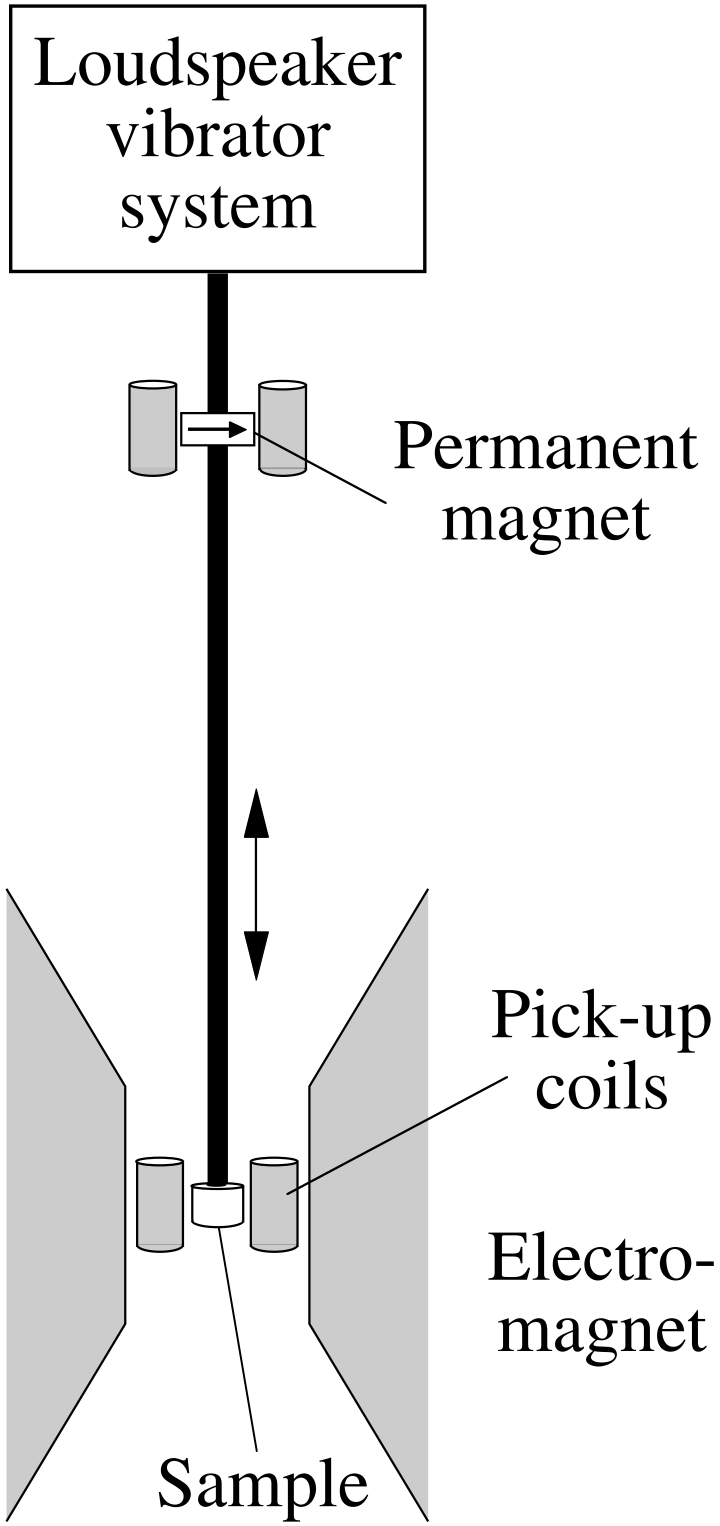

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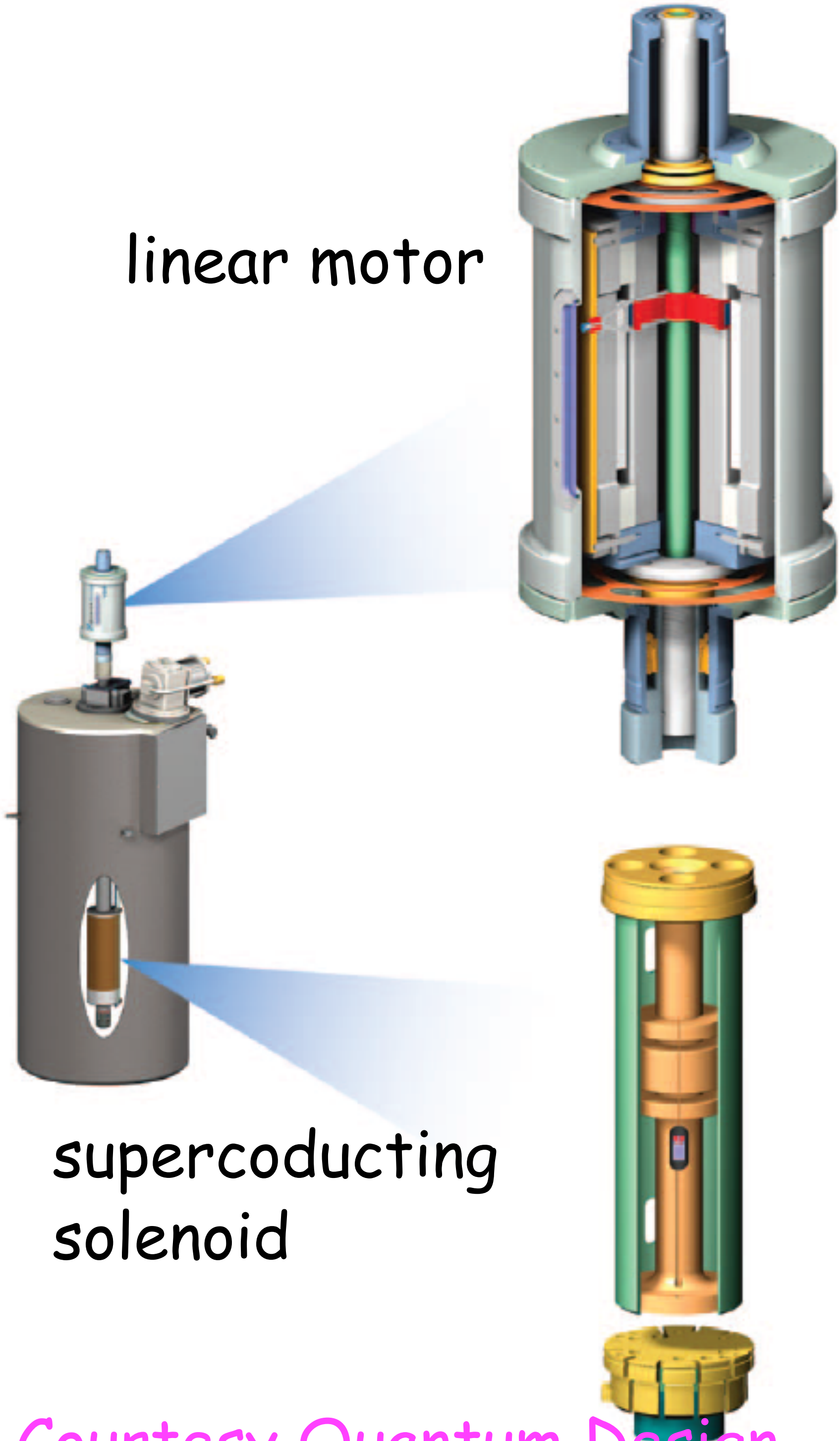


linear motor

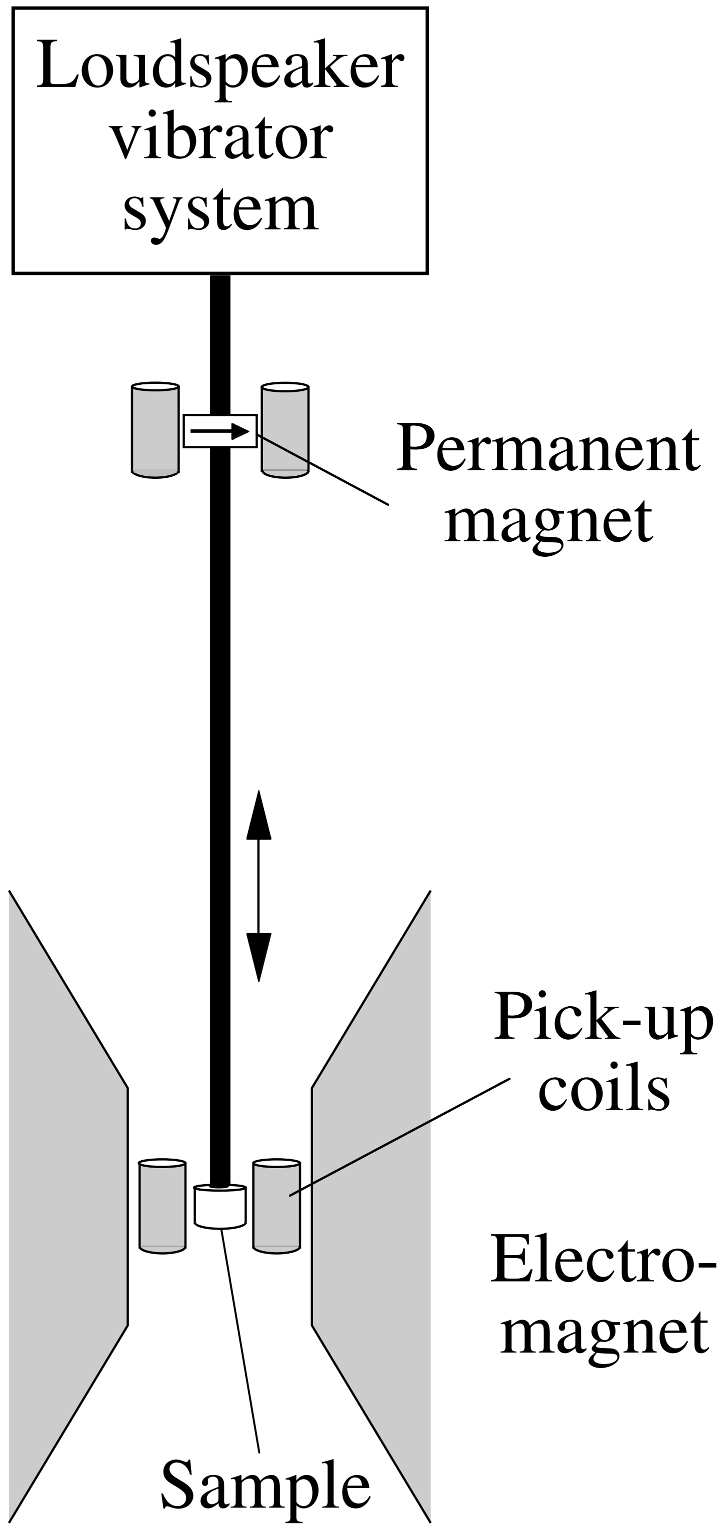
supercoducting solenoid

Courtesy Quantum Design.

3.1 Magnetic Measurements

a) Inductive methods

Vibrating Sample Magnetometer (VSM)



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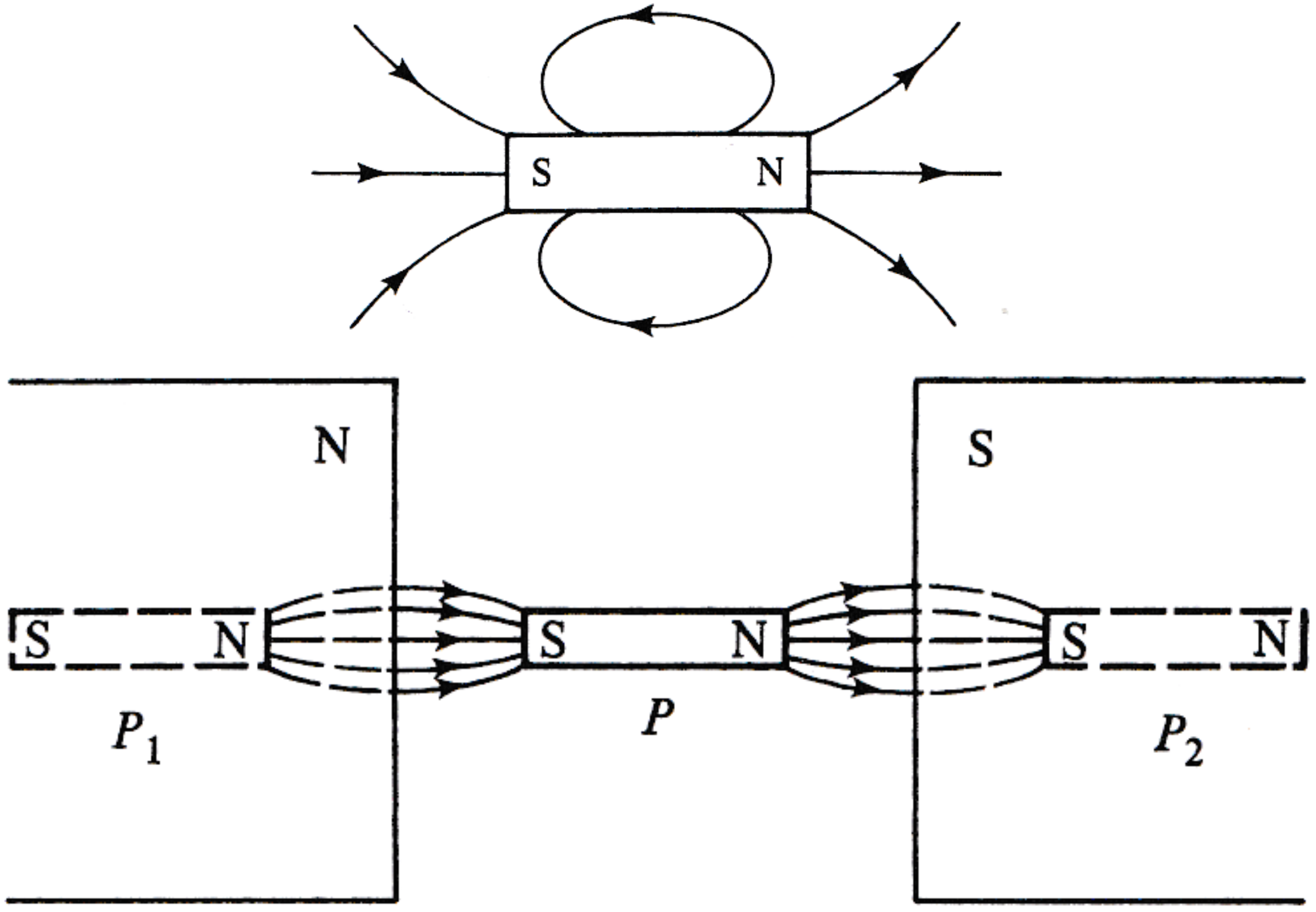
3.1 Magnetic Measurements

a) Inductive methods

Vibrating Sample Magnetometer (VSM)

- Direct method. Ideally, no calibration would be needed to derive the magnetization of the sample if pick-up coil geometry and sample volume are known and if magnetic field could be generated by a simple air coil
- However: for electromagnets the pole material interacts with the measuring process: "mirror images" of the sample are formed by the presence of soft magnetic iron yokes
- → Mirror images also induce voltage in pick-up coil
- Strength of mirror images depends on permeability of iron yoke that in turn depends on the induction level in magnet
- → VSM must be calibrated by replacing sample by Ni sample with accurately known saturation magnetization → determines *const* in $U_{\mathrm{ind}} = const \cdot V \cdot M_{\mathrm{sample}}$



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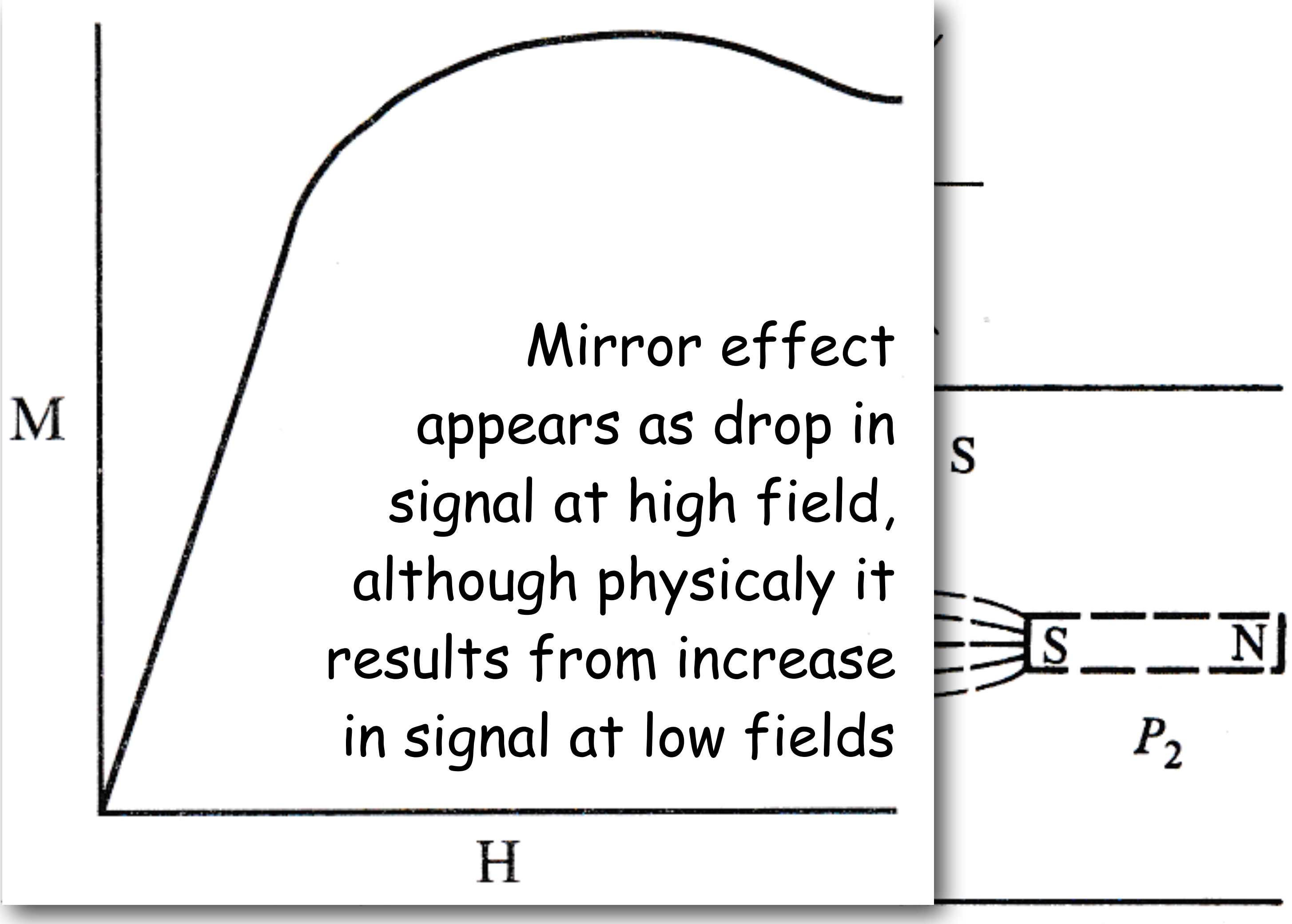

Mirror effect appears as drop in signal at high field, although physicaly it results from increase in signal at low fields

3.1 Magnetic Measurements

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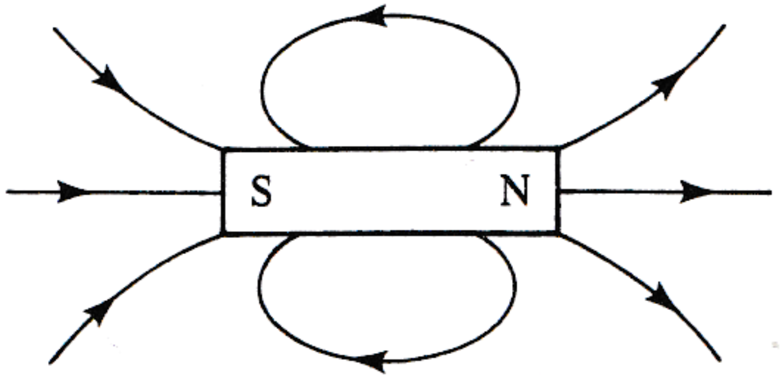
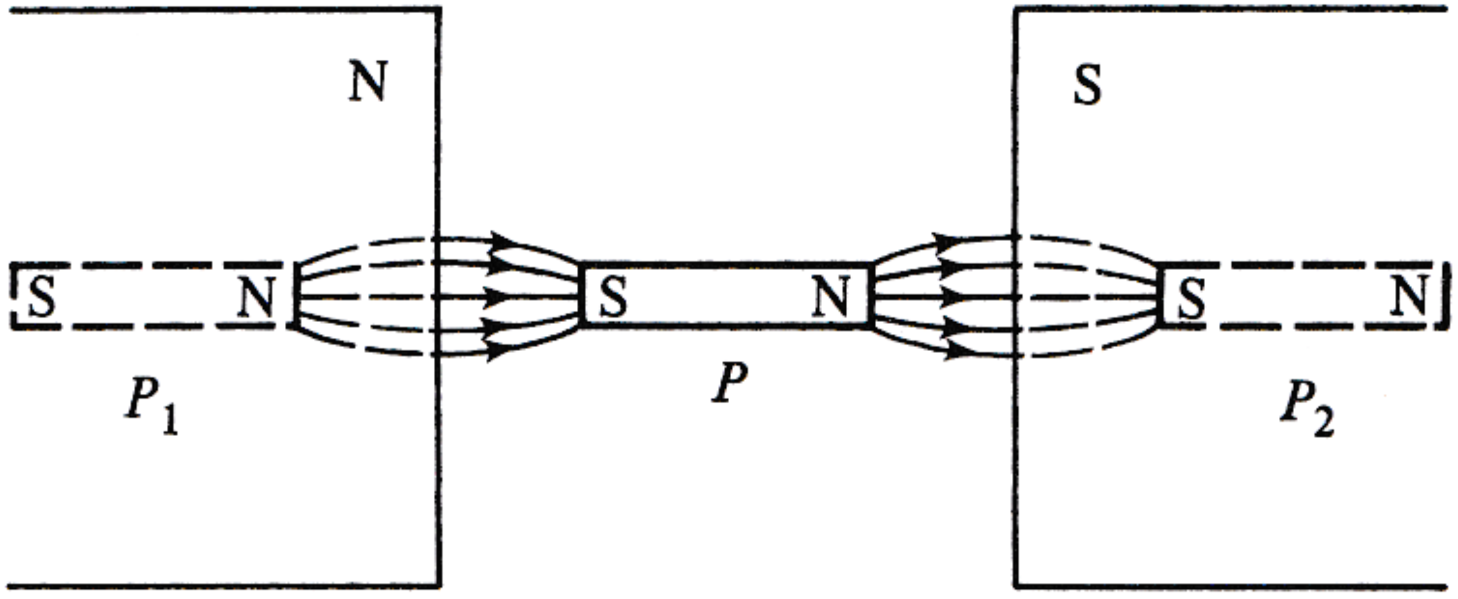


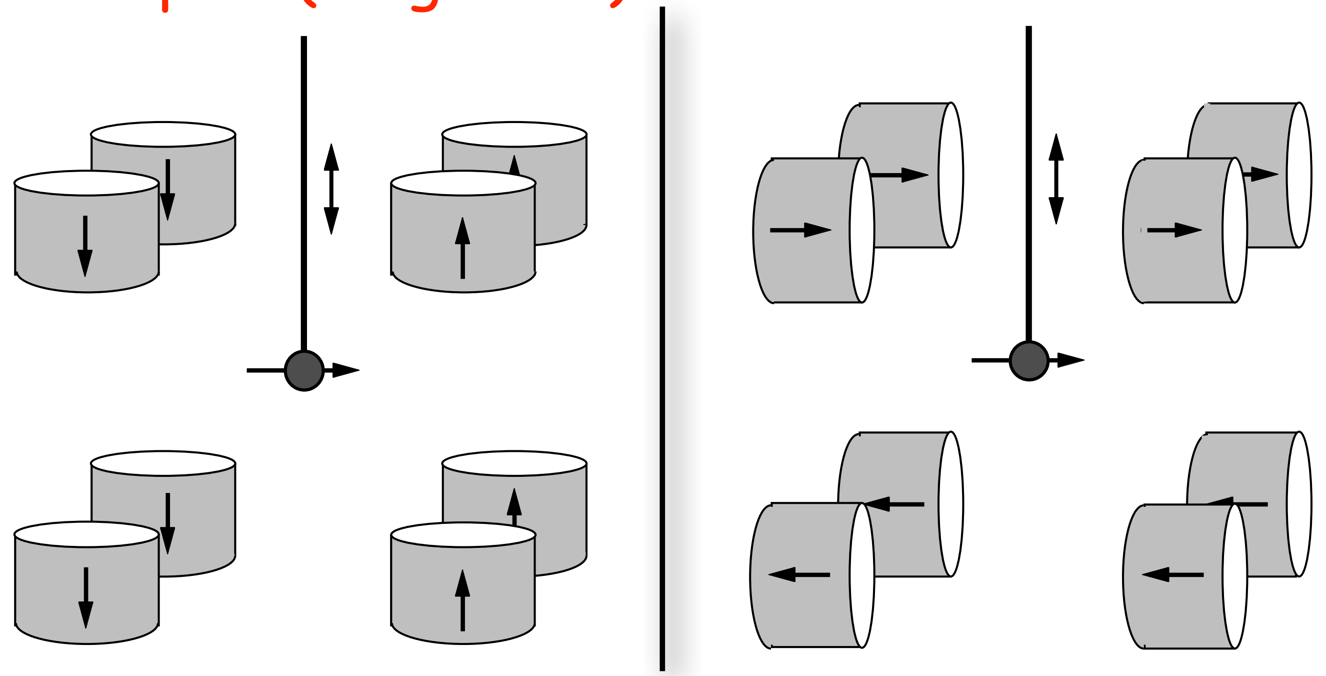


3.1 Magnetic Measurements

a) Inductive methods

Vibrating Sample Magnetometer (VSM)

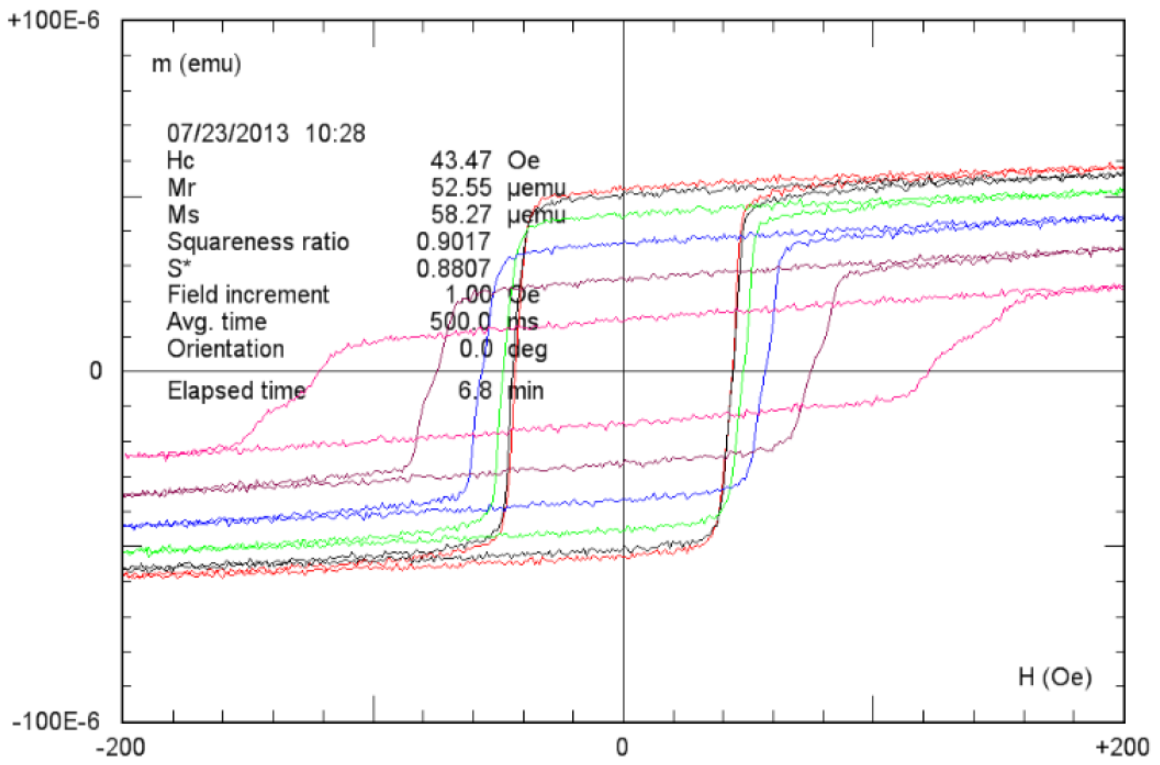
- VSM measures magnetic moment and therefore magnetization M – whereas fluxmeter methods measure flux density B
- VSM is best method to measure the saturation magnetization M_s of any material because high field can be applied to reach saturation
- VSM less useful for measurement of other parameters of magnetization curve in soft magnetic materials because of demagnetization effects (real field unknown). Exception: thin films. For high-anisotropy or hard magnetic materials, however, the VSM is the preferred instrument for many kinds of magnetic measurements
- Sensitivity: 10^{-5} emu $=$ 10^{-8} Am^2 → small samples (< 1 gramm)
- Pick-up coil arangements to measure longitdinal and transverse magnetization components
- SQUID magnetometer: high-sensitive variant of VSM. Pick-up signal trans-formed to SQUID device outside magnet



3.1 Magnetic Measurements

a) Inductive methods

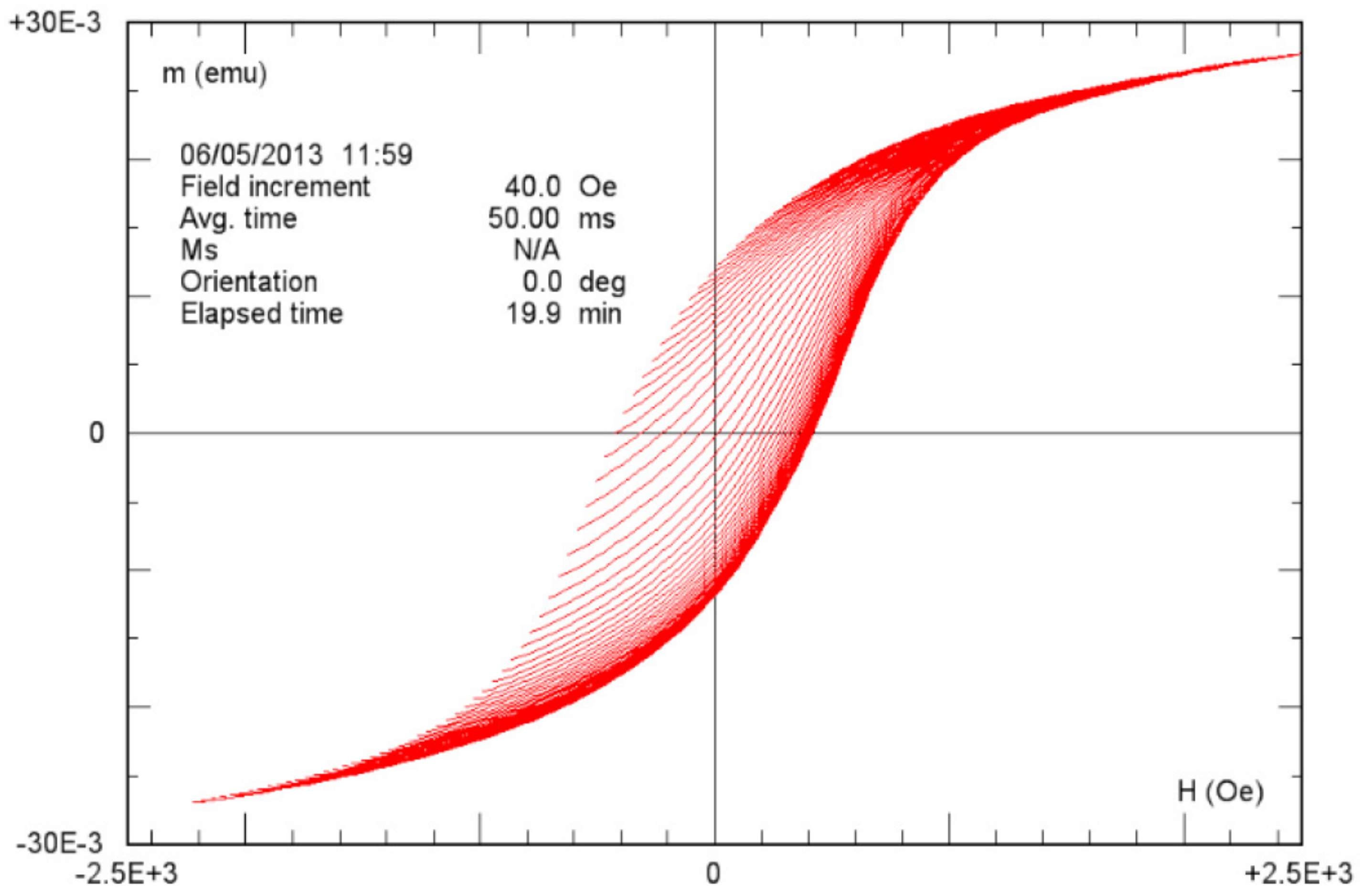
Vibrating Sample Magnetometer (VSM)



07/23/2013 10:28
Hc 43.47 Oe
Mr 52.55 μemu
Ms 58.27 μemu
Squareness ratio 0.9017
S* 0.8807
Field increment 1.00 Oe
Avg. time 500.0 ms
Orientation 0.0 deg
Elapsed time 6.8 min

M(H) vs out-of-plane angle for thin film

Examples



06/05/2013 11:59
Field increment 40.0 Oe
Avg. time 50.00 ms
Ms N/A
Orientation 0.0 deg
Elapsed time 19.9 min

parallel

File: #3 Fe3O4 irradiated.FORC.50 ms.19 min

Courtesy Lakeshore

3.1 Magnetic Measurements

a) Inductive methods

Sample surrounded by coil, in which voltage is induced when magnetization of sample is changed or when sample is moved. Voltage is integrated → signal proportional to magnetization

b) Magnetometric methods

For finite samples: demagnetizing field, which is proportional to mean magnetization ($H_{\mathrm{dem}} = -N\overline{M}$), is measured

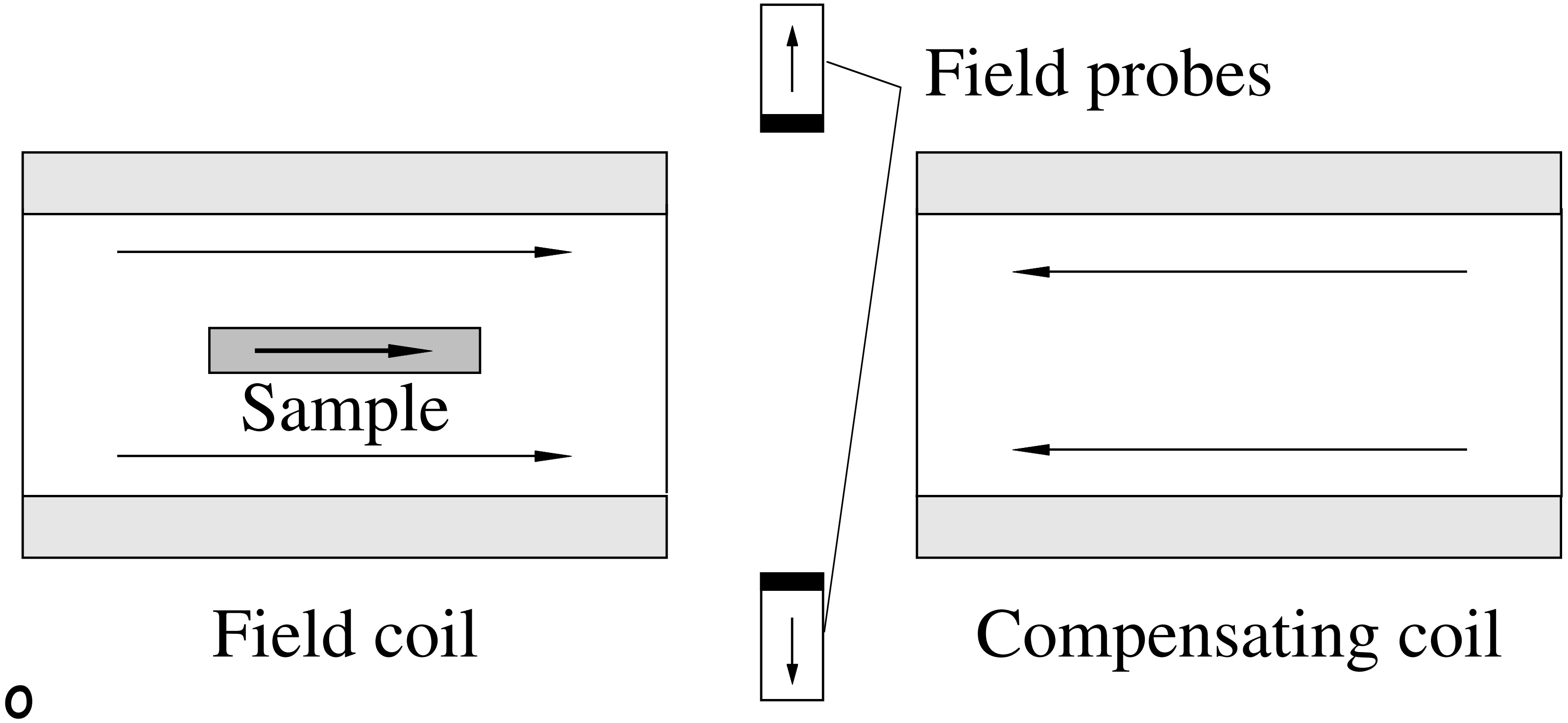
c) Optical magnetometry

Surface magnetization is measured by magneto-optic effect, useful for thin films where signal of inductive or magnetometric methods are too weak

3.1 Magnetic Measurements

b) Magnetometric measurements

- Magnetometer measures dipolar field generated by magnetized sample with the help of field detection device (e.g. Hall probe)
- In shown arrangement the difference signal of the two probes is proportional to the magnetic moment of the sample, and insensitive to the driving field



- Probes must be placed sufficiently far away from sample, so that only its dipolar field is detected → samples should be small and short, but can be of any shape → hard and high-anisotropy materials better suited than soft magnetic materials where demagnetization will dominate the intrinsic properties of short samples
- Advantages: (i) any sample shape, (ii) arbitrarily slow magnetization processes can be followed, (iii) sample can be easily exposed to various environmental conditions such as high or low temperature or mechanical stress.

3.1 Magnetic Measurements

c) Optical magnetometer

Magneto-optical Kerr effect

$$\boldsymbol{D} = \varepsilon \boldsymbol{E} = \varepsilon \begin{pmatrix} 1 & -\mathrm{i}\,Q\,m_3 & \mathrm{i}\,Q\,m_2 \\ \mathrm{i}\,Q\,m_3 & 1 & -\mathrm{i}\,Q\,m_1 \\ -\mathrm{i}\,Q\,m_2 & \mathrm{i}\,Q\,m_1 & 1 \end{pmatrix} \boldsymbol{E}$$

$$= \varepsilon \boldsymbol{E} + \mathrm{i}\,\varepsilon\,Q\,\boldsymbol{m} \times \boldsymbol{E}$$

$\boldsymbol{E}$: electric vector of light wave

$\boldsymbol{D}$: dielectric displacement vector

(= vector of light after reflection)

m_i: components of magnetization vector (cubic crystal)

$\boldsymbol{\varepsilon}$: dielectric tensor

Q : material constant (~M_s, complex, determines strength of rotation)

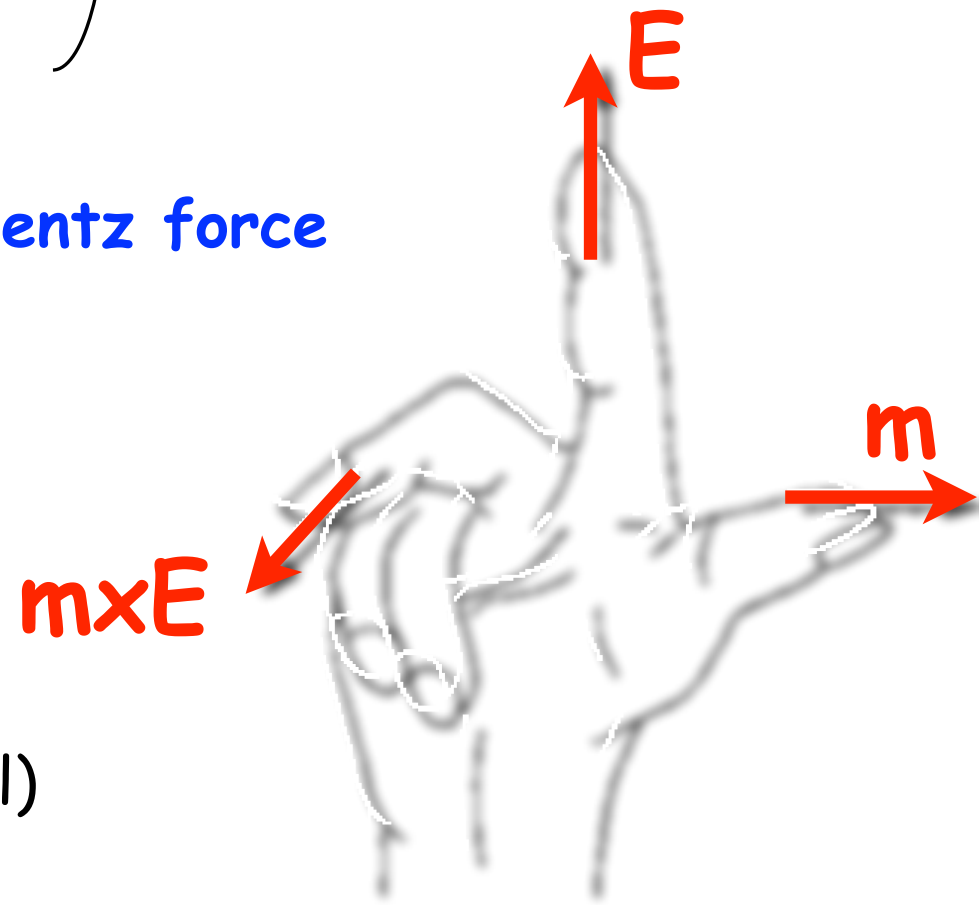
3.1 Magnetic Measurements

c) Optical magnetometer

Magneto-optical Kerr effect

$$\boldsymbol{D} = \varepsilon \boldsymbol{E} = \varepsilon \begin{pmatrix} 1 & -\mathrm{i}\,Q\,m_3 & \mathrm{i}\,Q\,m_2 \\ \mathrm{i}\,Q\,m_3 & 1 & -\mathrm{i}\,Q\,m_1 \\ -\mathrm{i}\,Q\,m_2 & \mathrm{i}\,Q\,m_1 & 1 \end{pmatrix} \boldsymbol{E}$$

$$= \varepsilon \boldsymbol{E} + \mathrm{i}\,\varepsilon\,Q\,\boldsymbol{m} \times \boldsymbol{E} \longrightarrow$$ concept of Lorentz force



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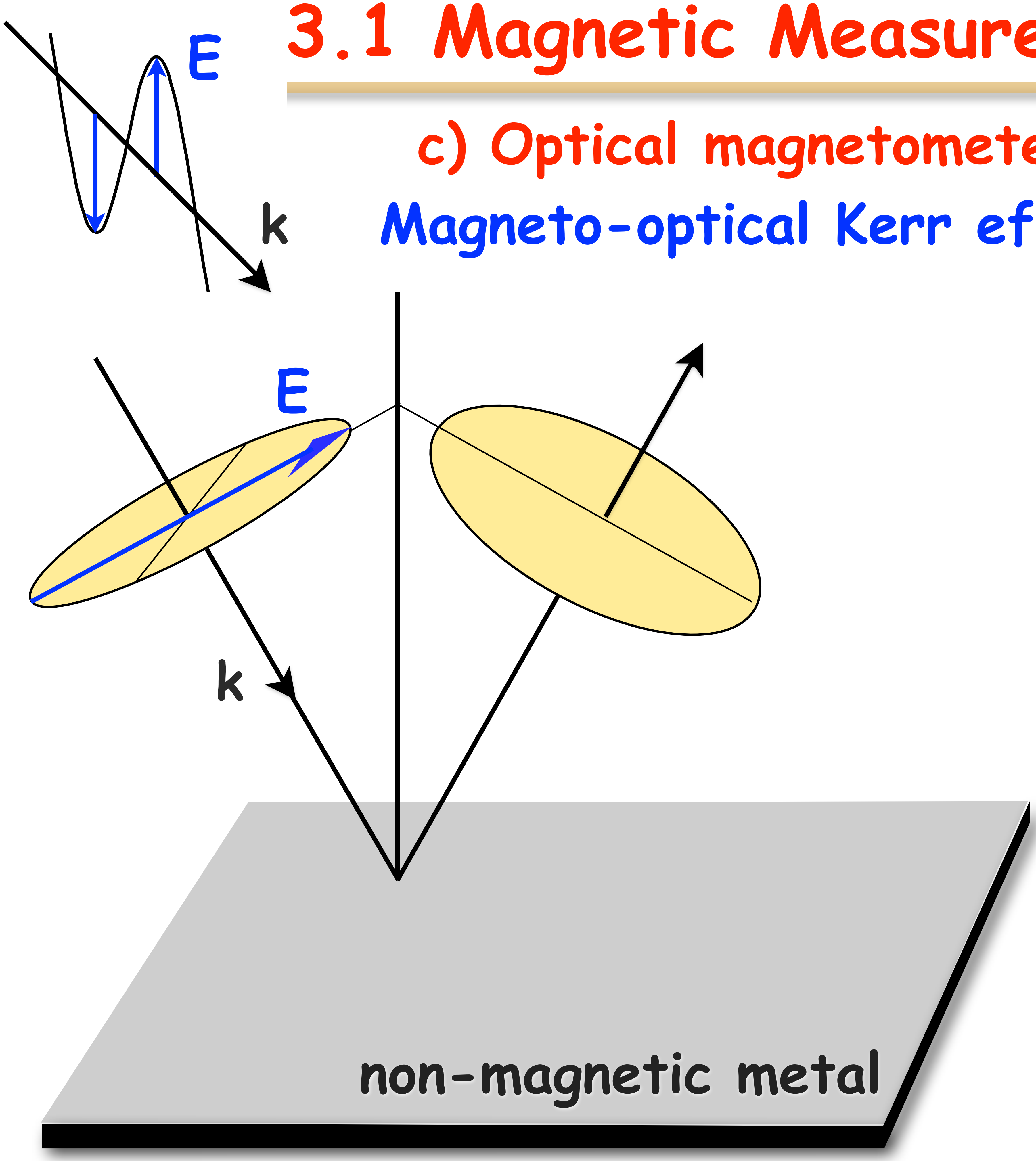

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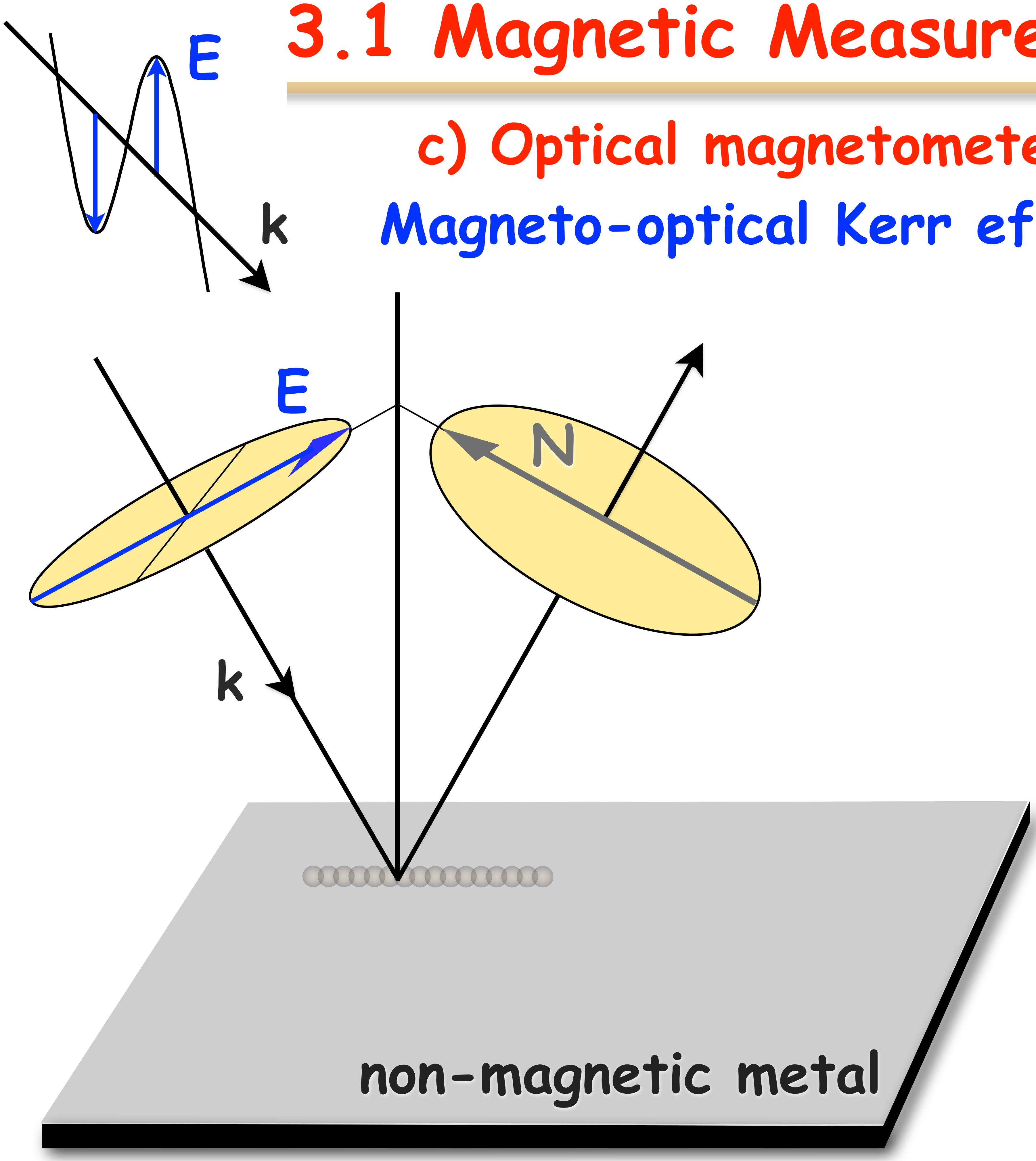

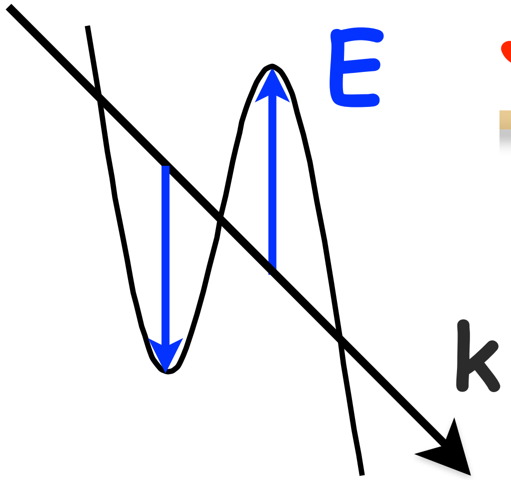

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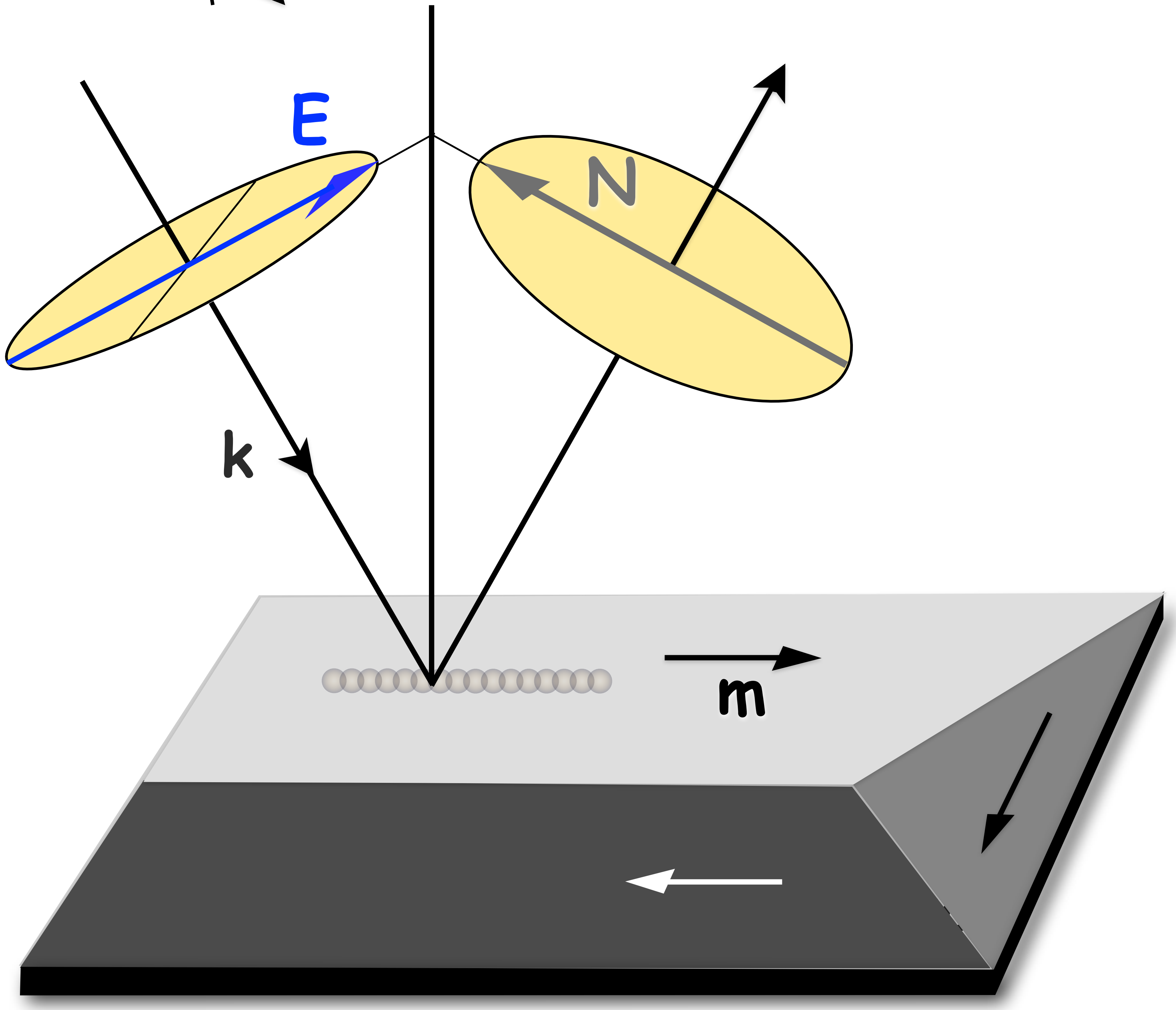

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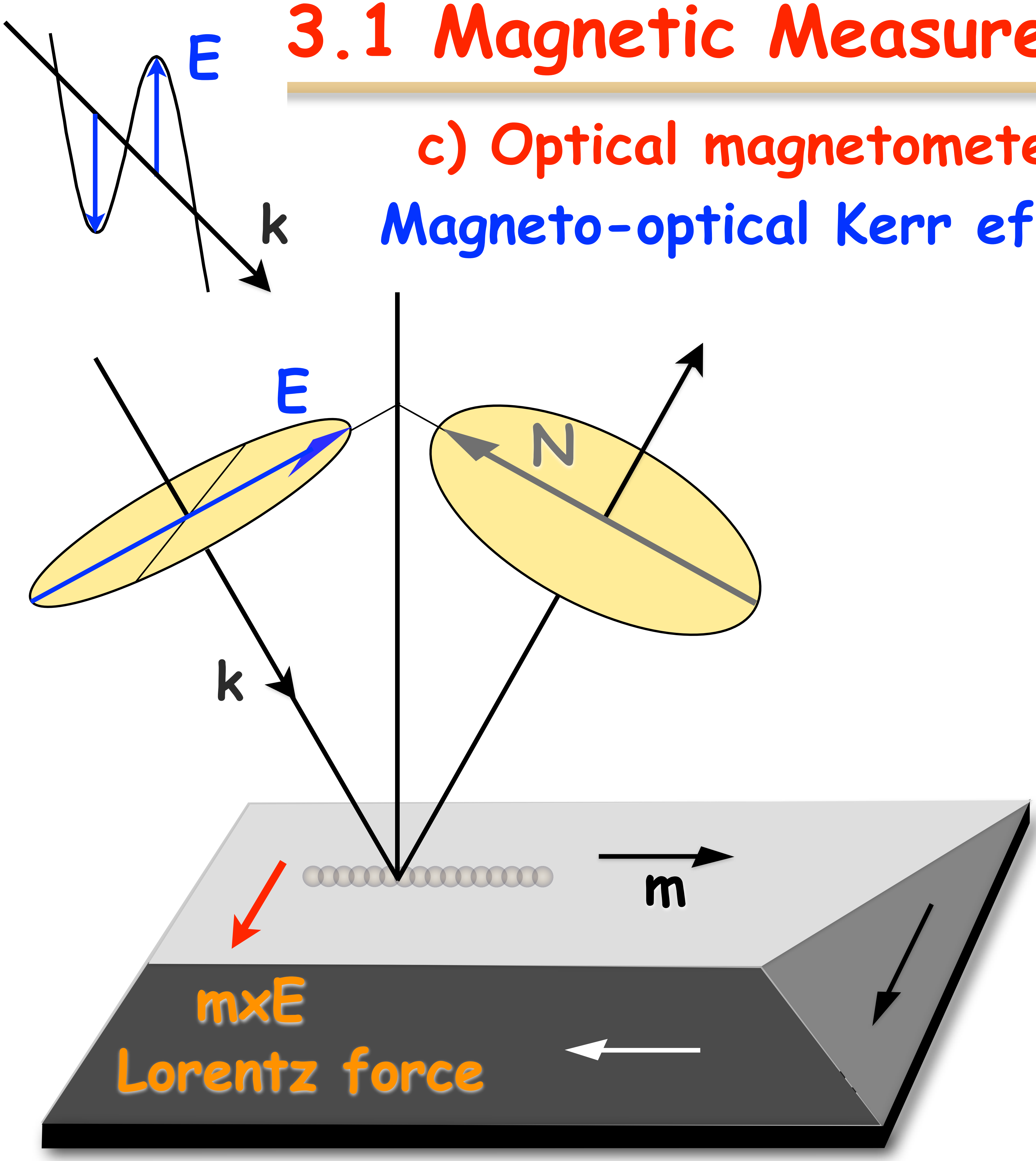

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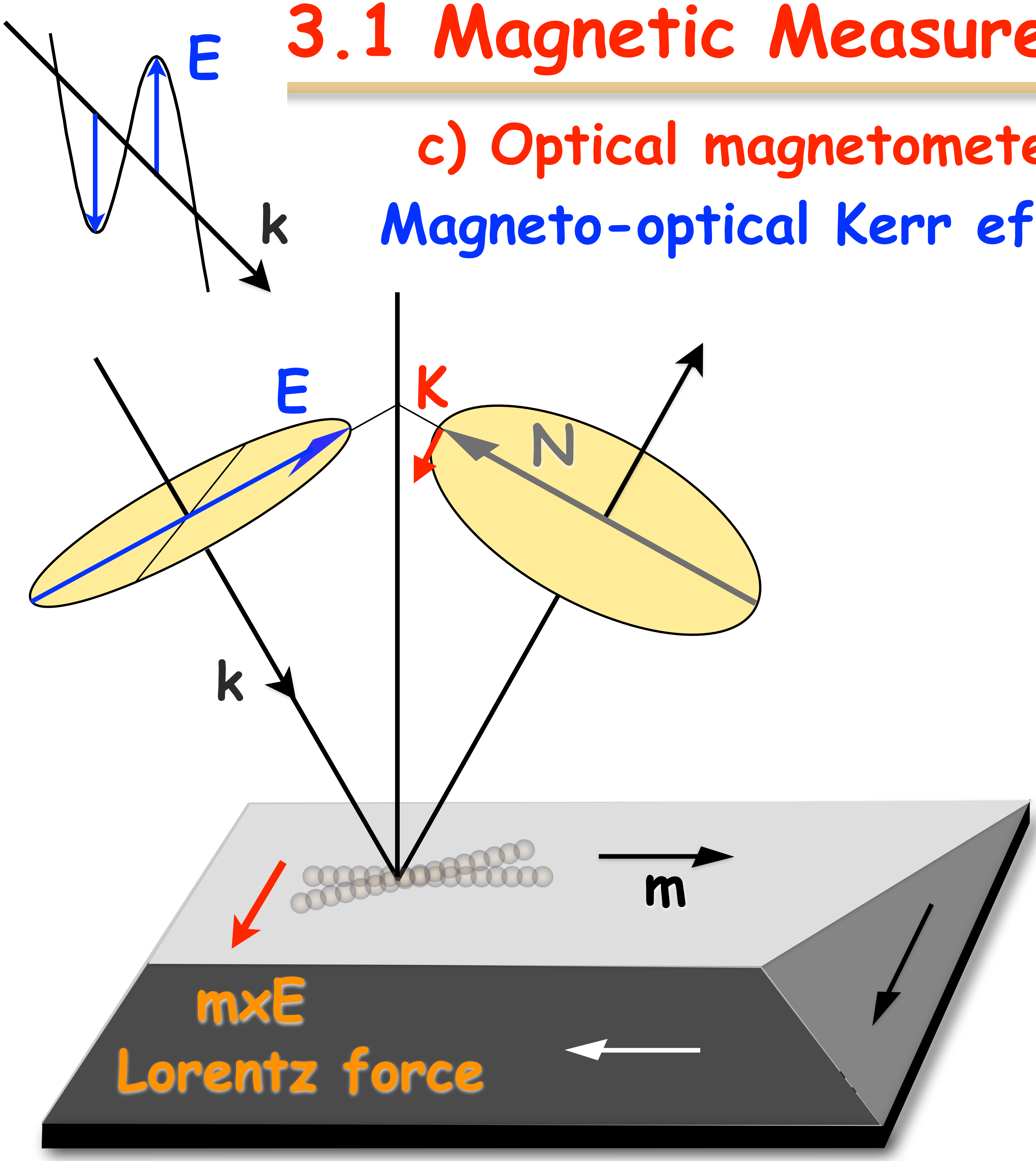

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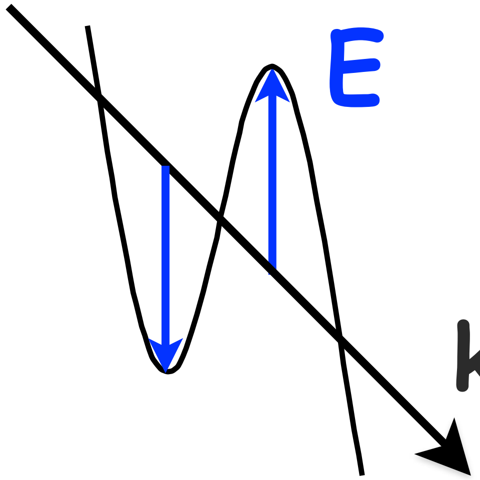

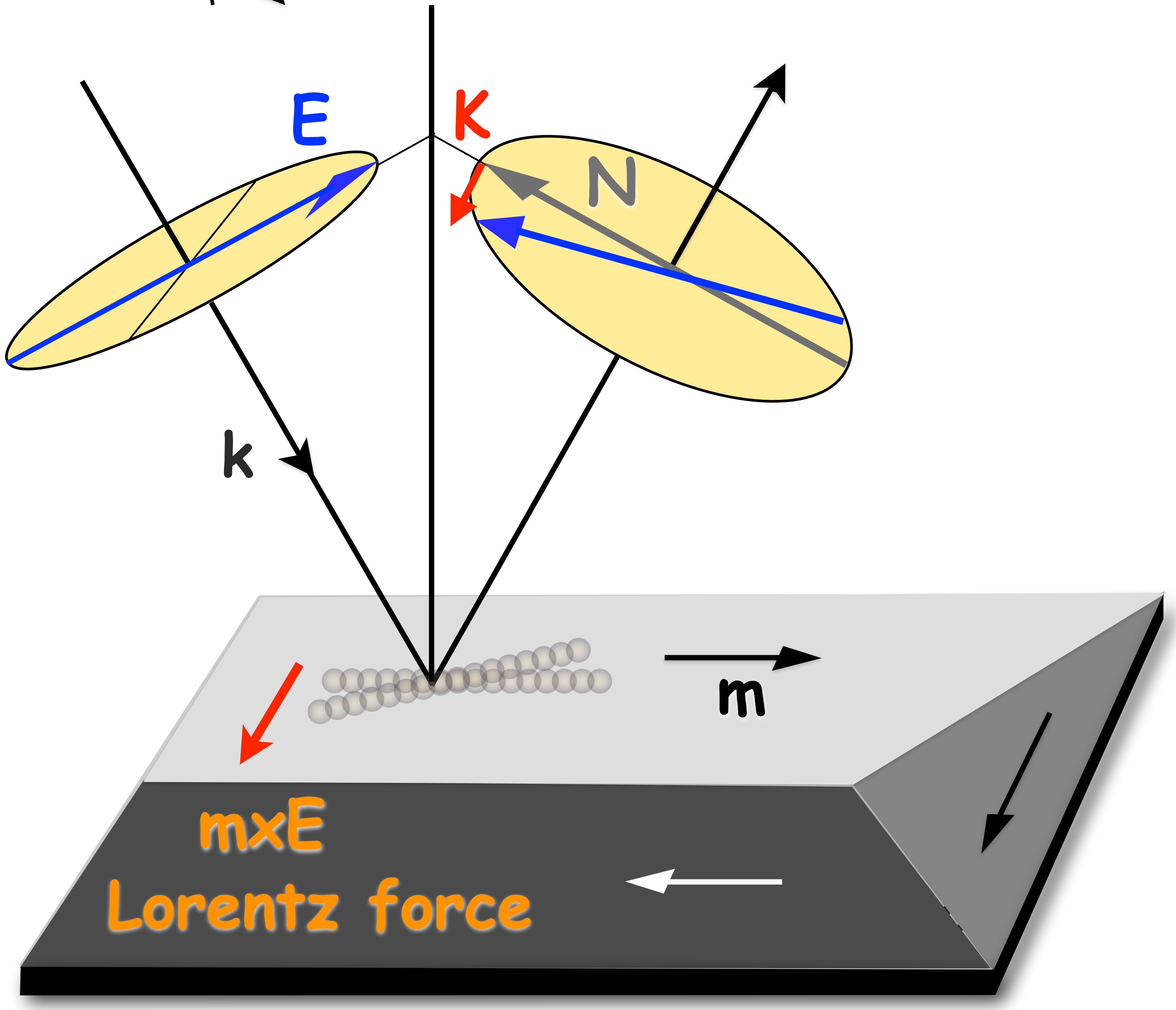

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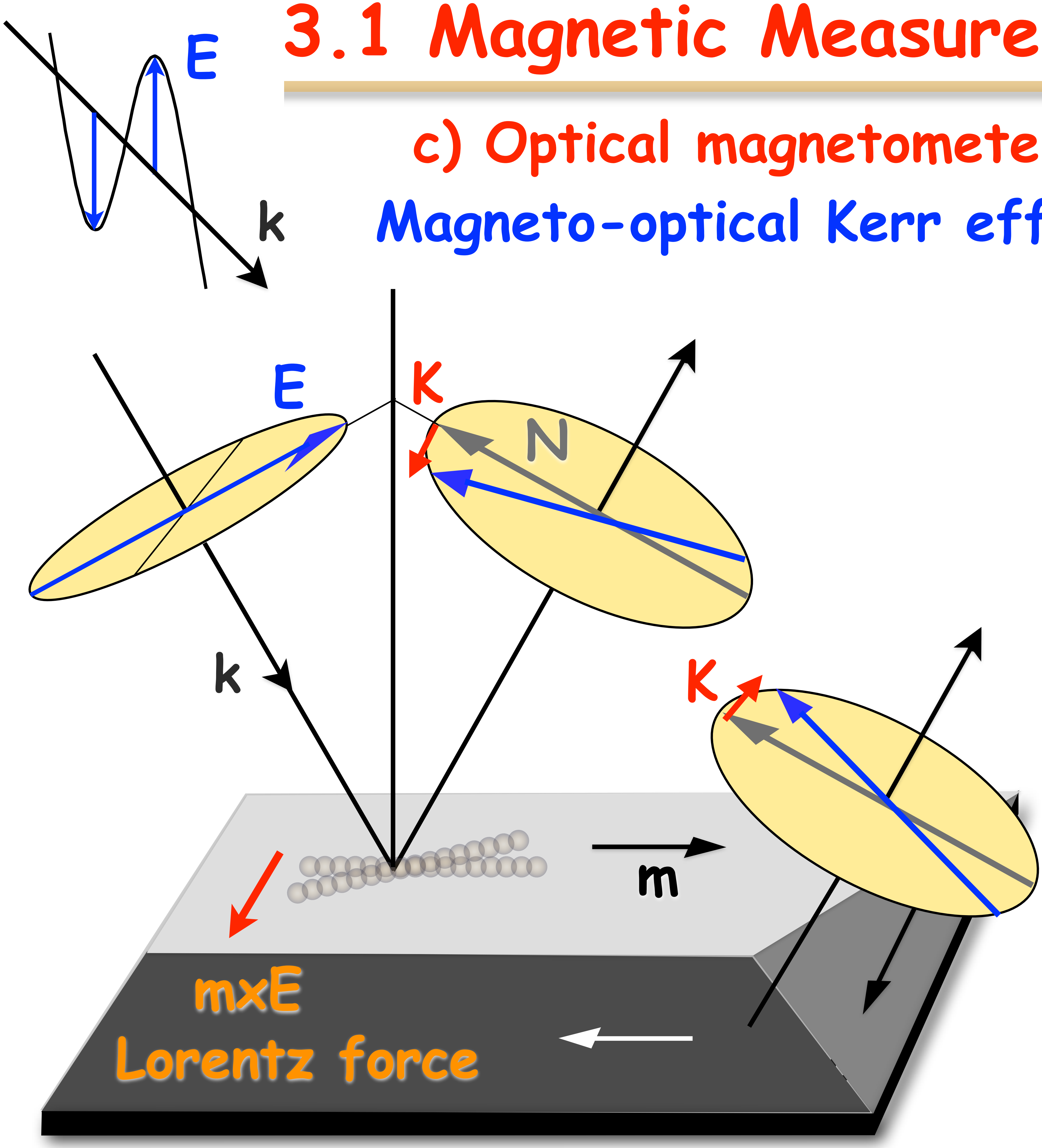

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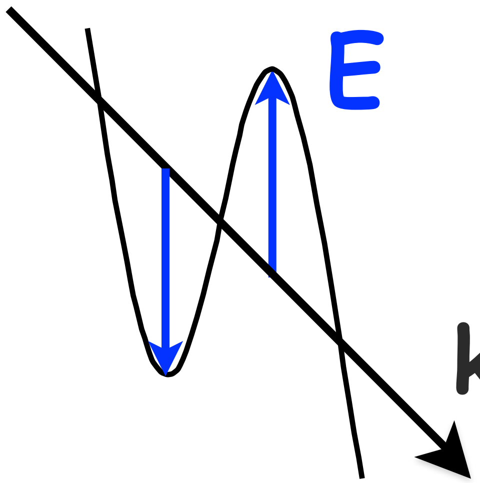

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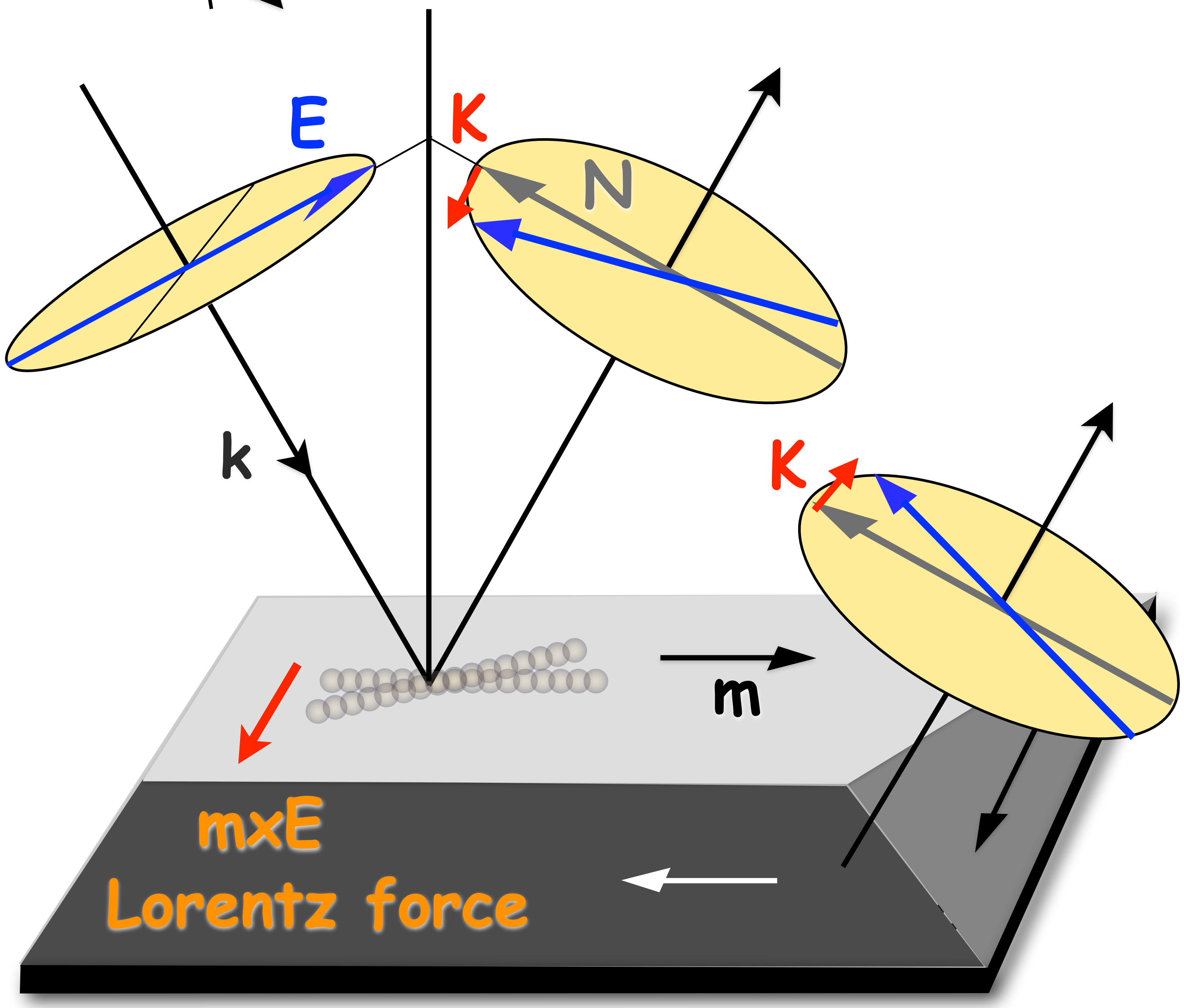

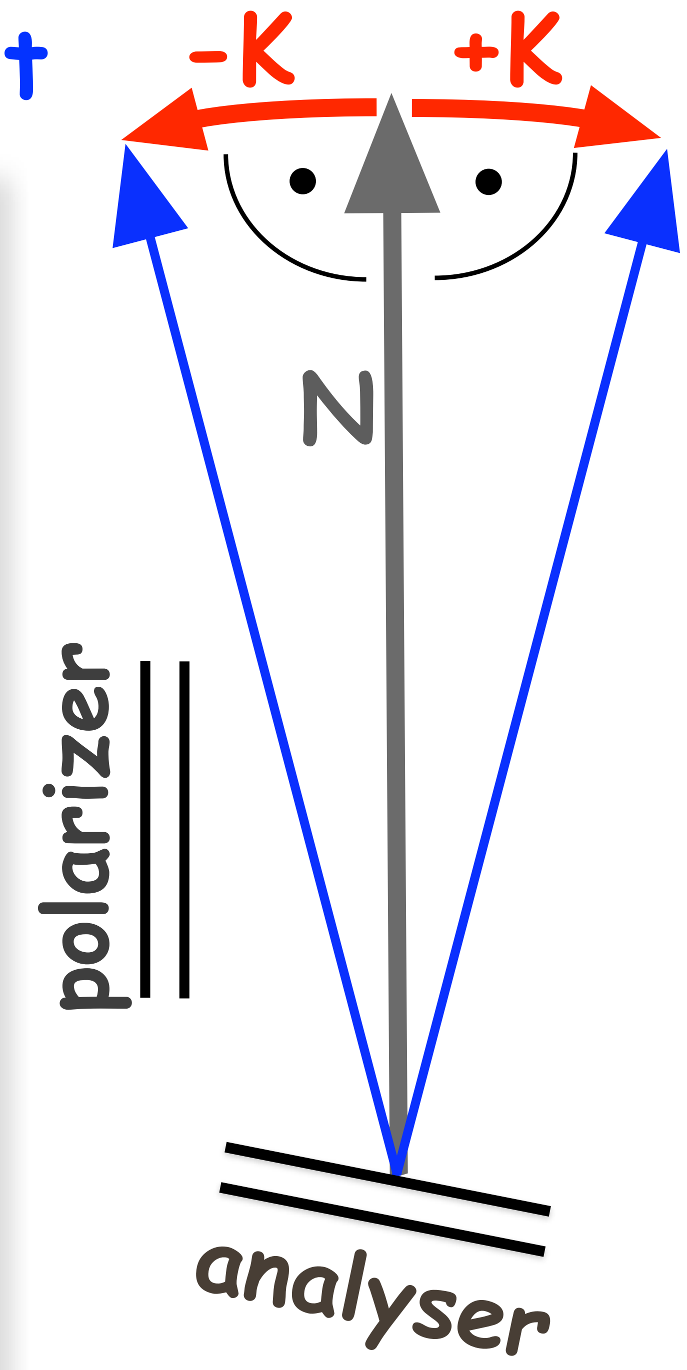

Kerr rotation
= K/N

3.1 Magnetic Measurements

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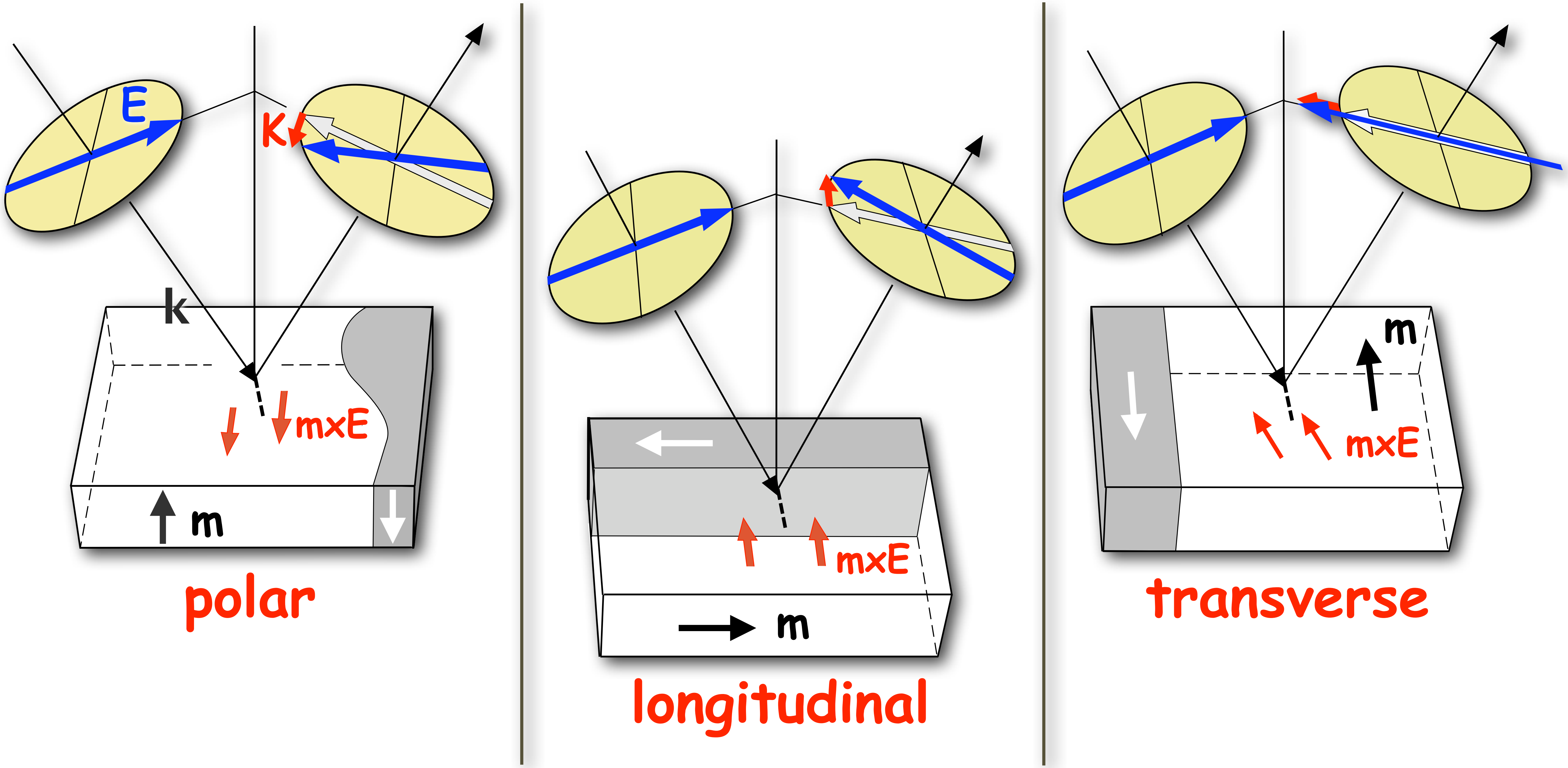
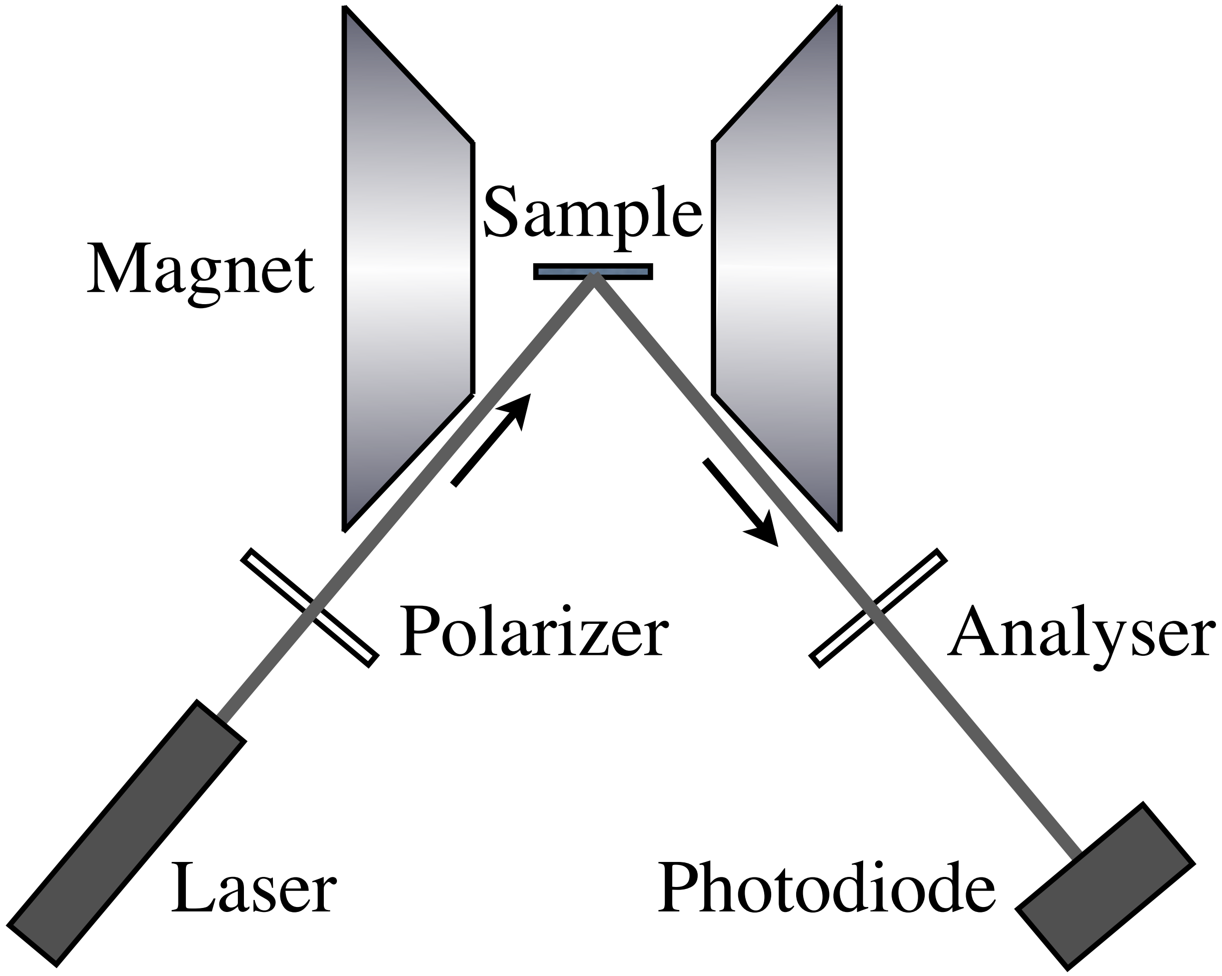
Magneto-optical Kerr effect



The Kerr effect causes a rotation of light, which is proportional to the magnetization component parallel to the reflected light beam

3.1 Magnetic Measurements

c) Optical magnetometer



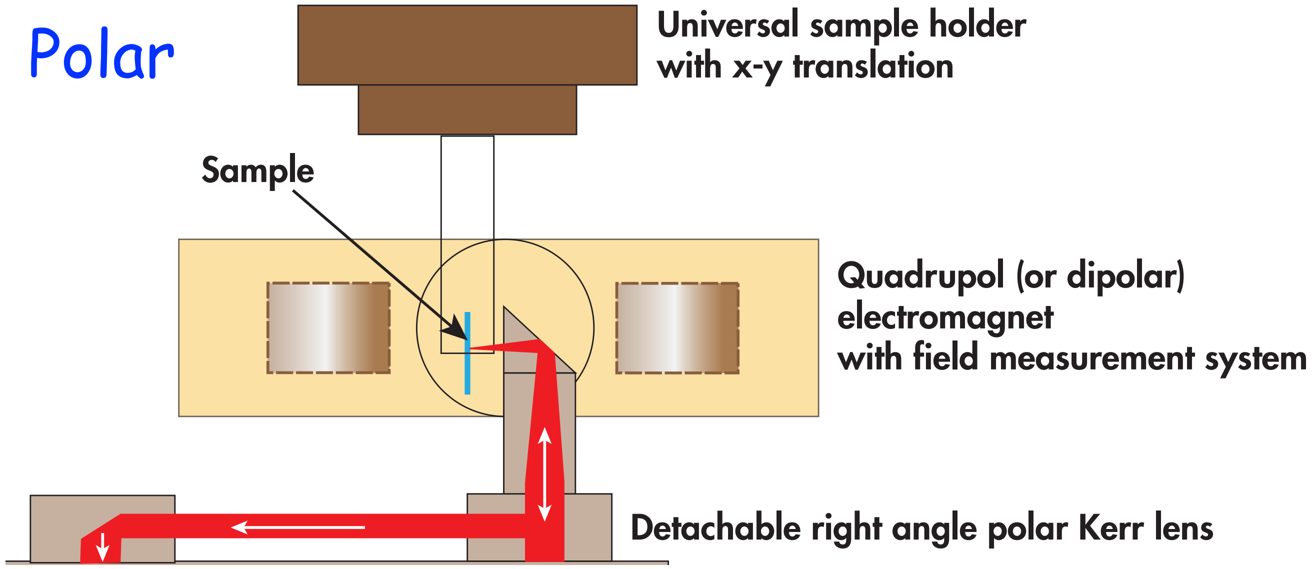
- Magneto-optical Kerr effect is linear function of magnetization and therefore well-suited for magnetometry
- For non-transparent material optical magnetometry makes sense only for thin films for which surface magnetization is representative
- Advantages: (i) direct, (ii) quasi-static and dynamic measurements, (iii) Space-resolved measurements are possible by scanning over the surface. (iv) Optical measurements can be performed on-line during preparation or treatment of a material for example inside vacuum chamber
- Noise suppression: feed split-off part of laser light as reference signal into amplifier. If polarization of light is modulated by a spinning analyser or electro-optical device, the magnetic signal can be detected by a lock-in amplifier, thus achieving virtually unlimited sensitivity

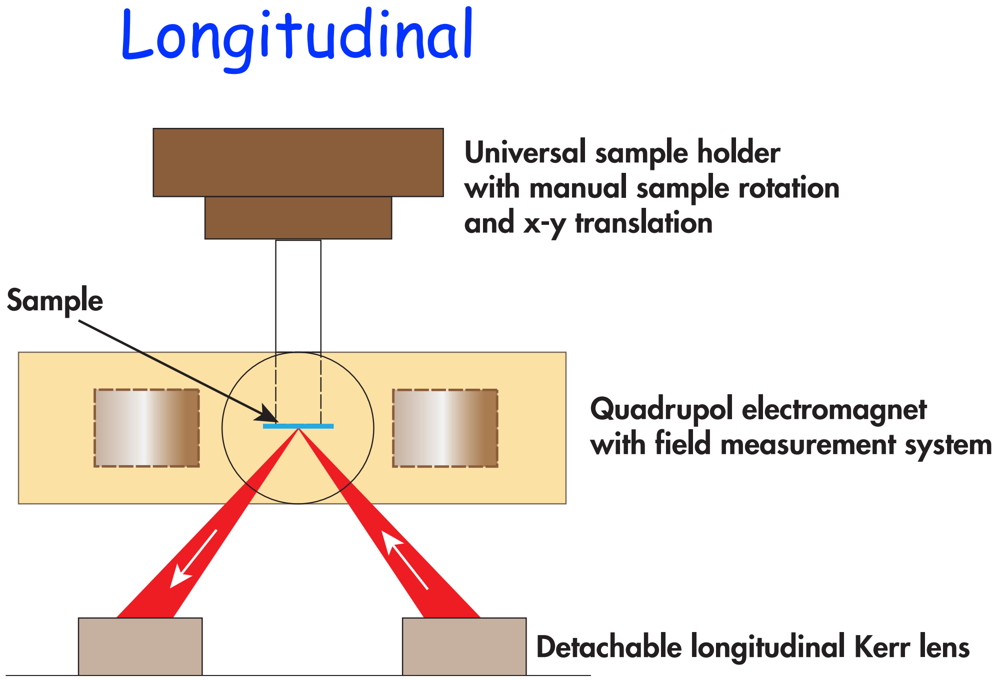
3.1 Magnetic Measurements

b) Optical magnetometer

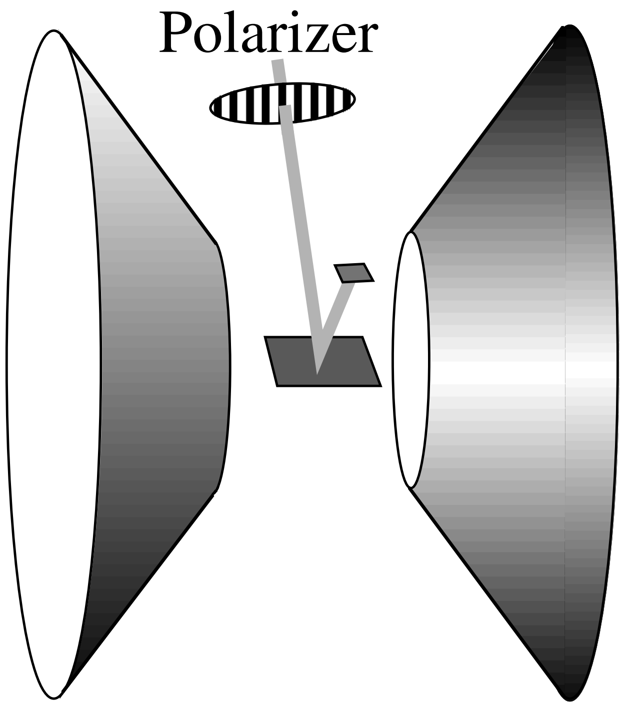
NANOMOKE3®
http://www.lot-qd.de

- Polar Kerr magnetometer (P-MOKE)





- Use of transverse Kerr effect (T-MOKE)
  - Polarizer set parallel to the plane of incidence and analyser omitted
  - M-component perpendicular to the plane of incidence causes variation of the reflected intensity, which can be detected electronically
  - Fits nicely into electromagnet



3.

Measurements to determine magnetic material parameters & properties

3.1 Magnetic measurements

3.2 Mechanical measurements

3.3 Resonance techniques

3.4 Dilatometric measurements

3.5 Domain methods

3.

Measurements to determine magnetic material parameters & properties

3.2 Mechanical Measurements

Principle:

Measurement of mechanical forces on magnetic sample

Two possibilities:

- A uniform field $\boldsymbol{H}$, acting on uniformly magnetized sample of magnetization $\boldsymbol{M}$ and volume V, generates a mechanical torque $T_{\mathrm{m}} = \mu_0 V\, H \times M$

- Gradient of non-uniform field generates a mechanical force $\boldsymbol{F}_{\mathrm{m}} = \mu_0 V\, \mathbf{grad}(\boldsymbol{M}\cdot\boldsymbol{H})$

a) Torque magnetometer

Most direct method to measure anisotropy

b) Field gradient mathods

Faraday Balance and Alternating Gradient Magnetometer

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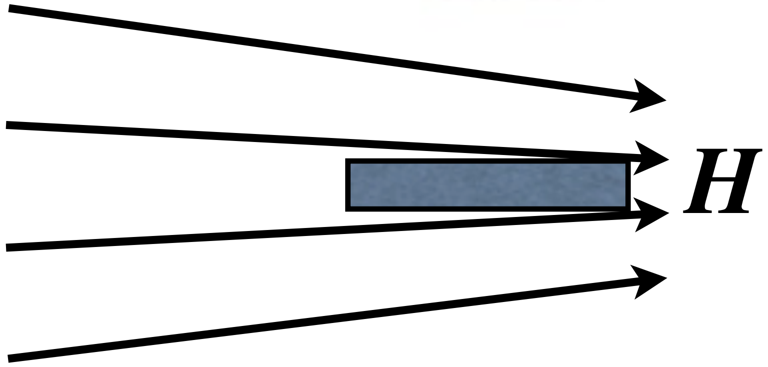

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Faraday Balance and Alternating Gradient Magnetometer

3.2 Mechanical Measurements

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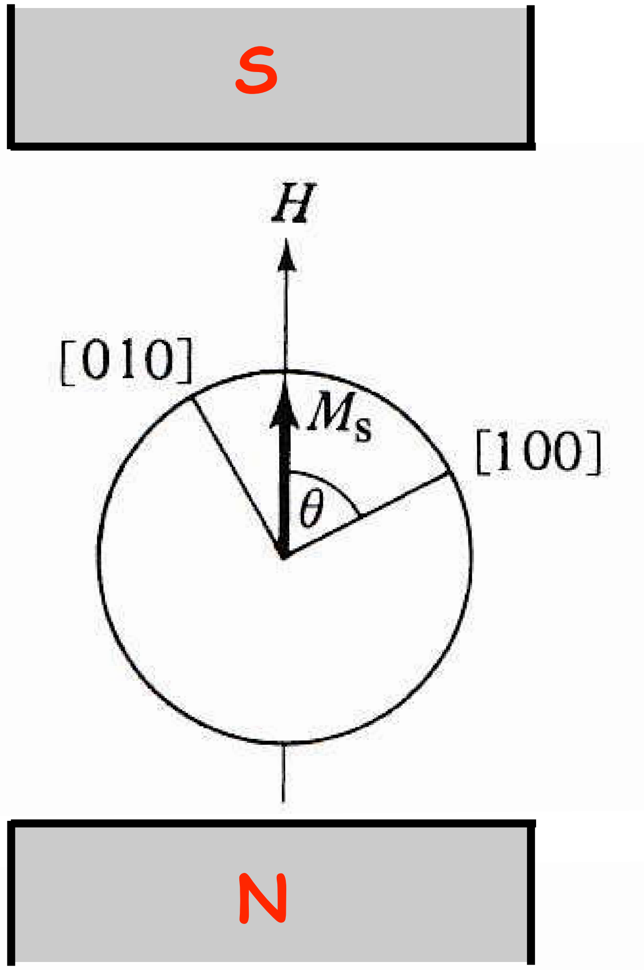


- Torque measurements offer most direct methods for measuring anisotropies
- Requires uniform, single-crystalline samples, preferably of spherical or disk shape
- Example: crystal with cubic anisotropy:

Anisotropy energy:

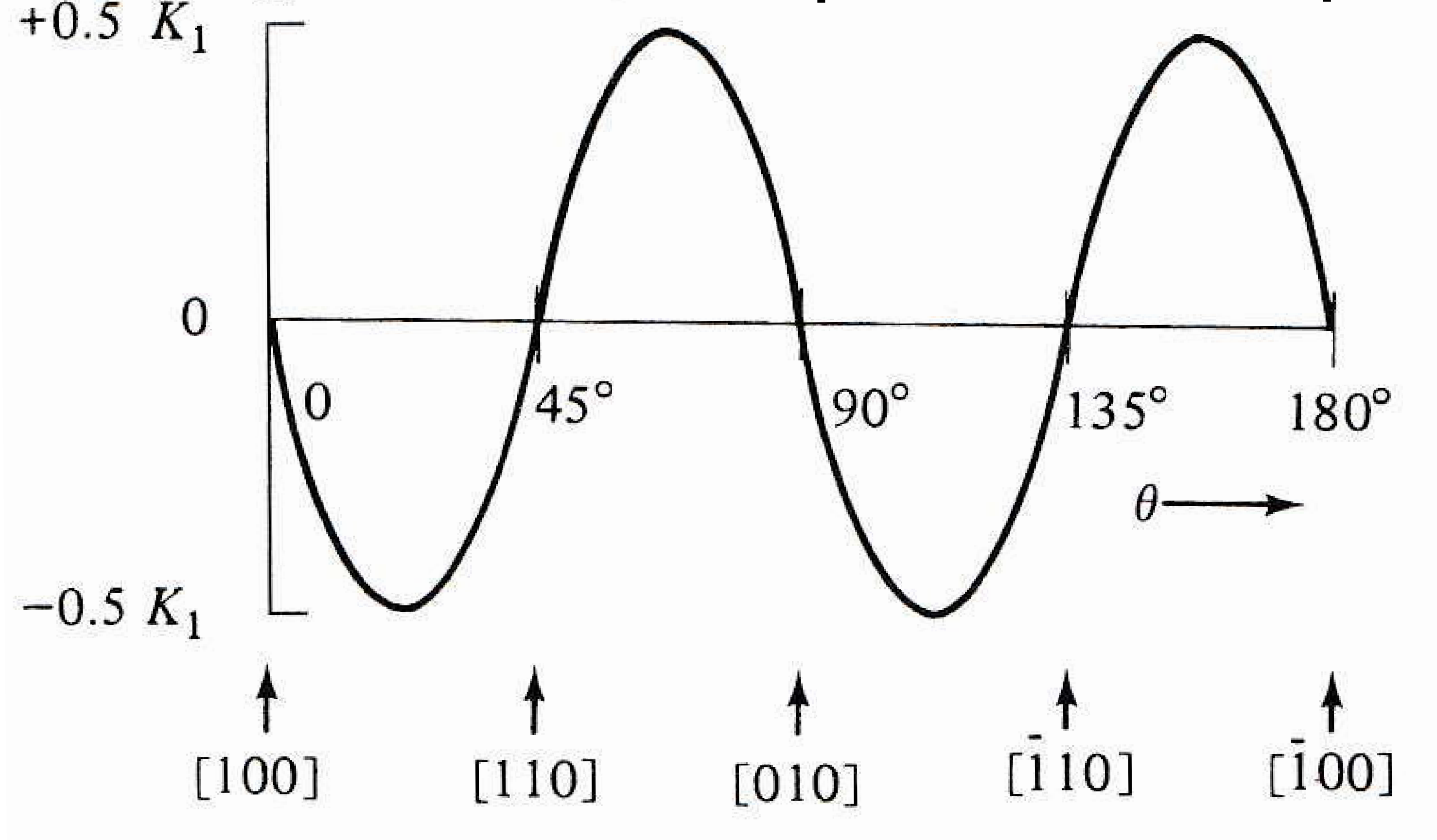
$$E = K_0 + K_1 \sin^2\theta \cos^2\theta,$$

Torque (assumption: $M_s \parallel H$):

$$L = -\frac{dE}{d\theta} = -K_1 \sin 2\theta \cos 2\theta$$

$$= -\frac{K_1}{2} \sin 4\theta.$$

- Anisotropy constant can be measured by torque measurement

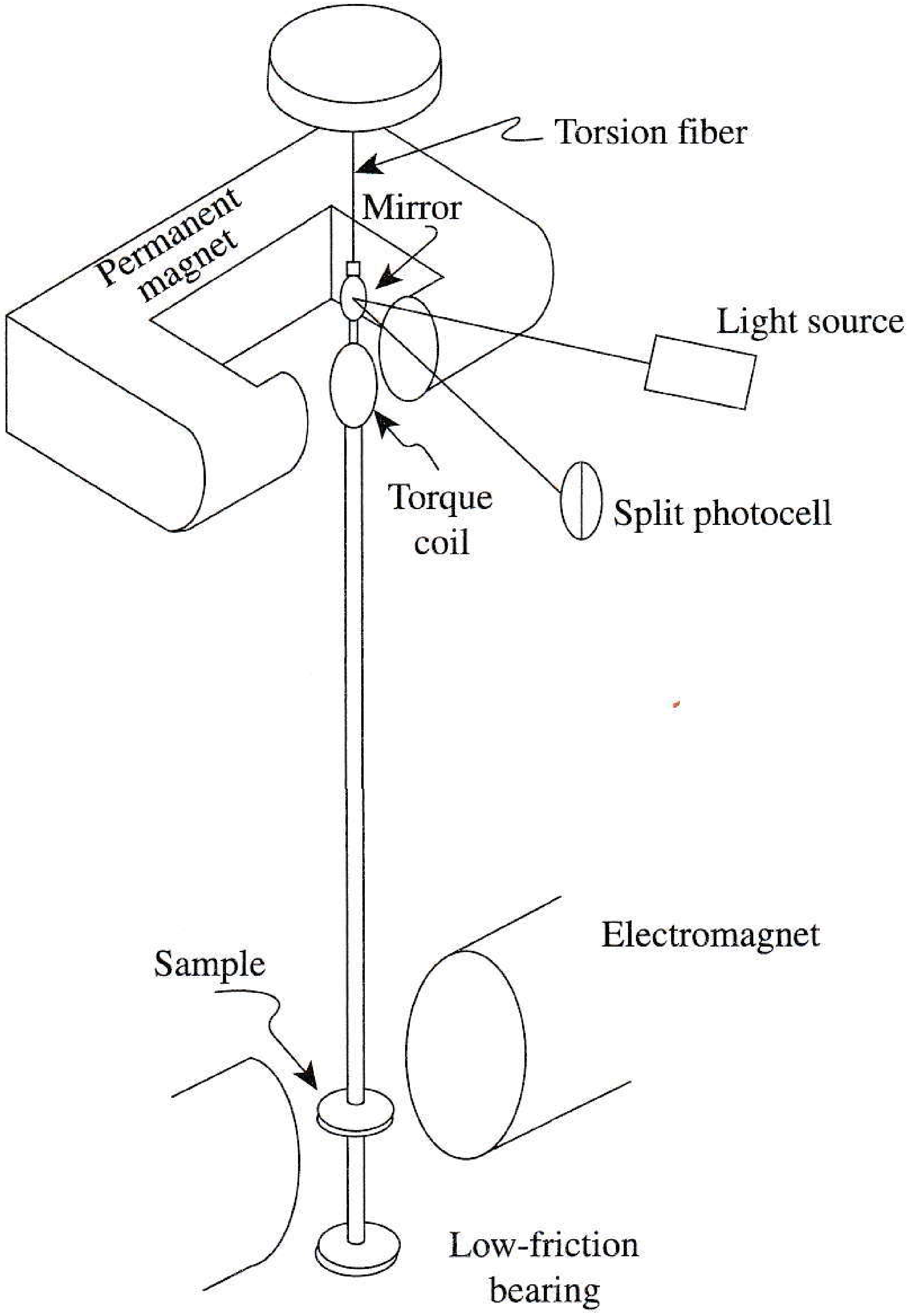


3.2 Mechanical Measurements

a) Torque magnetometer

Example: Torque meter with active sensing

- Sample hung from sensitive torsion fiber, placed in electromagnet that can be rotated
- Torque coil, placed in field of permanent magnet. Current through torque coil: coil experiences torque proportional to current
- Sensing circuit (light beam, mirror, photocell) provides feedback signal that drives a current through the torque coil to balance the anisotropy torque of sample
- Sample can be held at any angle to the field of the electromagnet
- Value of current through torque coil is proportional to torque on sample

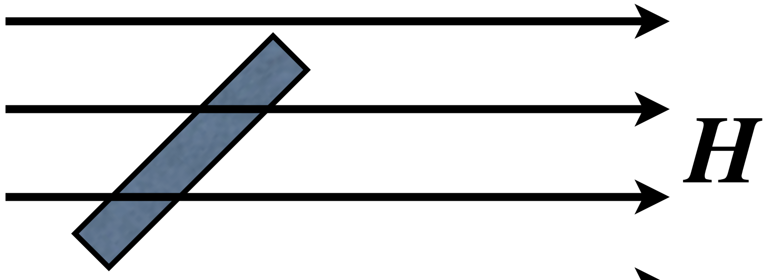


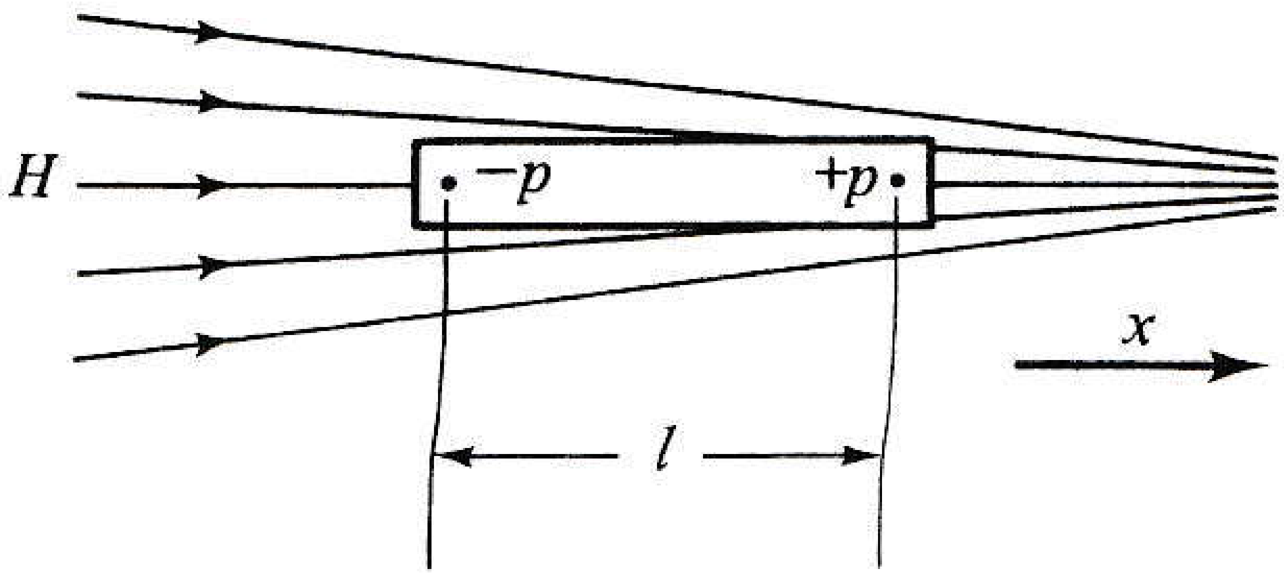
3.2 Mechanical Measurements

b) Field gradient methods

Faraday Balance

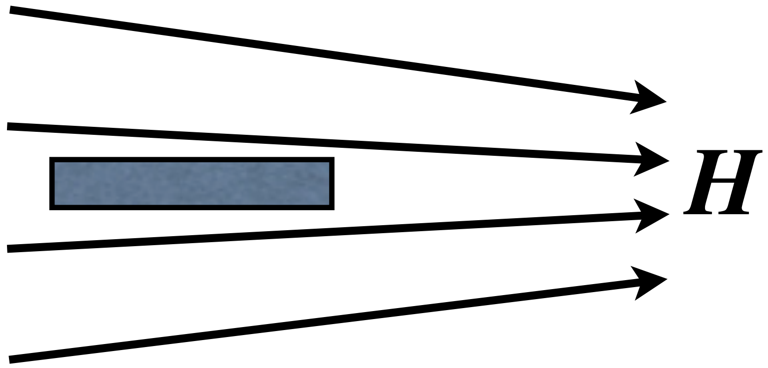
- Non-spherical body in homogeneous field will rotate till its long axis is parallel to field (compass needle)
- Field gradient: Field is stronger at north pole than at south pole → net force F_x to right, with m = magnetic moment and v = volume of body
- Force ~ M
- Body will move toward region of greater field strength (to right)





$$F_x = -pH + p\left(H + l\frac{dH}{dx}\right)$$

$$= pl\frac{dH}{dx} = m\frac{dH}{dx} = Mv\frac{dH}{dx}$$



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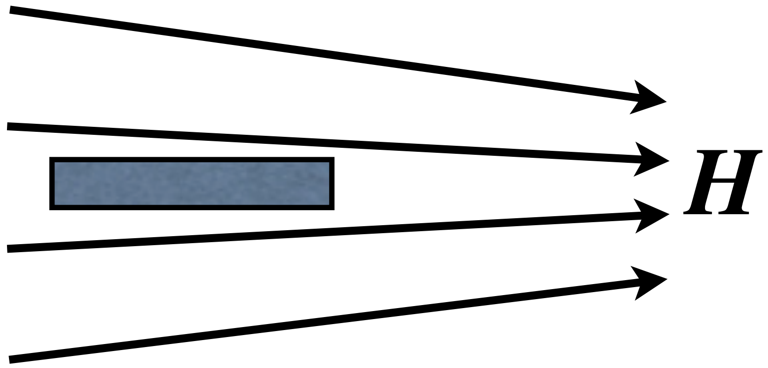

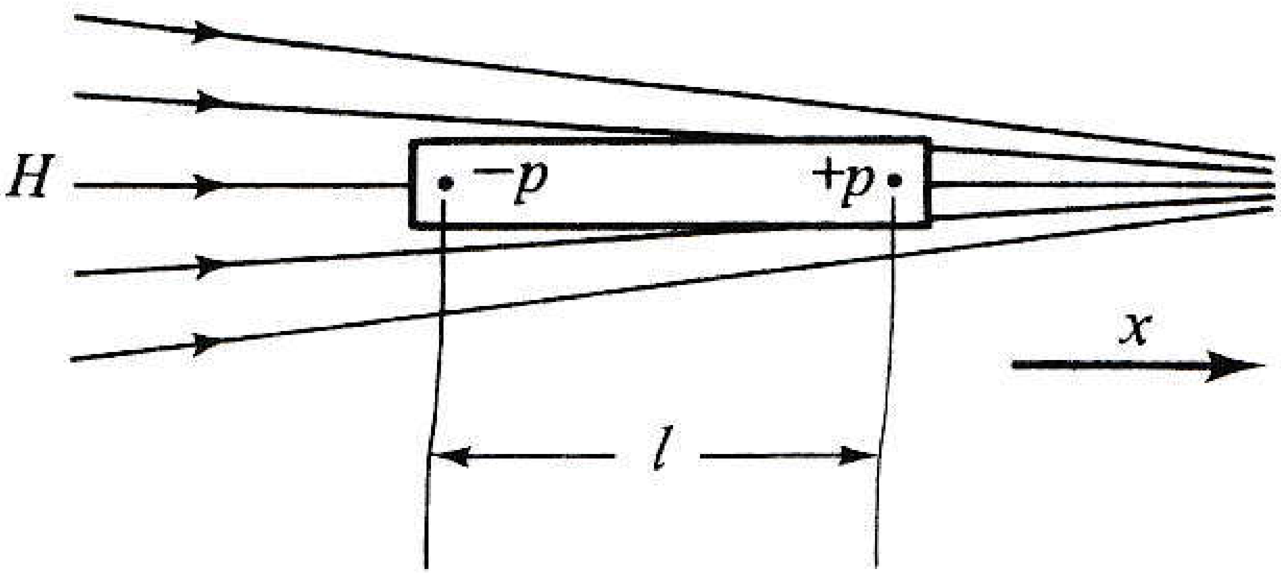

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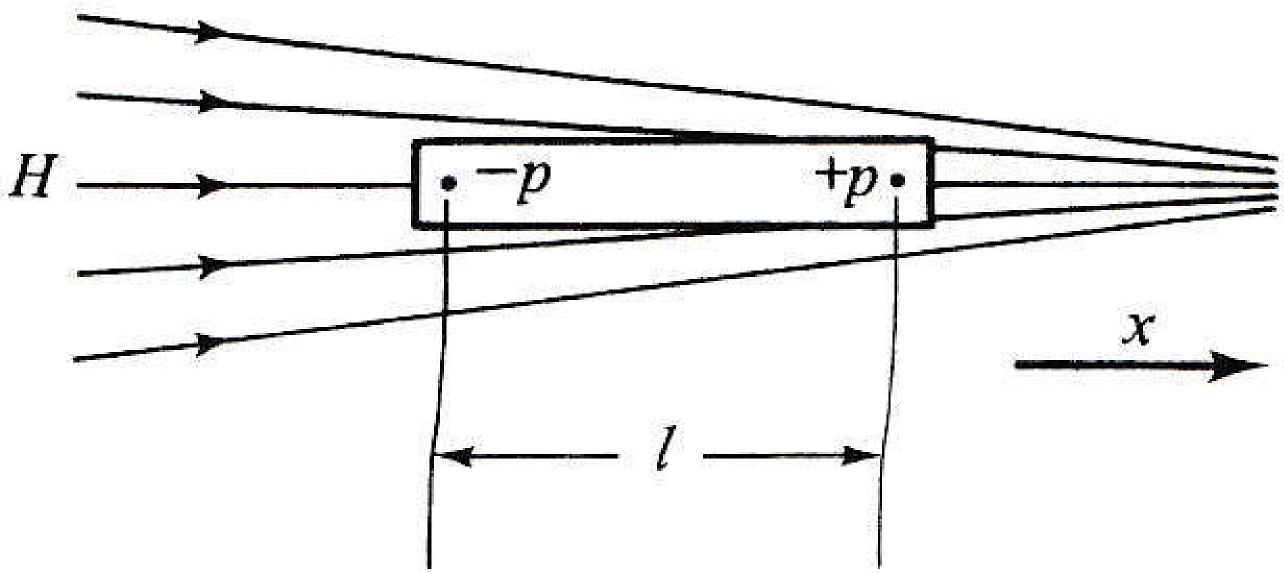

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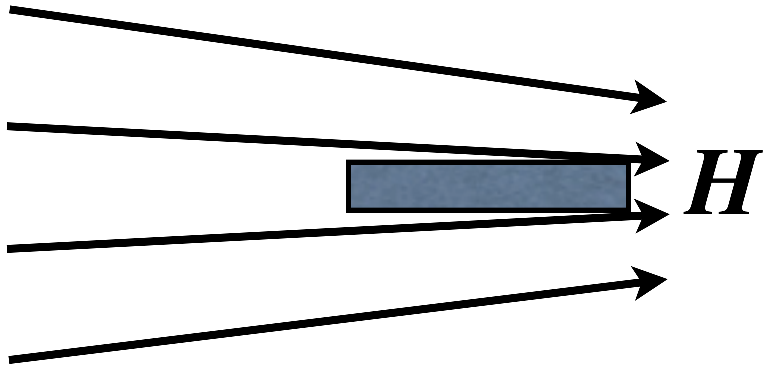
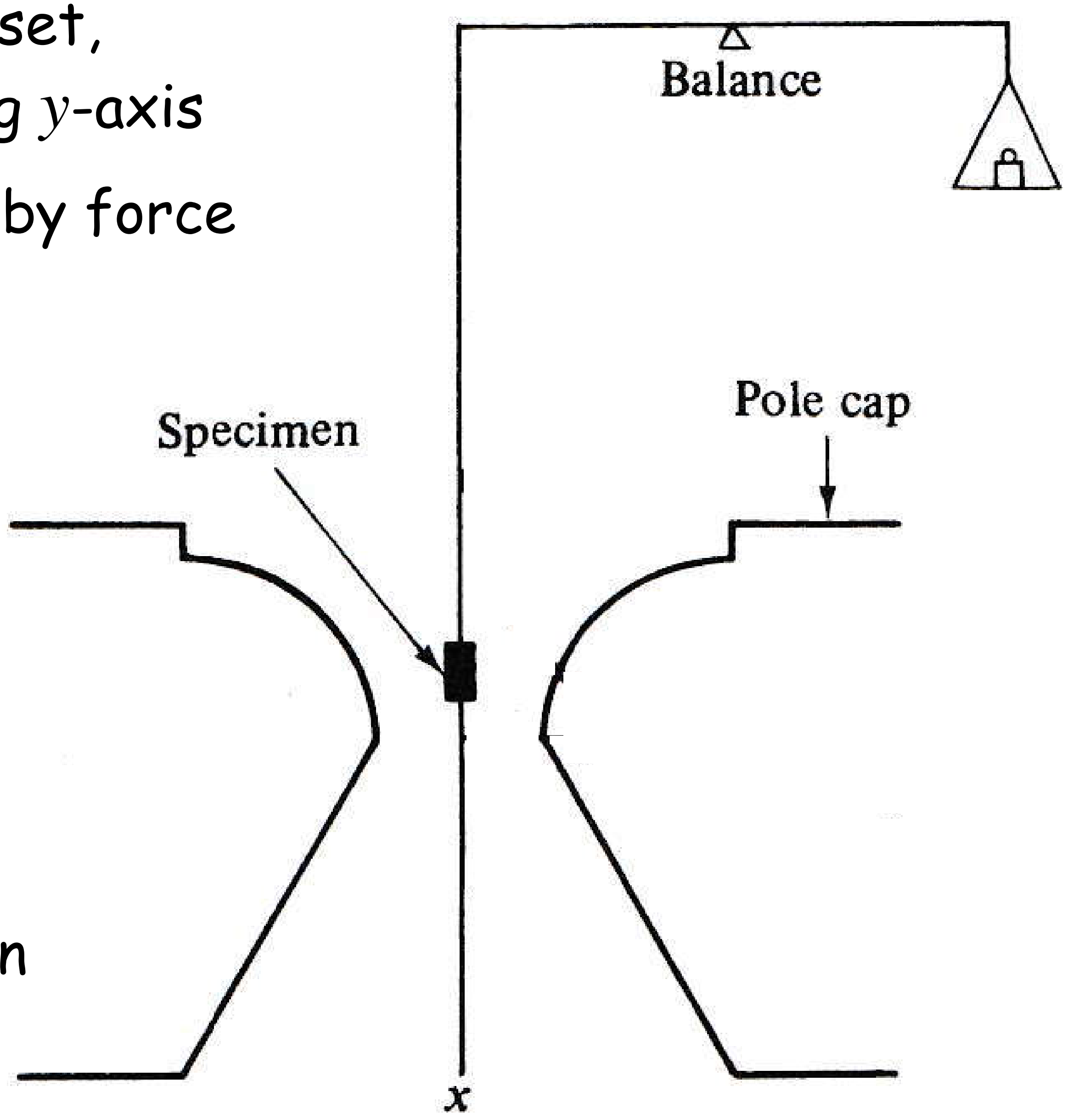


3.2 Mechanical Measurements

b) Field gradient methods

Faraday Balance

- Measures static force on sample in magnetic field gradient
- Uniform field is generated by electromagnet, and the gradient field is produced by an additional coil-set, optimized for a uniform magnetic gradient along y-axis
- If sample is mounted elastically, it is displaced by force
- Force can be detected and compensated by a calibrated electromagnetic counter-force. Compensation current is measure of force. Only quantity needed: sample volume
- Highly sensitive, can be applied to all kinds of magnetic substances
- In ferromagnetism it is best suited to measure the saturation magnetization with high precision



3.2 Mechanical Measurements

b) Field gradient methods

Alternating Gradient Magnetometer (AGM)

- AGM (or Vibrating Reed Magnetometer) resembles superficially the VSM
- VSM: sample agitated mechanically and electric signal is derived from motion.

  AGM: magnetically excited signal is recorded
- Sample mounted on elastic cantilever or "reed", which is excited into resonant vibration by alternating gradient field (produced by gradient coils in addition to dc magnet) by chosing resonance frequency
- Vibration is recorded by piezoelectric pick-up system: generates voltage proportional to vibration amplitude (proportional to force) = proportional to sample magnetic moment
- Sensitivity: 10^{-6} emu $=$ 10^{-9} Am^2 → higher than VSM

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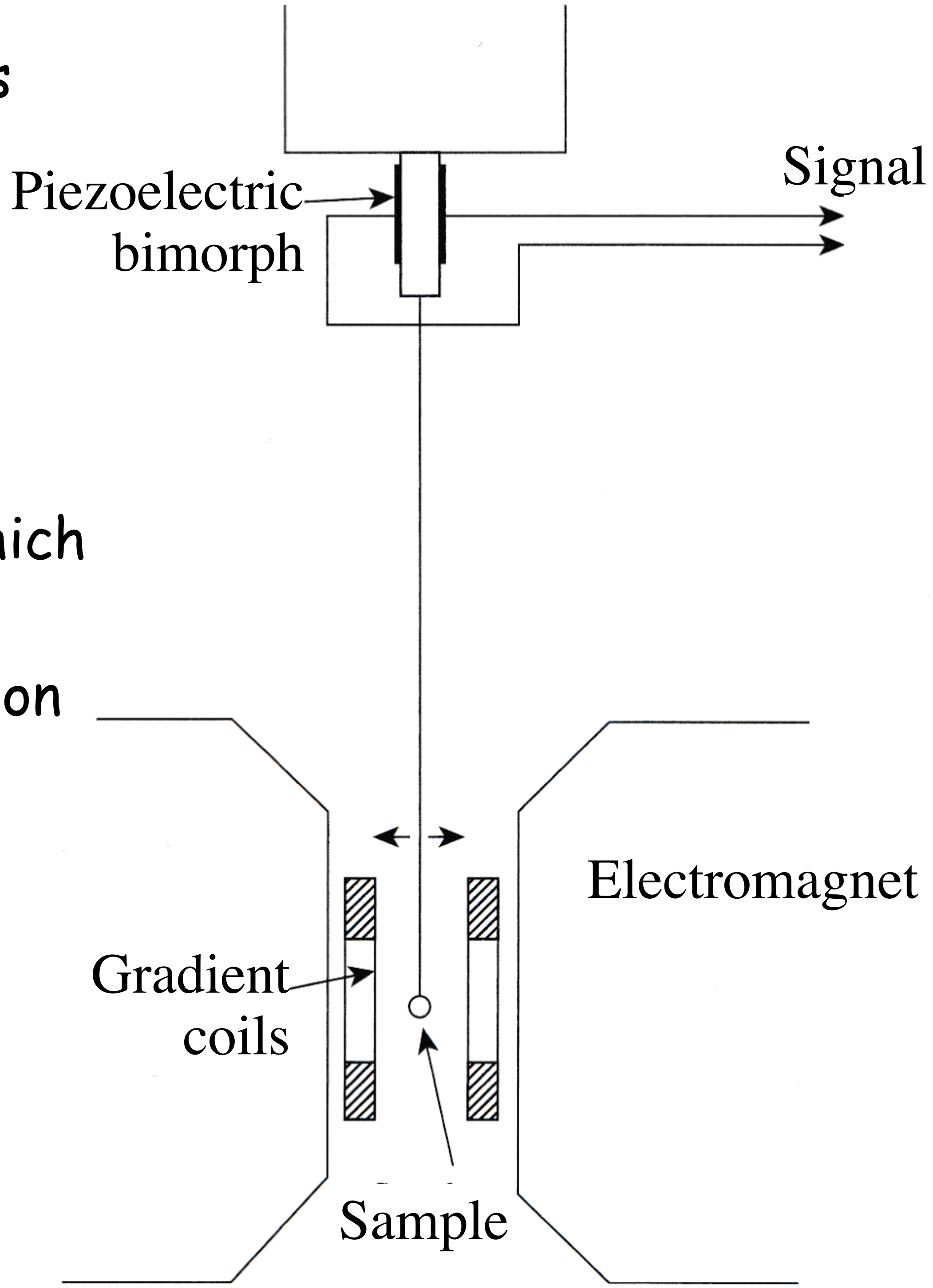

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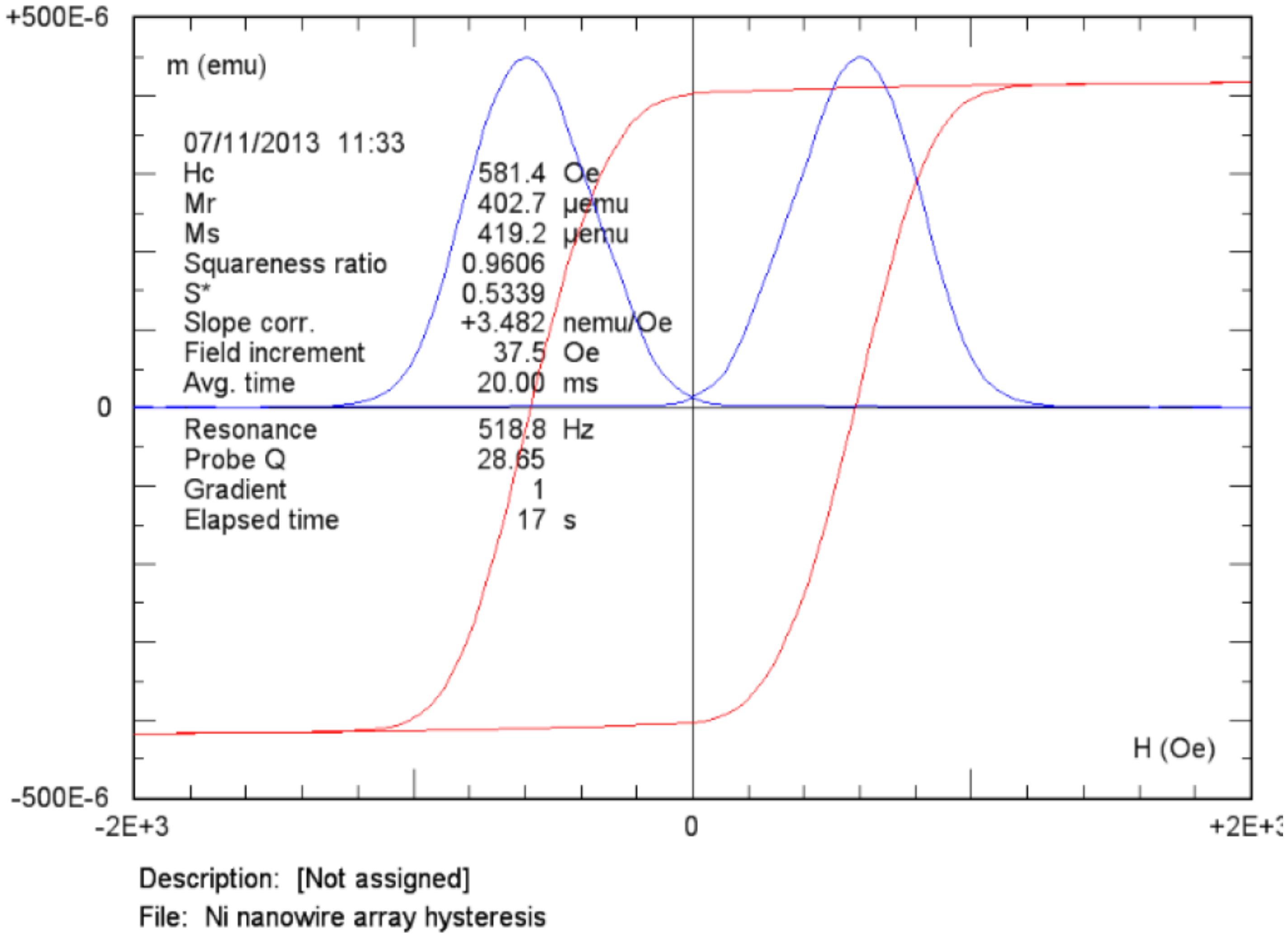

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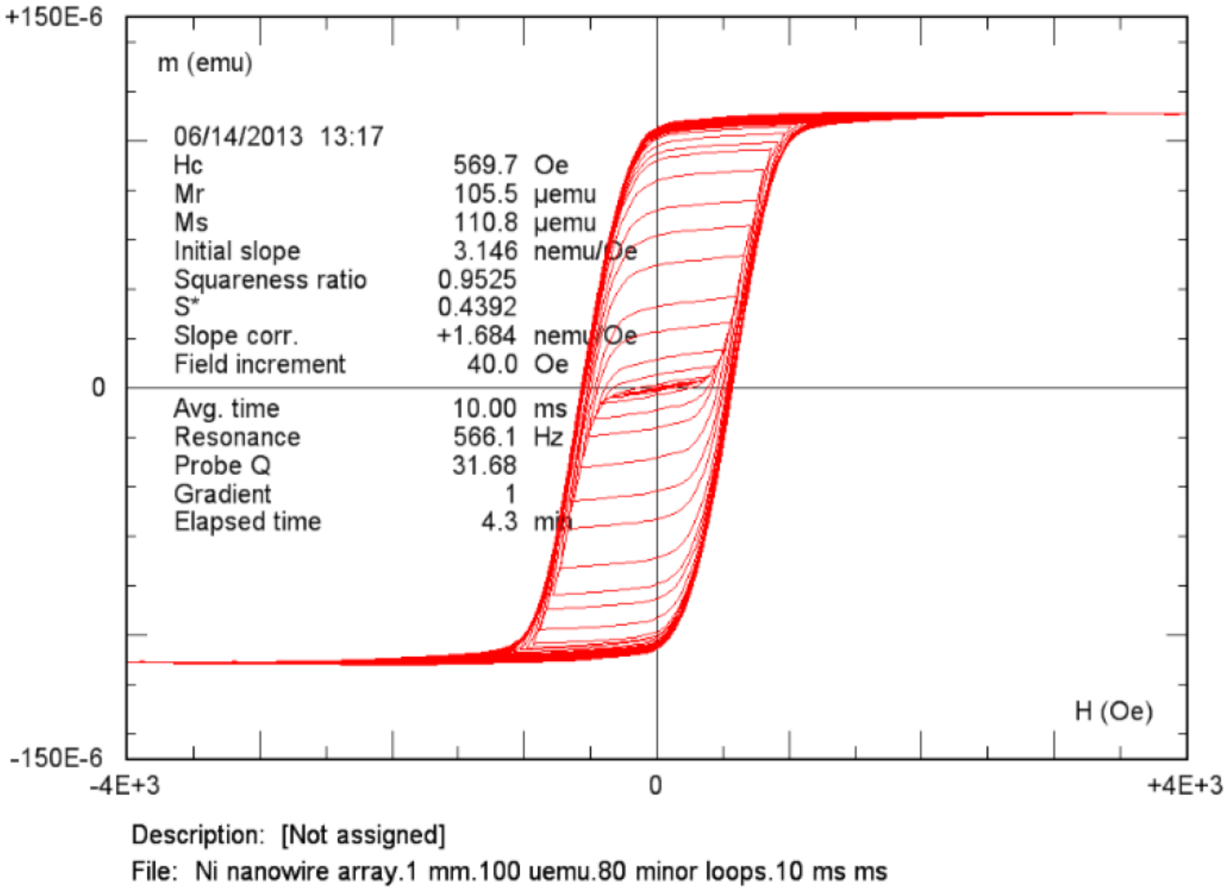
Hc	581.4 Oe
Mr	402.7 µemu
Ms	419.2 µemu
Squareness ratio	0.9606
S*	0.5339
Slope corr.	+3.482 nemu/Oe
Field increment	37.5 Oe
Avg. time	20.00 ms
Resonance	518.8 Hz
Probe Q	28.65
Gradient	1
Elapsed time	17 s

Description: [Not assigned]

File: Ni nanowire array hysteresis

Examples



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Hc	569.7 Oe
Mr	105.5 µemu
Ms	110.8 µemu
Initial slope	3.146 nemu/Oe
Squareness ratio	0.9525
S*	0.4392
Slope corr.	+1.684 nemu/Oe
Field increment	40.0 Oe
Avg. time	10.00 ms
Resonance	566.1 Hz
Probe Q	31.68
Gradient	1
Elapsed time	4.3 min

Description: [Not assigned]

File: Ni nanowire array.1 mm.100 uemu.80 minor loops.10 ms ms

Courtesy Lakeshore

3.

Measurements to determine magnetic material parameters & properties

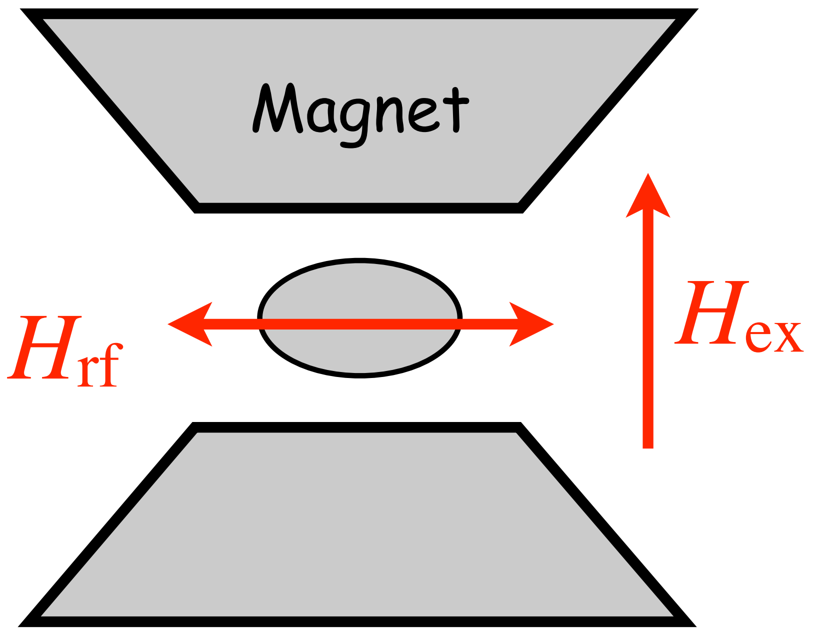
3.

Measurements to determine magnetic material parameters & properties

3.3 Resonance Techniques

Ferromagnetic Resonance (FMR)

- FMR is dynamic measurement method at microwave frequency (GHz-regime, order of Lamor frequency)
  - Because of high-frequency alternating magnetic field: eddy current shielding in metals is nearly complete (penetration depth of field ~ 200 nm)
    → resonance methods are not applicable to bulk metallic samples
    → can only be applied to non-conducting oxidic materials, thin films, and powdered materials
- Typical FMR resonance experiment:
  - Sample magnetized in strong **static** field H_{ex} to enforce uniformly magnetized state
  - To induce resonance phenomenen: alternating field with fixed GHz-frequency is superimposed at right angle to magnetization direction → stimulates precession of magnetization vector
  - The static field amplitude H_{ex} is swept till resonance is achieved at $H_{\mathrm{ex}} = H_{\mathrm{res}}$ (sweeping of field is easier than sweeping microwave frequency)



3.3 Resonance Techniques

Ferromagnetic Resonance (FMR)

- Resonance frequency:

$$\omega_{\mathrm{res}} = \gamma H_{\mathrm{res}} \qquad \text{with} \quad \gamma = \mu_0 g e / 2 m_{\mathrm{e}}$$

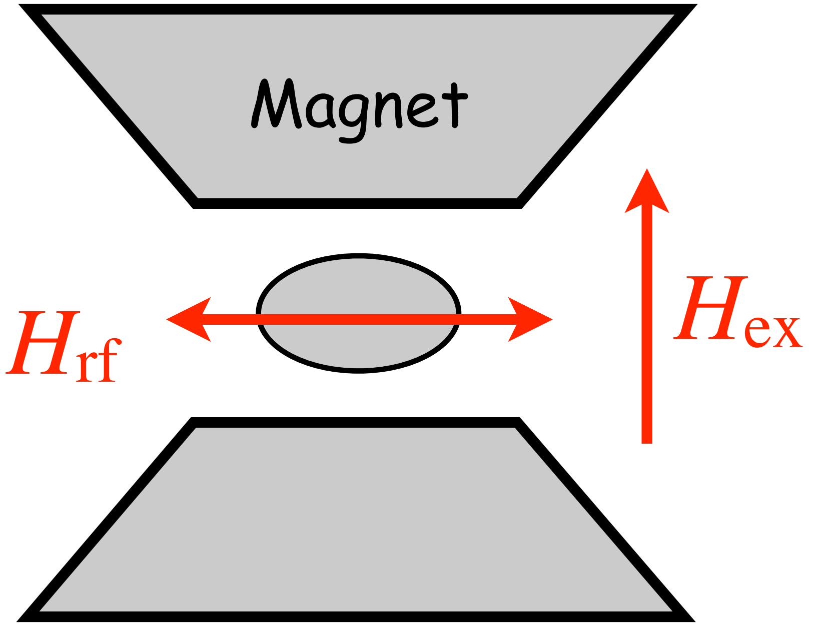
$$H_{\mathrm{res}} = \sqrt{\left[\frac{2K_{\mathrm{u}}}{\mu_0 M_{\mathrm{s}}} + H_{\mathrm{ex}} + M_{\mathrm{s}}\left(N_{\mathrm{b}} - N_{\mathrm{a}}\right)\right]\left[\frac{2K_{\mathrm{u}}}{\mu_0 M_{\mathrm{s}}} + H_{\mathrm{ex}} + M_{\mathrm{s}}\left(N_{\mathrm{c}} - N_{\mathrm{a}}\right)\right]}$$

Anisotropy field

External field

Demagnetizing field

- If M_{s} and N_{i} are known → anisotropy K_{u} can be derived from measuring H_{res}



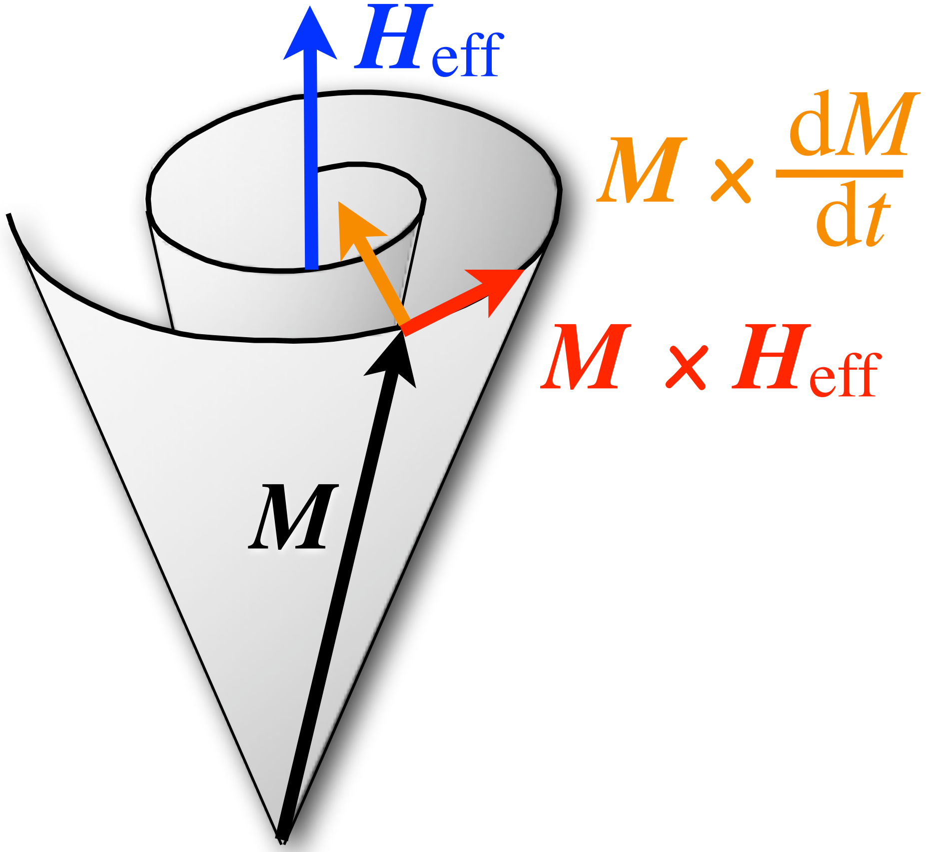
3.3 Resonance Techniques

Ferromagnetic Resonance (FMR)

- Based on LLG equation:

$$\frac{\mathrm{d}\boldsymbol{M}}{\mathrm{d}t} = -\gamma_0\ [\boldsymbol{M} \times \boldsymbol{H}_{\mathrm{eff}}] + \frac{\alpha}{M_{\mathrm{s}}}[\boldsymbol{M} \times \frac{\mathrm{d}\boldsymbol{M}}{\mathrm{d}t}]$$

  Desribes (damped) precession of $\boldsymbol{M}$ around effective field H_{eff}, caused by gyrotropic reaction of magnetic moment due to its angular momentum



- If the precession is uniform (does not depend on the position in sample): ferromagnetic resonance
- At higher frequencies, non-uniform precession modes may be excited:
  - If their wavelength is comparable with sample size: magnetostatic modes
  - Resonance phenomena at shorter wavelengths depend on the exchange stiffness constant and are called spin-wave modes
  - Finally, there are surface modes of magnetic resonance which can be excited for example by light instead of an alternating field. Then spectroscopy replaces the standard inductive detection methods

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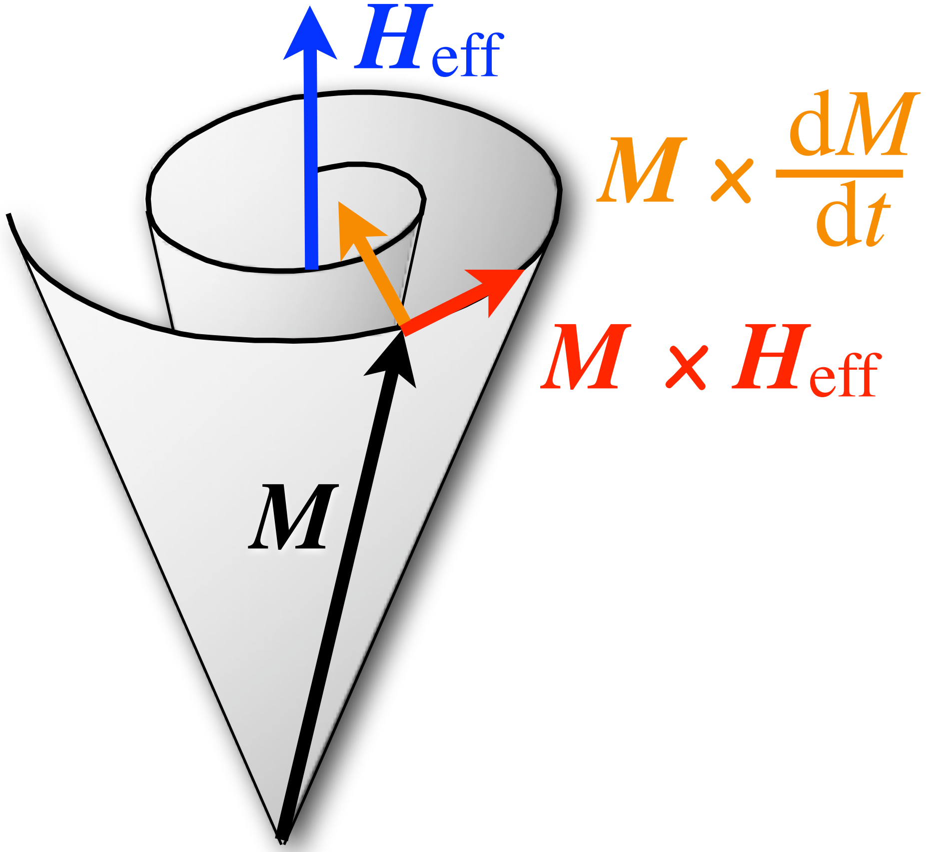

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→ **Lecture of B. Hillebrands**

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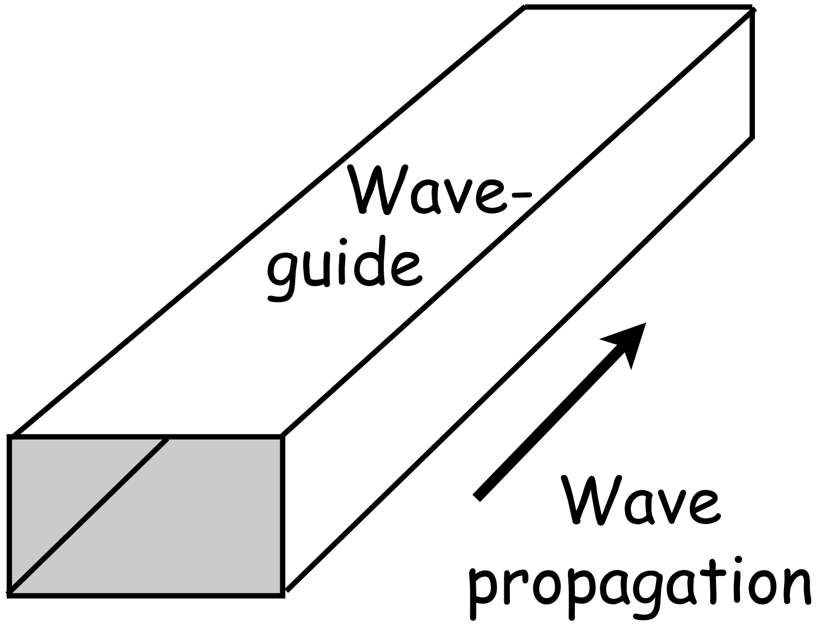
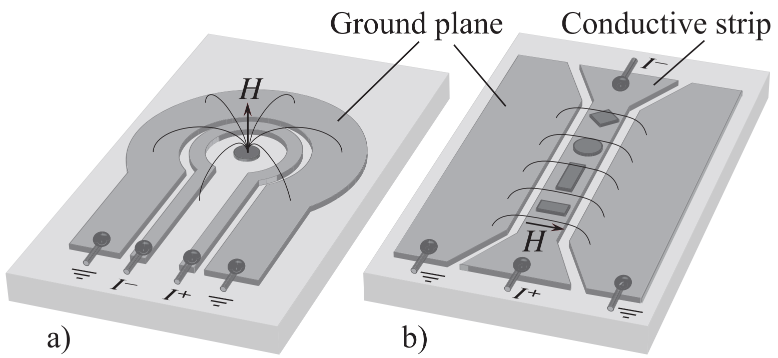




- FMR experiment:
  - Microwave field is generated by Klystron or Gunn diode and conducted to sample by wave guide (hollow metal tube)
  - Sample centered in microwave cavity (closed hollow metal structure), located in electromagnet

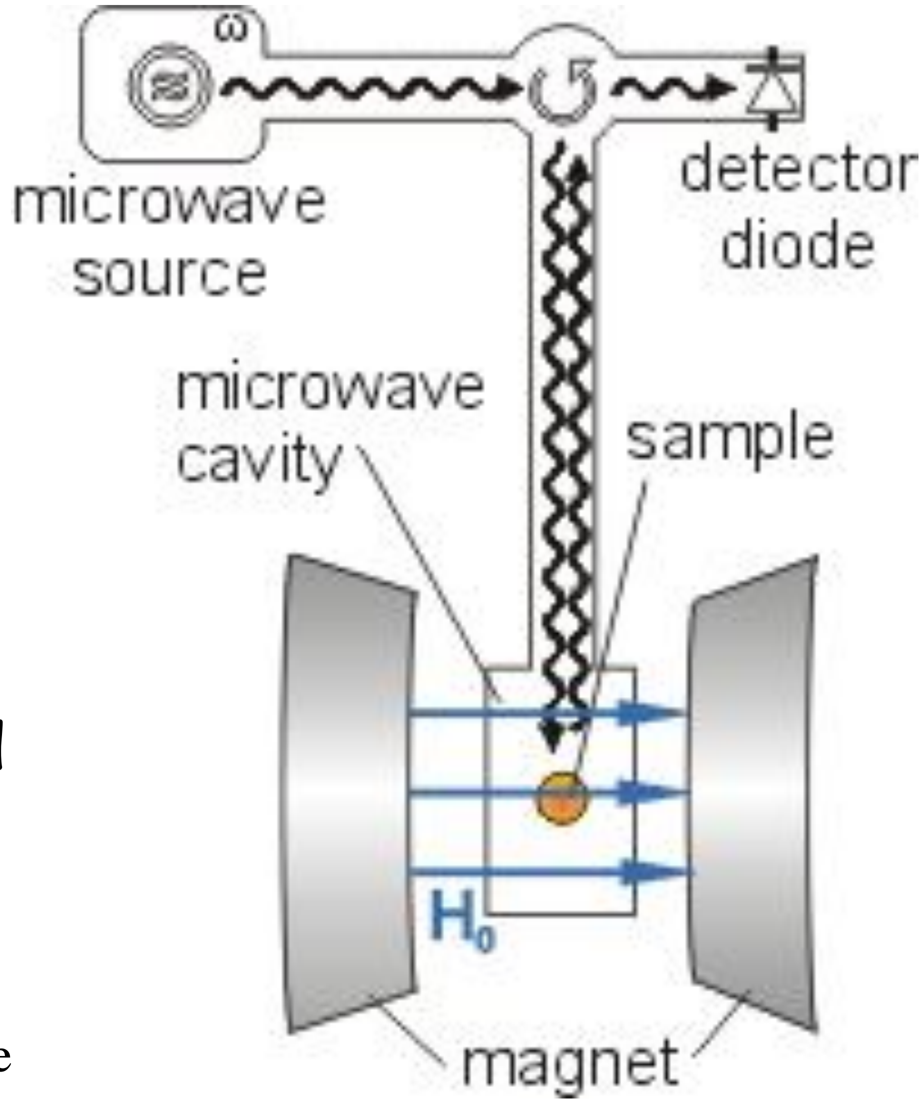
    → Microwave frequency fixed and determined by klystron and cavity. Alternativly sample can be placed on micro-stripline
  - Incident microwave signal couples to sample and is partially absorbed in dependence of field strength H_{ex}
  - If resonance condition is fulfilled → max. absorption of microwave power by sample
  - Crystal detector measures reflected microwave signal. This signal is input into lock-in amplifier where it is compared with reference signal
  - Output of lockin vs. external field: corresponds to derivative of absorption curve, i.e the change of absorbed intensity I as function of H_{ex}: $\mathrm{d}I/\mathrm{d}H_{\mathrm{ex}}$

3.3 Resonance Techniques

Ferromagnetic Resonance (FMR)

- FMR experiment:
  - Microwave field is generated by Klystron or Gunn diode and conducted to sample by wave guide (hollow metal tube)
  - Sample centered in microwave cavity (closed hollow metal structure), located in electromagnet

    → Microwave frequency fixed and determined by klystron and cavity. Alternativly sample can be placed on micro-stripline
  - Incident microwave signal couples to sample and is partially absorbed in dependence of field strength H_e



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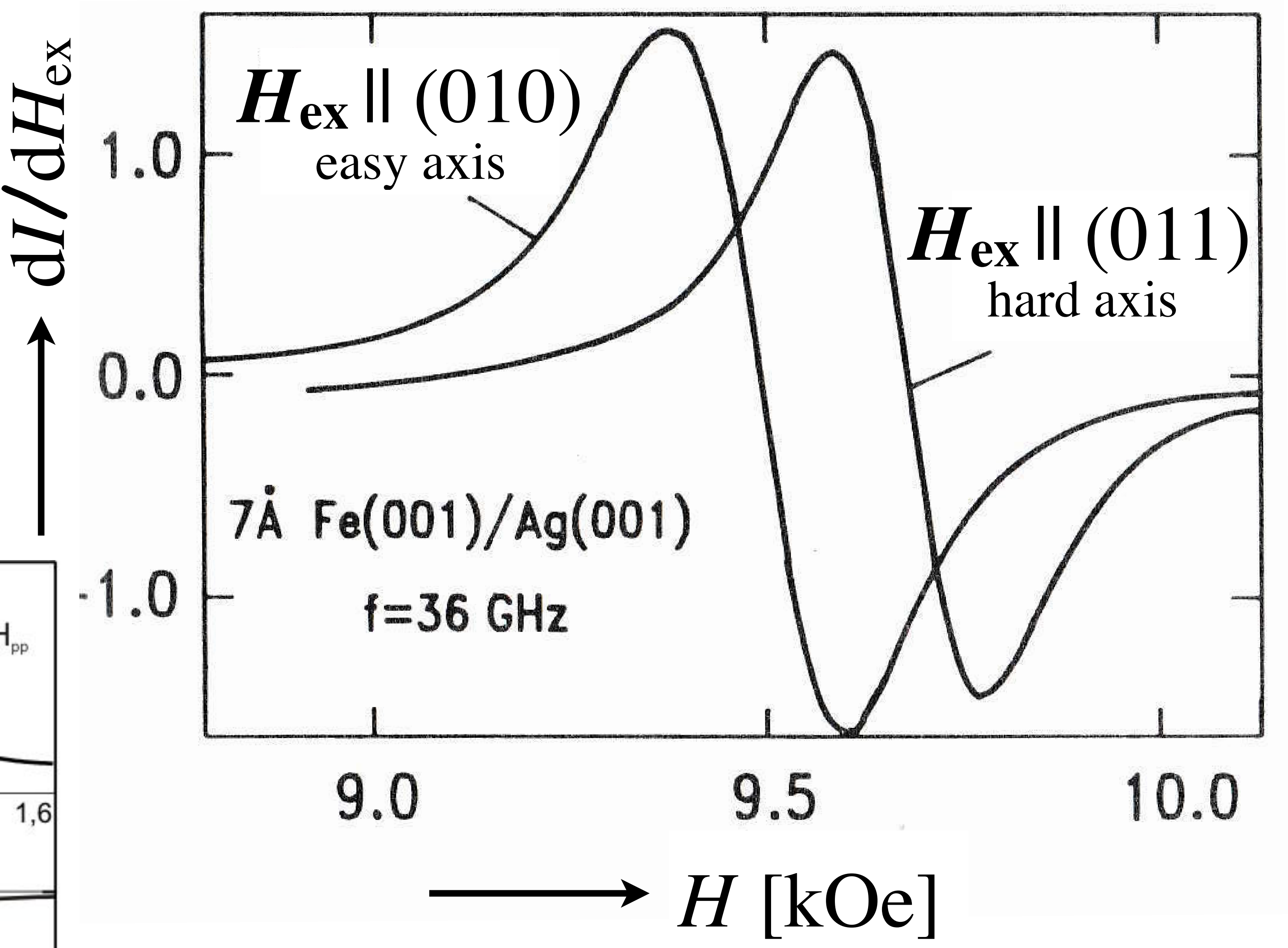
  - Output of lockin vs. external field: corresponds to derivative of absorption curve, i.e the change of absorbed intensity I as function of H_{ex}: dI/dH_{ex}

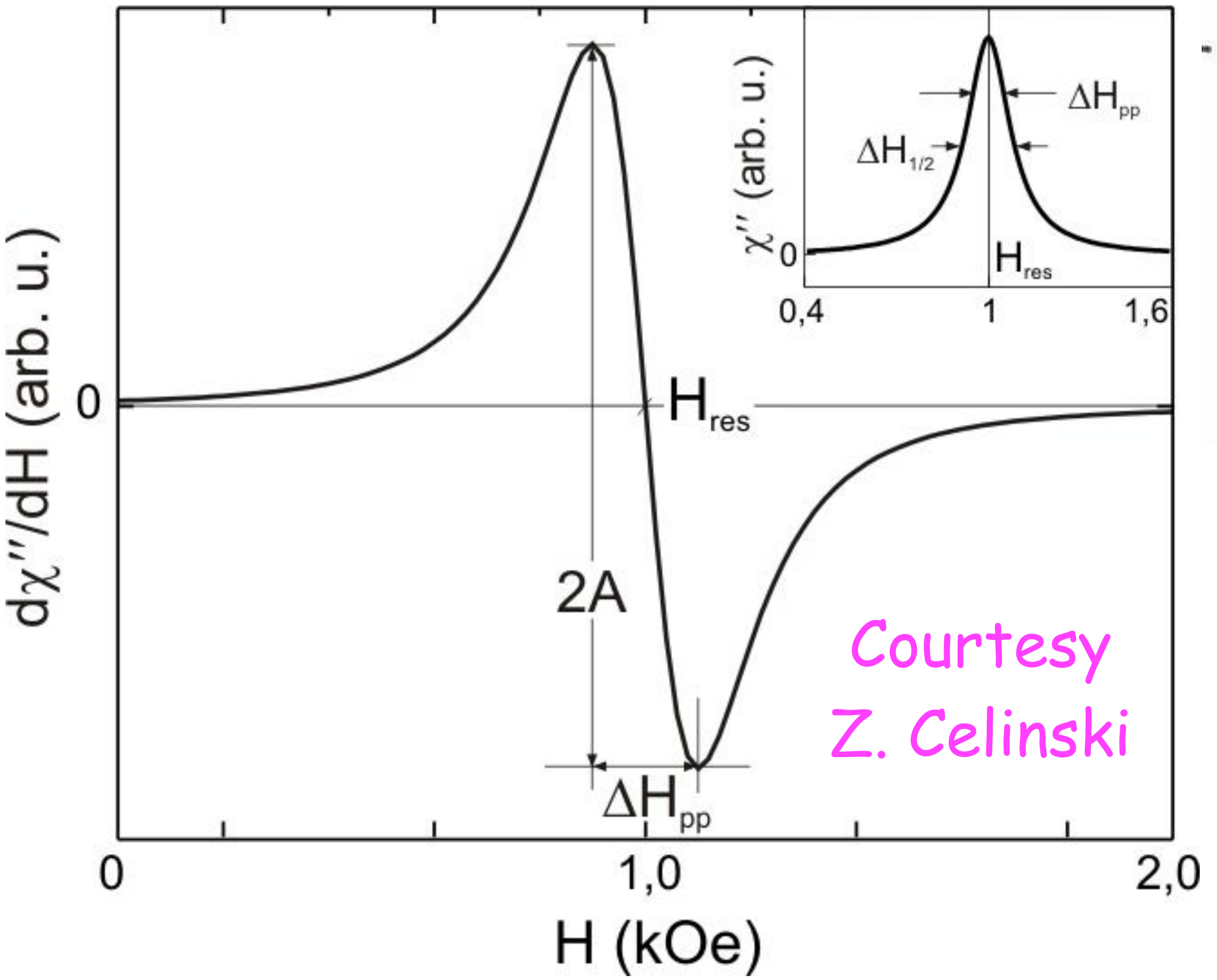
3.3 Resonance Techniques

Ferromagnetic Resonance (FMR)

- Resonance field H_{res} and half power line width ΔH can be determined
- H_{res} → magnetic anisotropy
- ΔH → relaxation
- Intensity → M_{s}



B. Heinrich et al., PRL **59**, 1756 (1987)



Courtesy Z. Celinski

3.3 Resonance Techniques

Spinwave Resonance

Light Scattering experiments

3.3 Resonance Techniques

Spinwave Resonance

Light Scattering experiments

→ Lecture of B. Hillebrands

3.

Measurements to determine magnetic material parameters & properties

3.

Measurements to determine magnetic material parameters & properties

3.1 Magnetic measurements

3.2 Mechanical measurements

3.3 Resonance techniques

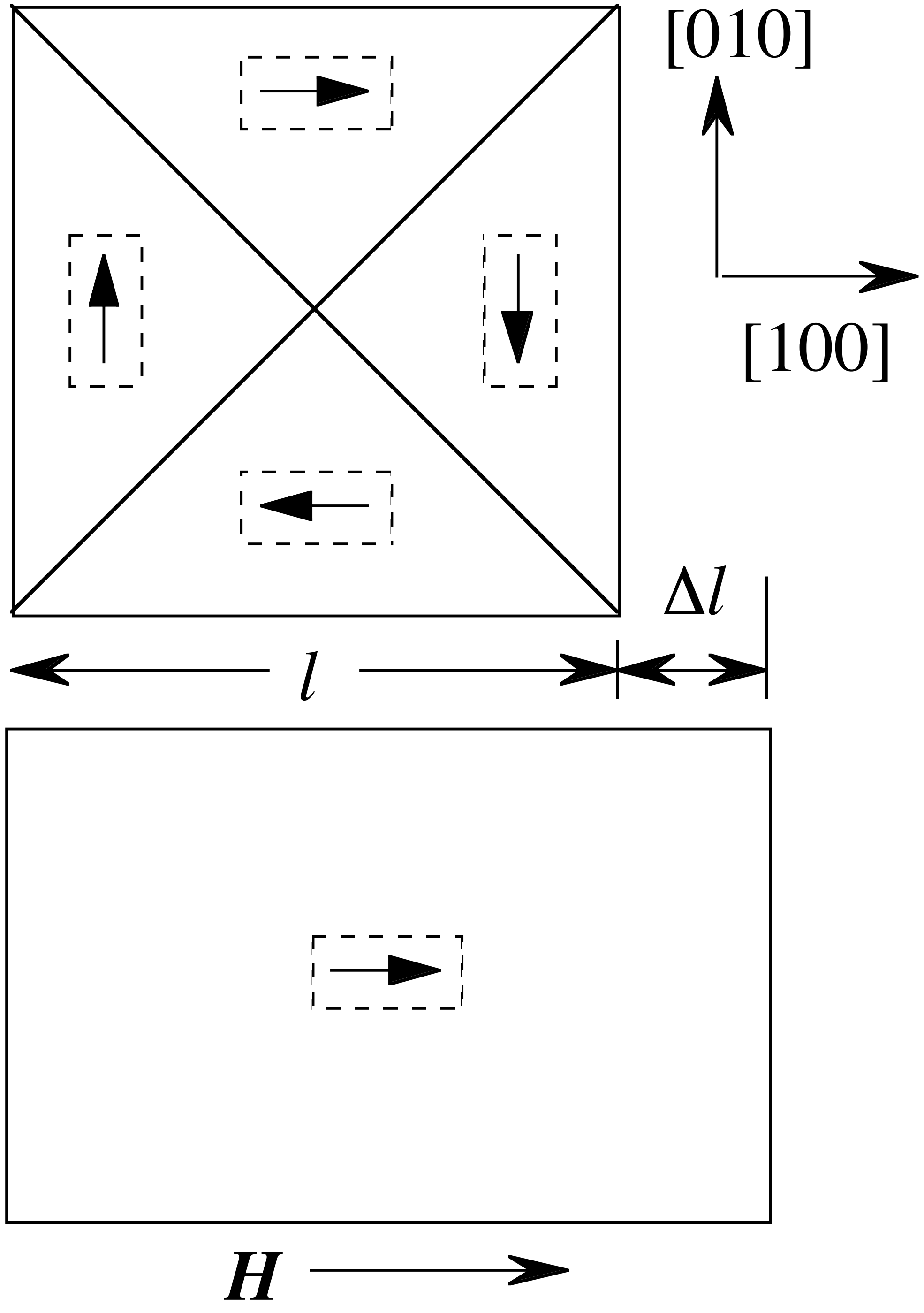
3.4 Dilatometric measurements

3.5 Domain methods

3.4 Dilatometric Measurements

Magnetostriction

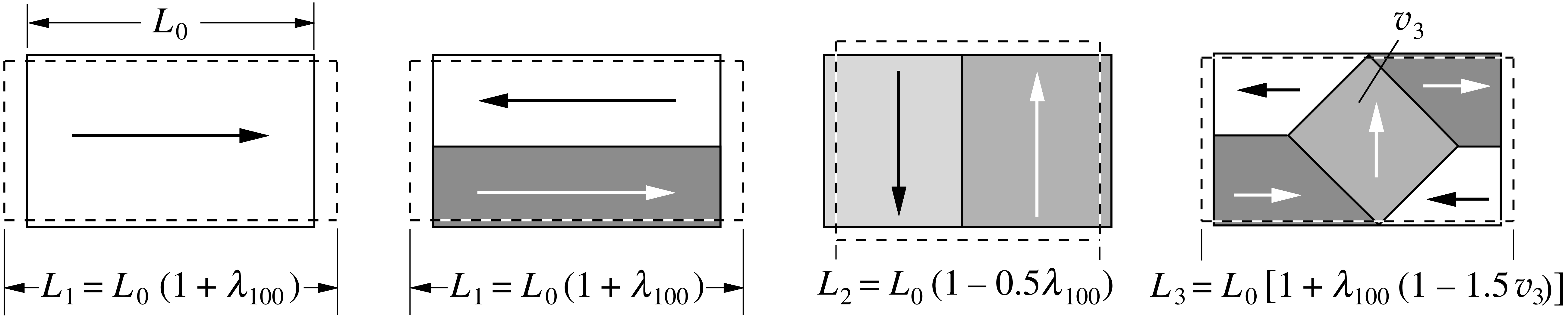
- Example: cubic crystal (iron) with 4 domain phases in ground state → no net-magnetization
- Each domain elongates crystal along M-direction → cubic lattice gets tetragonally distorted
- Application of magnetic field, saturation → elongation of total crystal
- Relative length change $\Delta l / l = \lambda_s$
  $\lambda_s =$ magnetostriction coefficient
- $\lambda_s > 0$: elongation in magnetization direction,
  $\lambda_s < 0$: compression
- Note: 180°-domains are elongated along same axis
- λ for iron: $20 \cdot 10^{-6}$ → very small effect



3.4 Dilatometric Measurements

Magnetostriction measurement, remark

- Every direct determination of magnetostriction constant requires measurement of length change between two different saturated states — usually parallel and perpendicular to measuring direction. Experiment with field applied along only one axis of crystal yields no useful information on magnetostriction constants, because then only some arbitrary demagnetized state and the saturated states are compared



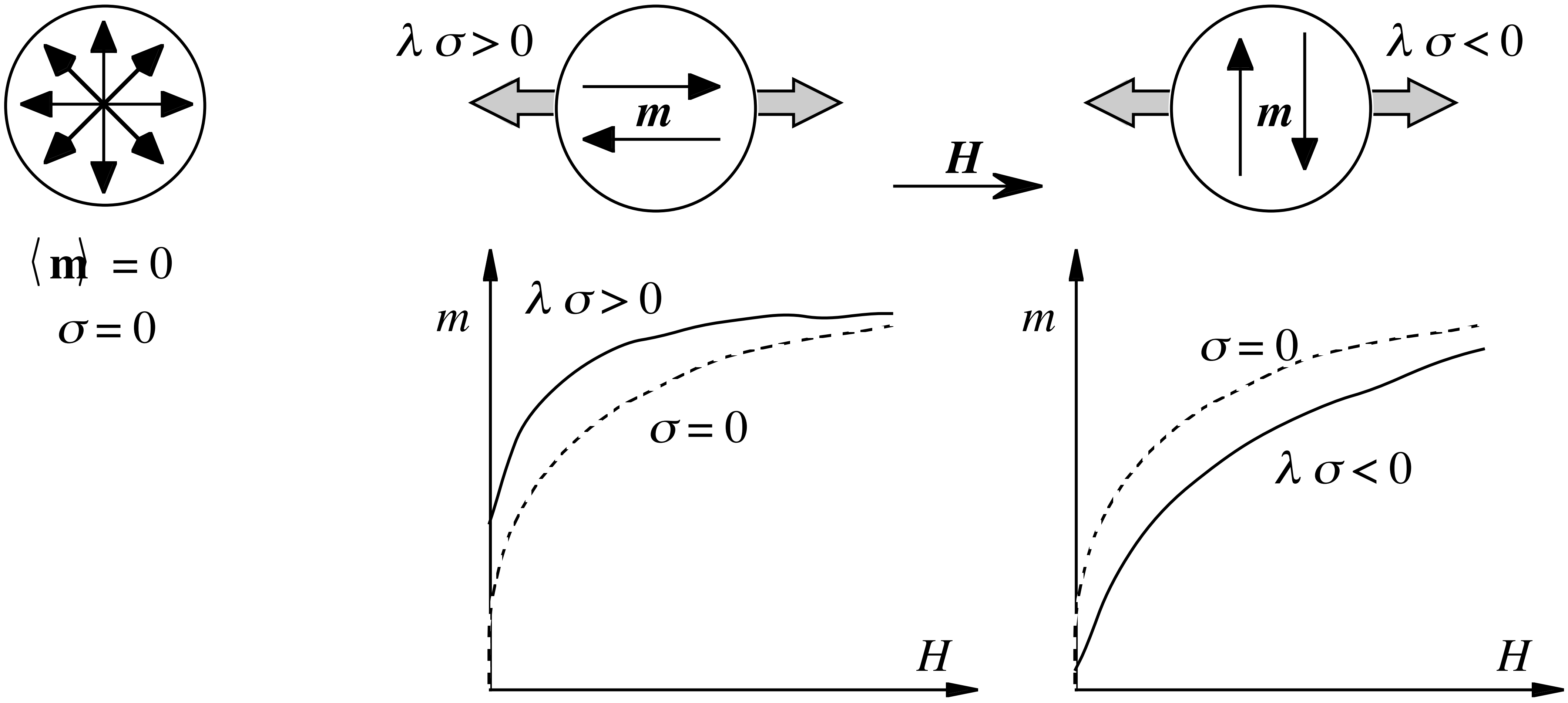
L_0: length of hypothetic non-magnetic state

3.4 Dilatometric Measurements

Magnetostriction measurement

- Saturation magnetostriction constants can be determined both directly or indirectly:
  - **Indirect methods**: stress sensitivity of a suitable magnetic property is analysed, more suitable for thin films and wires

    Example: magnetization curve and resonance measurements, if performed as function of external stress, can be evaluated in terms of magneto-elastic coefficients



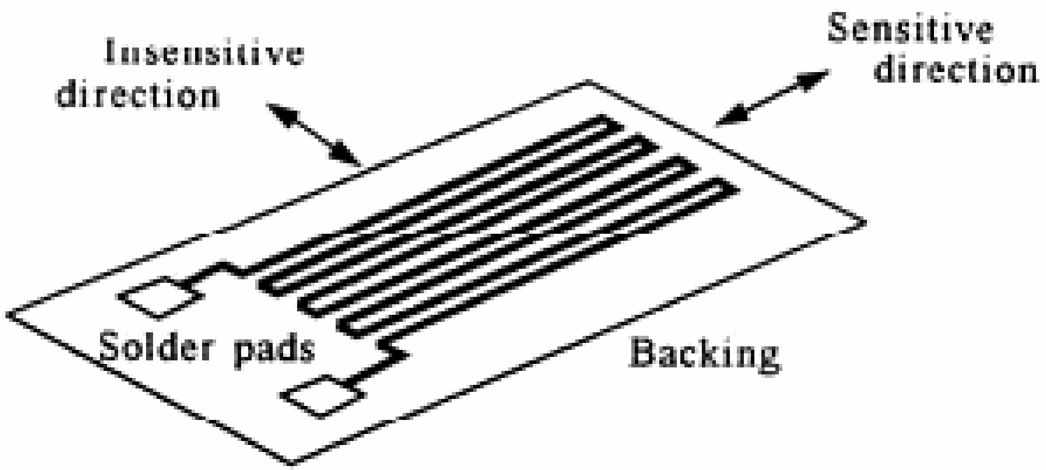
3.4 Dilatometric Measurements

Magnetostriction measurement

- Saturation magnetostriction constants can be determined both directly or indirectly:
  - **Direct measurements**: evaluate elongation of a magnet depending on the magnetization direction, preferred for bulk samples of sufficient size

Strain Gauges

- Plastic foil on which structured metal films act as sensors, cemented on sample
- Based on (small) change of el. resistance by elongation, measured with bridge circuit
- Strain gauge can be applied locally on favourable small region of single crystal or even on grain in a coarse-grained sample. Active area down to 1 mm^2
- It is always advisable to apply strain gauges on both sides of sample, and to connect them in series to avoid influence from sample bending induced by one-sided heating by the measuring current



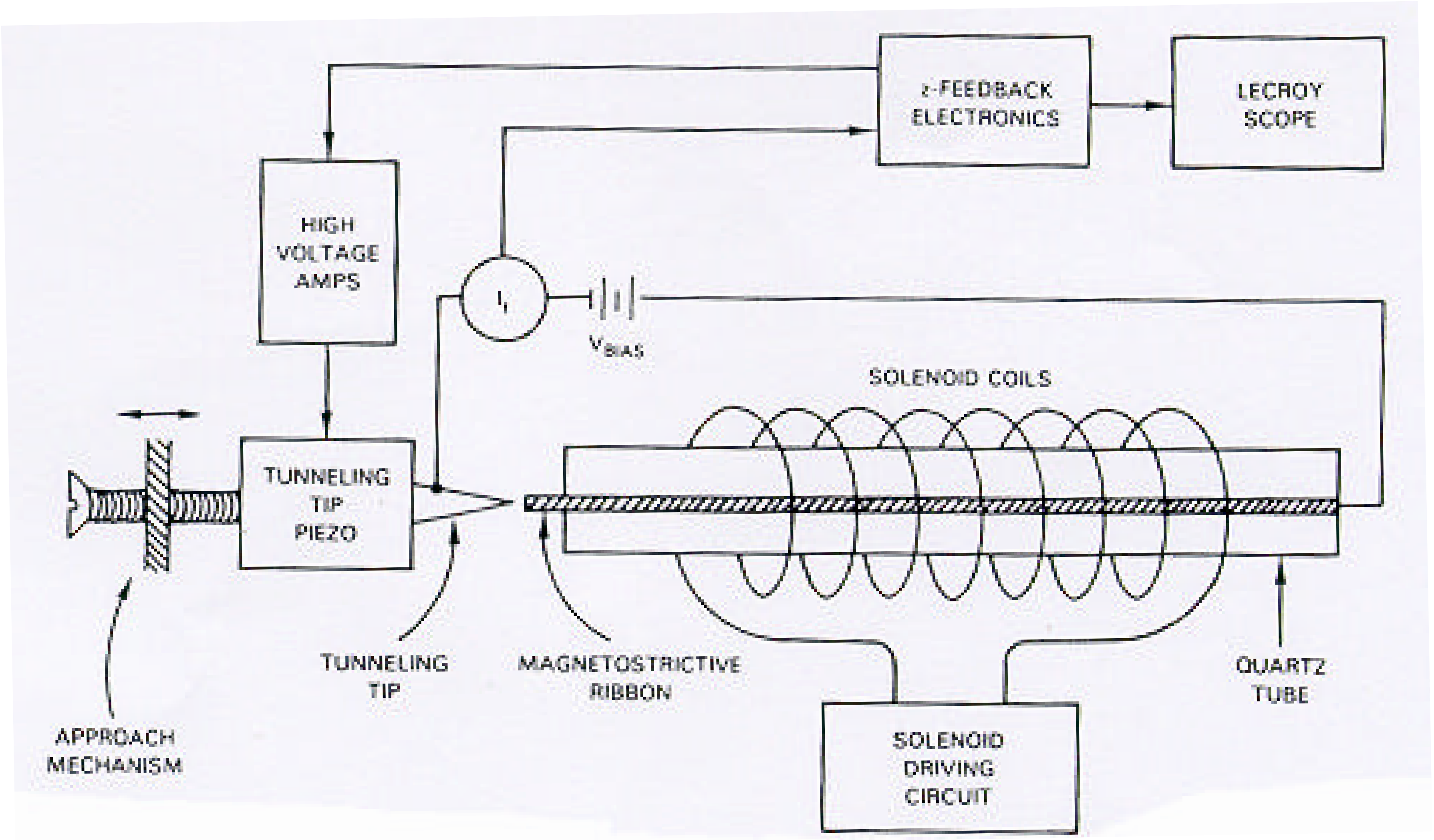
3.4 Dilatometric Measurements

Magnetostriction measurement

- Saturation magnetostriction constants can be determined both directly or indirectly:
  - **Direct measurements**: evaluate elongation of a magnet depending on the magnetization direction, preferred for bulk samples of sufficient size

  Dilatometers

  - Capacitive sensors, optical interferometers, tunnelling sensor, piezo sensor…
  - With normal samples of centimetre dimension a resolution in the nanometre range is required



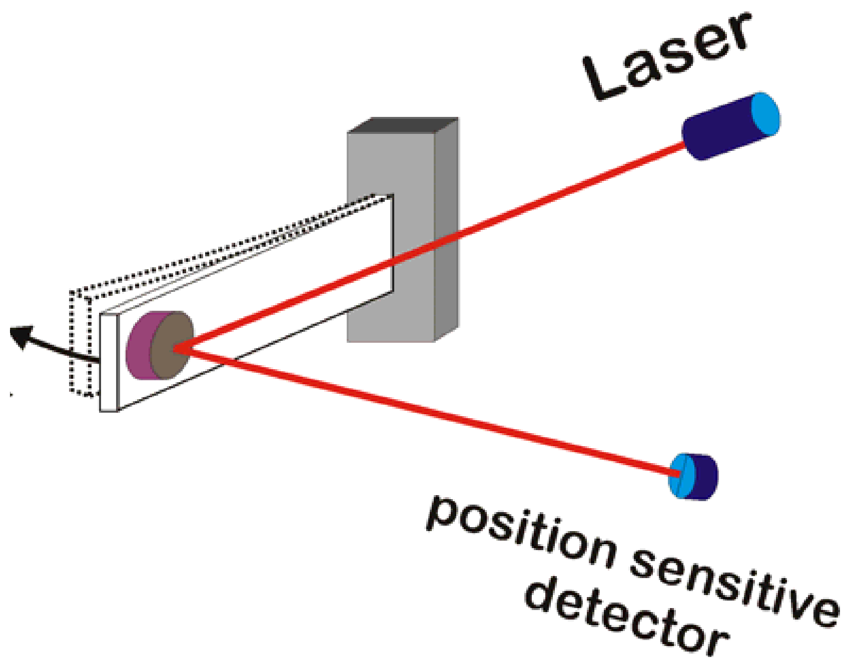
3.4 Dilatometric Measurements

Magnetostriction measurement

- Saturation magnetostriction constants can be determined both directly or indirectly:
  - **Direct measurements**: evaluate elongation of a magnet depending on the magnetization direction, preferred for bulk samples of sufficient size

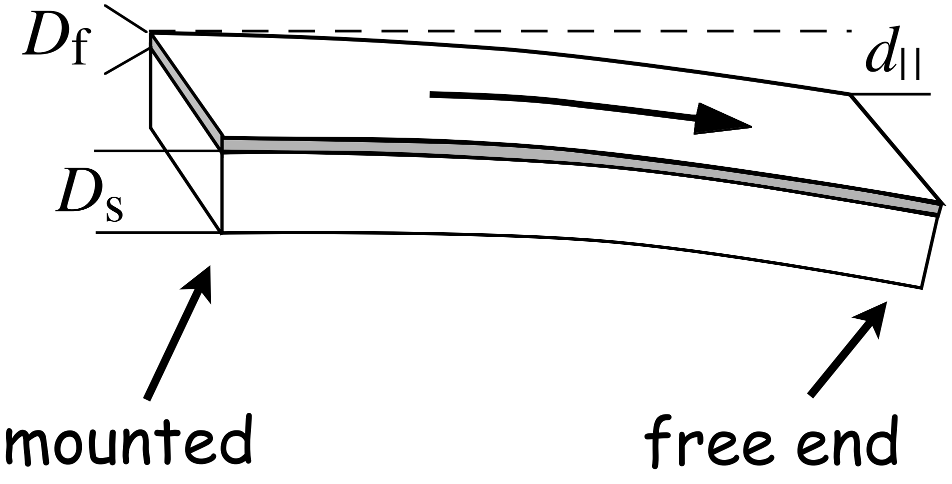
Cantilever for films

- Magnetostrictive bending of a sample-substrate-composite in magnetic field is optically detected as measuring signal by reflected laser beam, a quadrant detector and lock-in technique
- Sample esposed to rotating saturation field
- Deflection difference d_f of free end of cantilever for longitudinal and transverse magnetization:



$$d_\mathrm{f} = d_{||} - d_{\perp} = \frac{3 D_\mathrm{f} L^2}{D_\mathrm{s}^2} \frac{E_\mathrm{f}(1+\nu_\mathrm{s})}{E_\mathrm{s}(1+\nu_\mathrm{f})} \cdot \frac{3}{2} \lambda_\mathrm{s} \sin^2 \vartheta$$

E_x and ν_x : Young's moduli and Poisson's ratios of film and substrate, θ : magnetization angle ($\theta = 0$ for magnetization along cantilever)



3.

Measurements to determine magnetic material parameters & properties

3.

Measurements to determine magnetic material parameters & properties

3.1 Magnetic measurements

3.2 Mechanical measurements

3.3 Resonance techniques

3.4 Dilatometric measurements

3.5 Domain methods

3.5 Domain Methods

Magnetic energies

Energy term	Coefficient	Definition	Range
Exchange energy	A [J/m]	Material constant	$10^{-12} - 2 \cdot 10^{-11}$ J/m
Anisotropy energies	K_u, K_c ... [J/m^3]	Material constants	$\pm(10^2 - 2 \cdot 10^7)$ J/m^3
External field energy	$H_{ex} J_s$ [J/m^3]	H_{ex} = external field J_s = saturation magnetization	Open, depending on field magnitude
Stray field energy	K_d [J/m^3]	$K_d = J_s^2/2\,\mu_0$	$0 - 3 \cdot 10^6$ J/m^3
External stress energy	$\sigma_{ex}\lambda$ [J/m^3]	σ_{ex} = external stress λ = magnetostriction constant	Open, depending on stress magnitude
Magnetostrictive self energy	$C\,\lambda^2$ [J/m^3]	C = shear modulus	$0 - 10^3$ J/m^3

3.5 Domain Methods

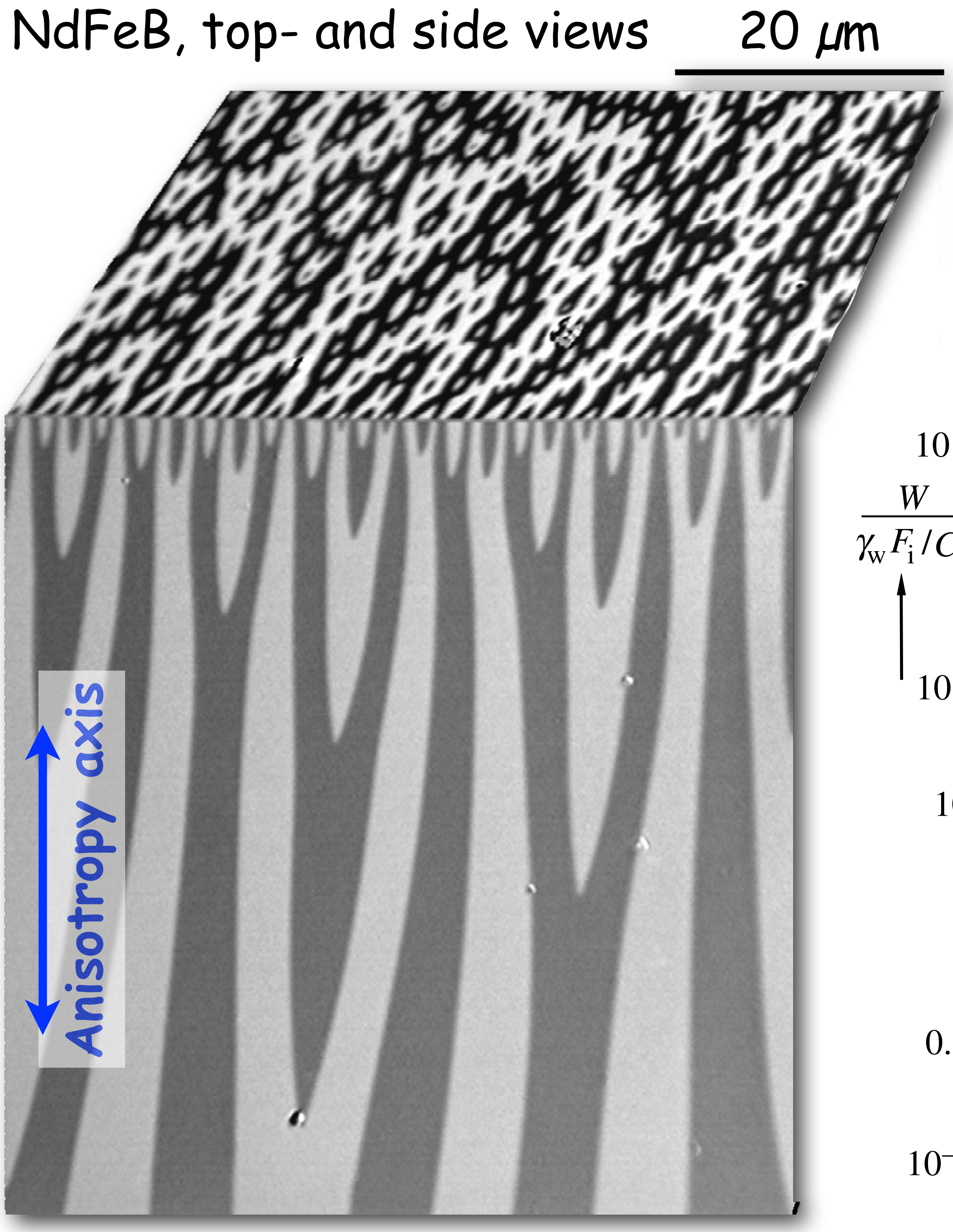
Principle:

Under favourable circumstances material constants may be derived directly from observed domains. Such approach requires an equilibrium situation for which a reliable theoretical treatment is possible.

3.5 Domain Methods

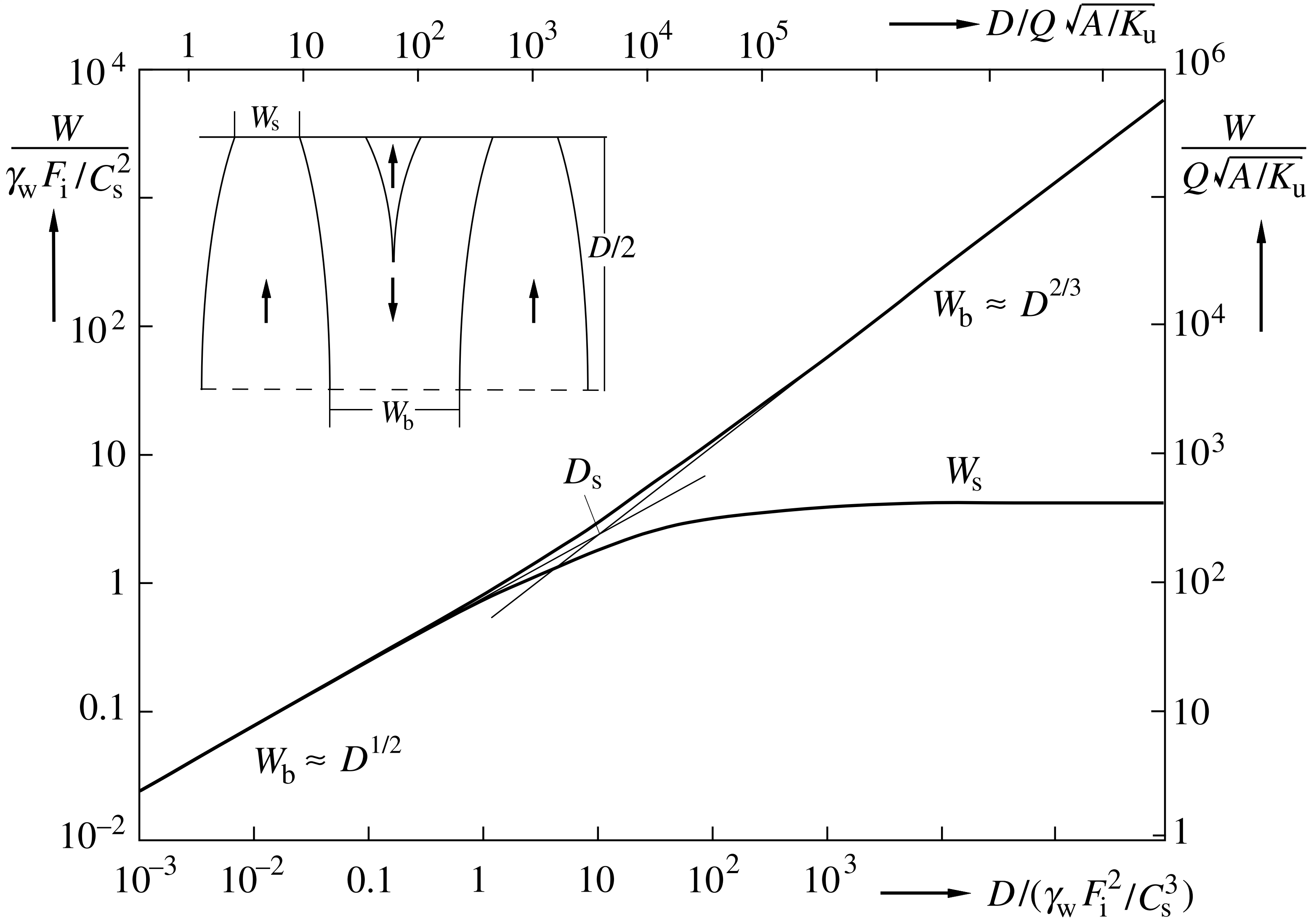
Example: Surface domain width in bulk uniaxial crystals

NdFeB, top- and side views



For sufficiently thick crystals: surface domain width W_s constant, independent of sample dimensions, depends only on wall energy γ_W and stray field energy constant $K_d = \mu_0 M_S^2 / 2$:

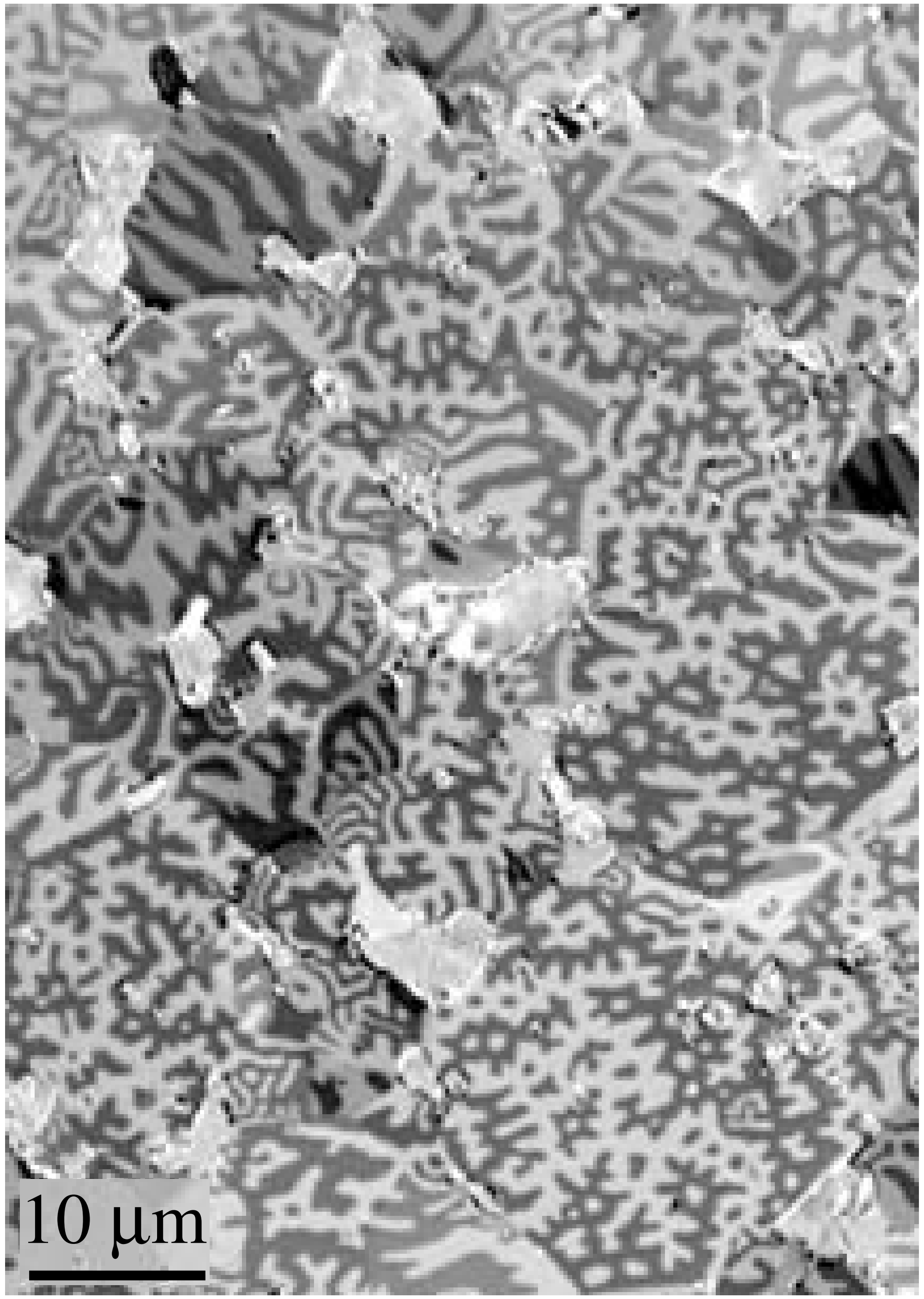
$$\boxed{W_s = 108\ \gamma_W / K_d} \qquad \gamma_W = 4\sqrt{(AK_u)}$$



3.5 Domain Methods

Example: Surface domain width in bulk uniaxial crystals

NdFeB magnet,
c-axis perpendicular



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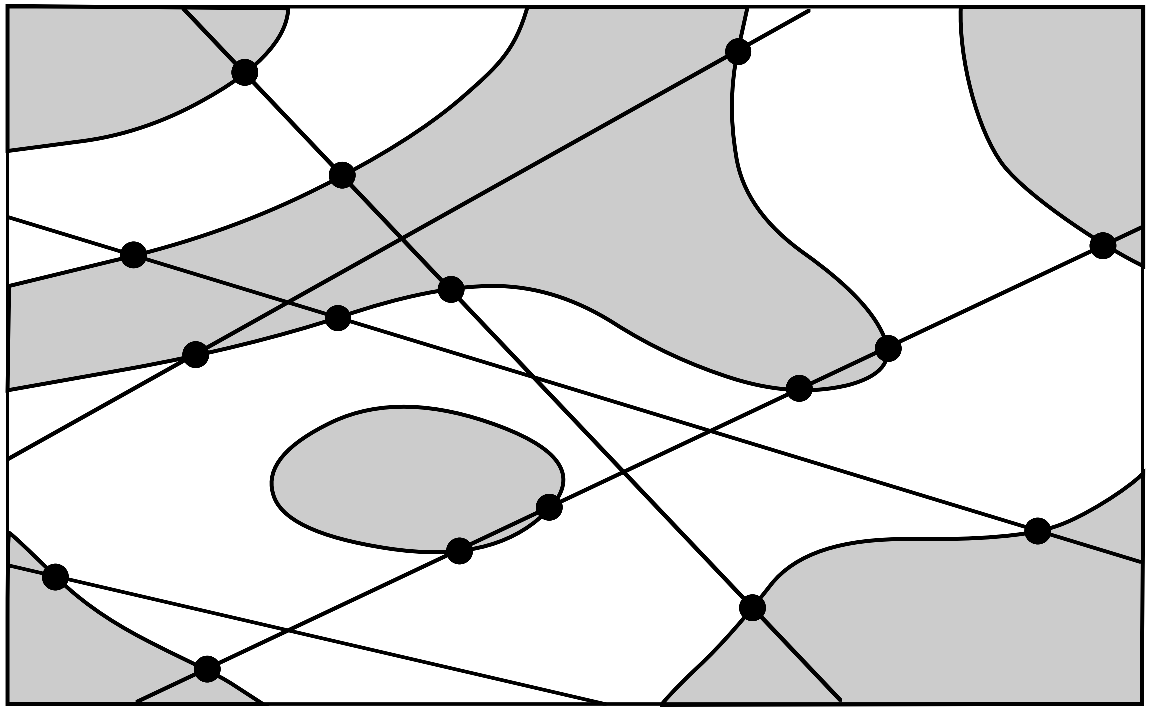
$$W_s = 108\ \gamma_W/K_d$$

$$\gamma_W = \pi\,\sqrt{A/K_u}$$

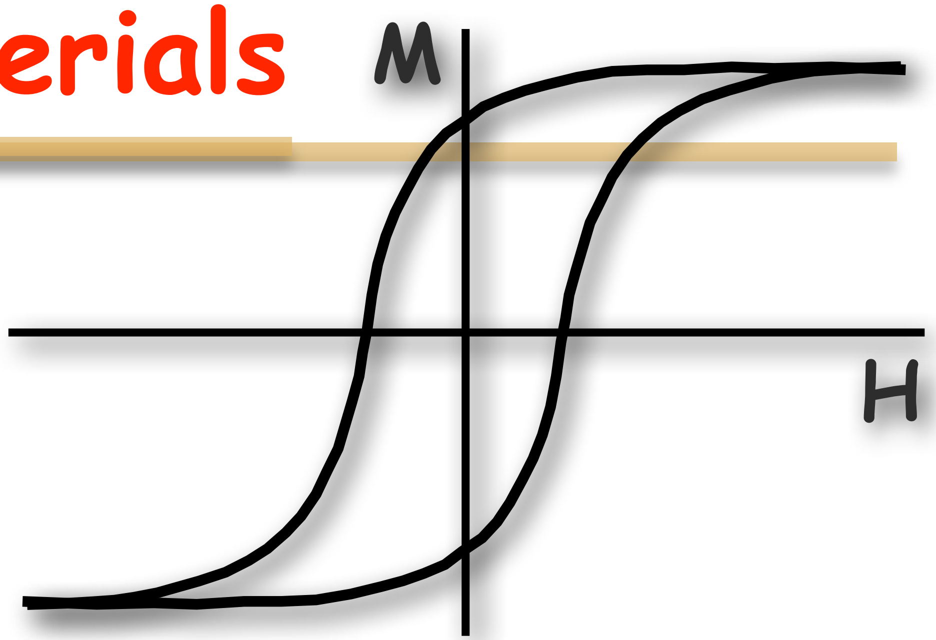


$$\text{Domain width} = \frac{2 \cdot \text{Total test line length}}{\pi \cdot \text{Number of intersections}}$$

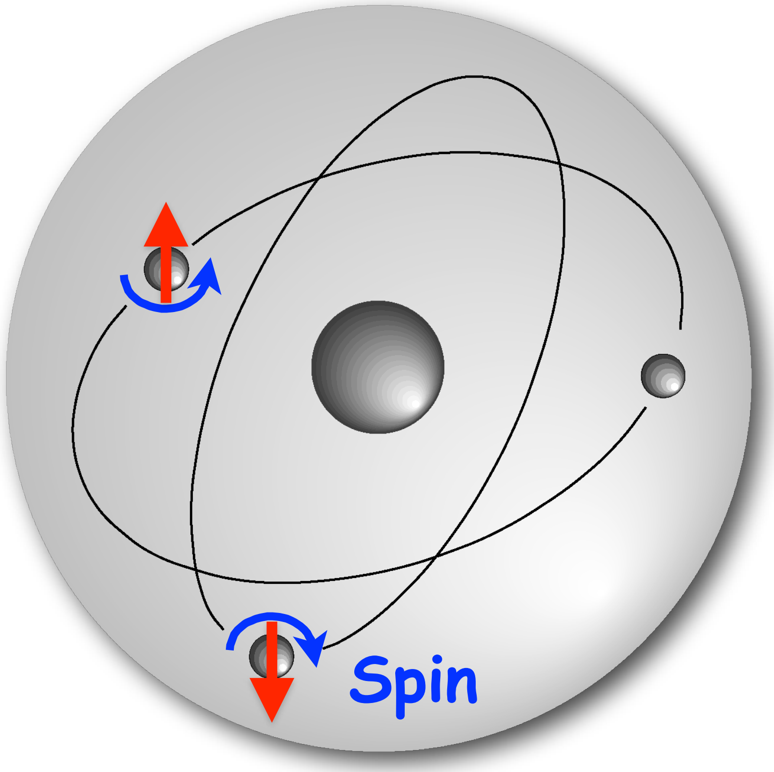
4.

Domain scale measurements (Magnetic Imaging)

Descriptive levels of magn. materials

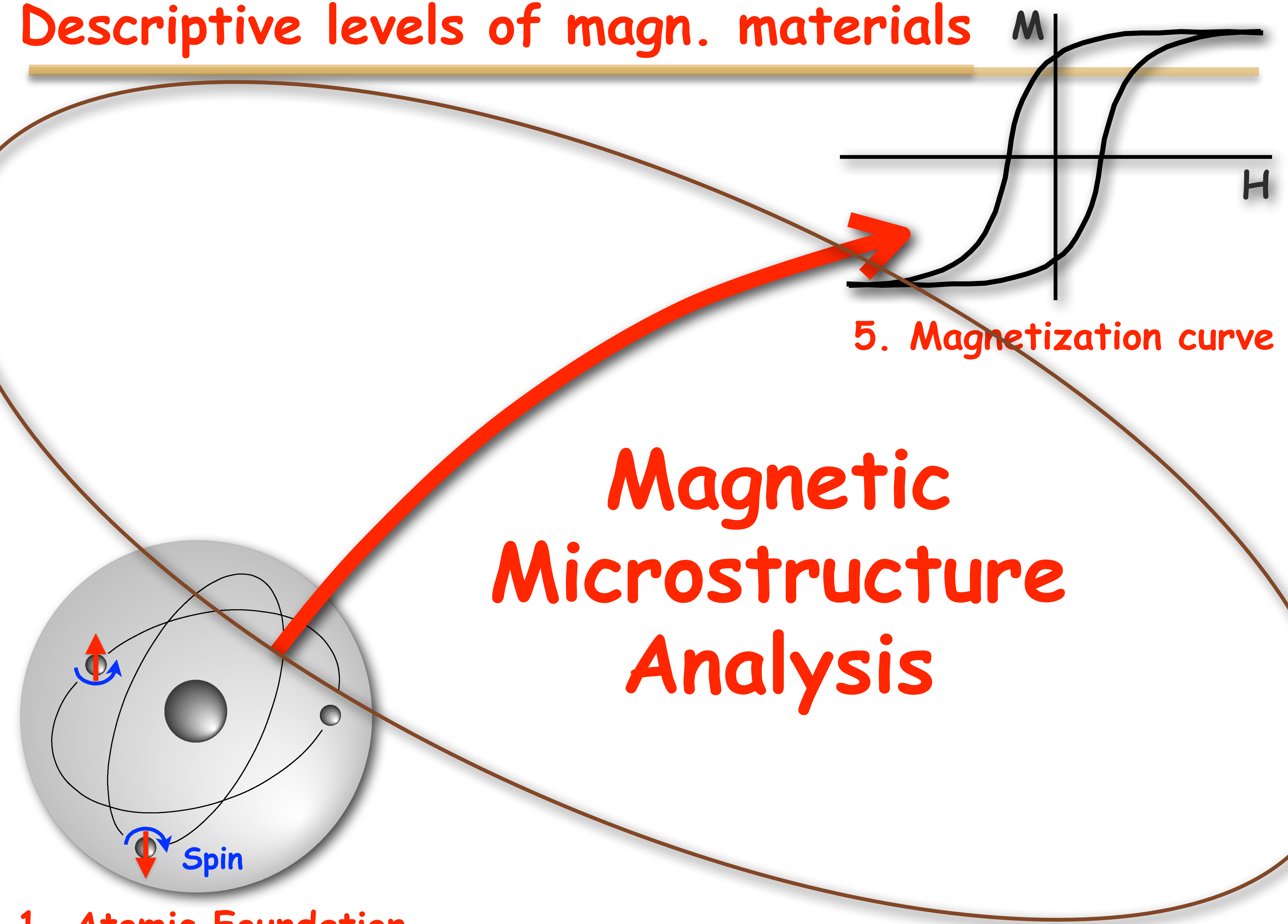


5. Magnetization curve



1. Atomic Foundation

Descriptive levels of magn. materials

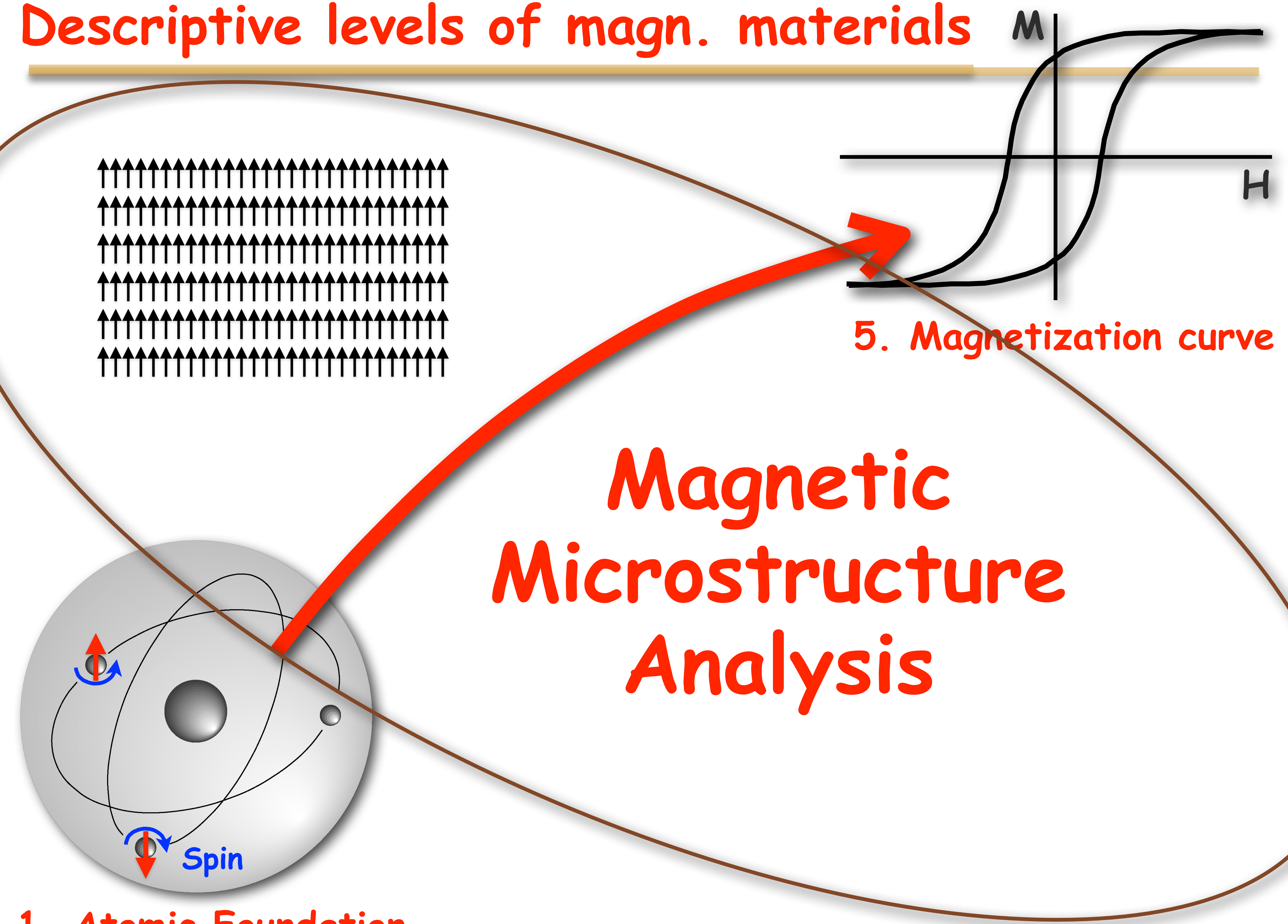


5. Magnetization curve

Magnetic Microstructure Analysis

1. Atomic Foundation

Descriptive levels of magn. materials

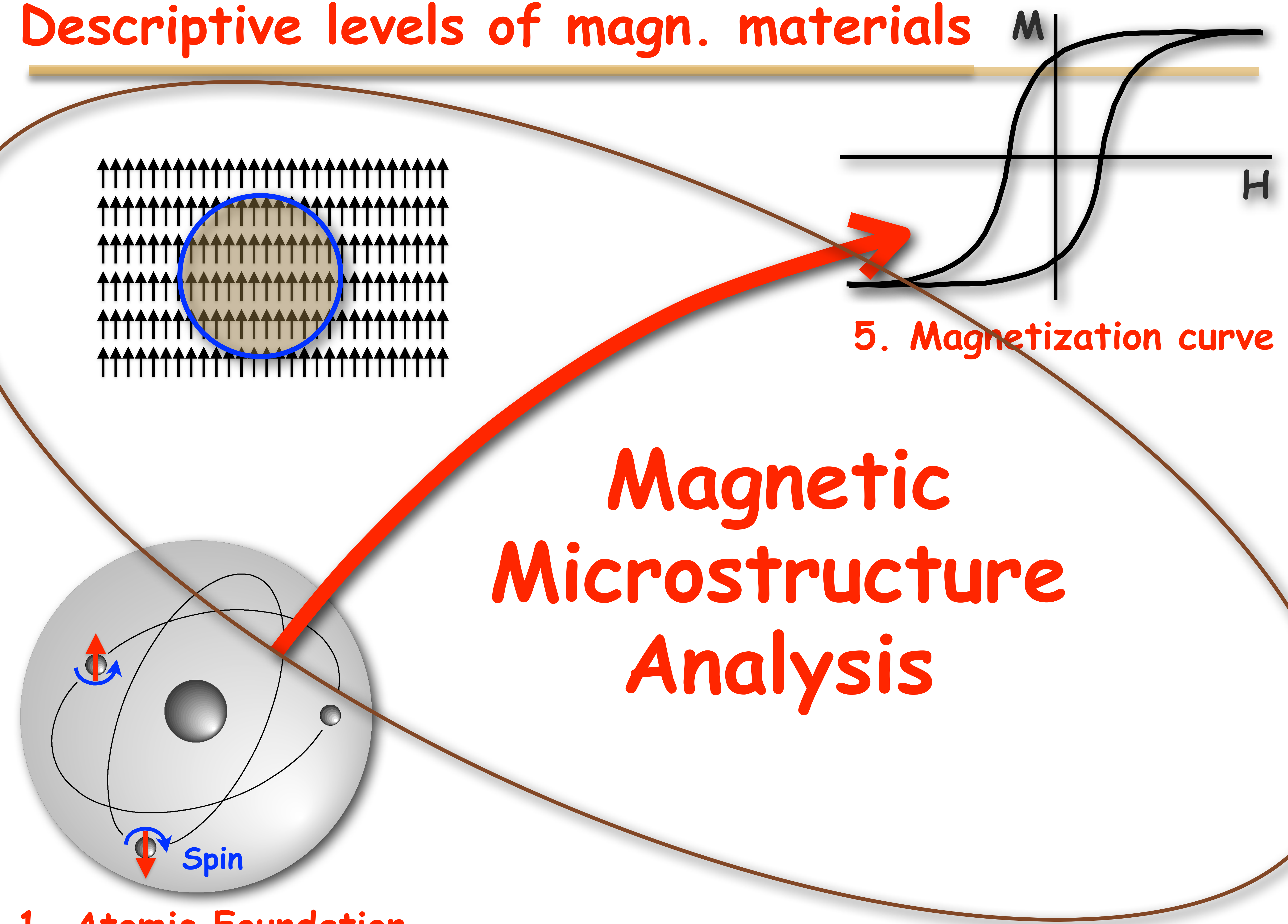


5. Magnetization curve

Magnetic Microstructure Analysis

1. Atomic Foundation

Descriptive levels of magn. materials

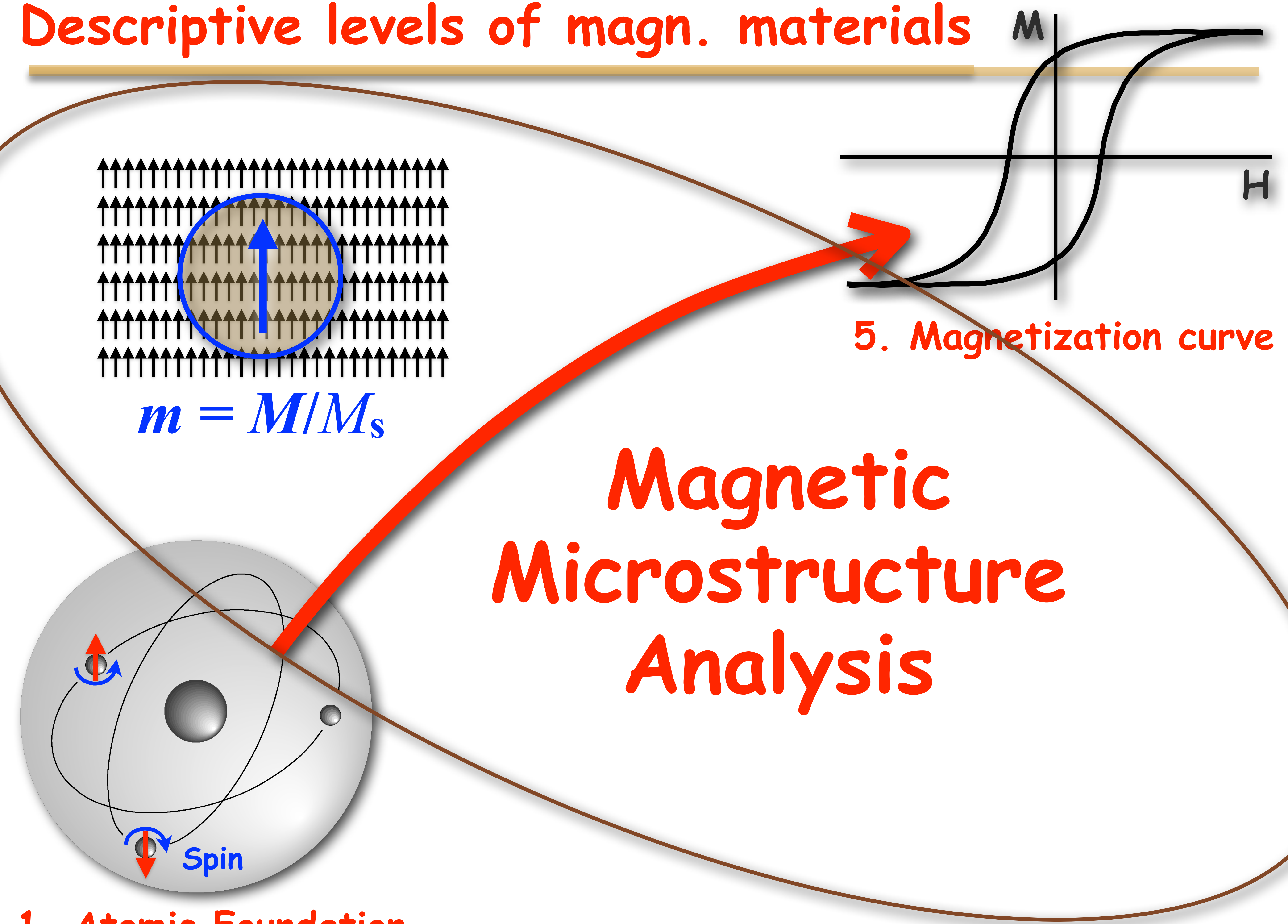


5. Magnetization curve

Magnetic Microstructure Analysis

1. Atomic Foundation

Descriptive levels of magn. materials



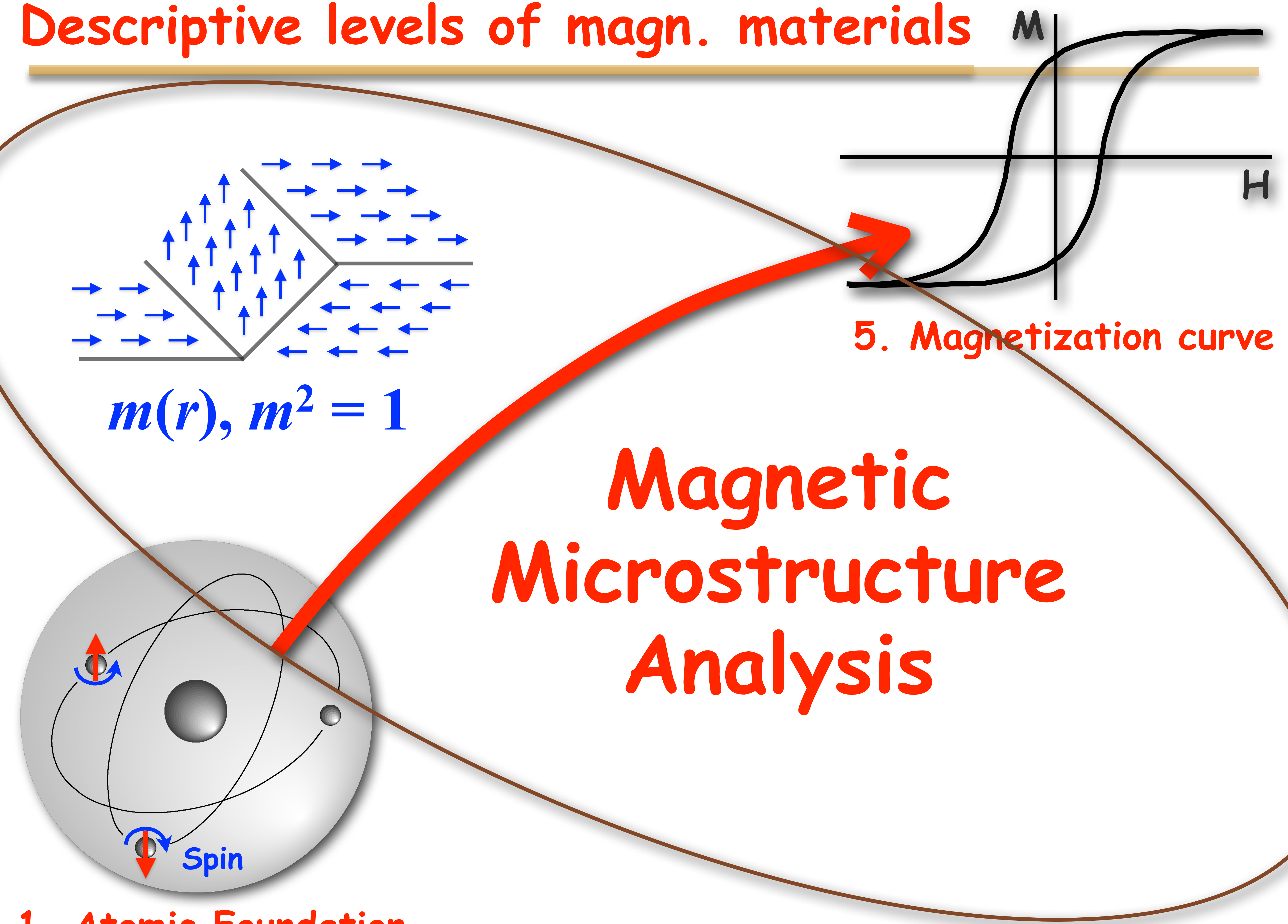
$m = M/M_s$

5. Magnetization curve

Magnetic Microstructure Analysis

1. Atomic Foundation

Descriptive levels of magn. materials



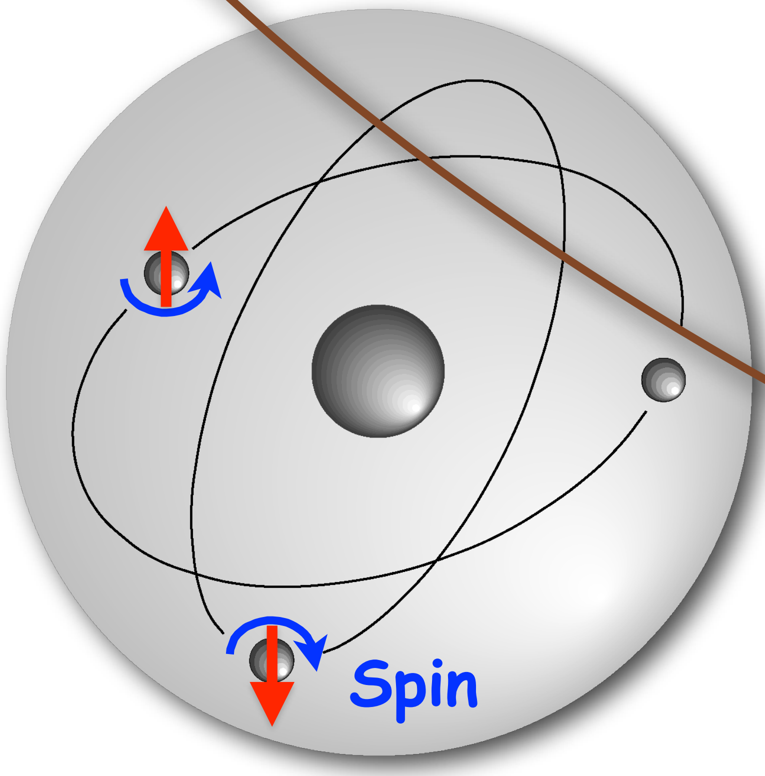
$m(r), m^2 = 1$

5. Magnetization curve

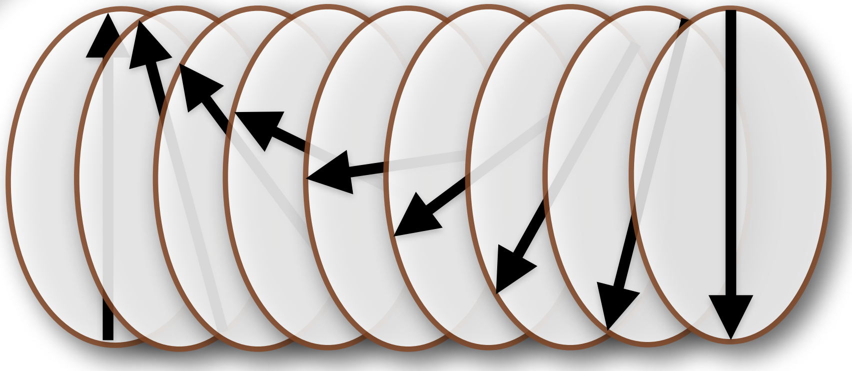
Magnetic Microstructure Analysis

1. Atomic Foundation

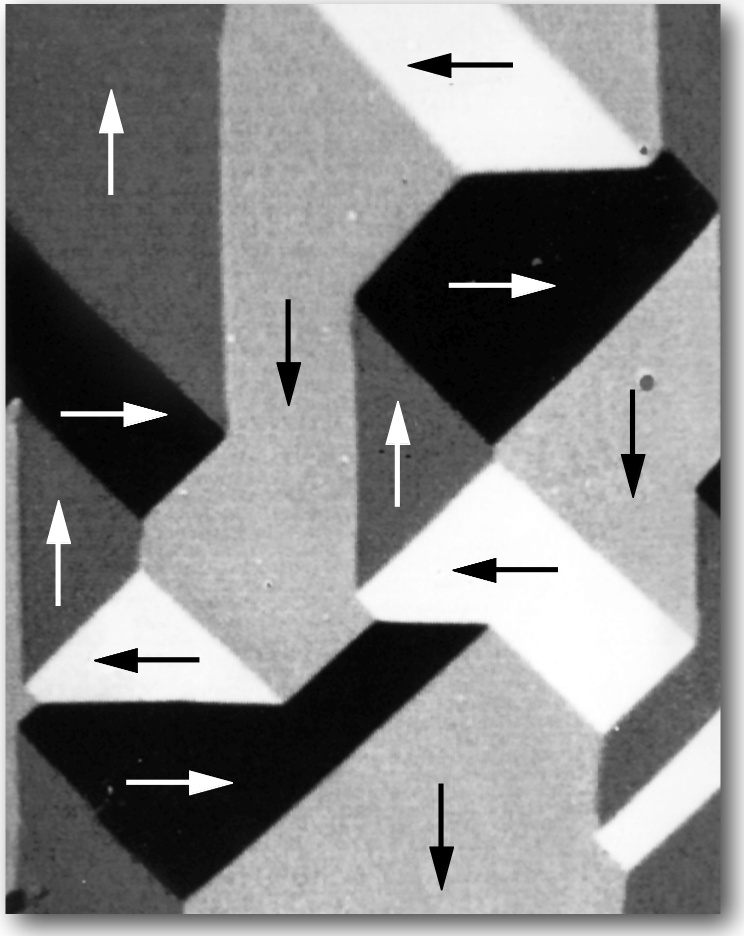
Descriptive levels of magn. materials



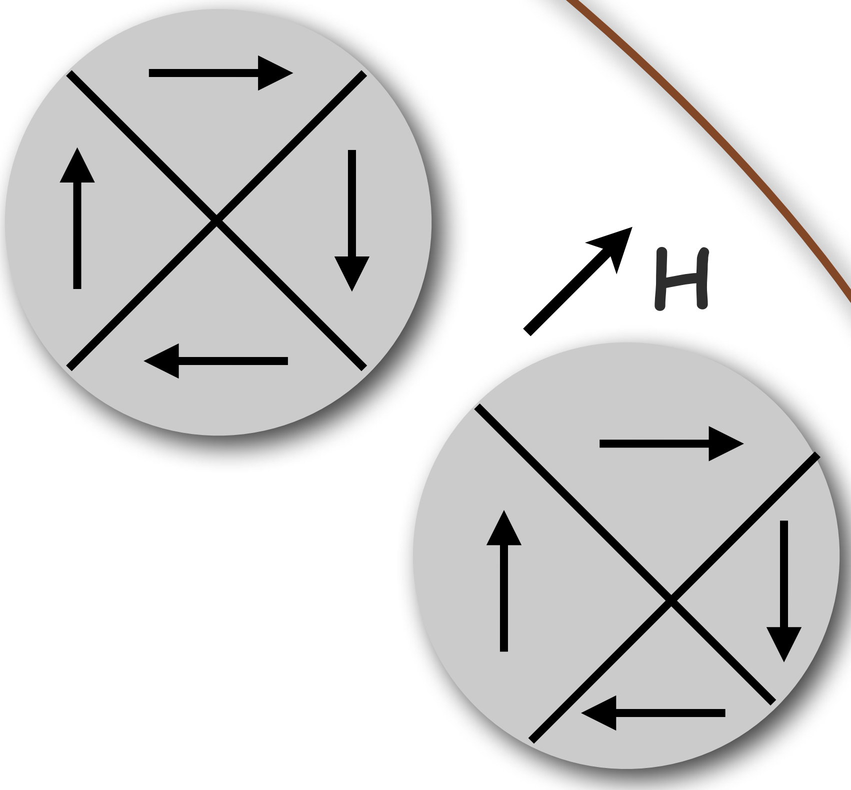
1. Atomic Foundation



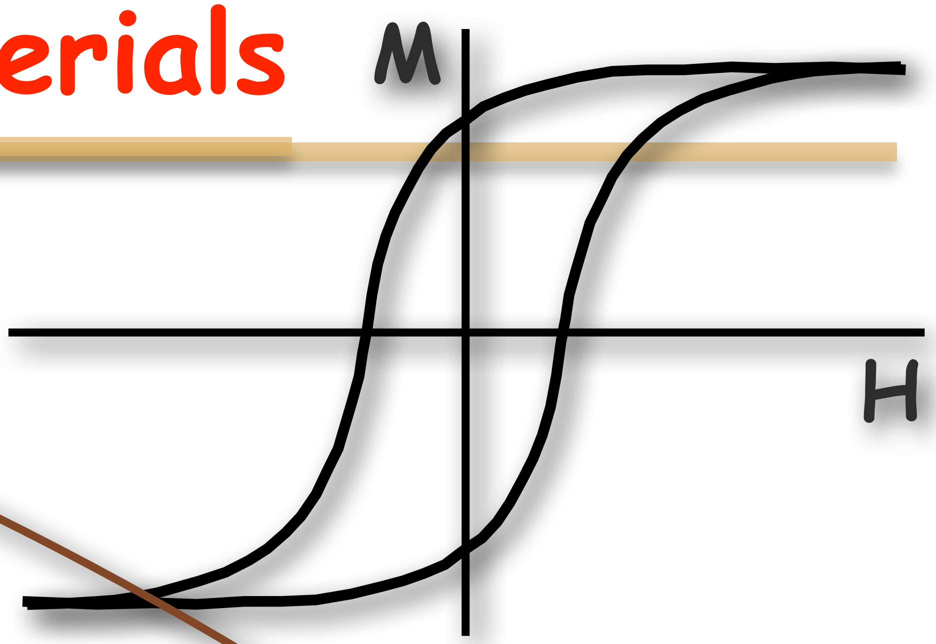
2. Micromagnetic Analysis



3. Domain Analysis

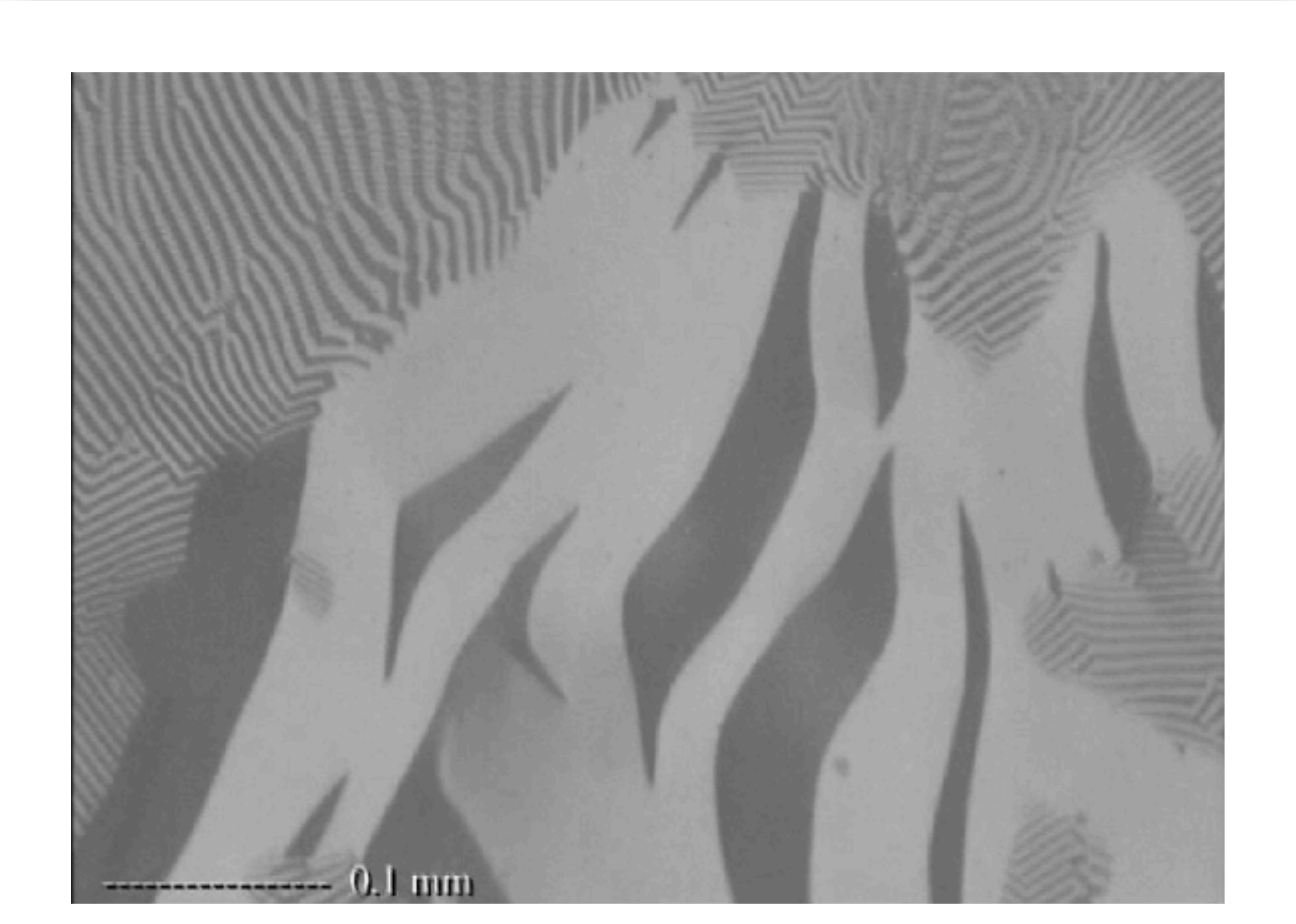


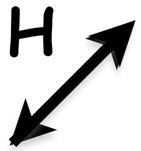
4. Phase Analysis

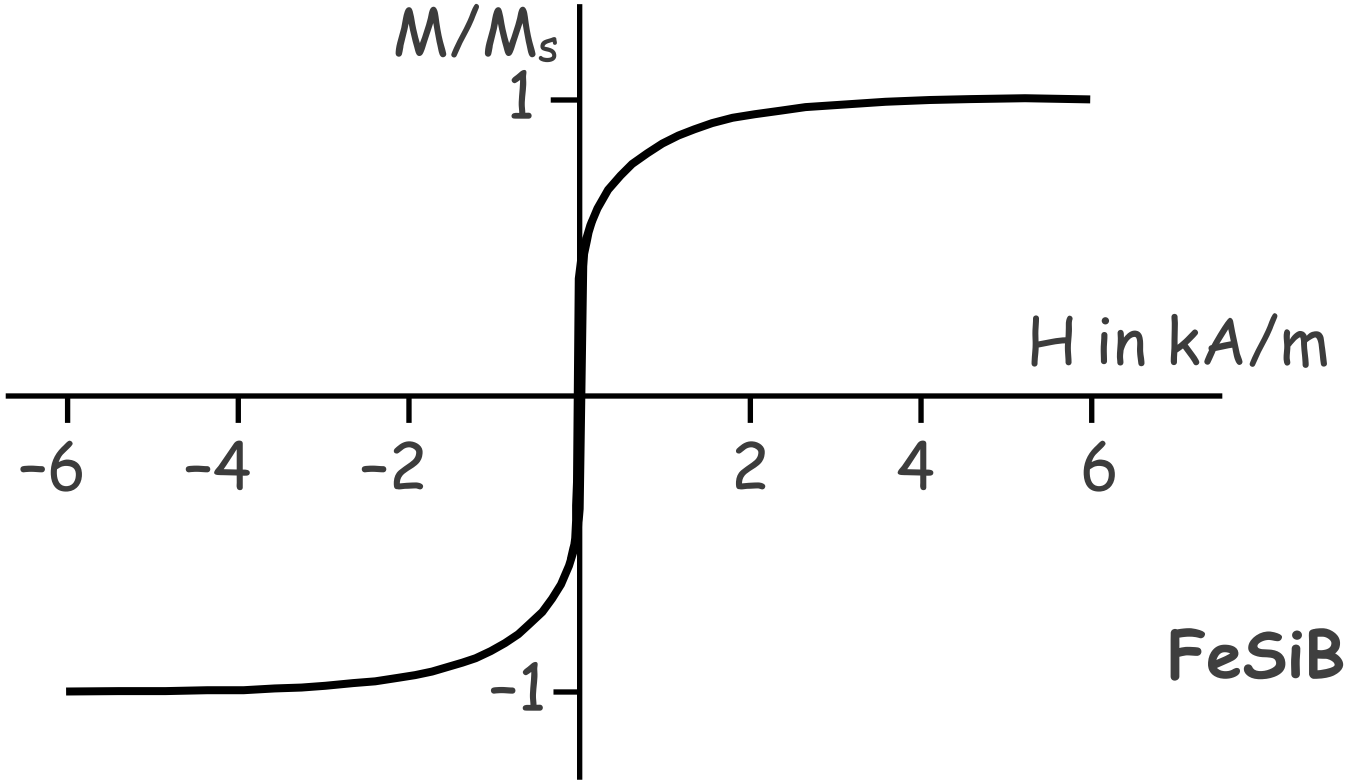


5. Magnetization curve

M(H) loop and domains

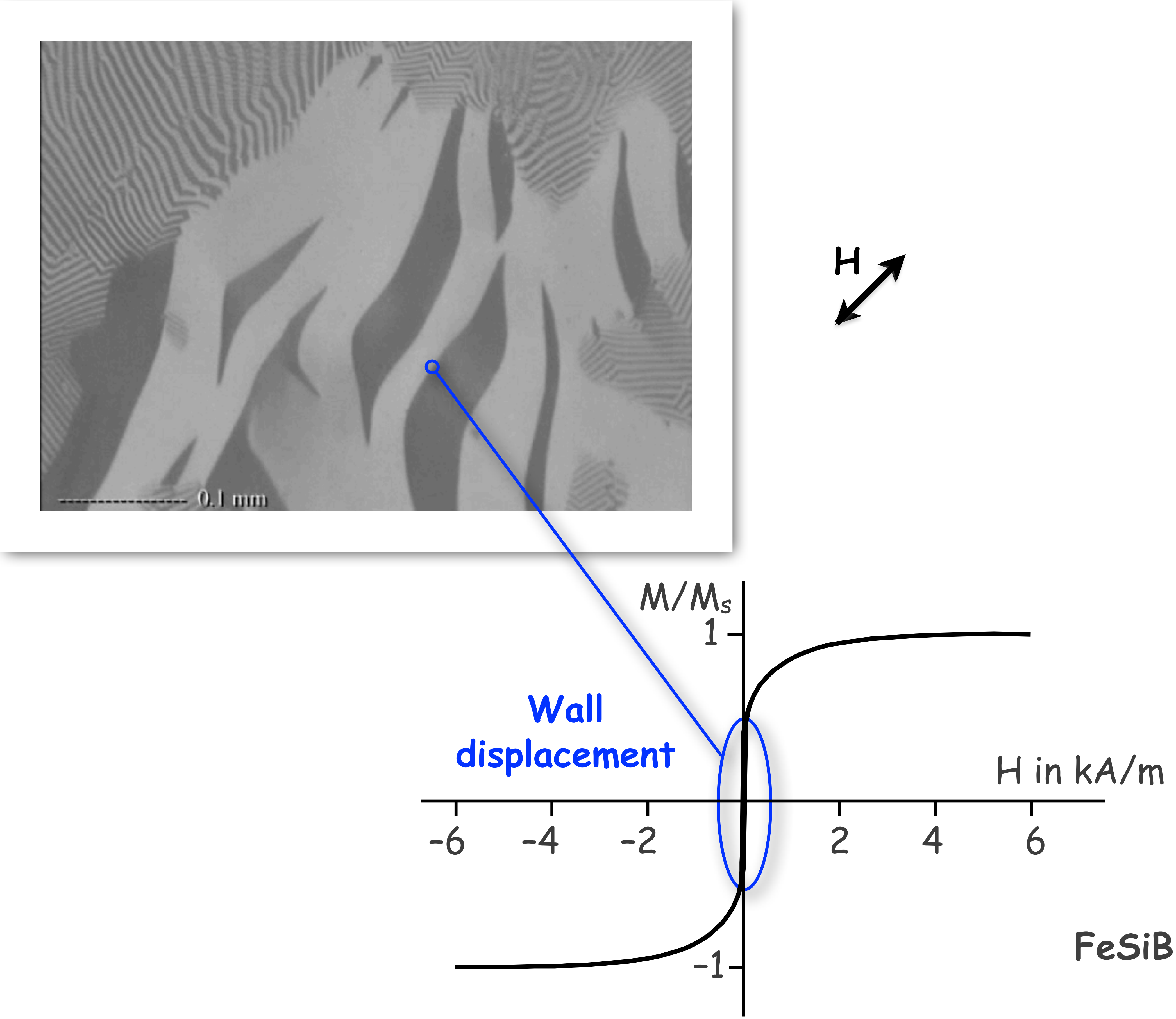






FeSiB amorphous ribbon
(rapidly quenched)

M(H) loop and domains



FeSiB amorphous ribbon
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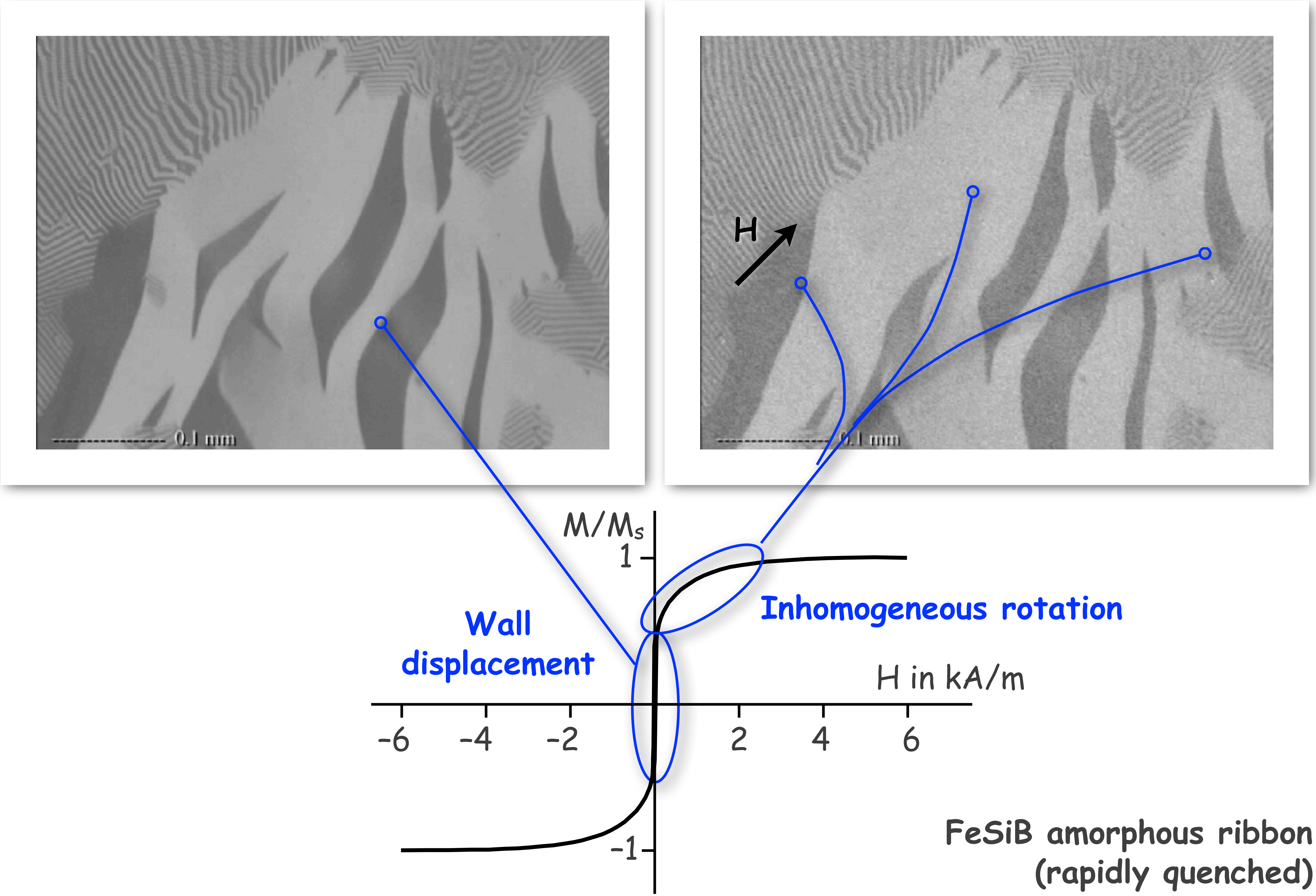

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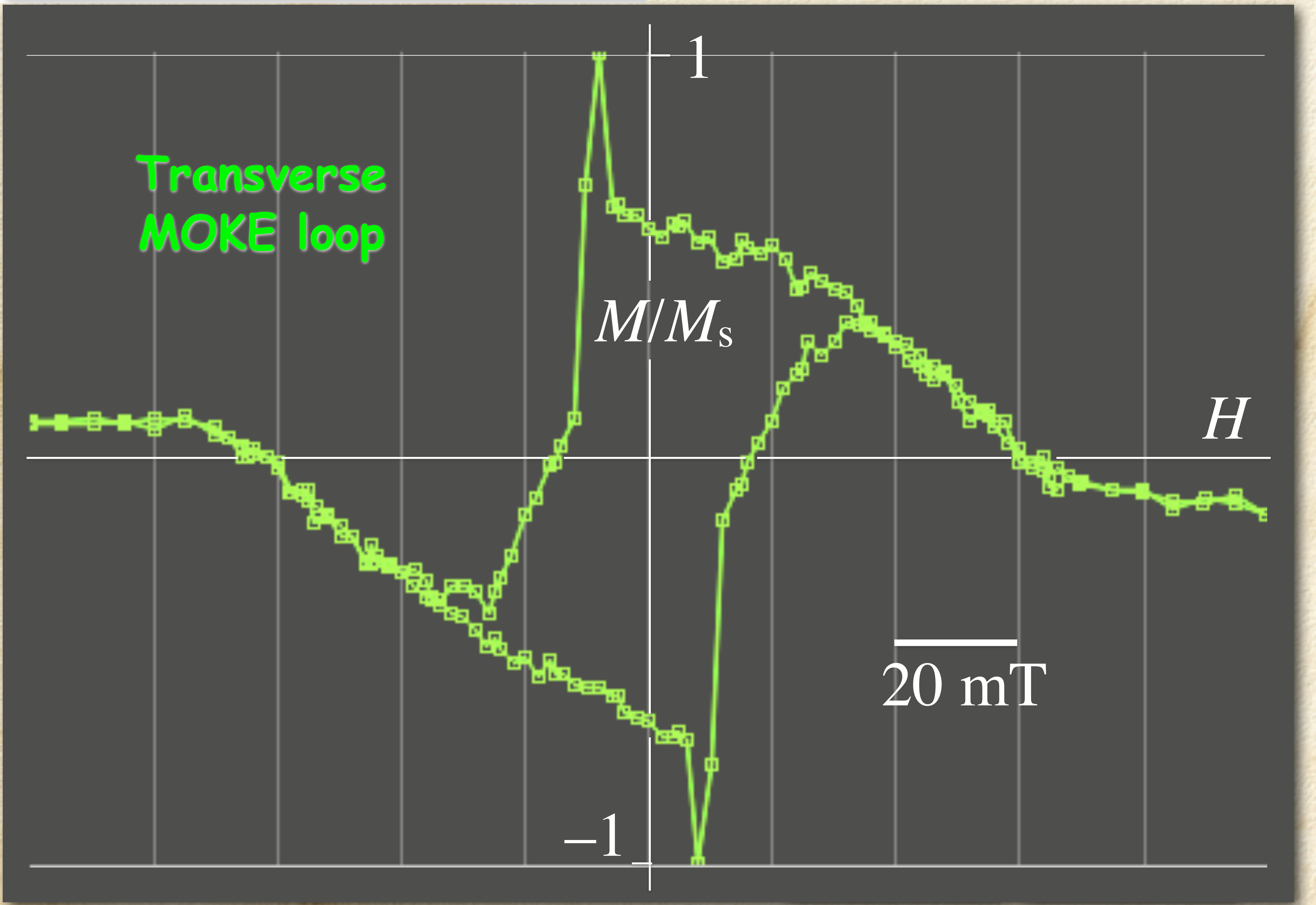
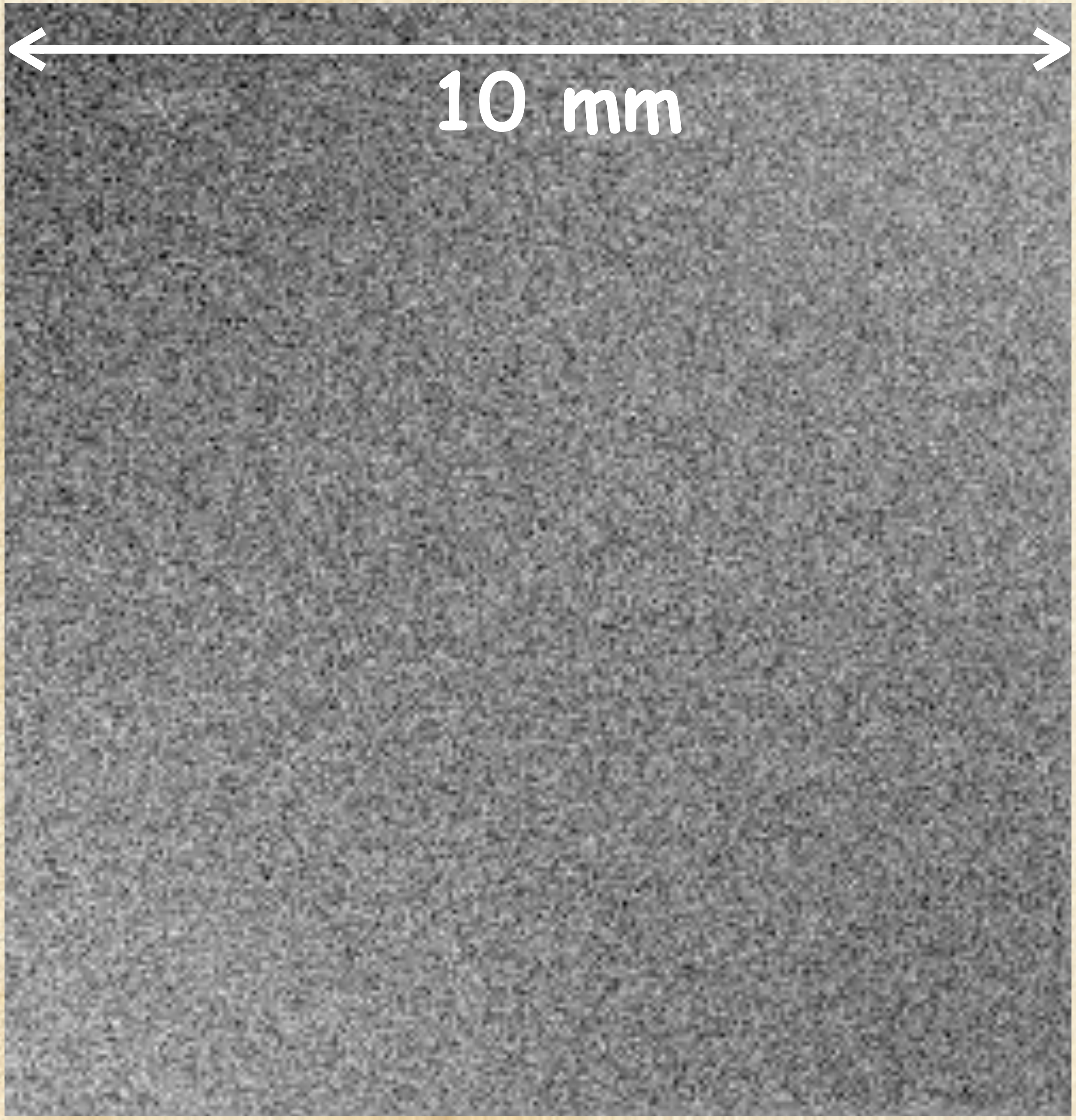
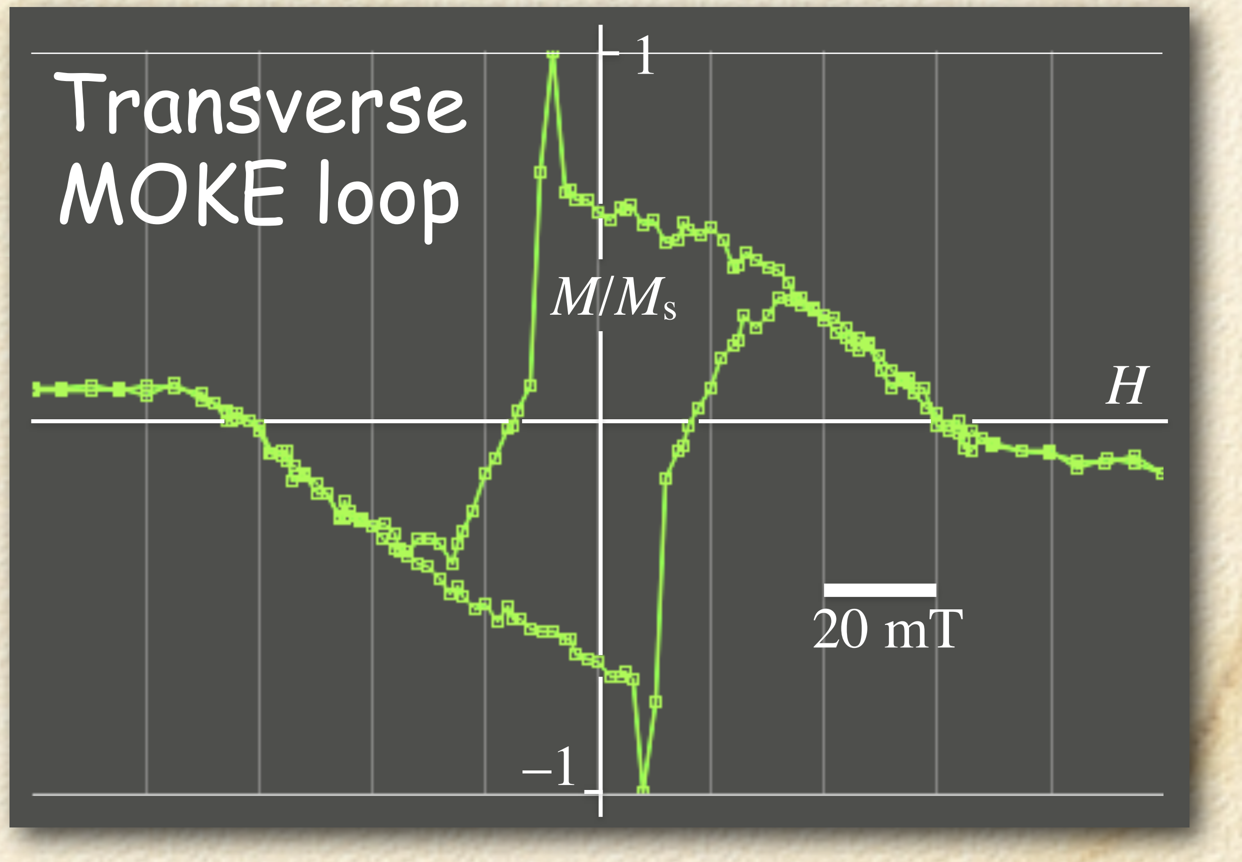


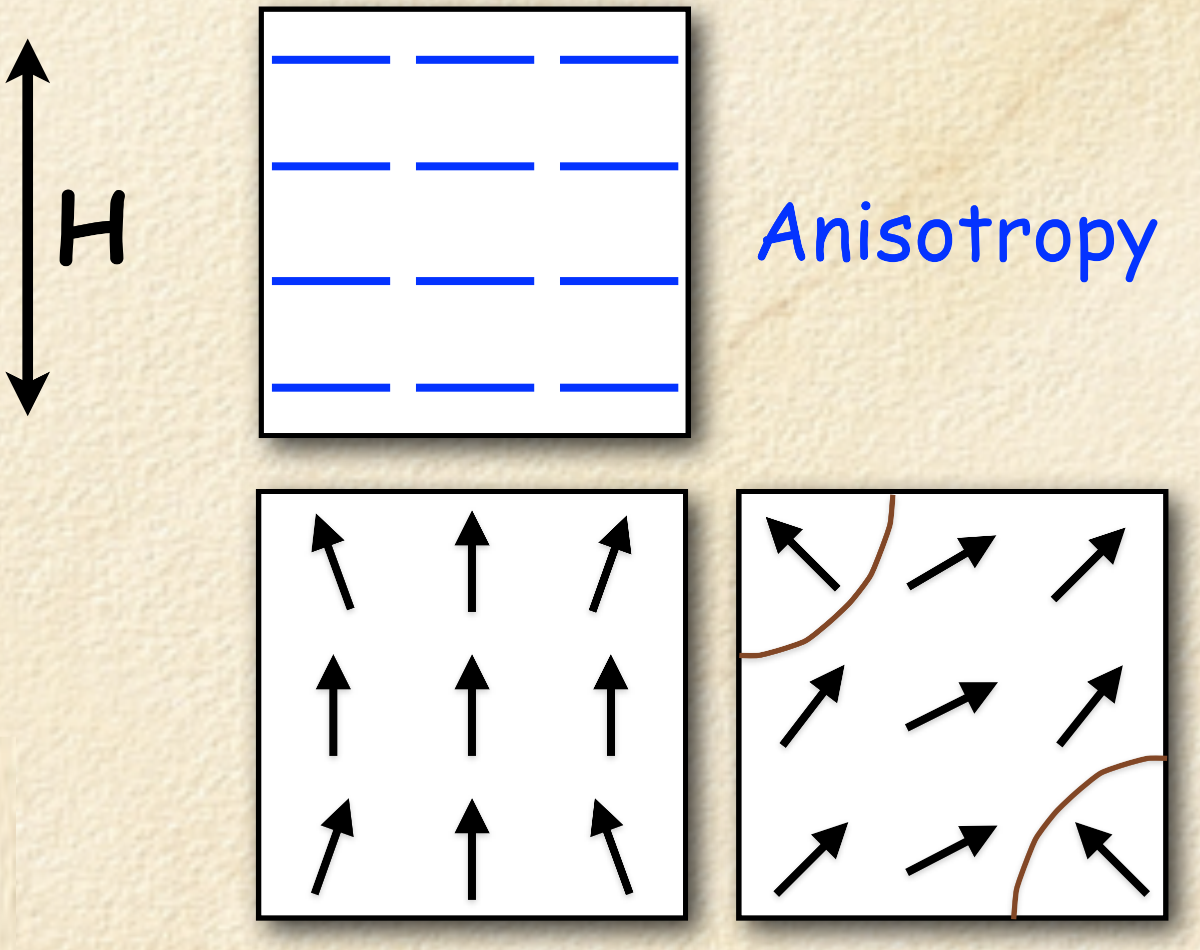
M(H) loop and domains

$Co_{27}Sm_{73}$ amorphous film
(thickness 200 nm)

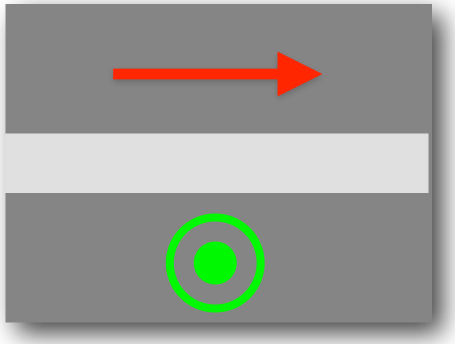


Sample: F. Magnus and B. Hjörvarsson,
Uppsala University (unpublished)

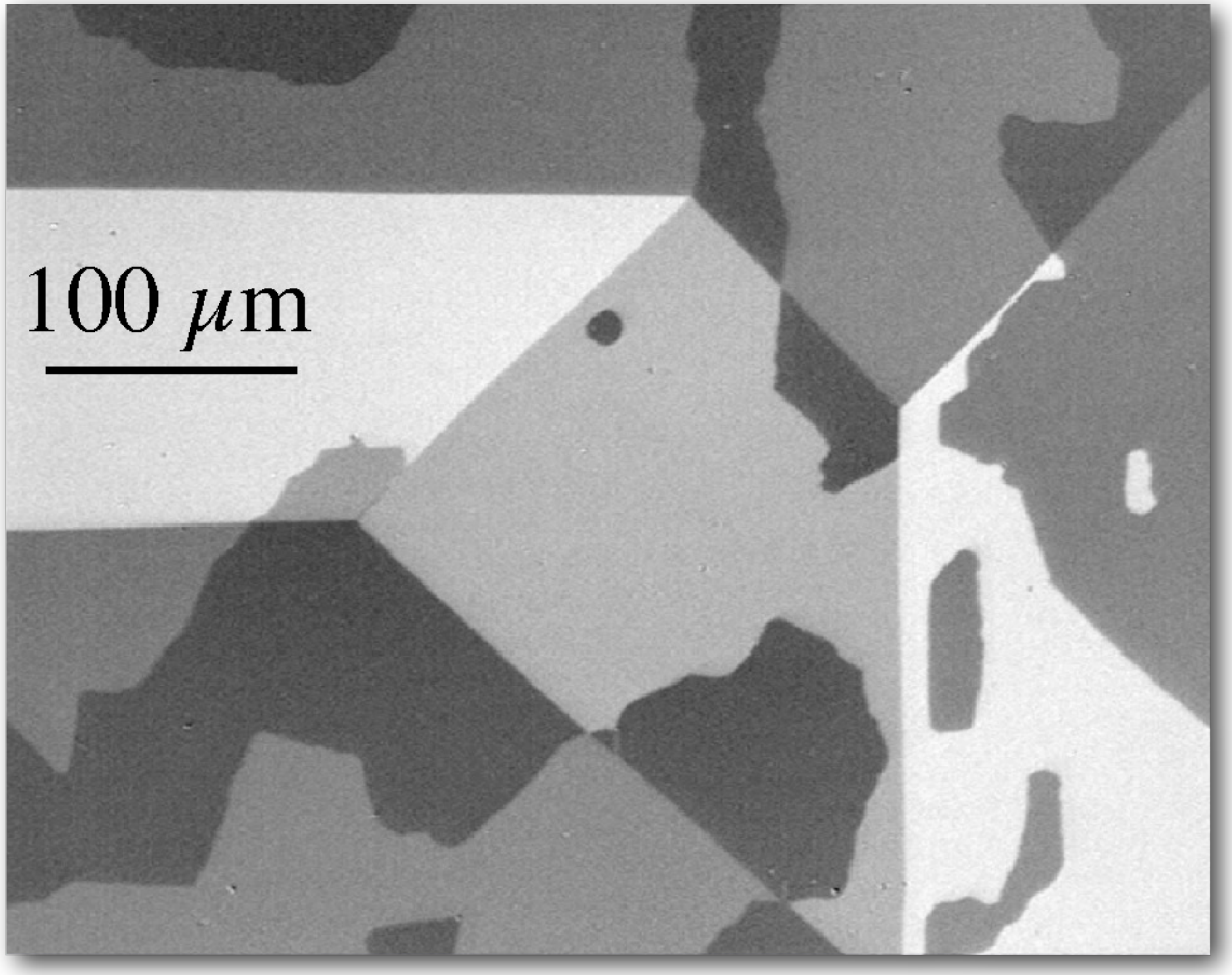


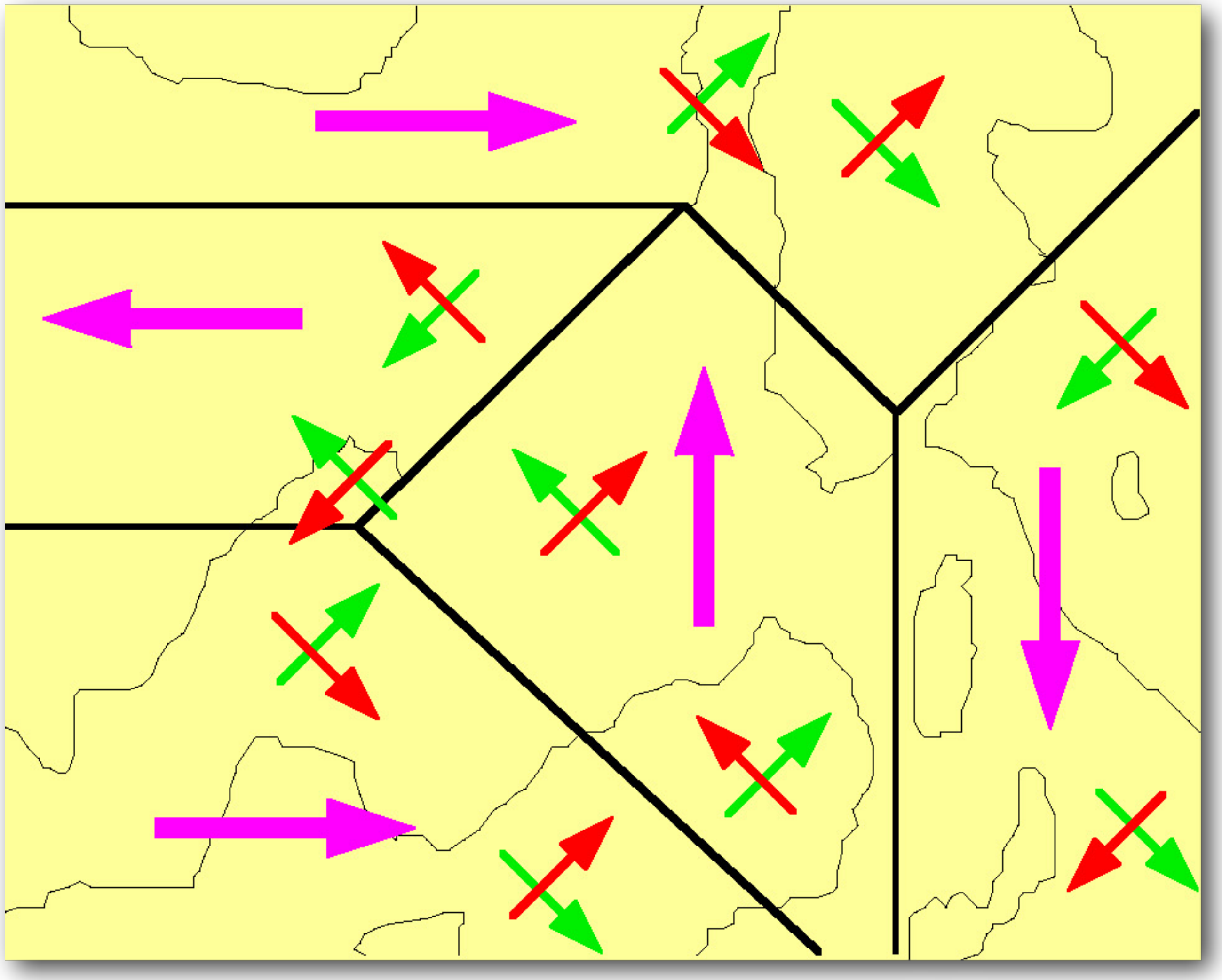


Biquadratic coupling in multilayers



Fe (30 nm)
Cr (1.6 nm)
Fe (30 nm)

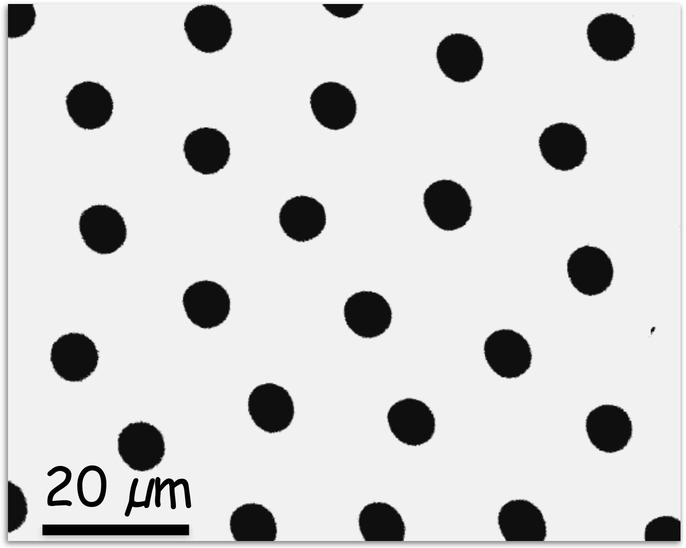




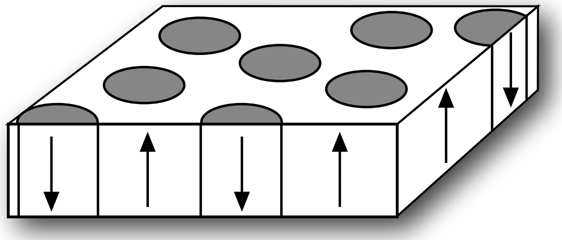
M. Rührig, RS, et al., Phys. Stat. Sol. 125 (1991)

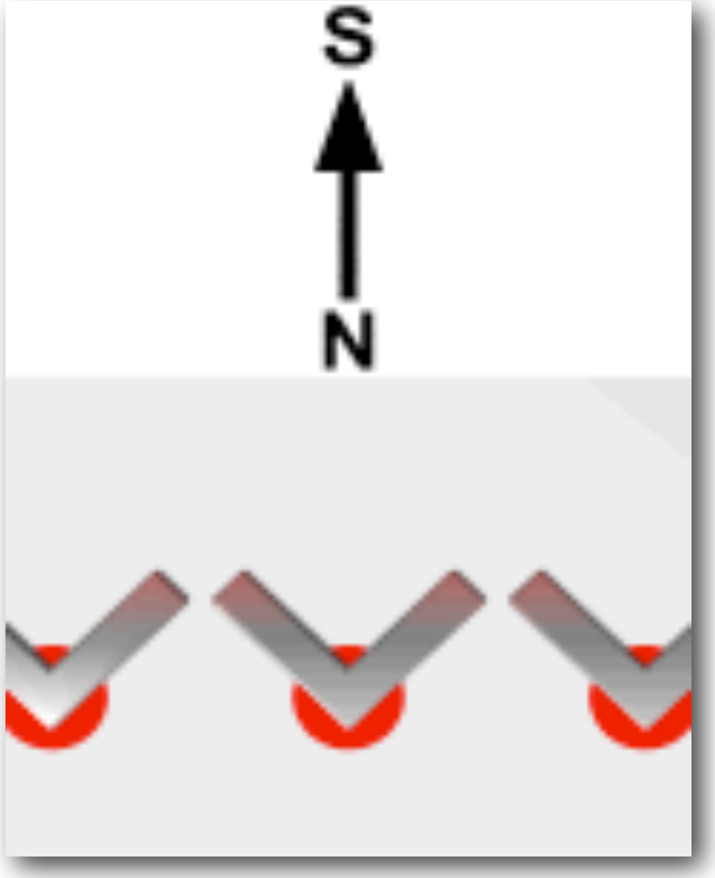
Domain Shift Register Devices

Bubble Memory



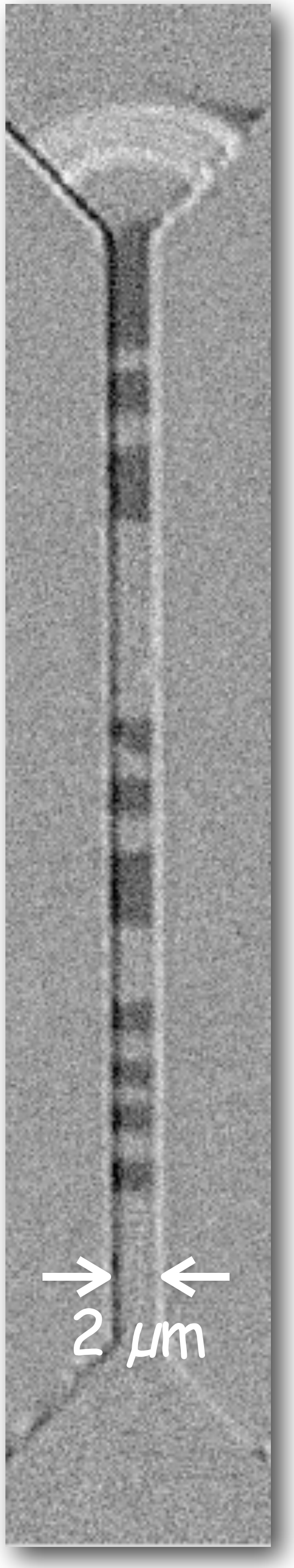
Bubble domains

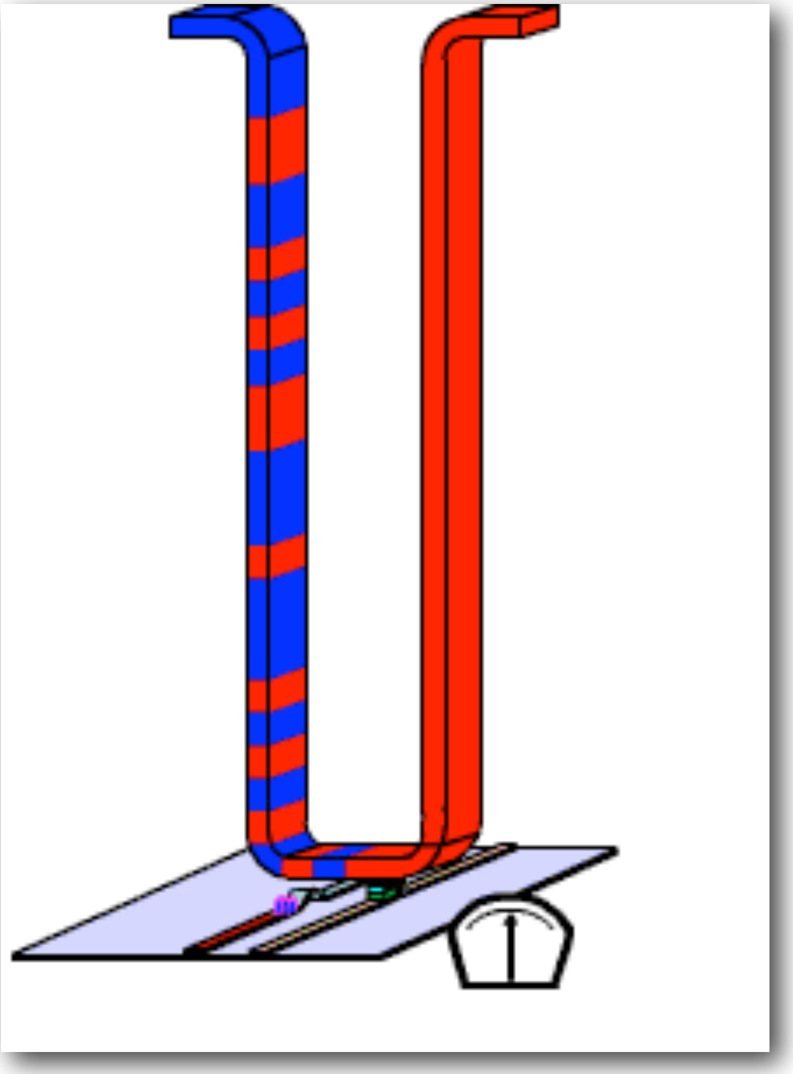




http://commons.wikimedia.org/wiki/How_bubble_memory_works

Race Track Memory





Kerr-movie of Co/Ni PMA multilayer

courtesy
S. Parkin, IBM

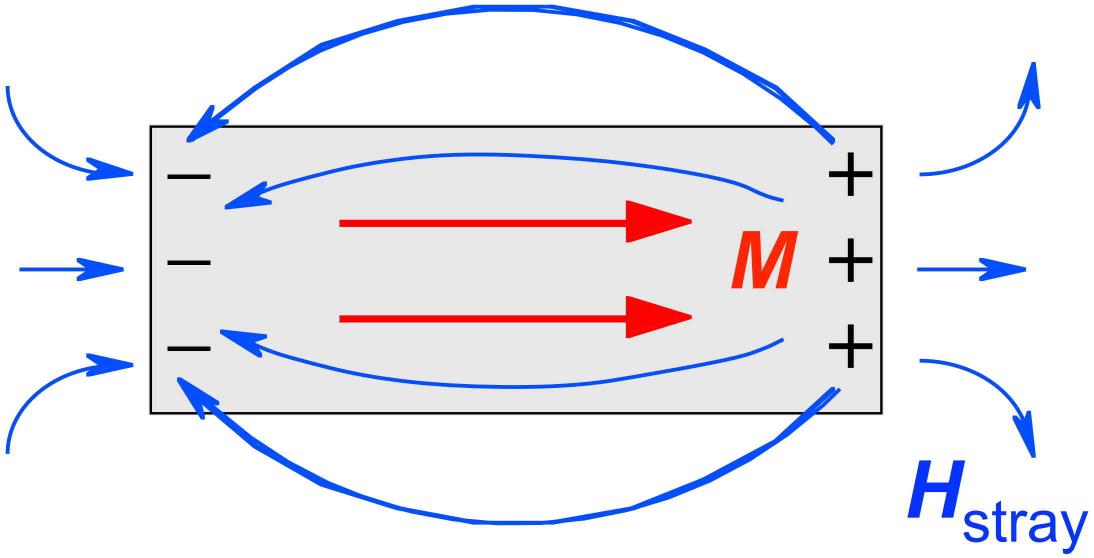
Sensitivity of imaging methods

$\boldsymbol{B} = \mu_0 (\boldsymbol{H} + \boldsymbol{M})$ $(\boldsymbol{H} = \boldsymbol{H}_{ext} + \boldsymbol{H}_{stray})$

$\mathrm{div}\, \boldsymbol{B} = 0$

$\downarrow$

$\mathrm{div}\, \boldsymbol{H}_{stray} = - \mathrm{div}\, \boldsymbol{M}$



- Sensitive to $\boldsymbol{H}_{stray}$
  - Bitter technique
  - Magnetic force microscopy
  - Hall probe microscopy
- Sensitive to ***M***
  - Magneto-optical microscopy
  - X-ray spectroscopy
  - Polarized electrons (SEMPA, SPT)
- Sensitive to ***B***
  - Transmission electron microscopy
- Sensitive to distortions
  - X-ray, neutron scattering

Sensitivity of imaging methods

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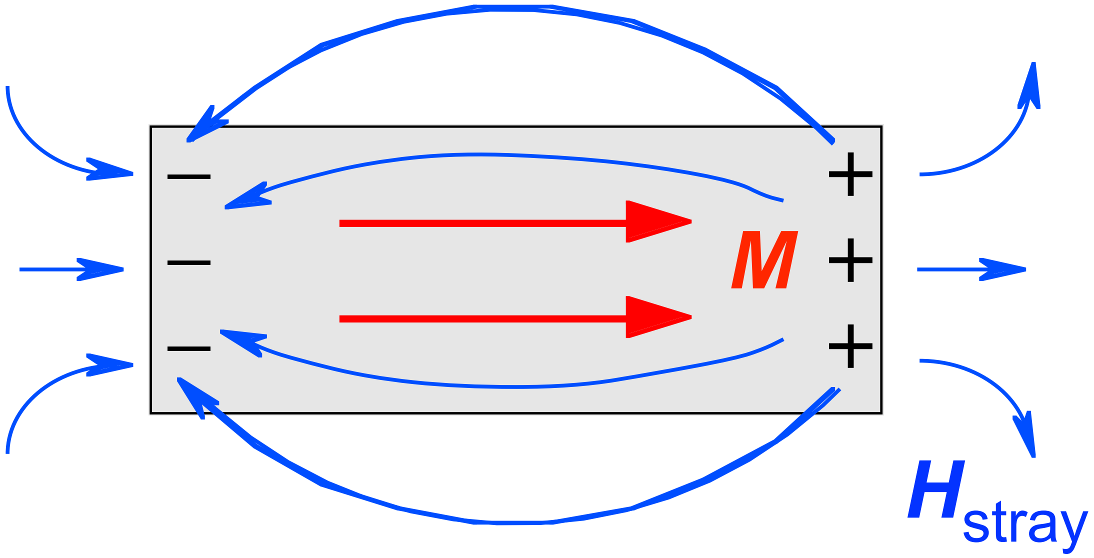

- Sensitive to $\boldsymbol{H}_{stray}$
  1. Bitter technique
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  3. Hall probe microscopy
- Sensitive to $\boldsymbol{M}$
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  - X-ray spectroscopy
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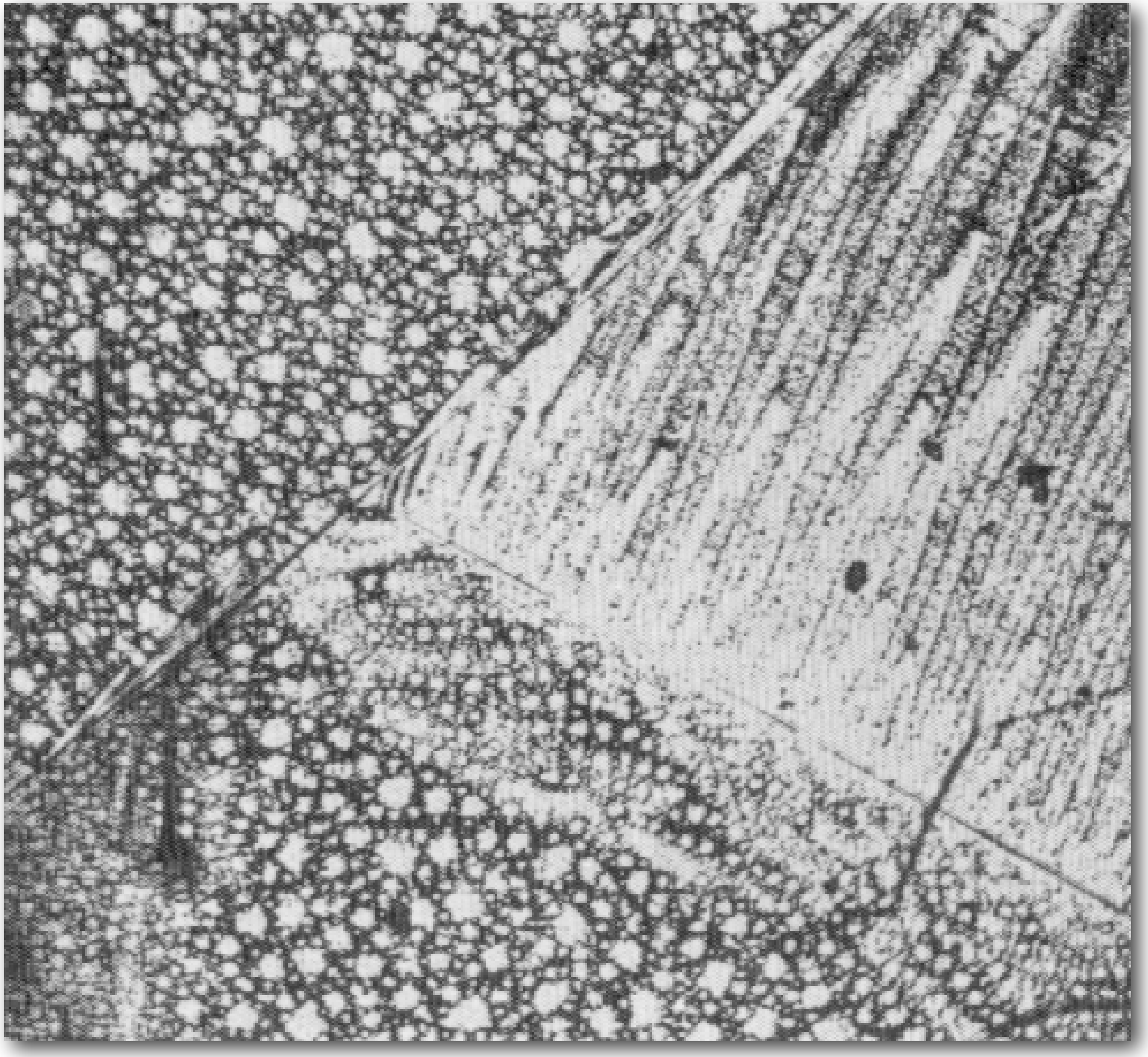
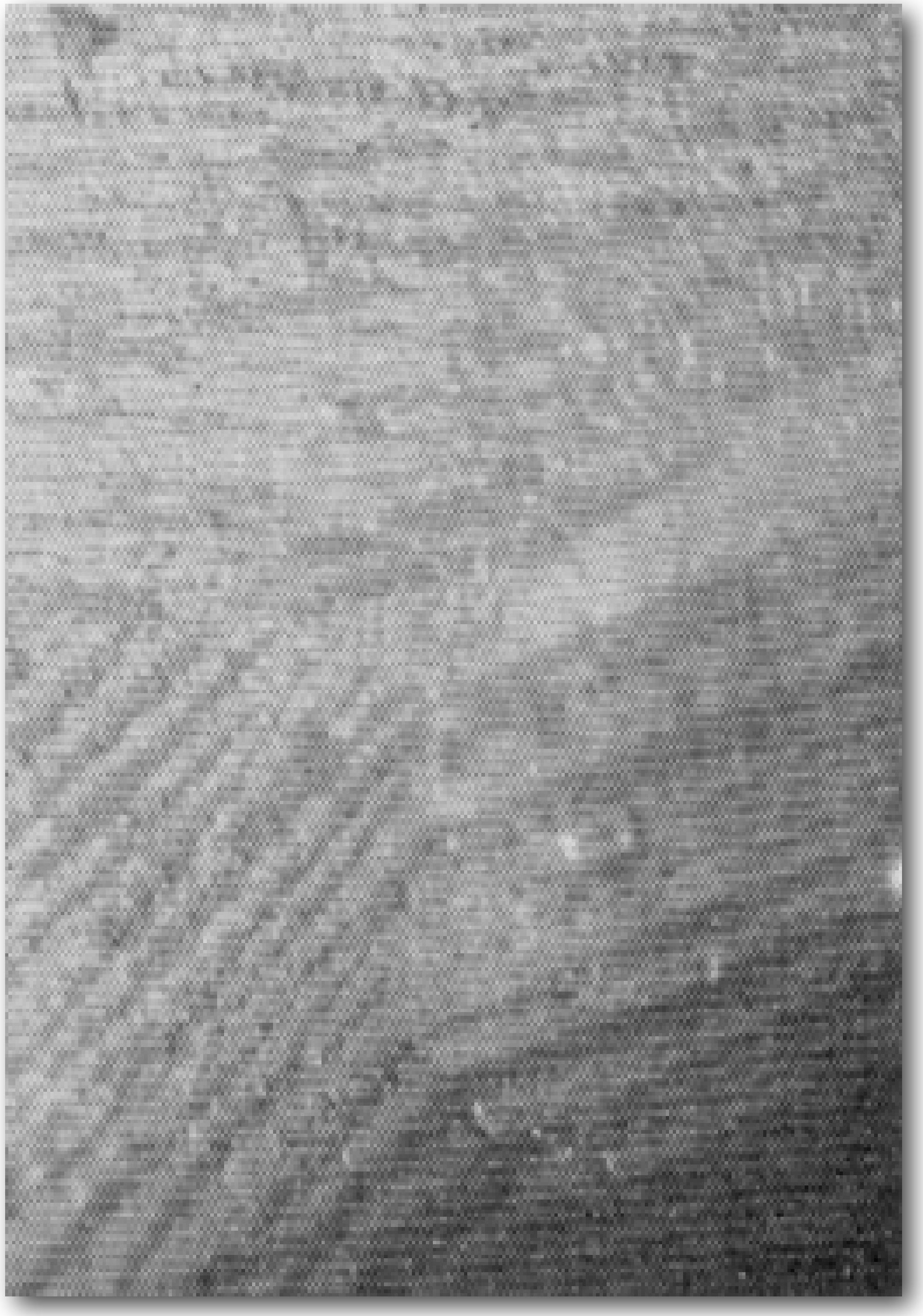
- Sensitive to $\boldsymbol{B}$
  - Transmission electron microscopy
- Sensitive to distortions
  - X-ray, neutron scattering

1. Bitter technique

History: first imaging of domains by F. Bitter, 1931)



Cobalt

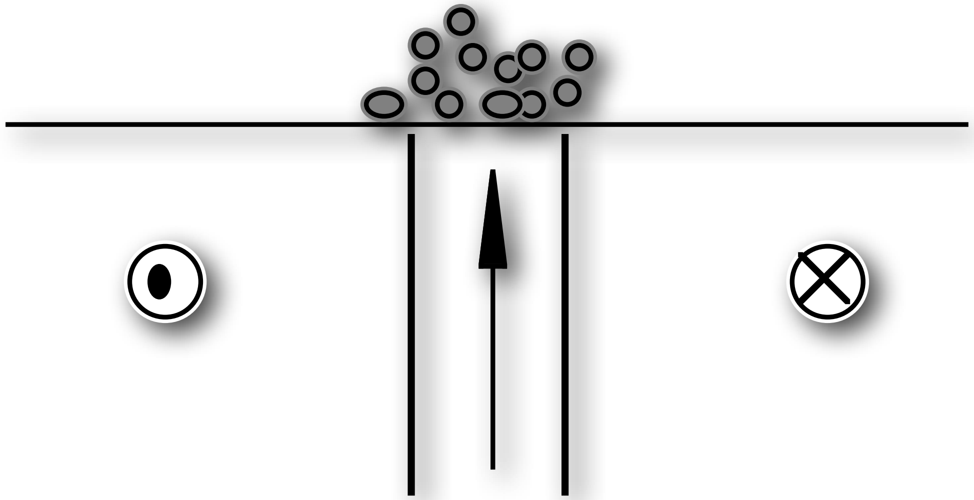


FeSi

1. Bitter technique

Principle

- Magnetic colloid: Magnetite particles (diameter about 10 nm) in water
- Accumulation in stray field at sample surface



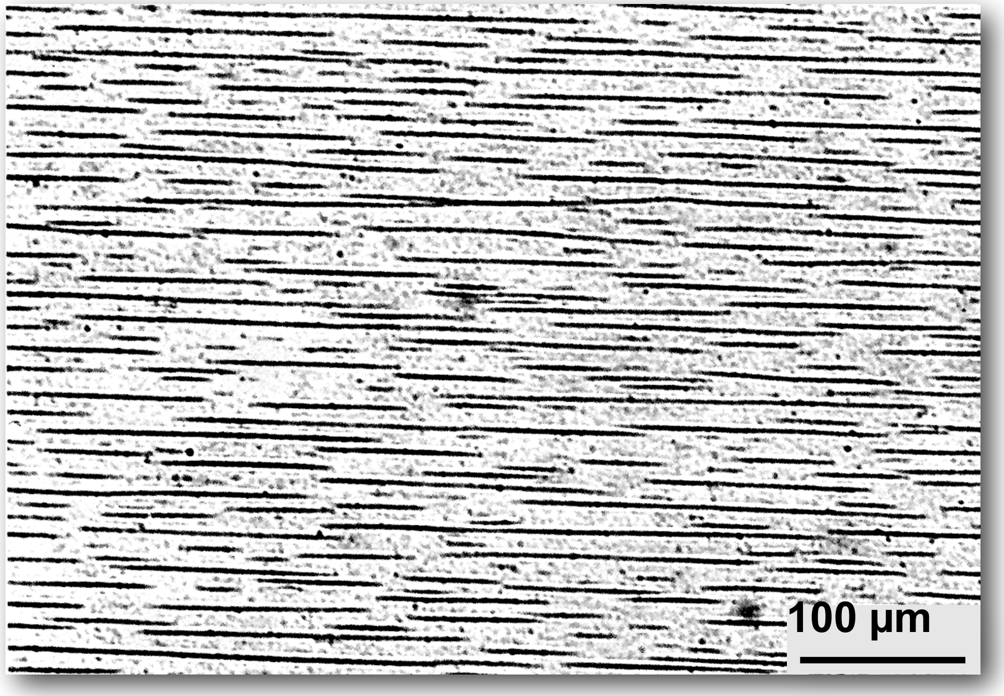
Sensitivity

- Reversible agglomeration in weak magnetic field

  → Increase of volume, elongated shape

  → Large susceptibility

  → Large sensitivity to stray fields
  in order of a few 100 A/m



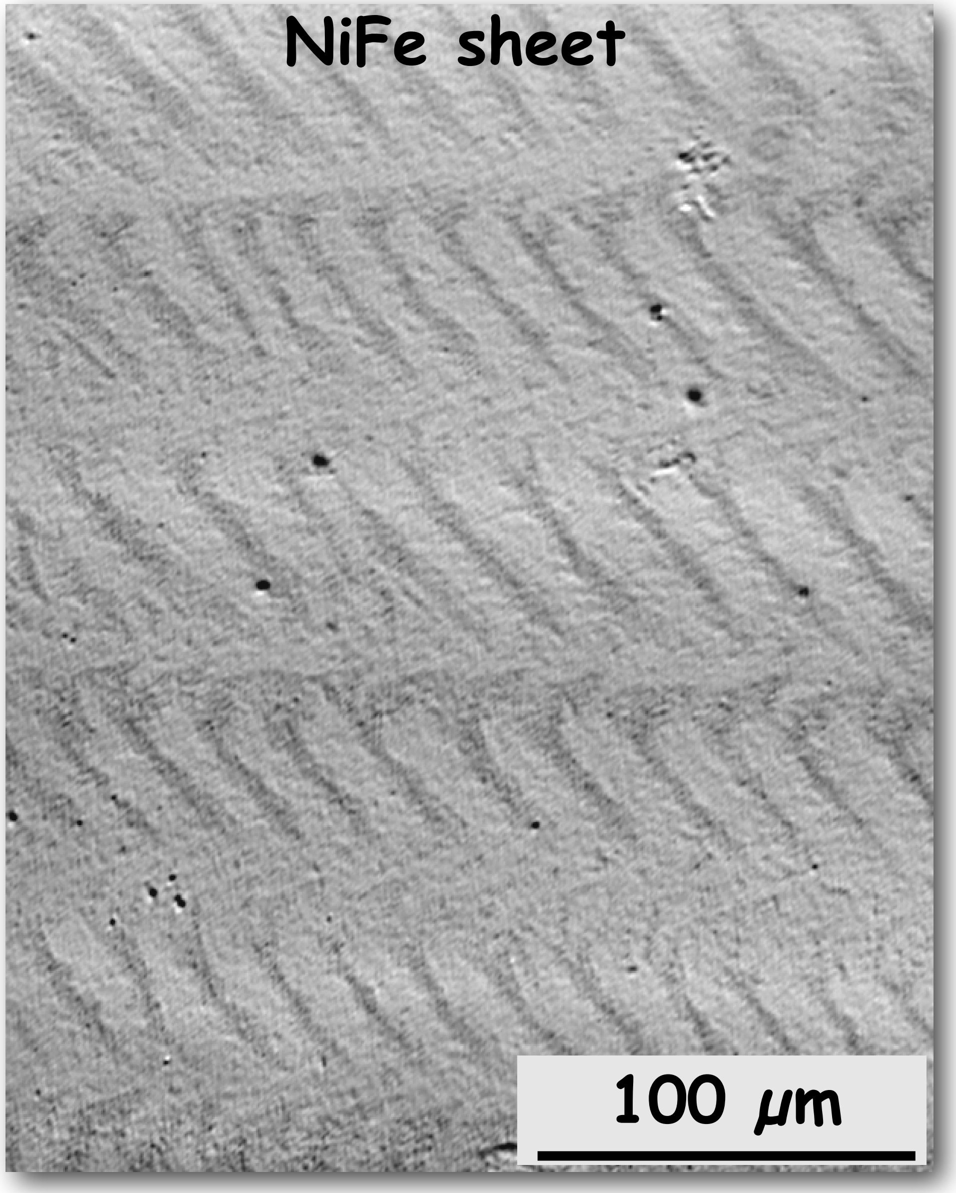
Agglomeration in magnetic field
(560 A/m)

1. Bitter technique

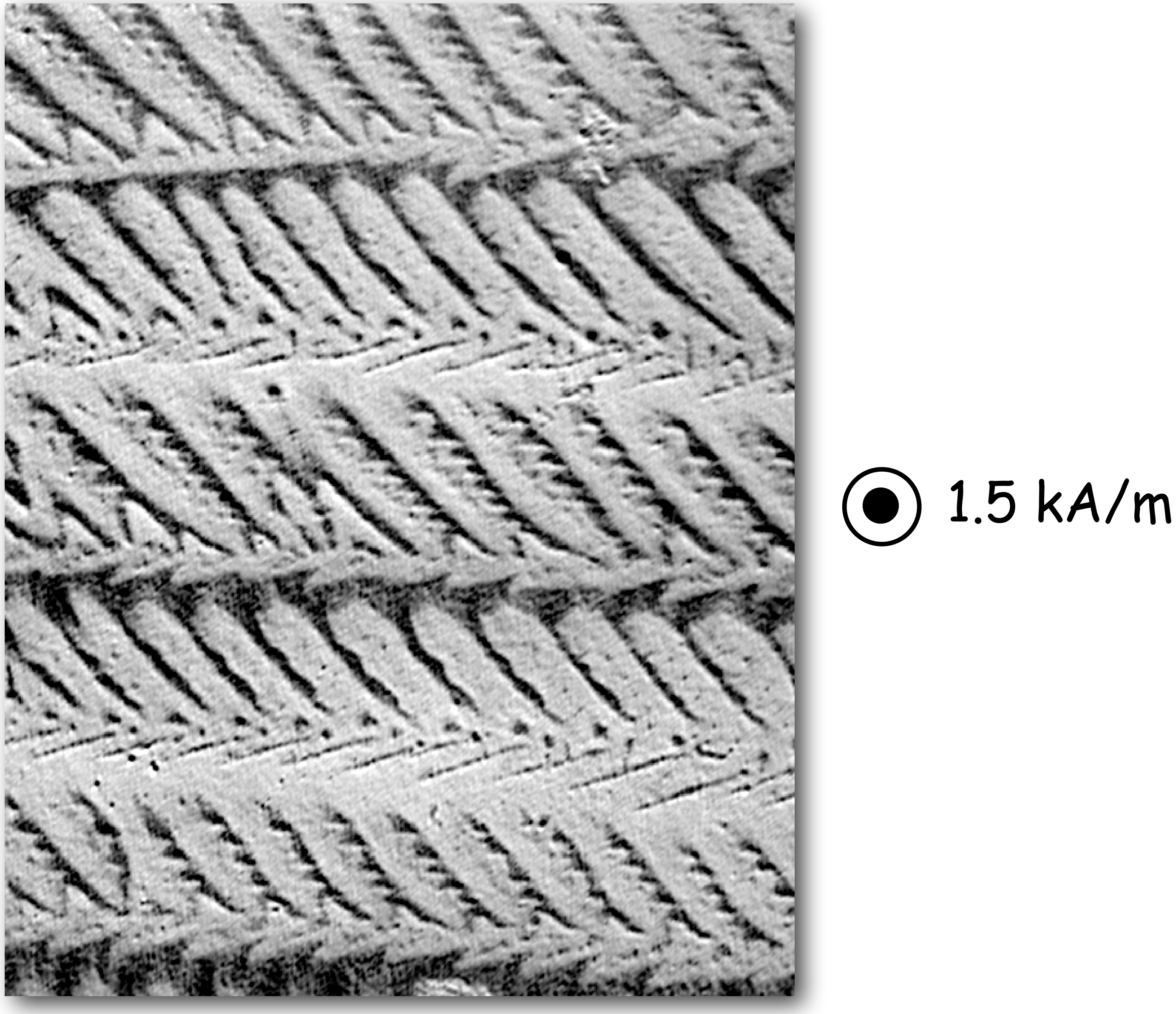
Sensitivity

Increase of sensitivity in weak perpendicular field

⟶ Domain imaging in soft magnetic materials



Without auxiliary field



With perpendicular field

1. Bitter technique

Dry colloid technique

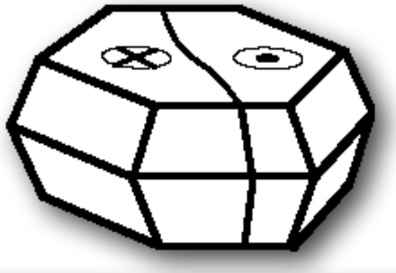
Allowing colloid to dry on surface

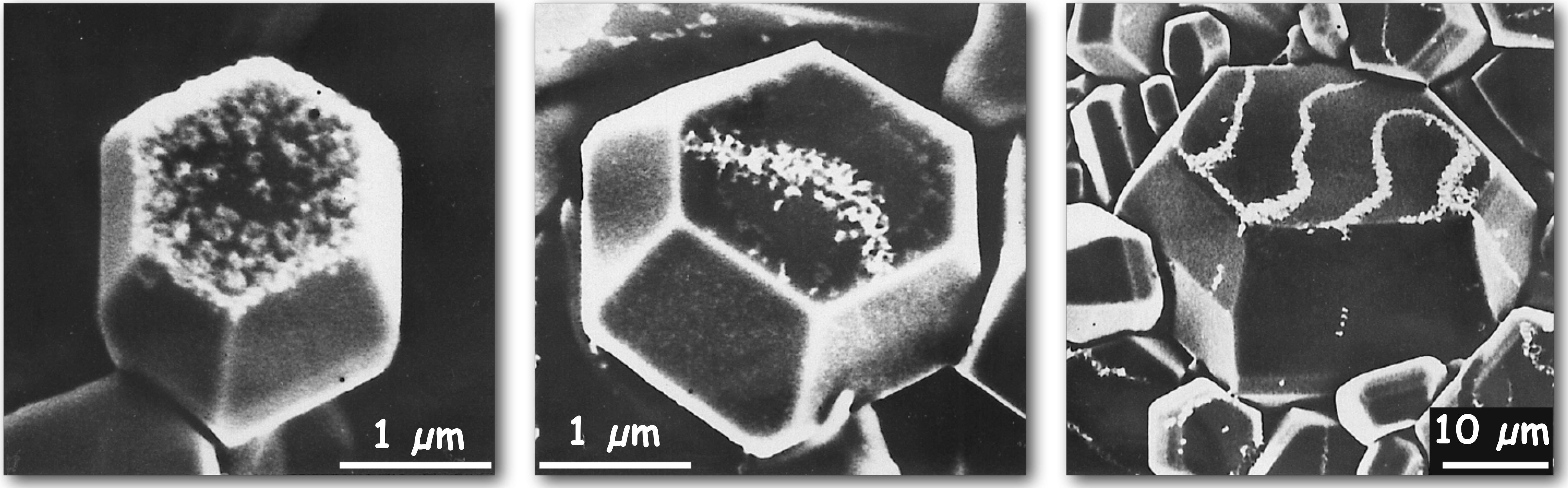
Adding agent

→ Strippable film

→ Imaging in electron microscope

Ba-Ferrite particles (courtesy K. Goto, Sendai)





Dry colloid technique: Static domain observation on rough, 3-dimensional surfaces at high resolution of some 10 nm

1. Bitter technique

Dry colloid technique

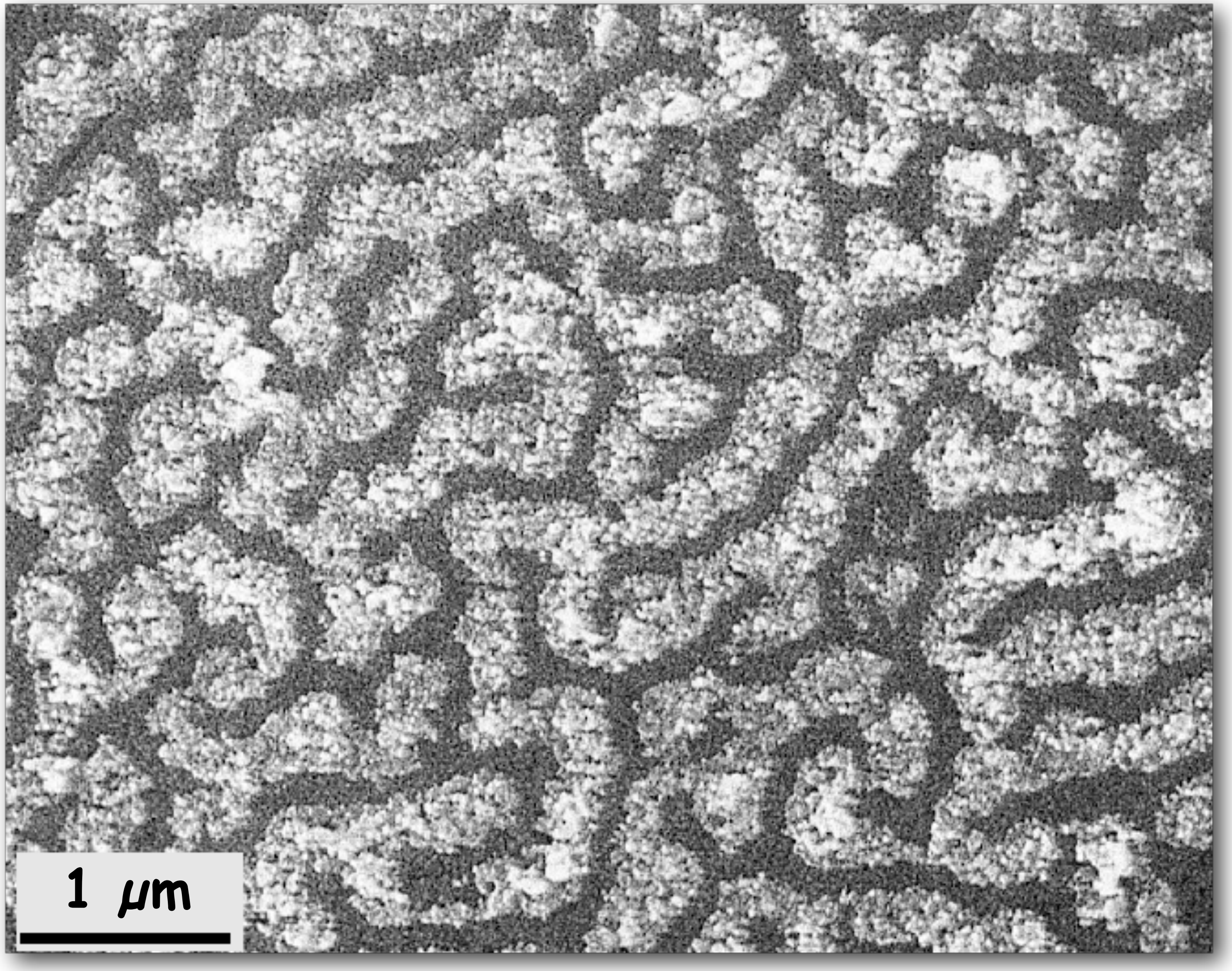
Allowing colloid to dry on surface

Adding agent

→ Strippable film

→ Imaging in electron microscope

CoCr recording medium
(courtesy J. Simsová, Prague)

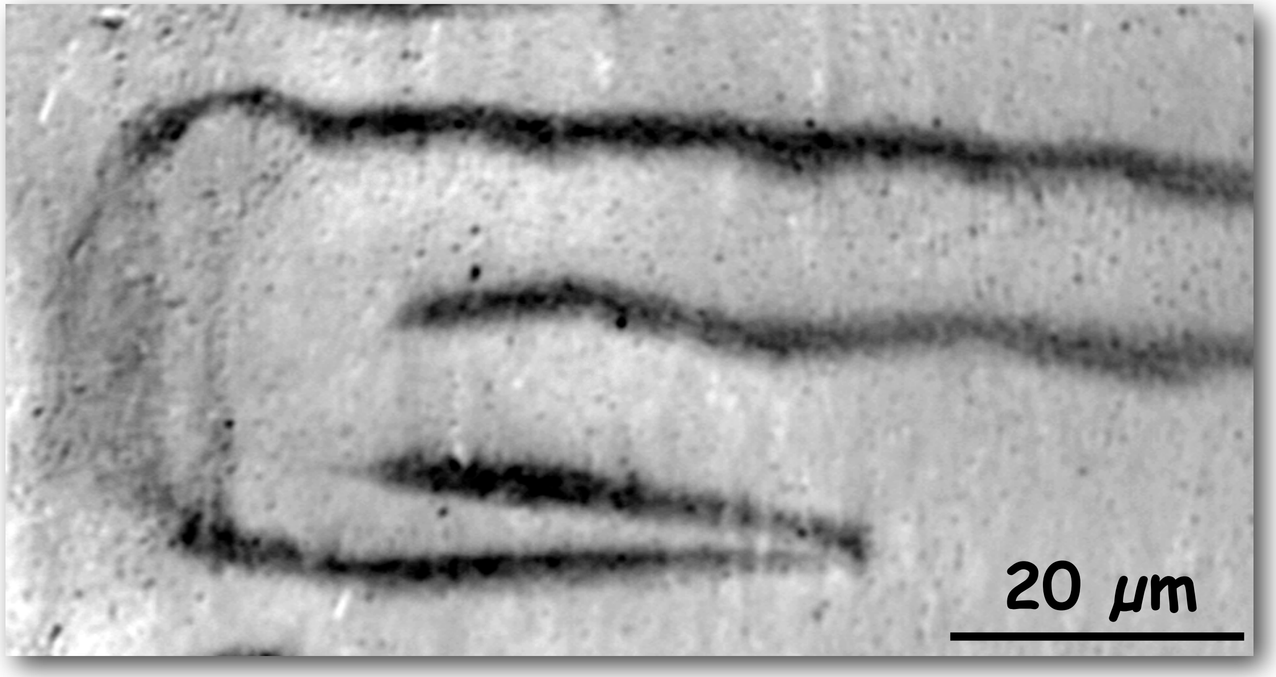


Dry colloid technique: Static domain observation on rough, 3-dimensional surfaces at high resolution of some 10 nm

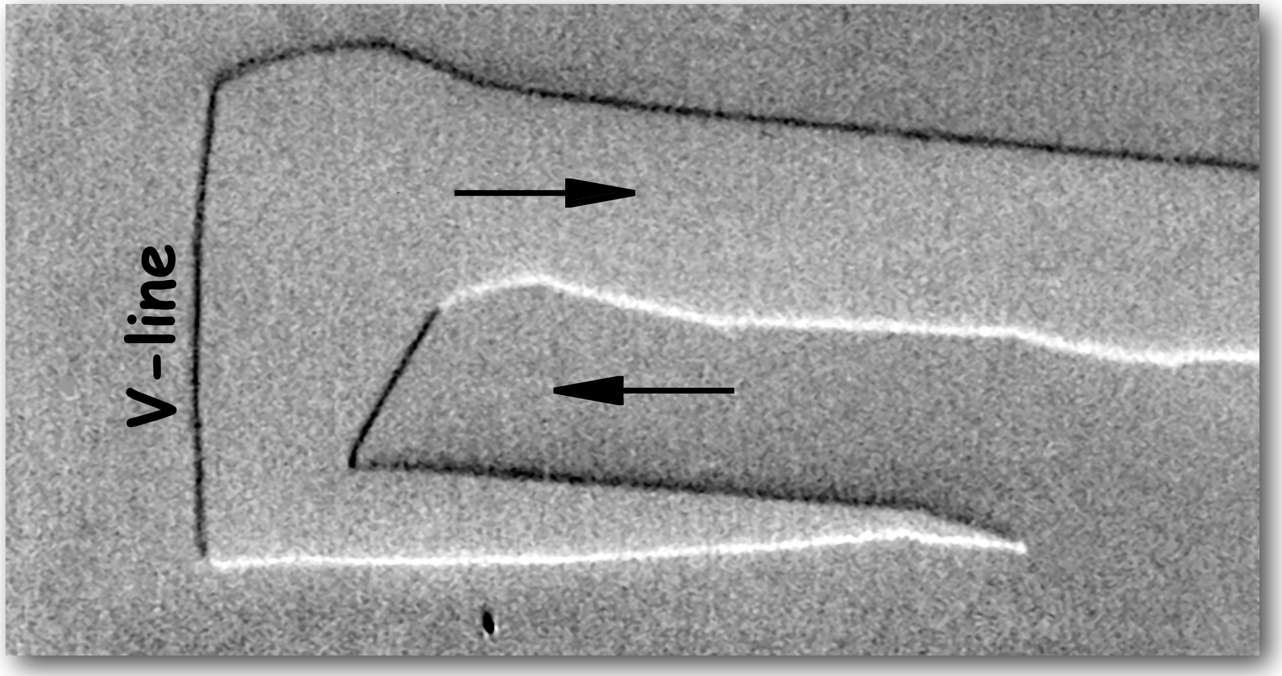
1. Bitter technique

V-lines

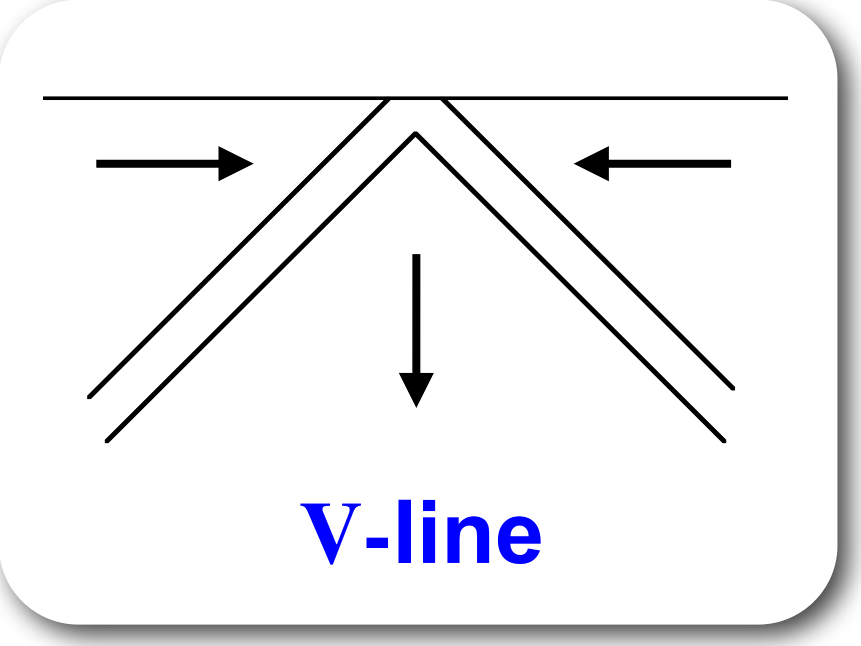
Visible and invisible features



Bitter image



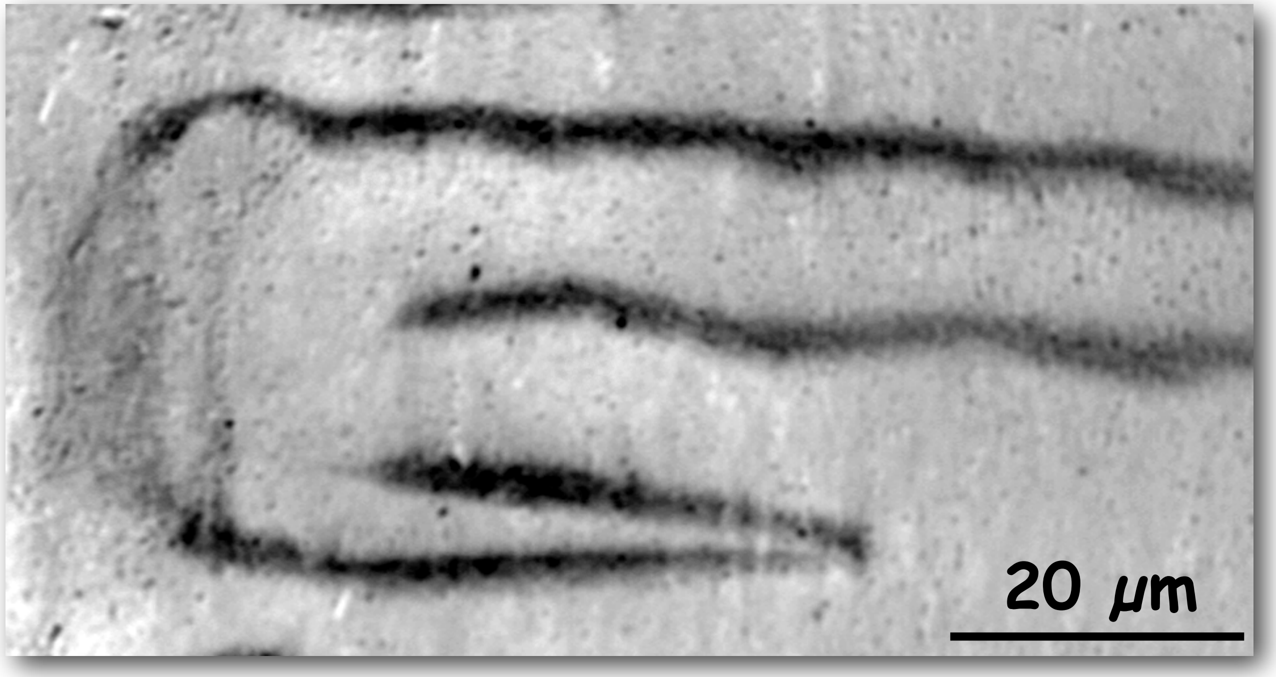
Kerr image



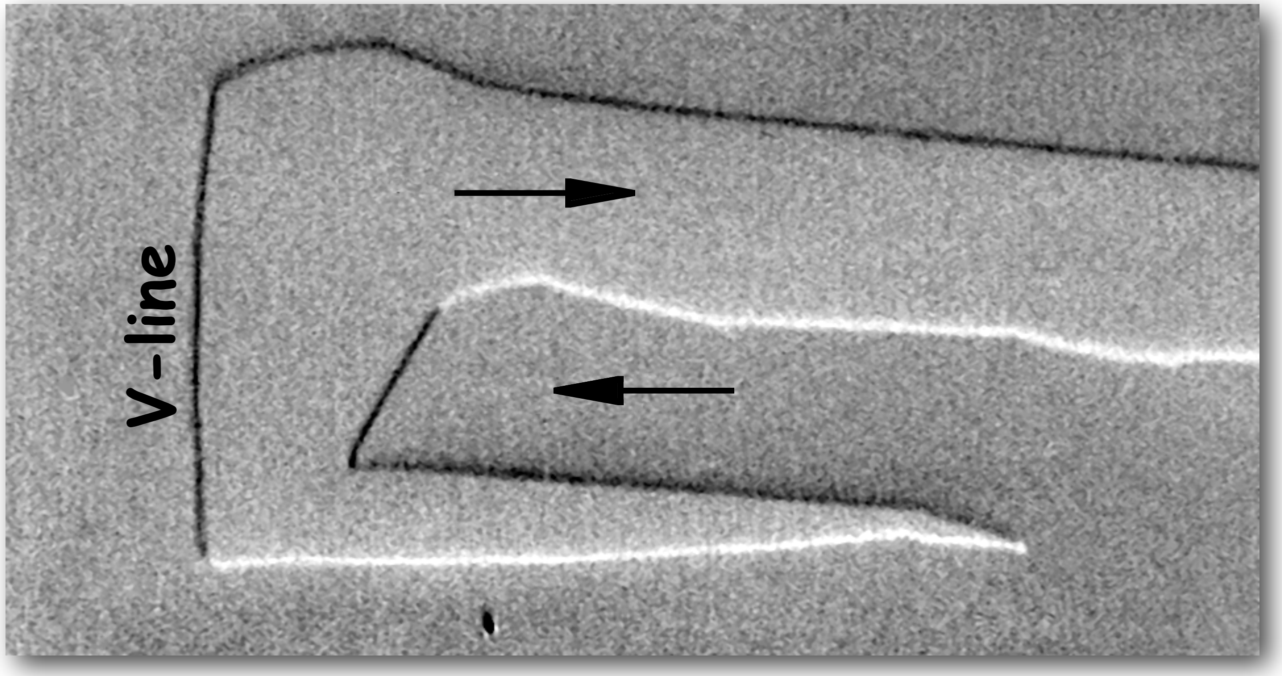
1. Bitter technique

Visible and invisible features

V-lines

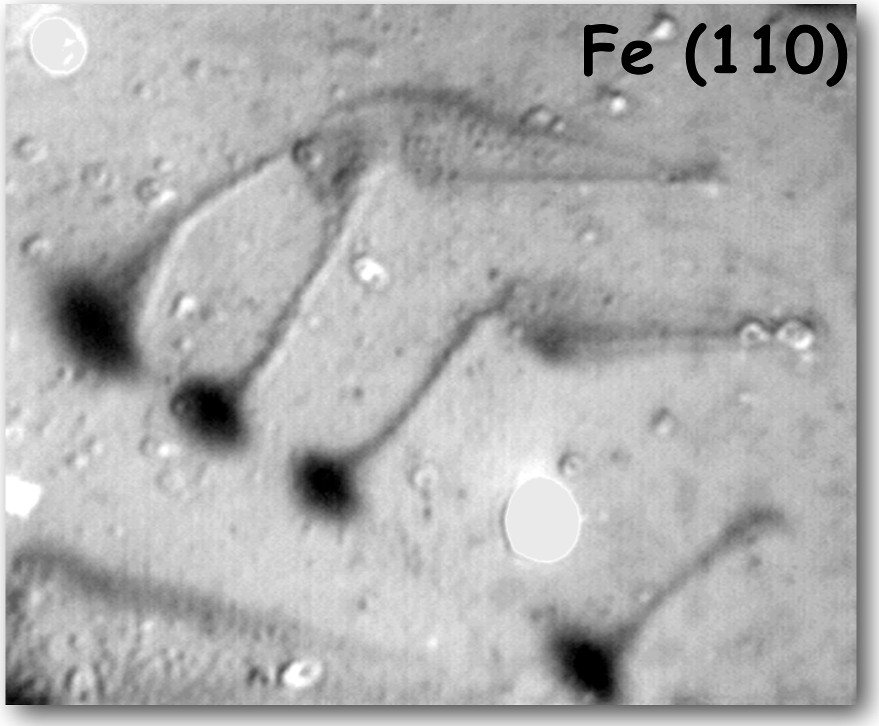


Bitter image

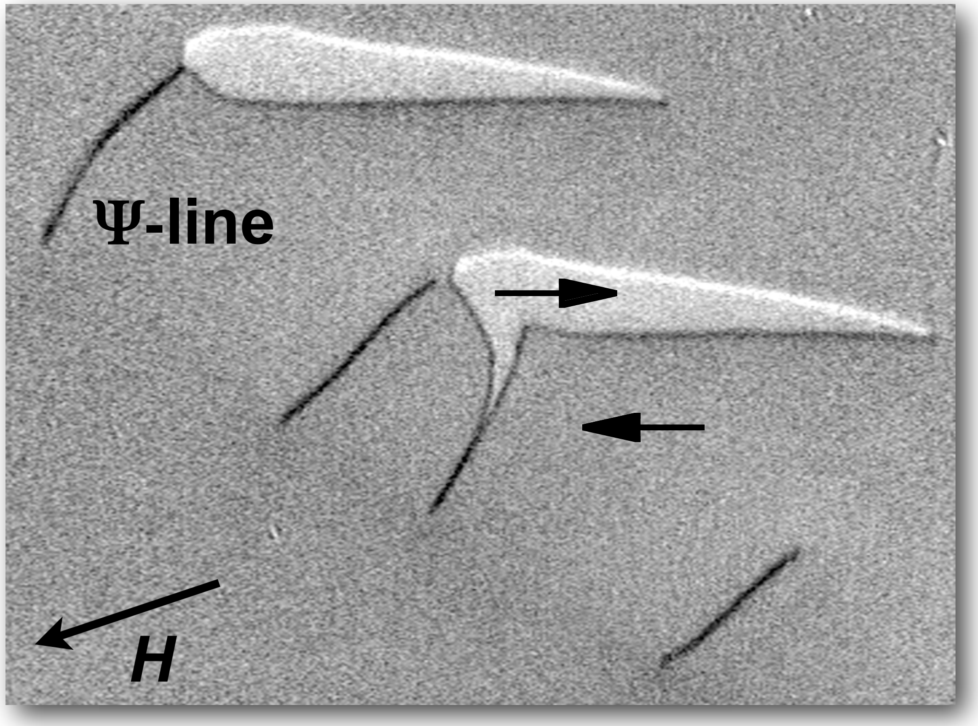


Kerr image

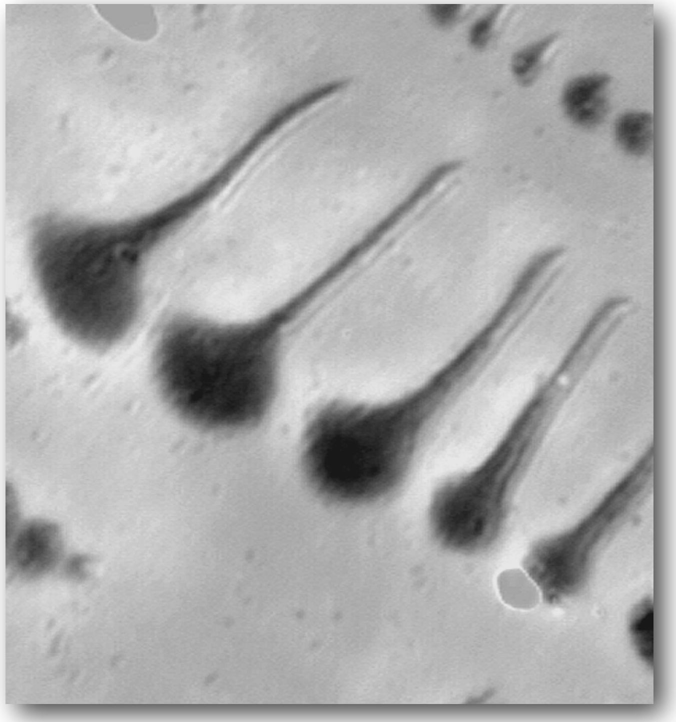
Ψ-lines



Bitter image



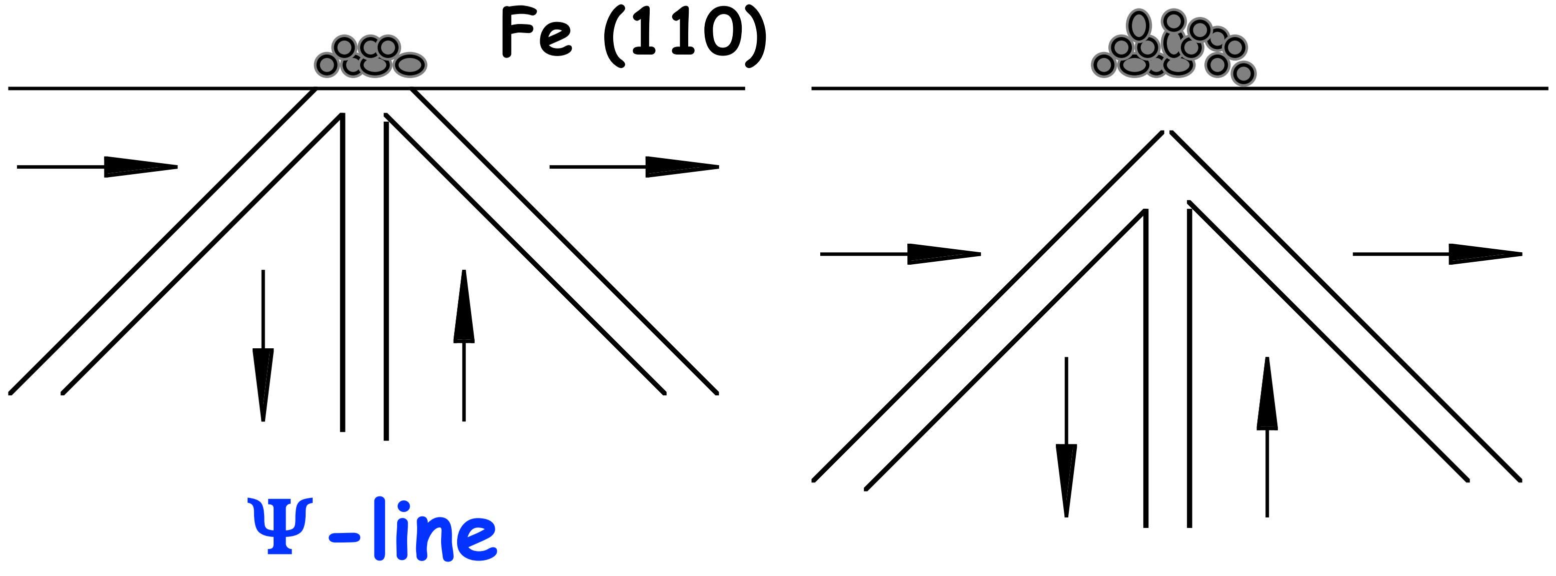
Kerr image



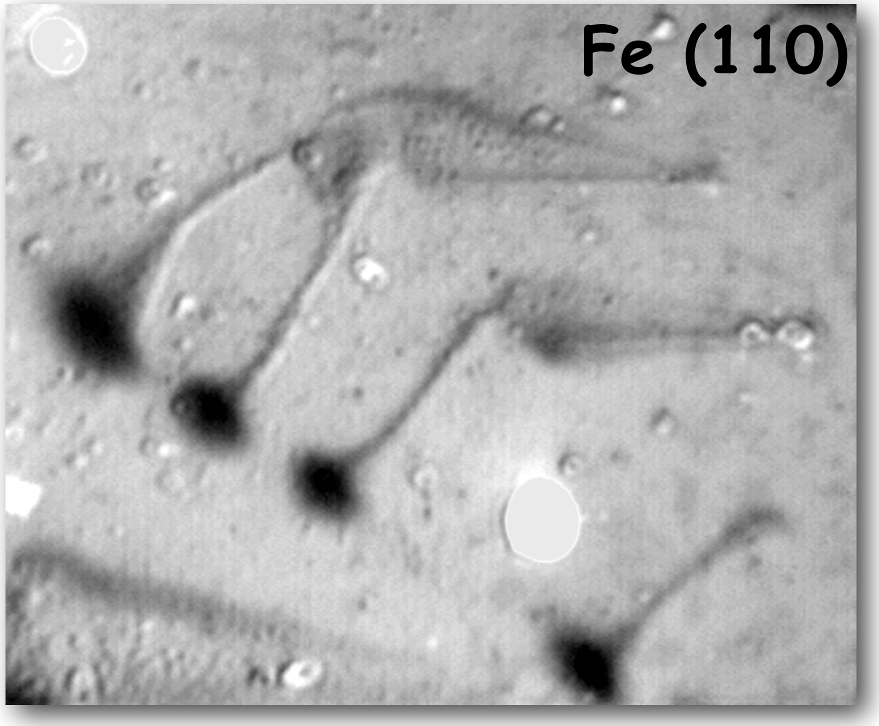
Bitter image

1. Bitter technique

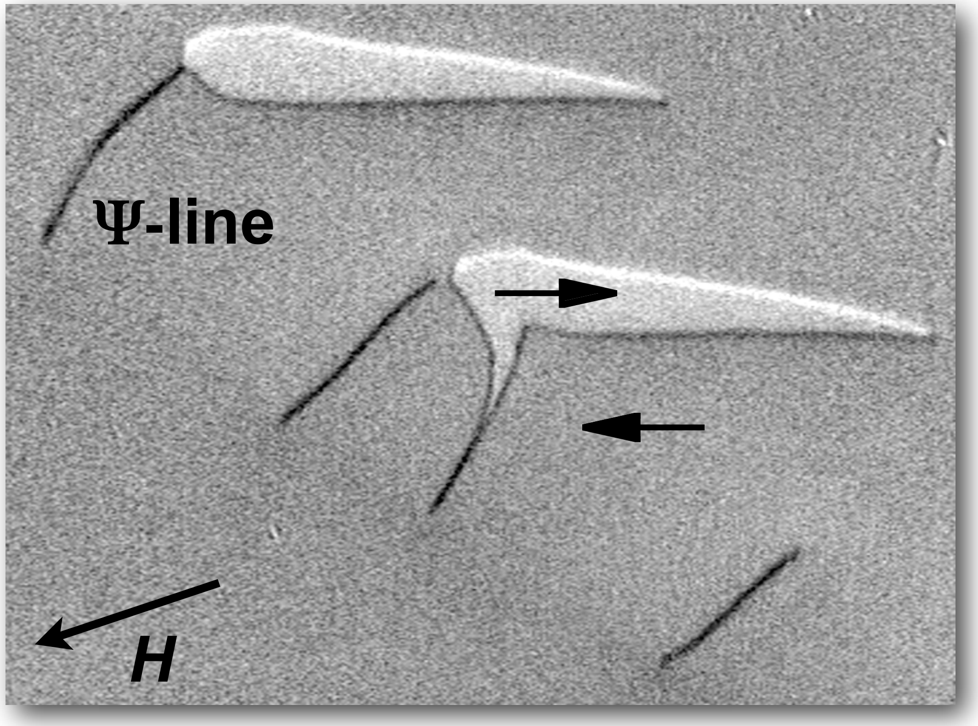
Visible and invisible features



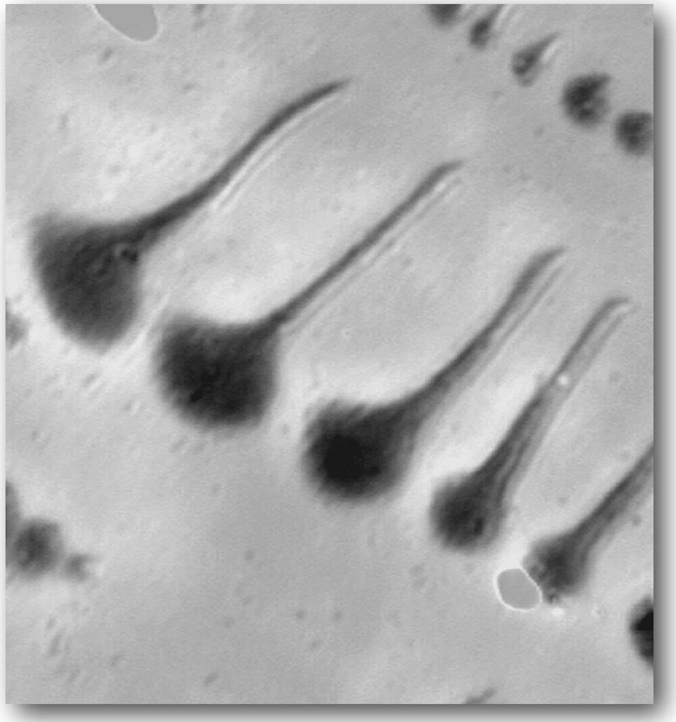
Ψ-lines



Bitter image



Kerr image



Bitter image

1. Bitter technique

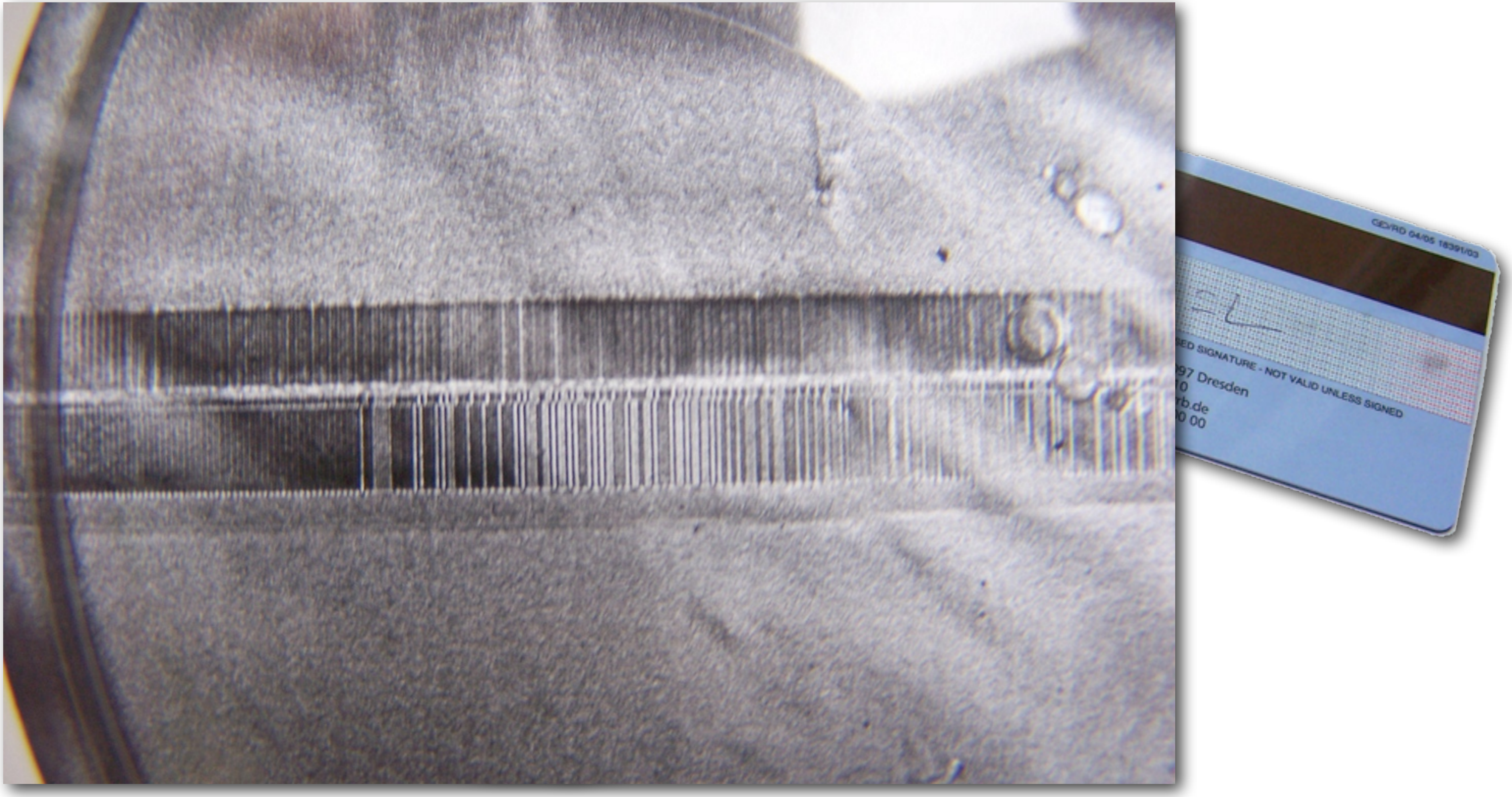
Toner powder emulsion

Laser printer toner + water + household detergent

1. Bitter technique

Toner powder emulsion

Laser printer toner + water + household detergent

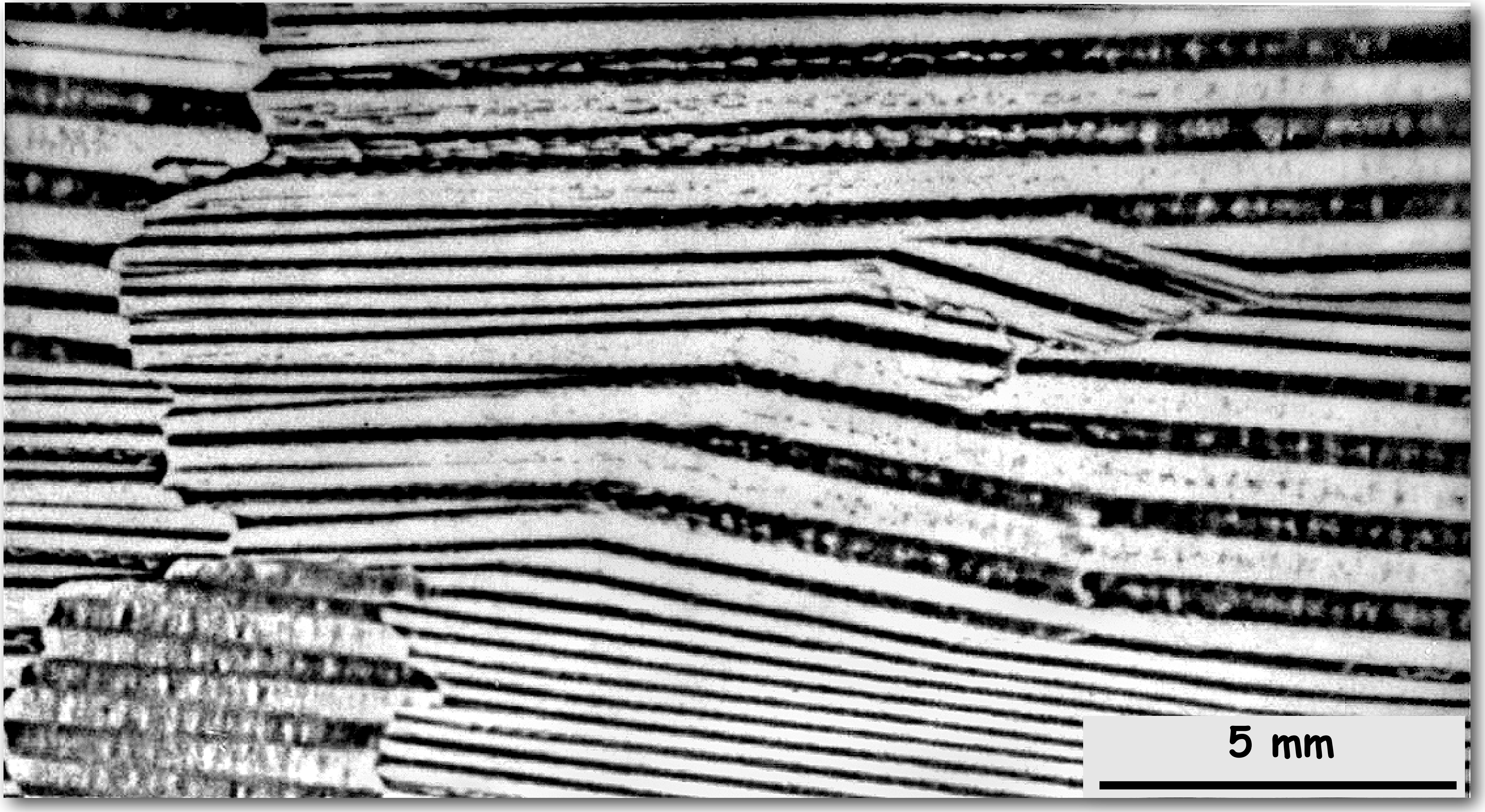


1. Bitter technique

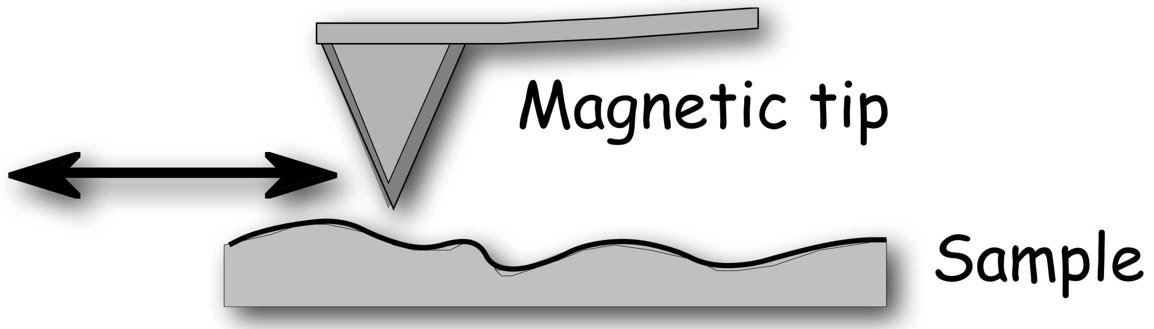
Toner powder emulsion

Laser printer toner + water + household detergent

Transformer steel (courtesy S. Arai, Nippon steel)

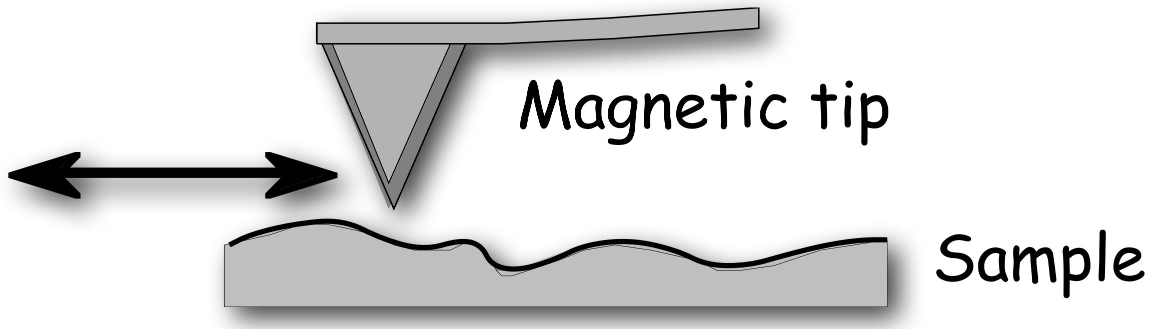


2. Magnetic Force Microscopy



- Spin-off of scanning tunnelling microscope
- Tip-shaped probe at free end of cantilever (= flexible beam). Tip position detected (e.g.) by optical interference between tip of a fibre and cantilever
- 2 modes:
  - Static mode: MFM is run at constant force (equivalent to constant deflection of the cantilever) and the height necessary to obtain this state is used as the imaging information
  - Dynamic mode: Cantilever is operated at frequency close to its mechanical resonance, and change in resonance amplitude or shift in phase due to stray field interaction are detected. Since a magnetic force gradient is equivalent to an additional contribution to the spring constant of the cantilever, profiles of constant force gradient are recorded this way

2. Magnetic Force Microscopy



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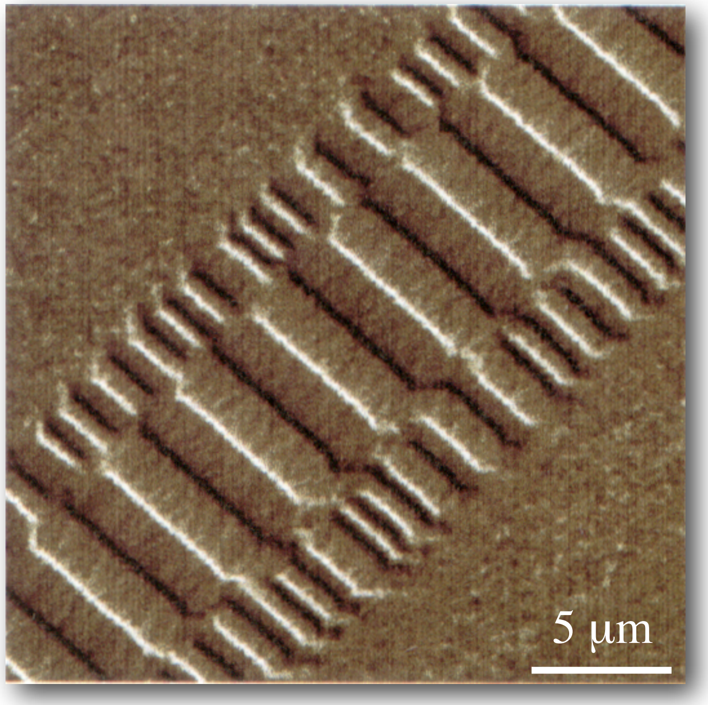
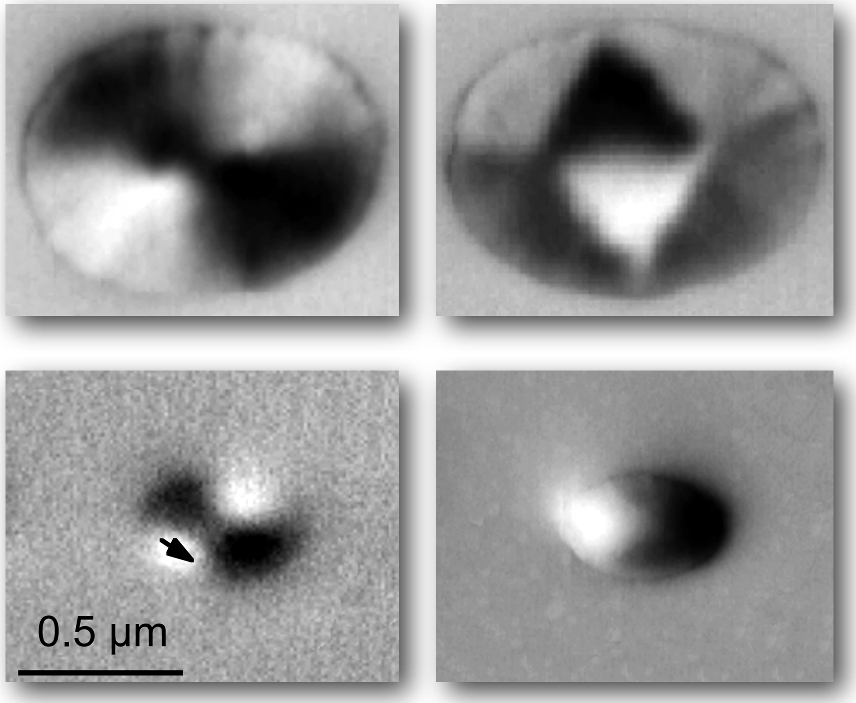
- 2 modes:
  - Static mode: MFM is run at constant force (equivalent to constant deflection of the cantilever) and the height necessary to obtain this state is used as the imaging information

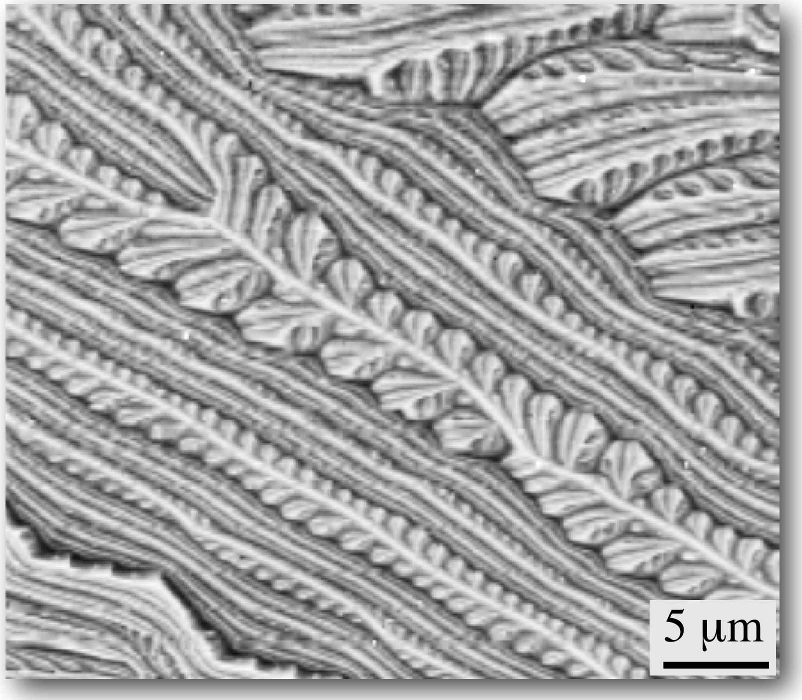
Data track on hard disk



Co elements



(111) Fe surface



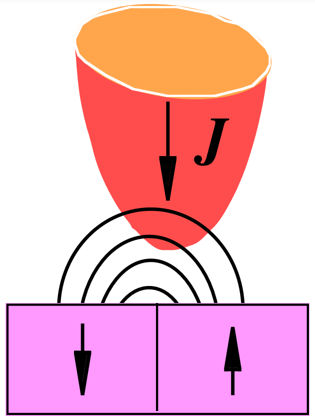
2. Magnetic Force Microscopy

Conventional interpretation of MFM: $F = -\partial E_{\text{inter}}/\partial z$; $\partial F/\partial z = -\partial E^{2}_{\text{inter}}/\partial z^{2}$

$$E_{\text{inter}} = -\int_{\text{tip}} \boldsymbol{J}_{\text{tip}} \cdot \boldsymbol{H}_{\text{sample}}\, dV = -\int_{\text{sample}} \boldsymbol{J}_{\text{sample}} \cdot \boldsymbol{H}_{\text{tip}}\, dV$$

$\boldsymbol{H}_{\text{tip}} = -\,\text{grad}\,\bar{\Phi}_{\text{tip}}$, partial integration

Φ_{tip} = tip potential



tip magnetization = stray field sensor

Alternative interpretation *Hubert, Rave, Tomlinson:* Phys. Stat. Sol. B204, 817 (1997)

$$E_{\text{inter}} = \int_{\text{surface}} \sigma_{\text{sample}}\, \Phi_{\text{tip}}\, dS + \int_{\text{sample}} \rho_{\text{sample}}\, \Phi_{\text{tip}}\, dV$$

$\sigma_{\text{sample}} = \boldsymbol{n} \cdot \boldsymbol{J}_{\text{sample}}$ (surface charge)

$\rho_{\text{sample}} = -\,\text{div}\, \boldsymbol{J}_{\text{sample}}$ (volume charge)

Force

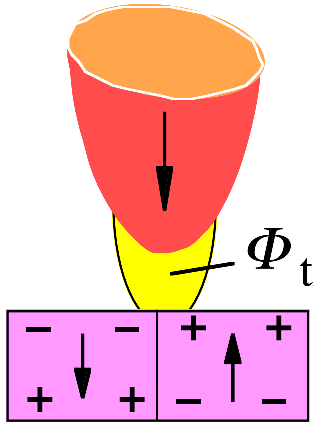
$$F = -\frac{\partial E_{\text{inter}}}{\partial z} = -\int \left(\cancel{\frac{\partial \sigma}{\partial z}}\, \Phi + \sigma \frac{\partial \Phi}{\partial z}\right) dS - \int \left(\cancel{\frac{\partial \rho}{\partial z}}\, \Phi + \rho \frac{\partial \Phi}{\partial z}\right) dV$$

for weak interaction

$$\approx -\int \sigma \frac{\partial \Phi}{\partial z}\, dS - \int \rho \frac{\partial \Phi}{\partial z}\, dV$$

Force gradient

$$\frac{\partial F}{\partial z} = -\frac{\partial E^{2}_{\text{inter}}}{\partial z^{2}} \approx -\int \sigma \frac{\partial^{2} \Phi}{\partial z^{2}}\, dS - \int \rho \frac{\partial^{2} \Phi}{\partial z^{2}}\, dV$$



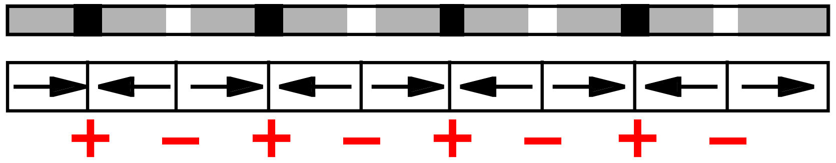
tip potential = charge sensor

In the limit of weak interaction: MFM is *Charge Microscopy*

2. Magnetic Force Microscopy

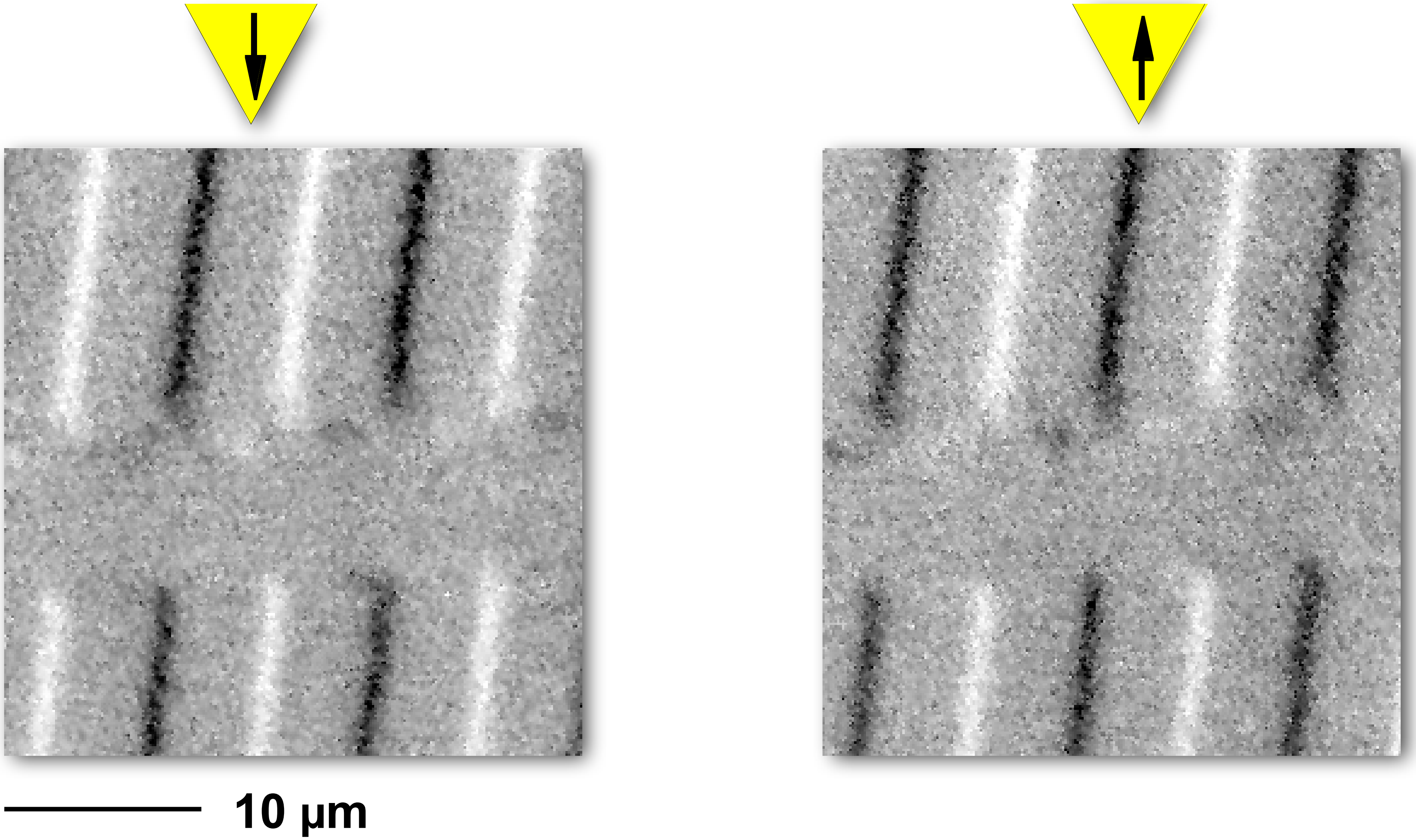
Charge contrast

Longitudinal data track



Tip down

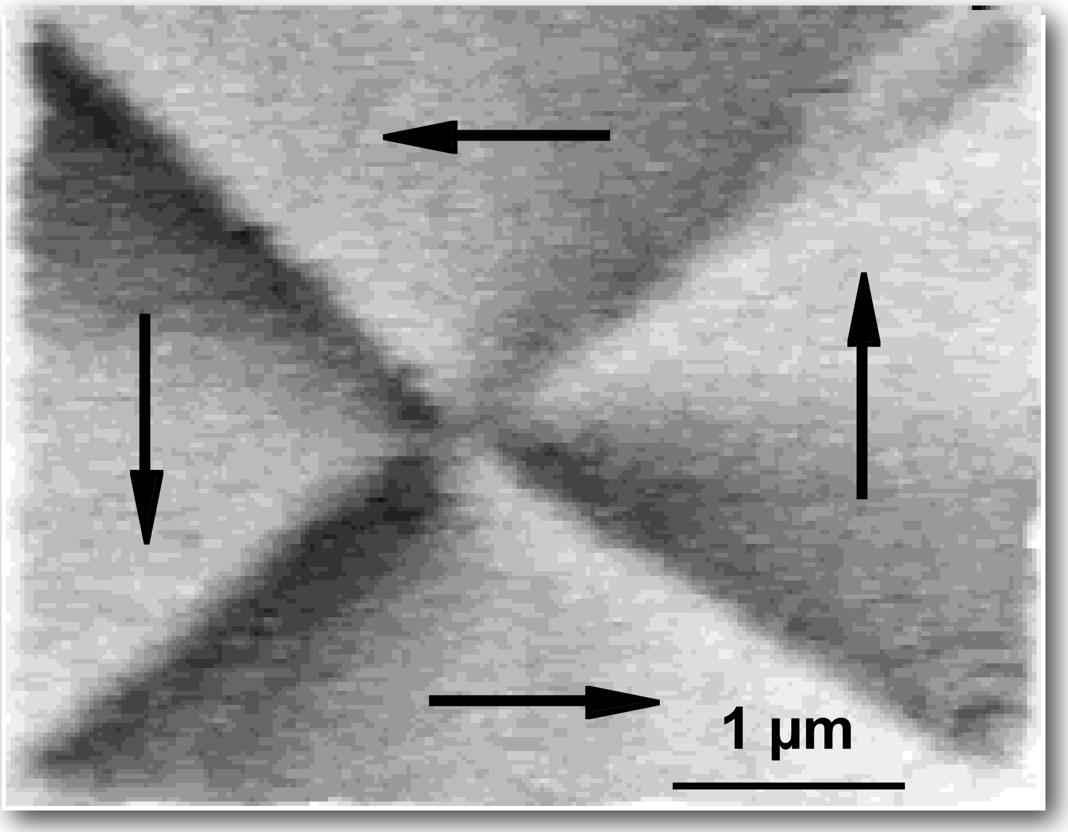
Tip up



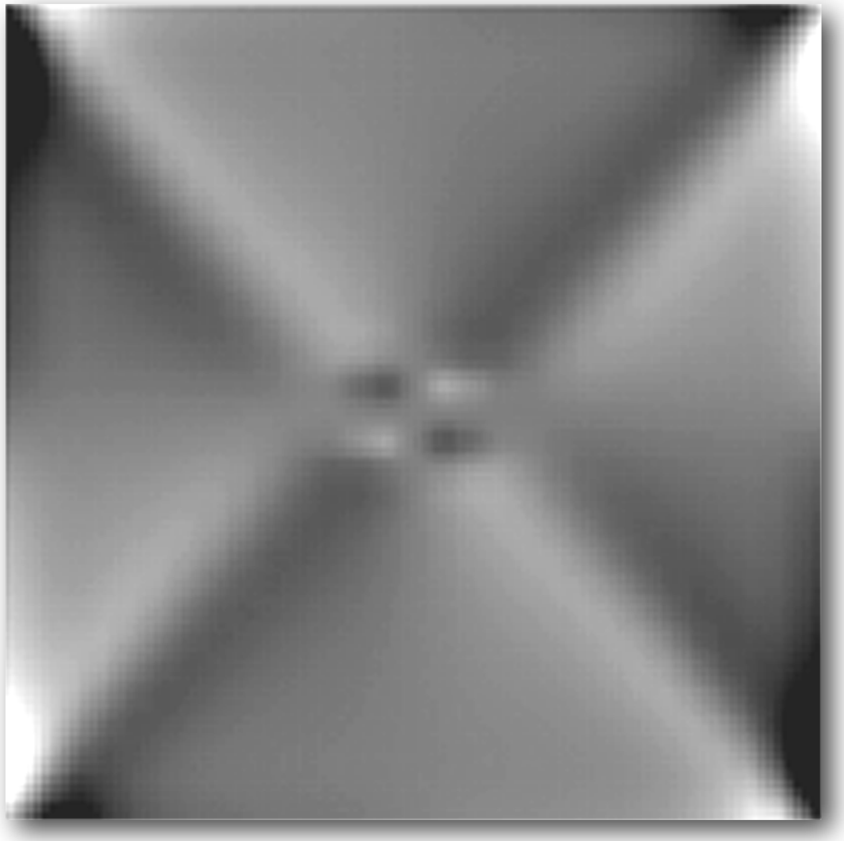
2. Magnetic Force Microscopy

Charge contrast

MFM image

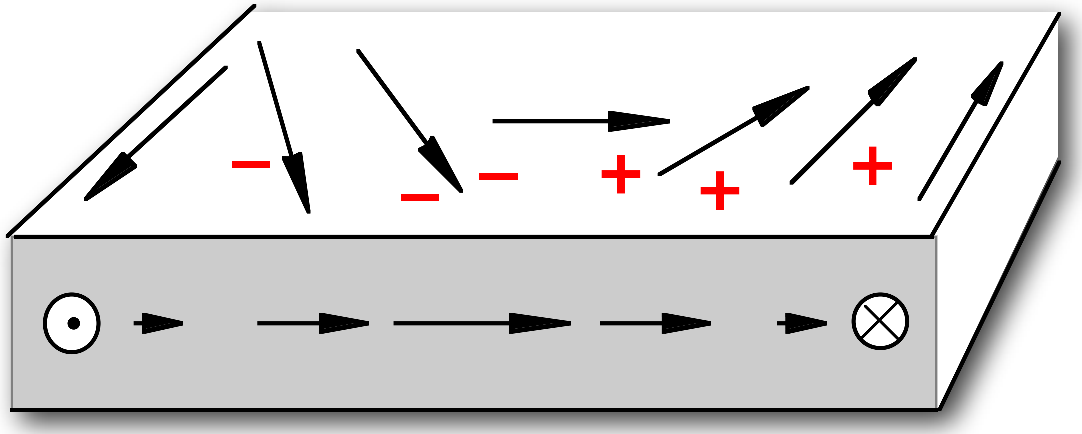


FeTnN element (30 nm)
(courtesy J. Miltat)



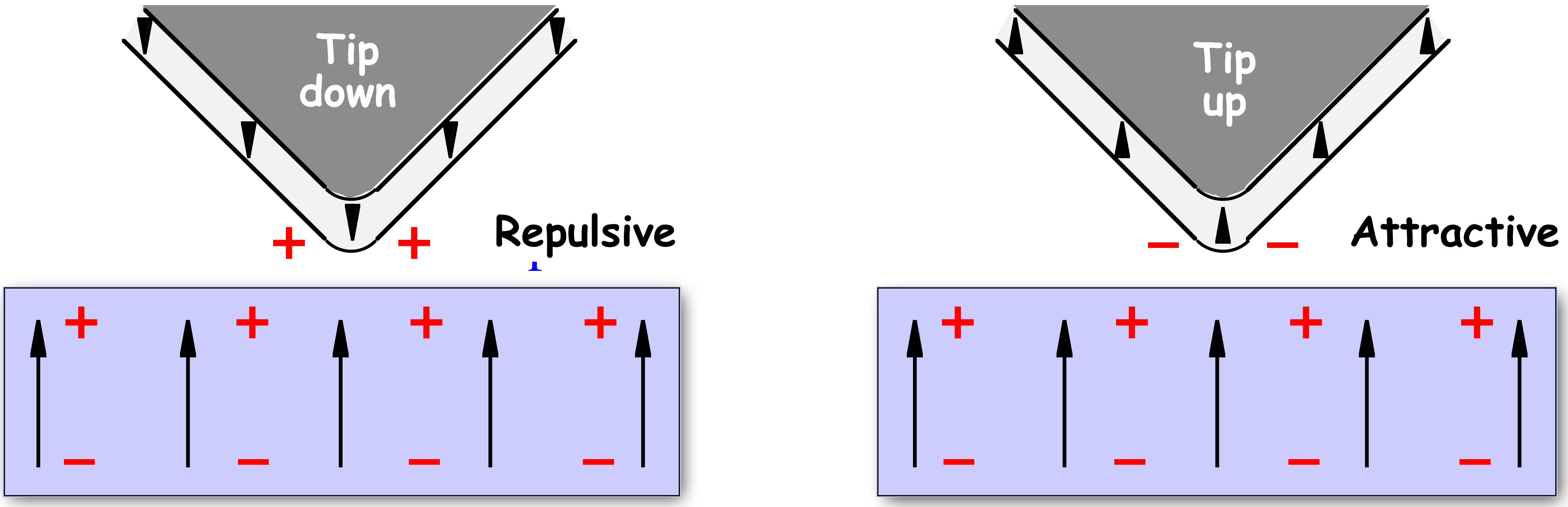
Simulated charge distribution

A. Hubert, W. Rave, S. Tomlinson:
Phys. Stat. Sol. B 204, 817 (1997)



2. Magnetic Force Microscopy

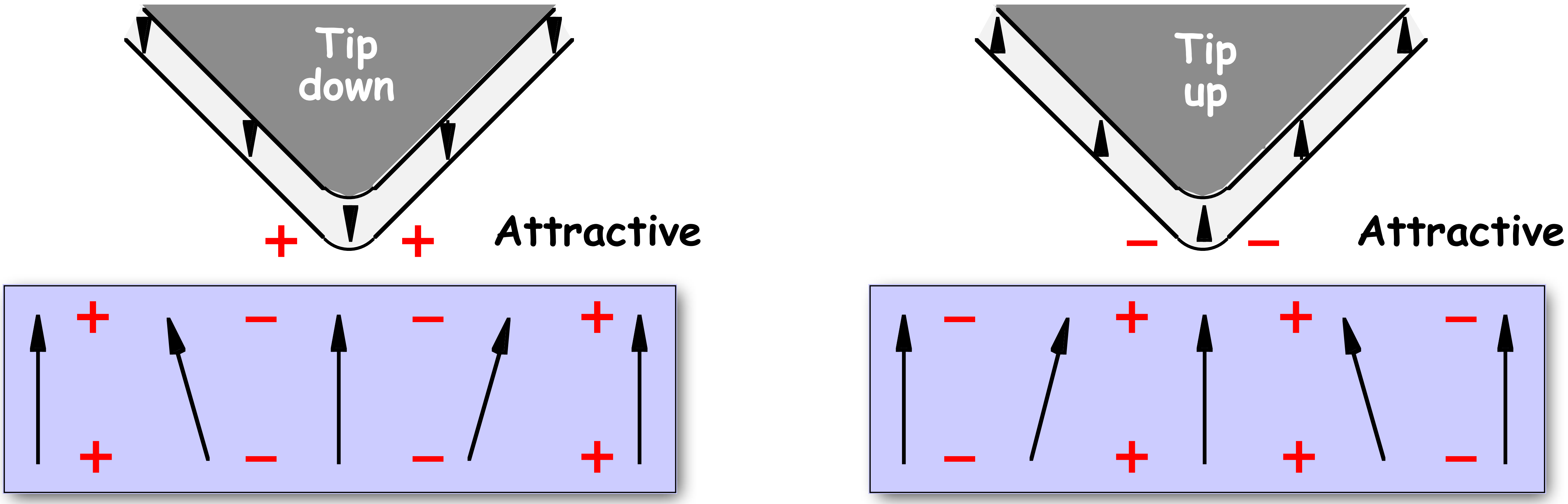
Charge contrast



Charge contrast is inverted when tip polarity is inverted

2. Magnetic Force Microscopy

Induced charges: susceptibility contrast



→ Tip induces charges of opposite polarity in each case

→ Always attractive interaction (independent of tip polarity)

→ Strength of attraction depends on local susceptibility

2. Magnetic Force Microscopy

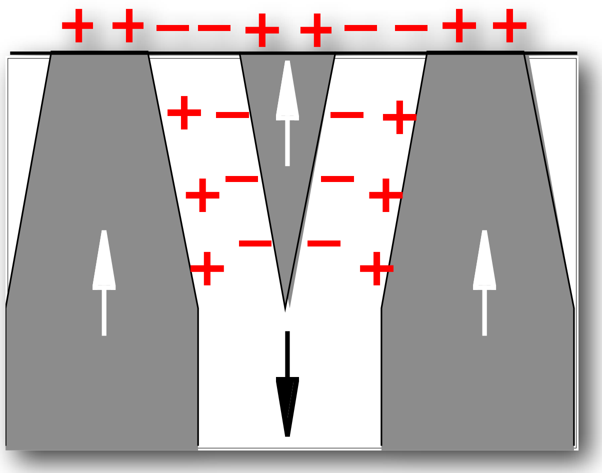
Charge & susceptibility contrast

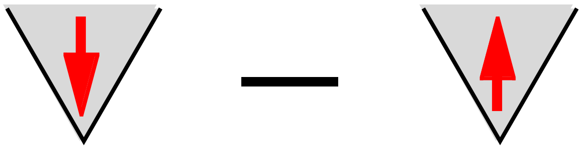
- Charge contrast is inverted by inversion of tip magnetization,
- Suscept. contrast: not inverted
- → Separation of charge and susceptibility contrast by difference and sum images

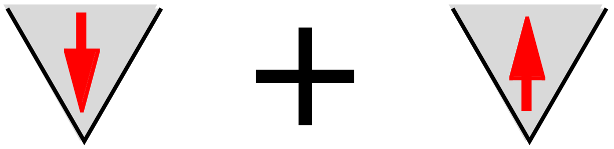
E. Zueco et al.:
JMMM 190, 42 (1998)

□ Stronger attraction

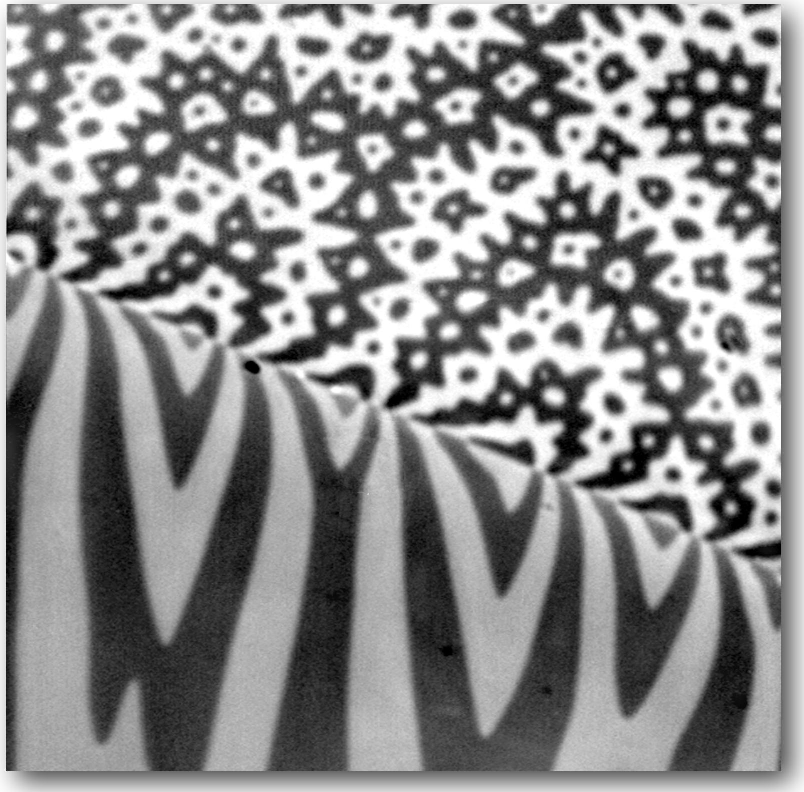
■ Weaker attraction



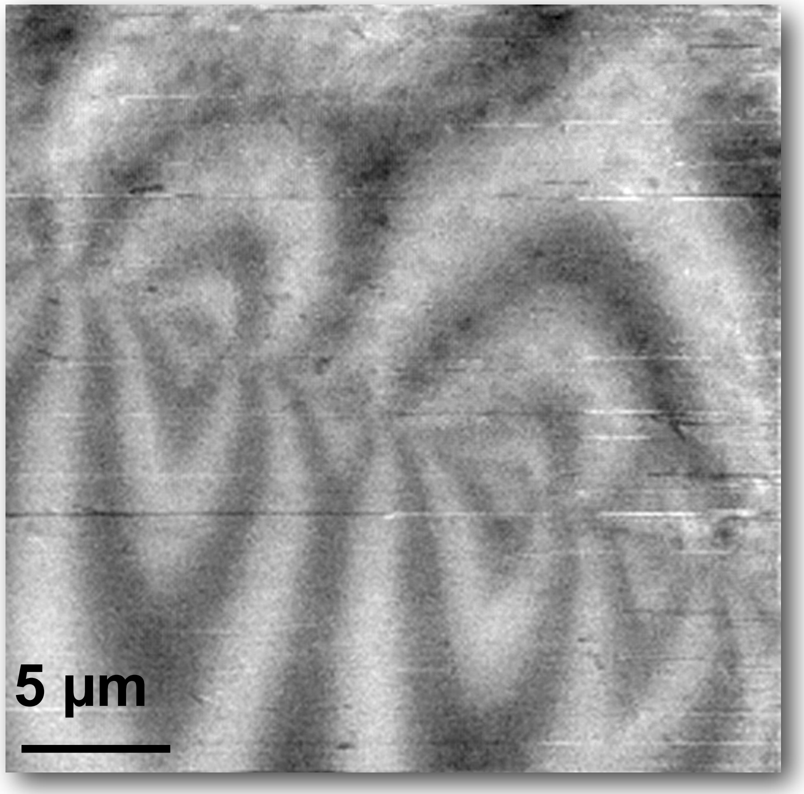




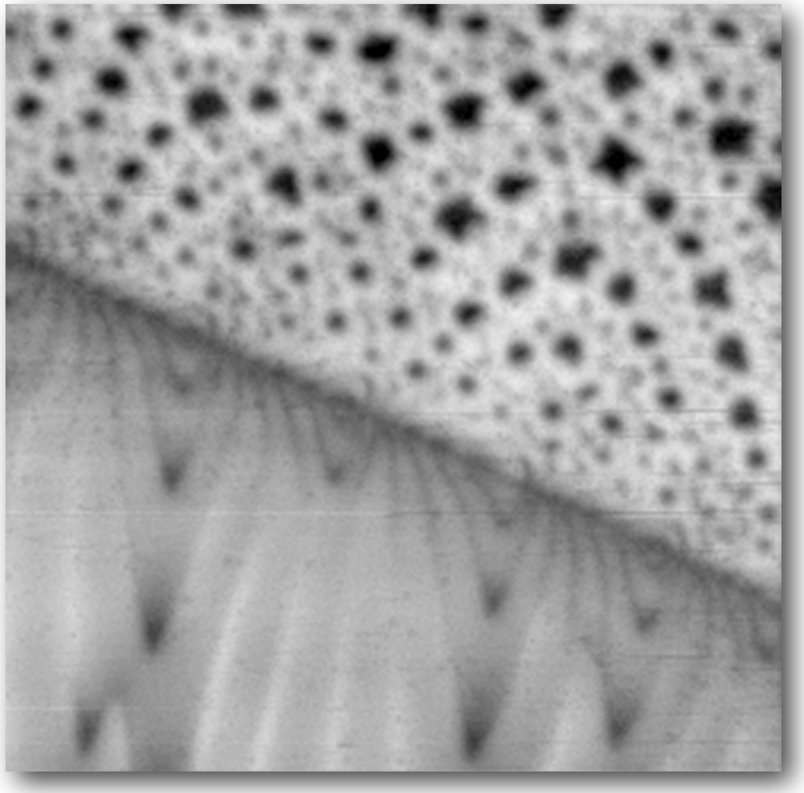
NdFeB twin boundary



Kerr image



MFM: charge contrast

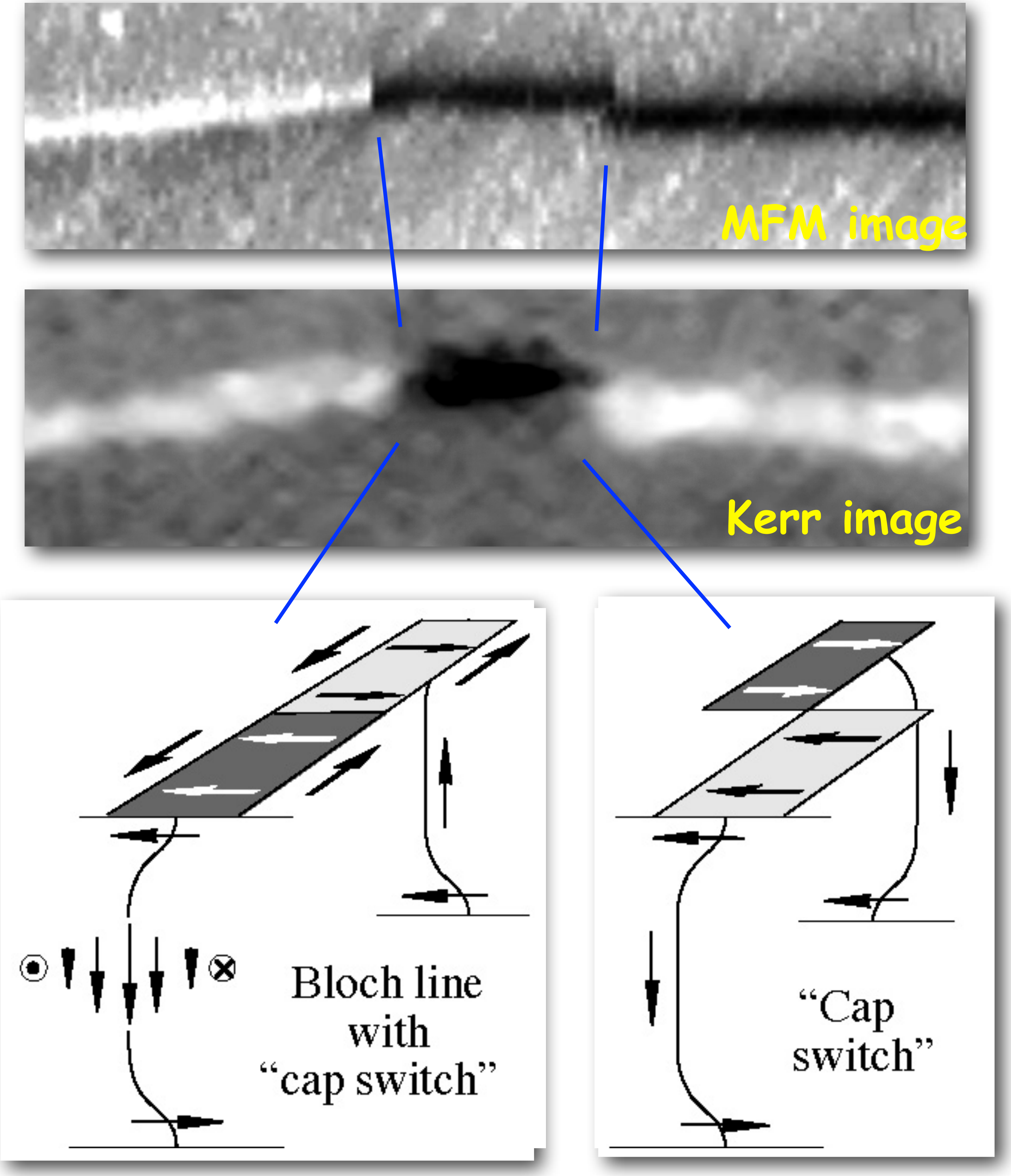


Susceptibility contrast

2. Magnetic Force Microscopy

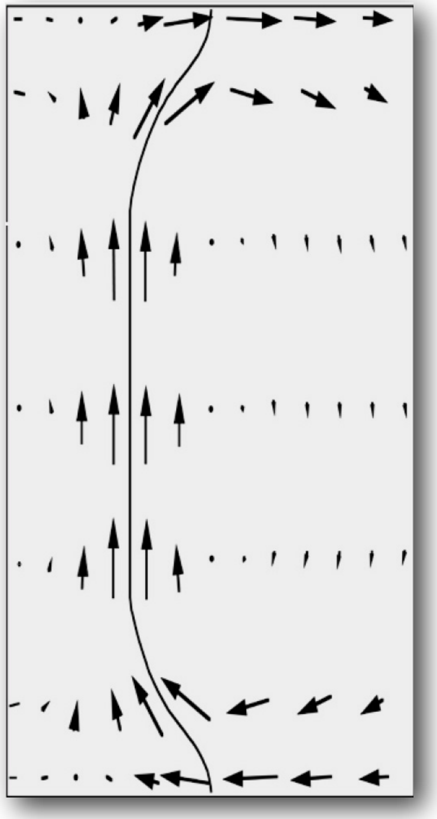
Depth sensitivity

Fe whisker



MFM:
sensitive to interior
magnetization of the wall

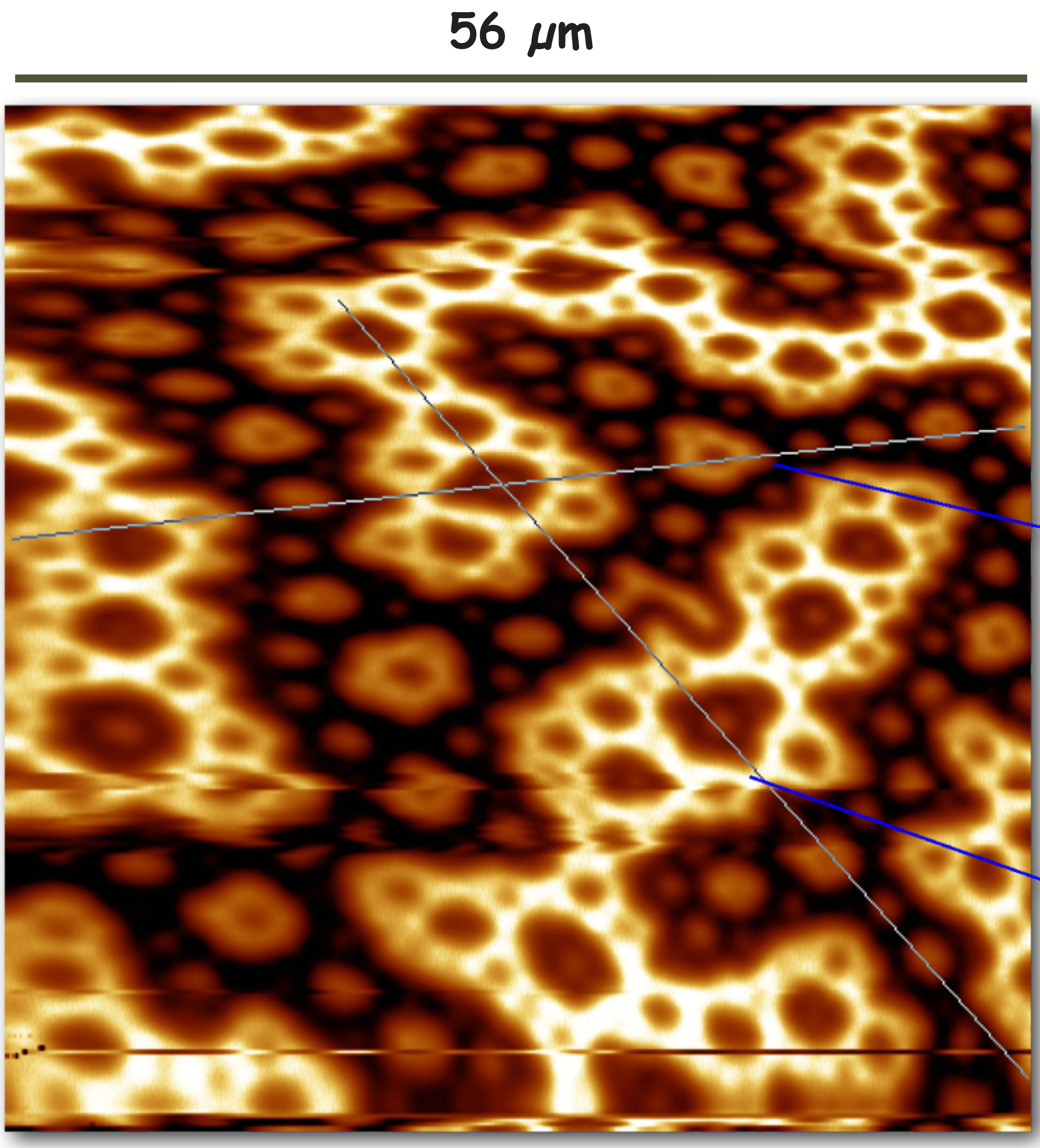
Kerr:
sensitive to surface
magnetization



Asymmetric
vortex wall

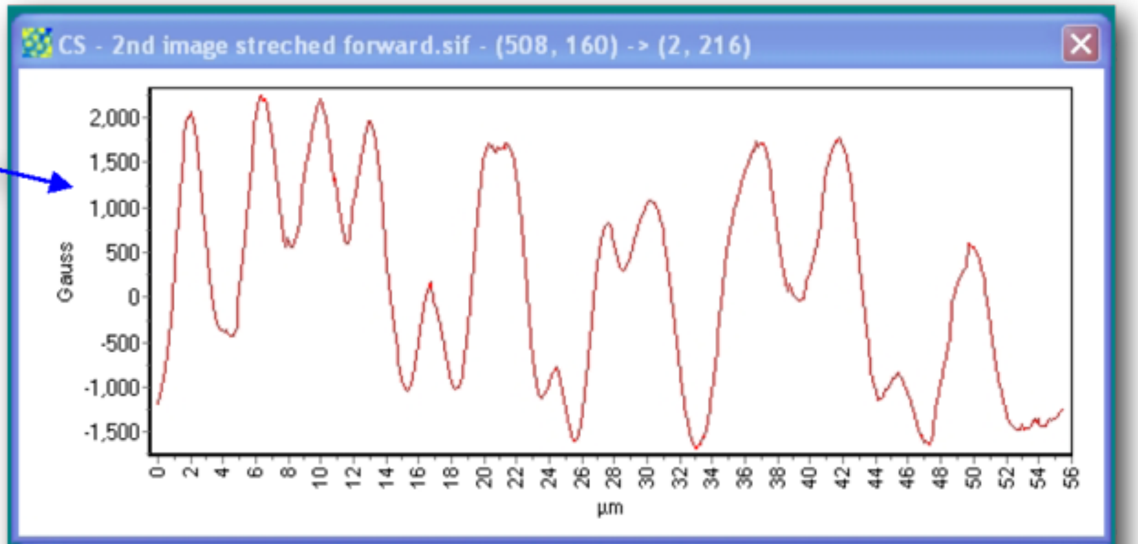
E. Zueco et al.,
JMMM 196, 115 (1999)

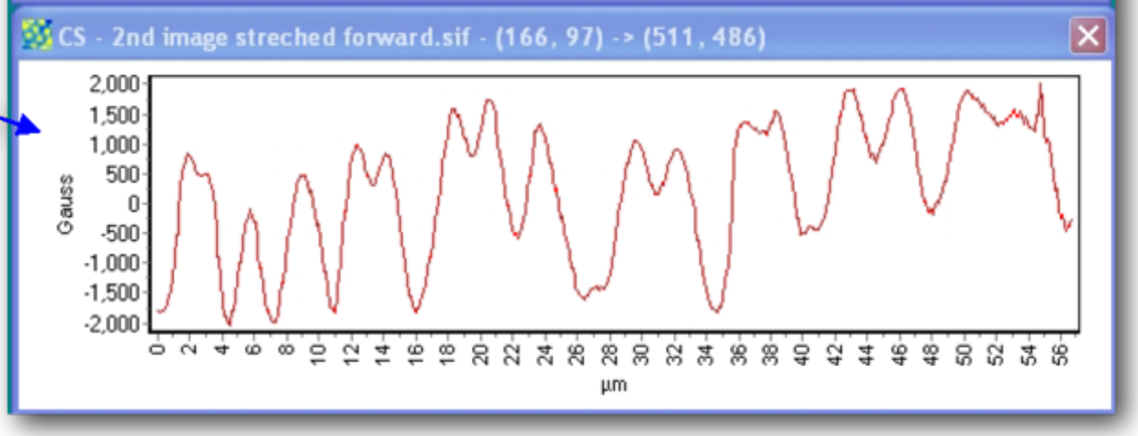
3. Hall-Probe Microscopy



NdFeB crystal

Micro-Hallprobe
is scanned across surface


CS - 2nd image streched forward.sif - (508, 160) -> (2, 216)


CS - 2nd image streched forward.sif - (166, 97) -> (511, 486)

Together with
J. McCord and U. Wolff, IFW

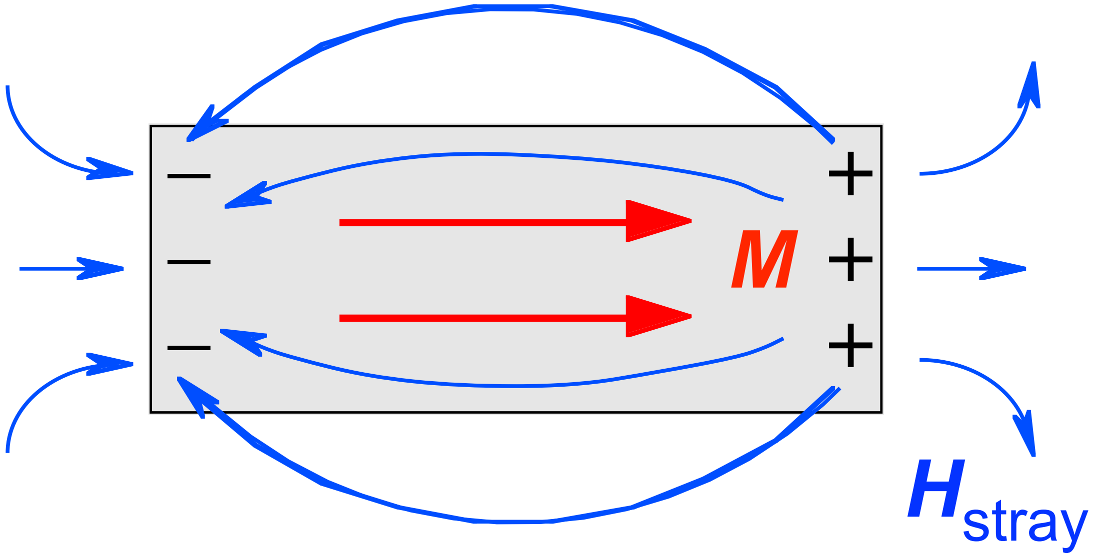
Sensitivity of imaging methods

$\boldsymbol{B} = \mu_0 (\boldsymbol{H} + \boldsymbol{M}) \quad (\boldsymbol{H} = \boldsymbol{H}_{ext} + \boldsymbol{H}_{stray})$

$\text{div}\, \boldsymbol{B} = 0$

$\downarrow$

$\text{div}\, \boldsymbol{H}_{stray} = -\, \text{div}\, \boldsymbol{M}$



- Sensitive to $\boldsymbol{H}_{stray}$
  1. Bitter technique
  2. Magnetic force microscopy
  3. Hall probe microscopy
- Sensitive to $\boldsymbol{M}$
  - Magneto-optical microscopy
  - X-ray spectroscopy
  - Polarized electrons (SEMPA, SPT)
- Sensitive to $\boldsymbol{B}$
  - Transmission electron microscopy
- Sensitive to distortions
  - X-ray, neutron scattering

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$\downarrow$

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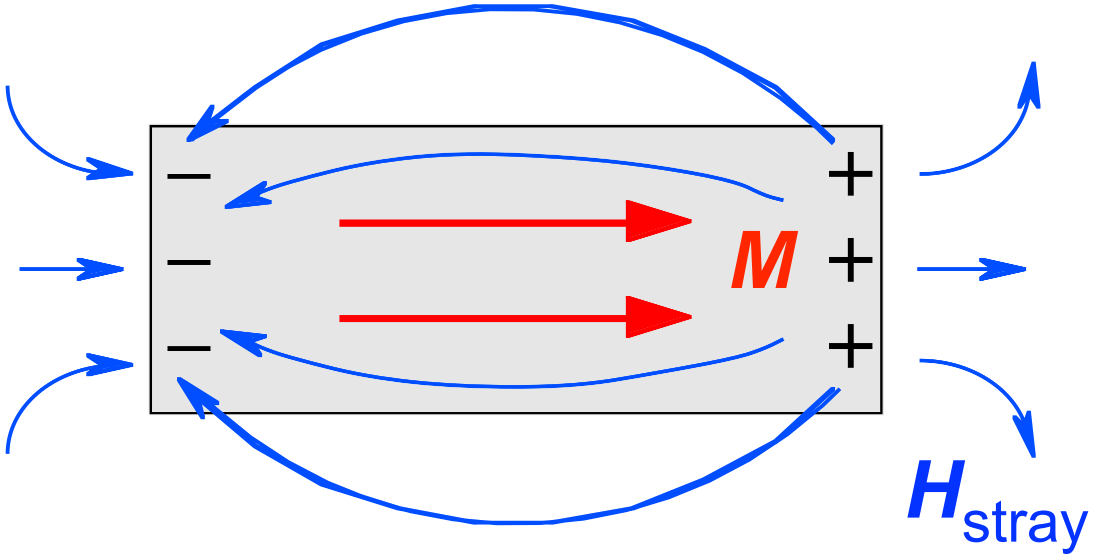
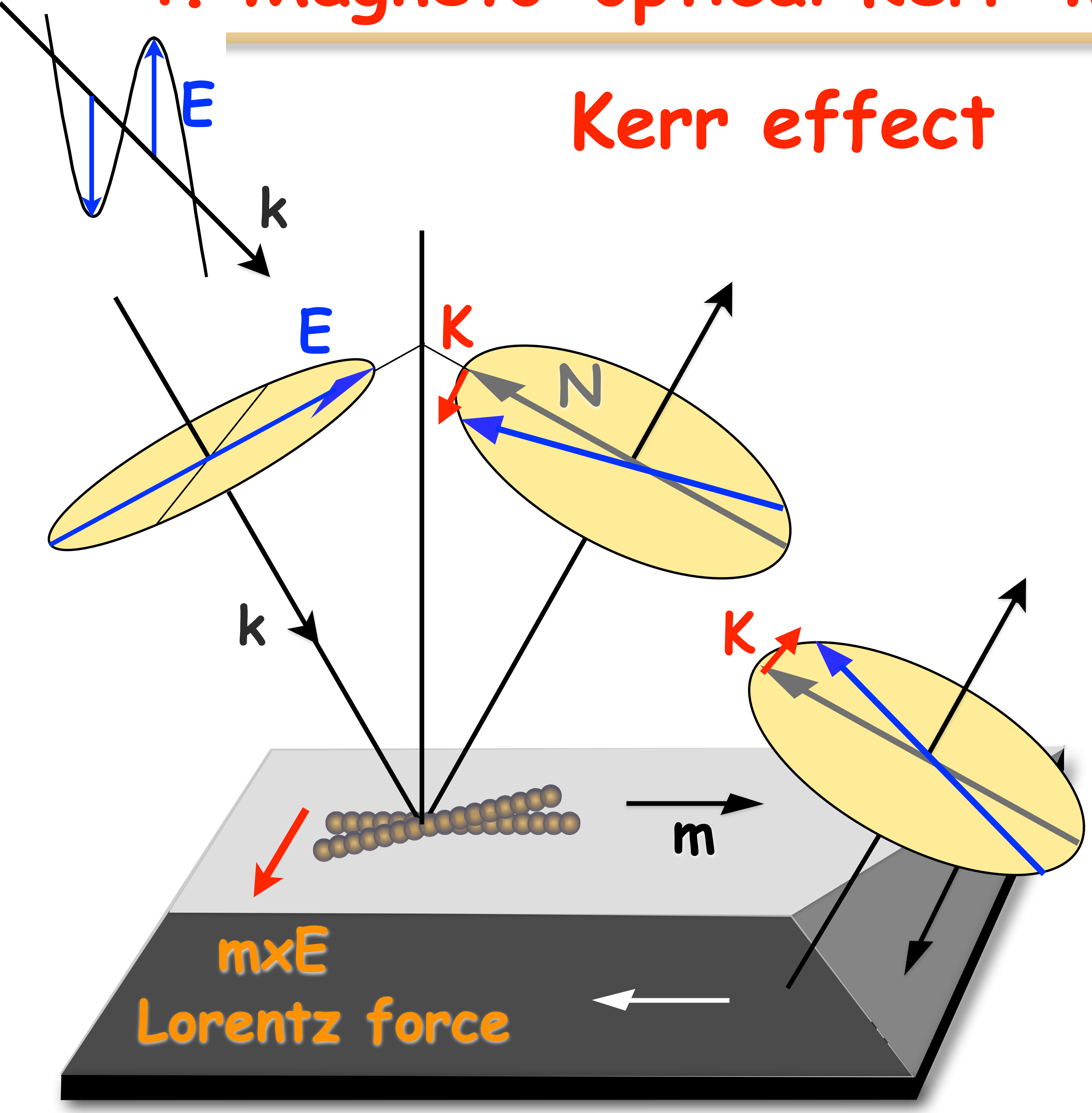
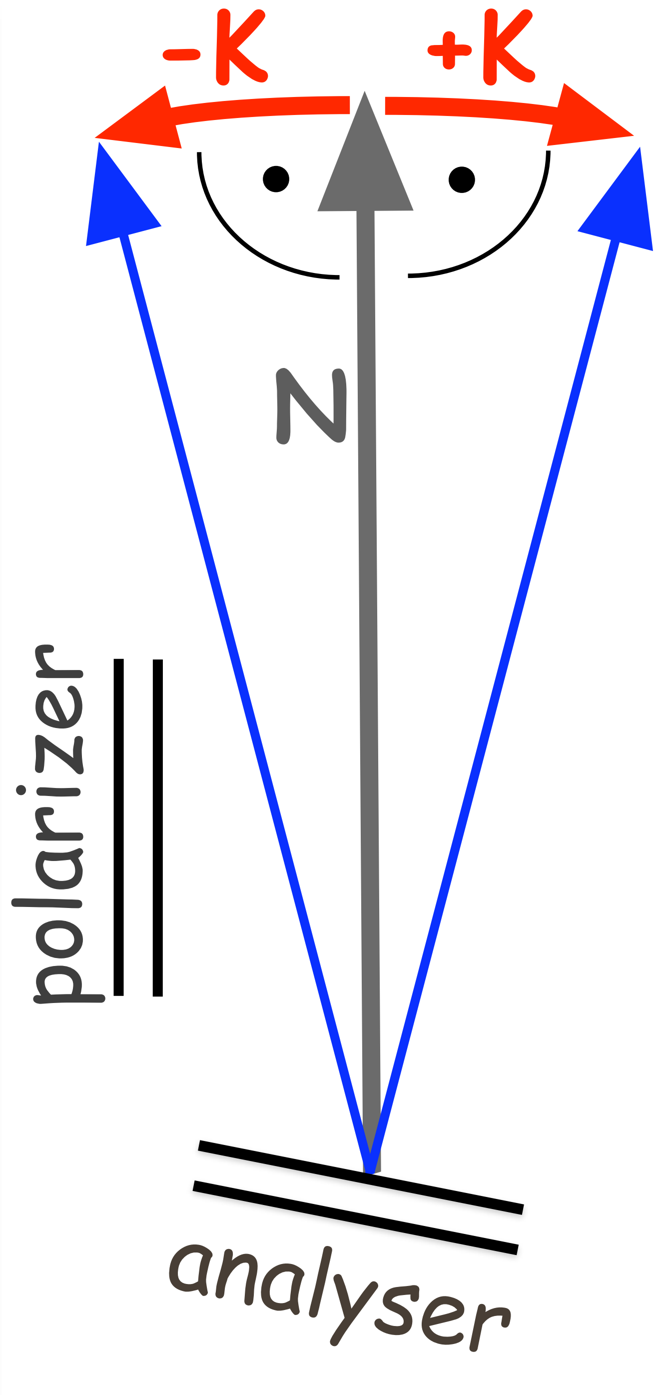


- Sensitive to $\boldsymbol{H}_{stray}$
  1. Bitter technique
  2. Magnetic force microscopy
  3. Hall probe microscopy
- Sensitive to $\boldsymbol{M}$
  4. Magneto-optical microscopy
  5. X-ray spectroscopy
  6. Polarized electrons (SEMPA, SPT)
- Sensitive to $\boldsymbol{B}$
  - Transmission electron microscopy
- Sensitive to distortions
  - X-ray, neutron scattering

4. Magneto-optical Kerr-Microscopy

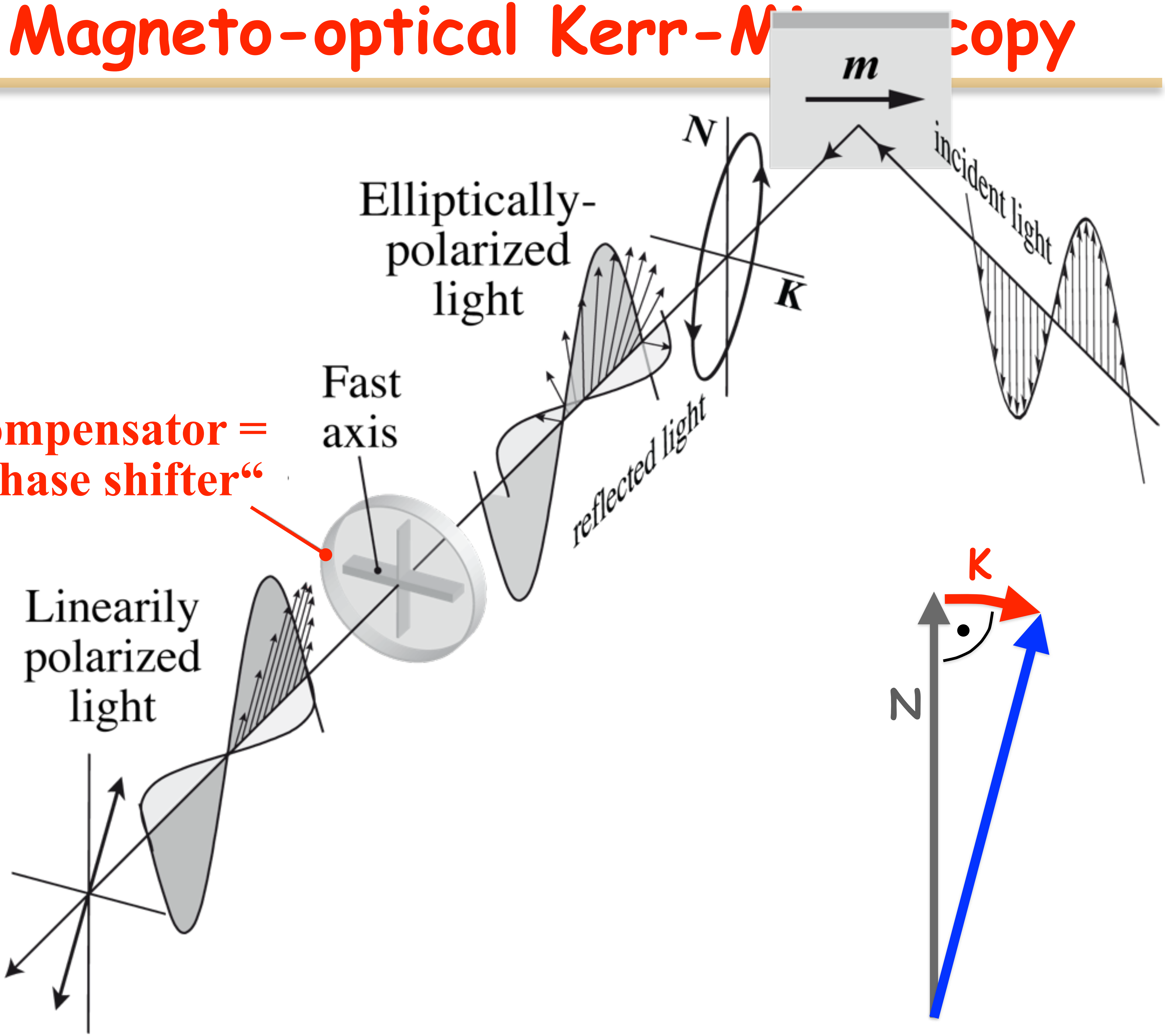
Kerr effect





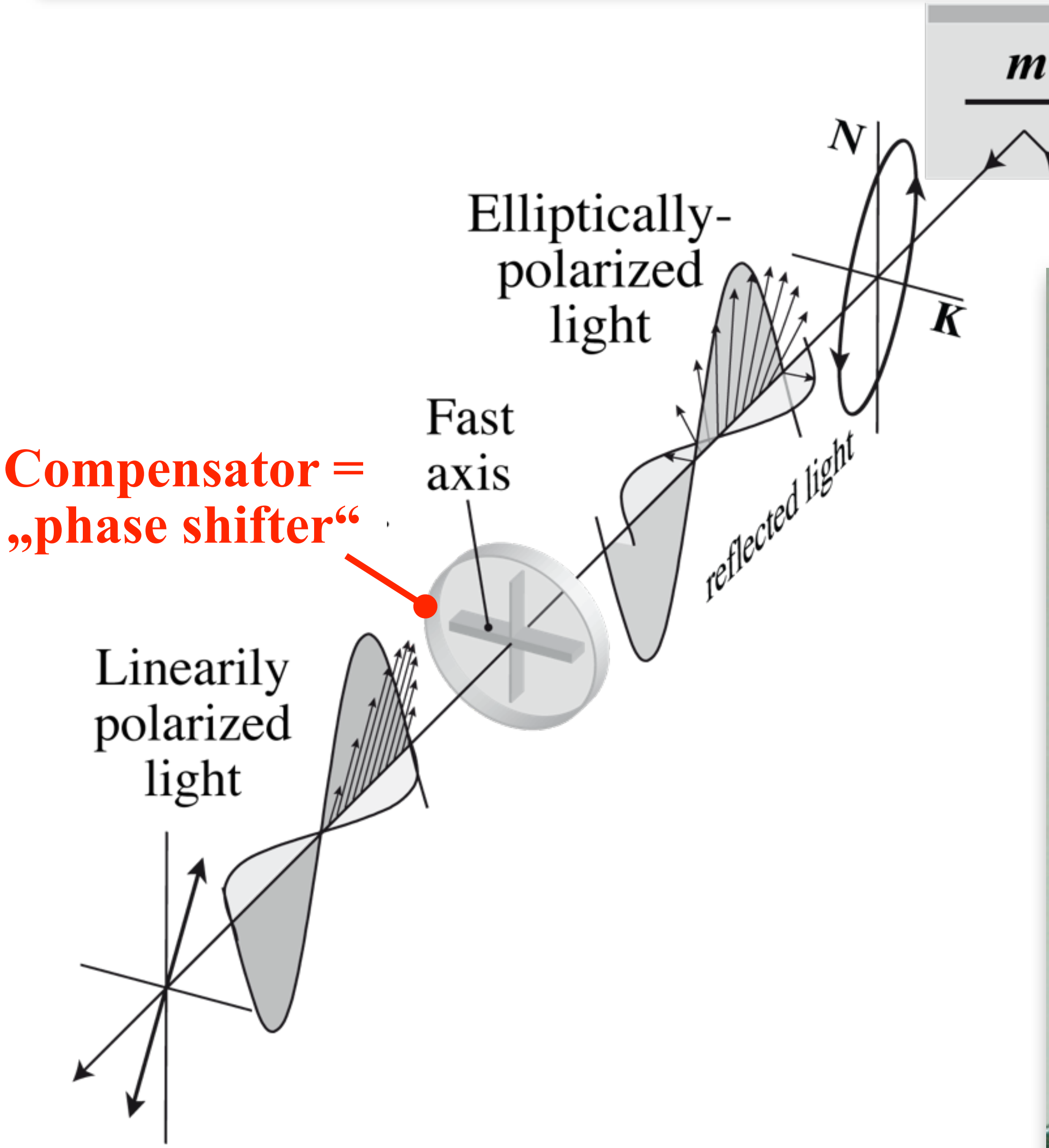
Kerr rotation
= K/N

4. Magneto-optical Kerr-M copy

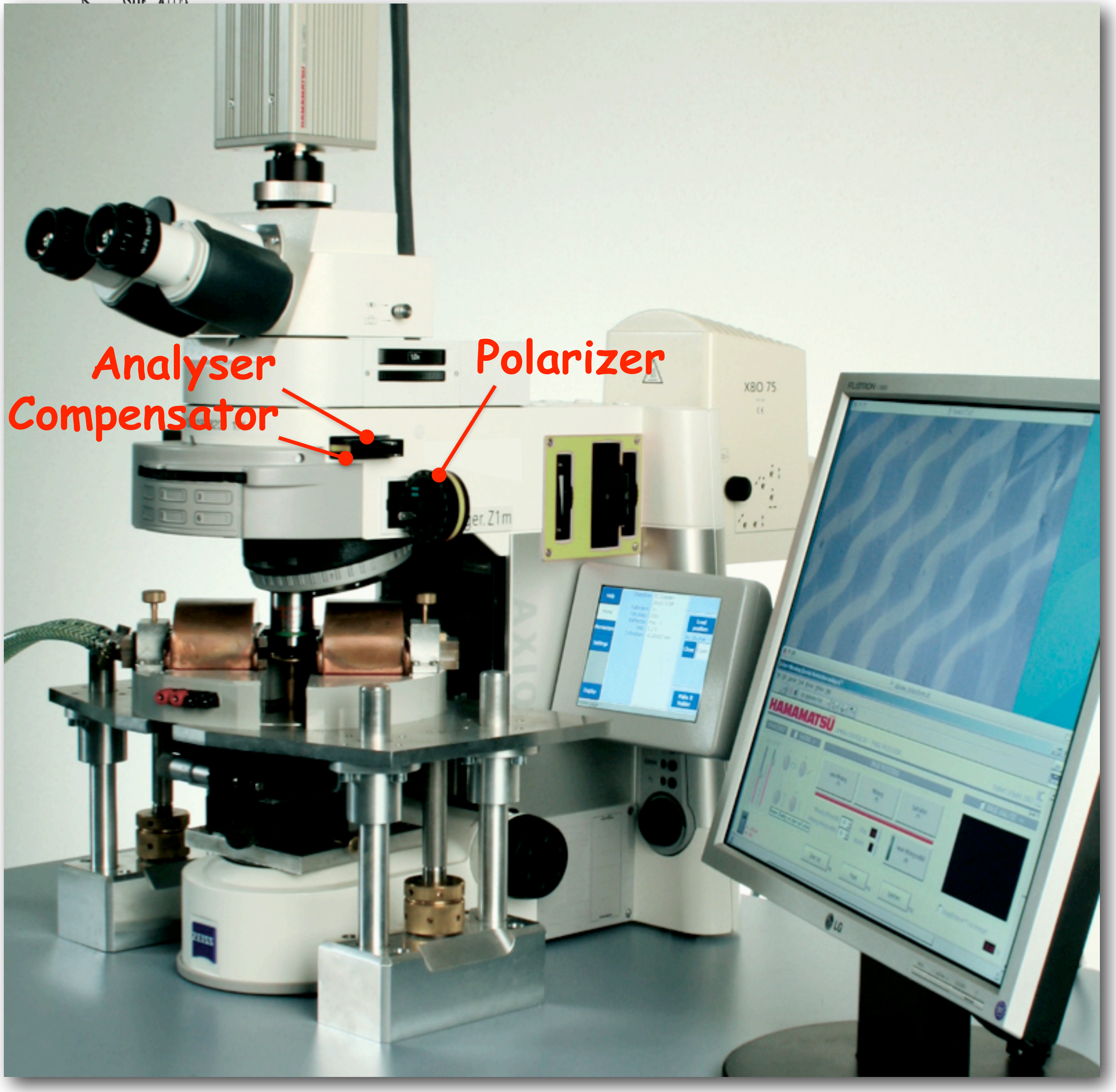


Compensator = „phase shifter“

4. Magneto-optical Kerr-Microscopy



Compensator = „phase shifter“



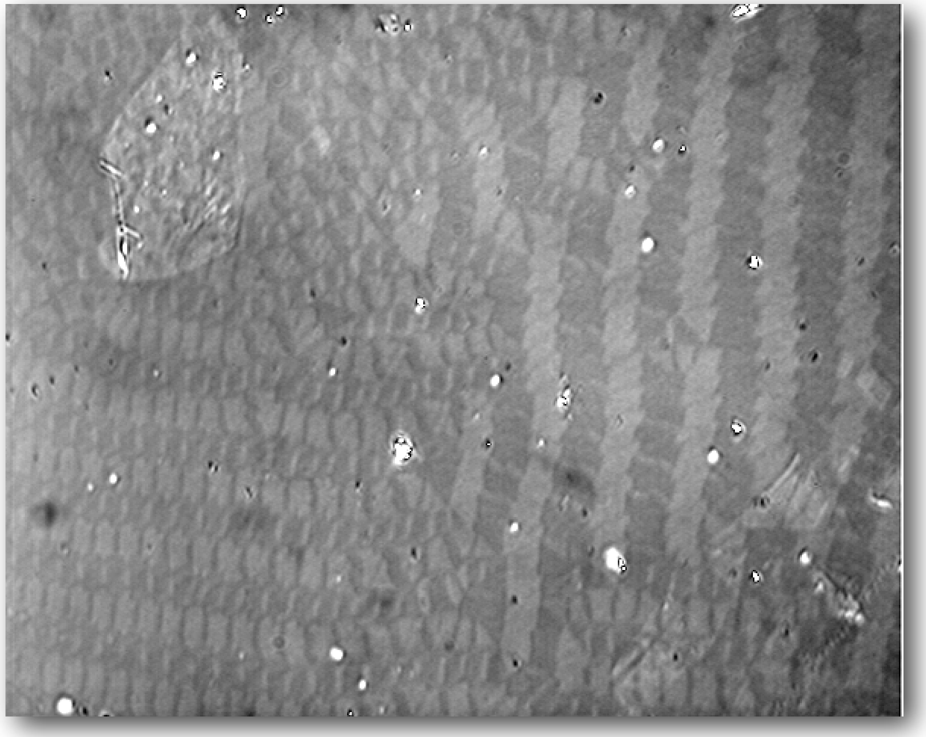
Analyser

Compensator

Polarizer

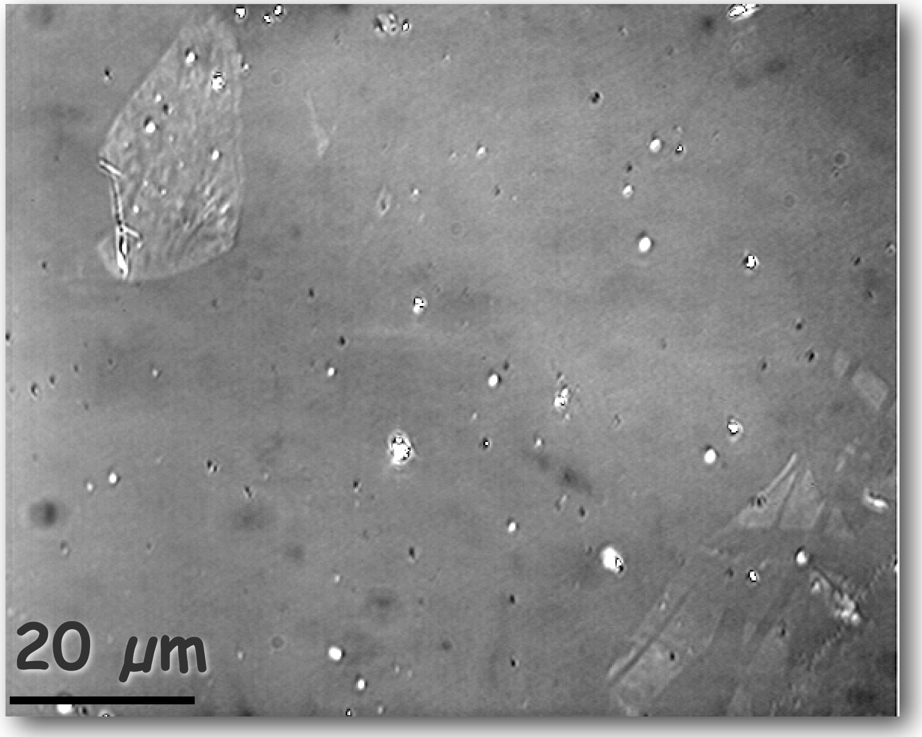
4. Magneto-optical Kerr-Microscopy

Difference image technique



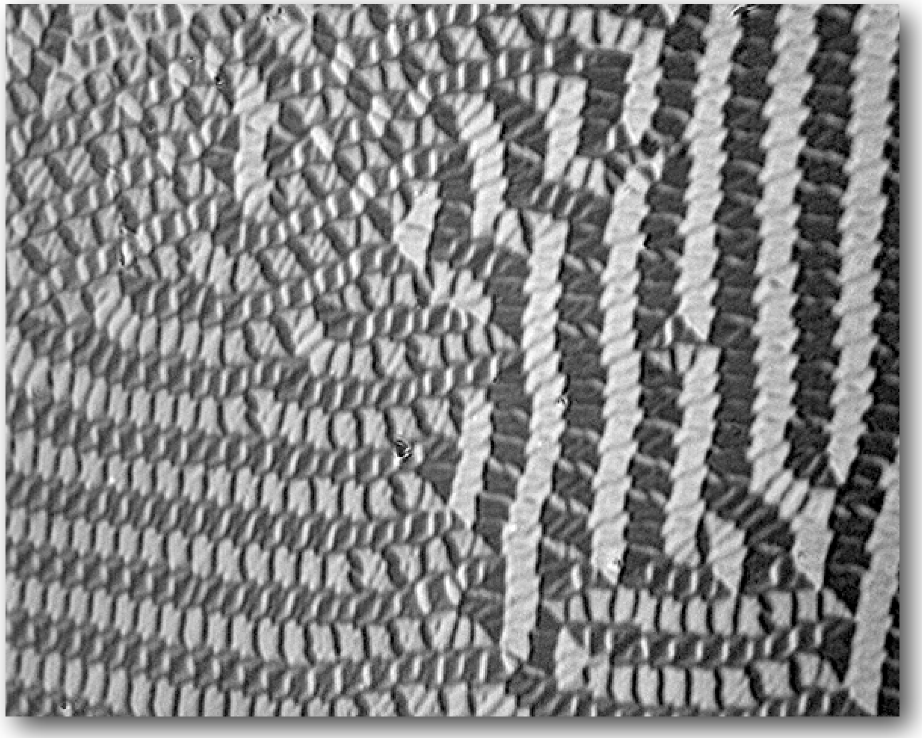
Original image

−



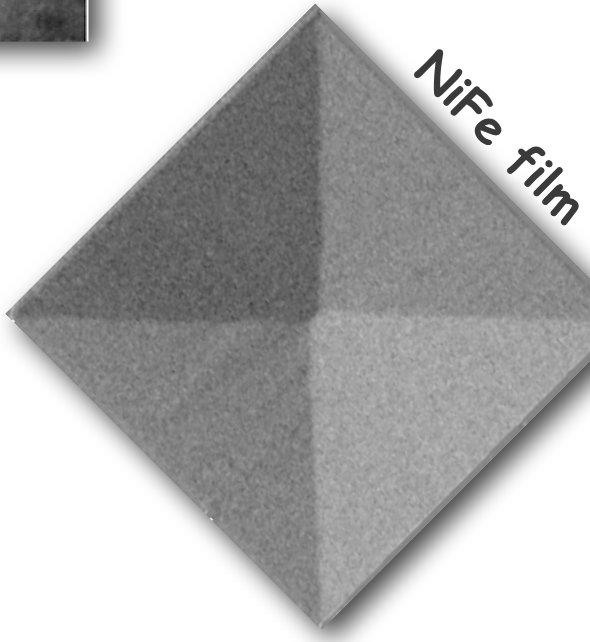
Reference image

↓



amorphous
ribbon

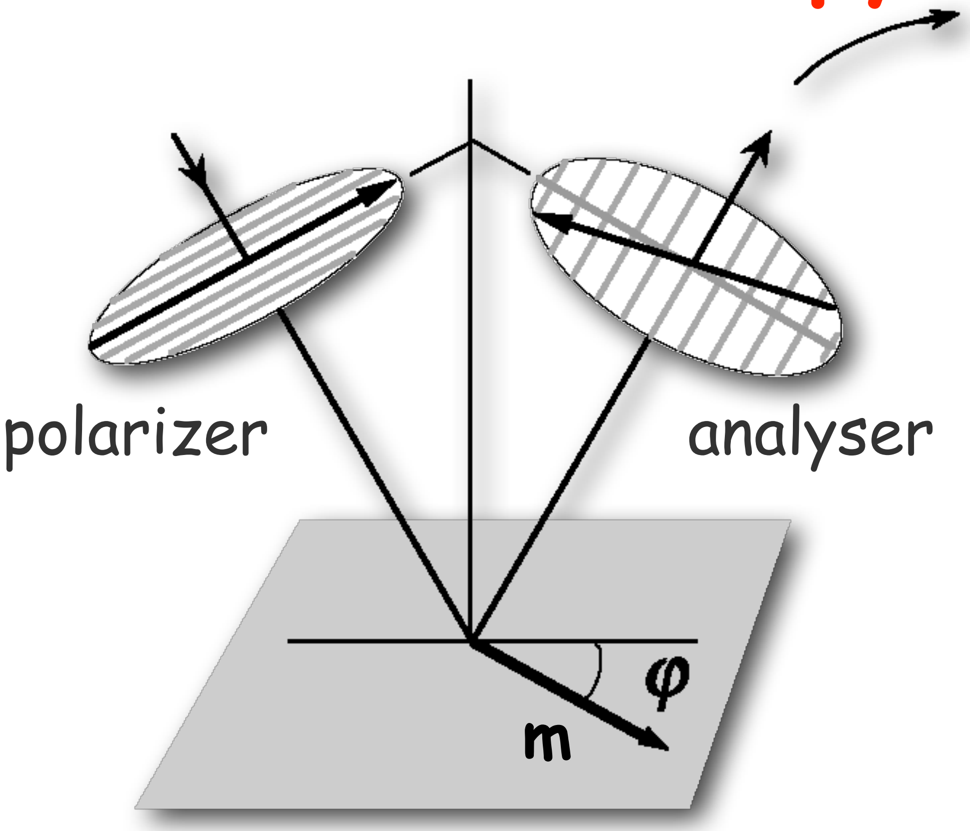
Difference image

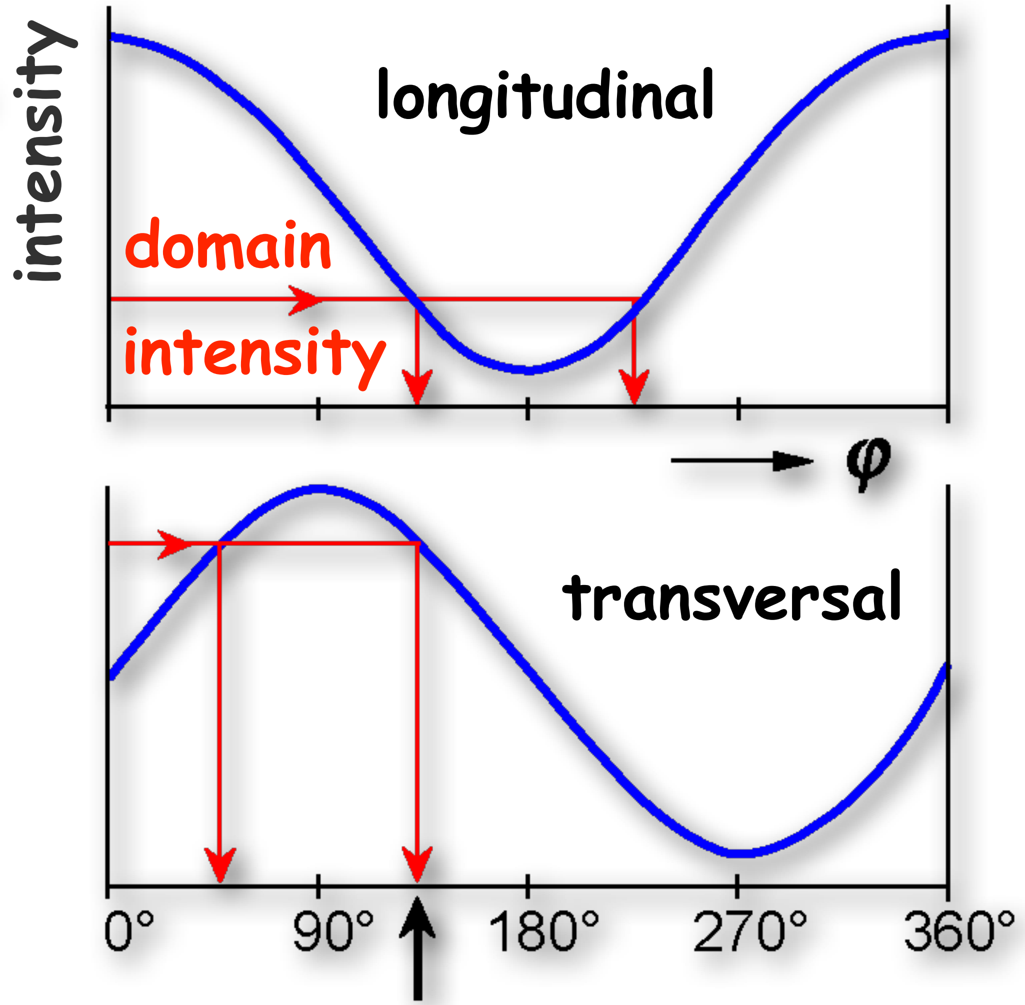


Important:
Difference Imaging
in real time !

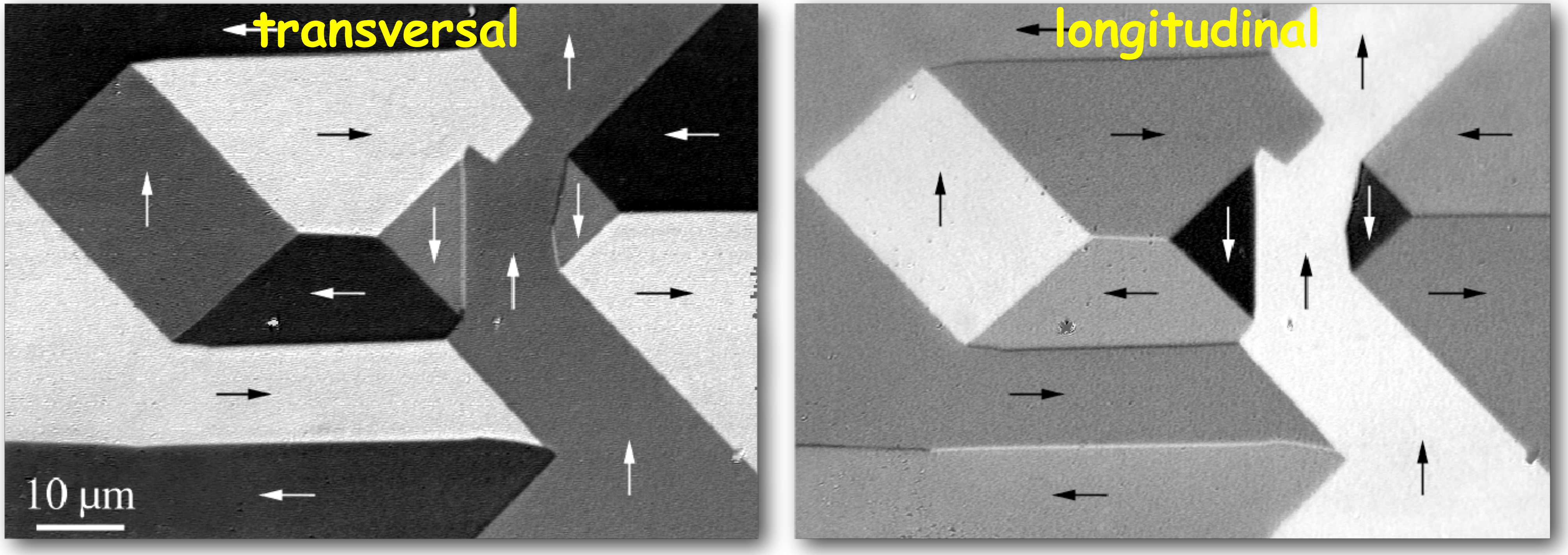
4. Magneto-optical Kerr-Microscopy

Quantitative Kerr microscopy





Domains on (100)-FeSi sheet

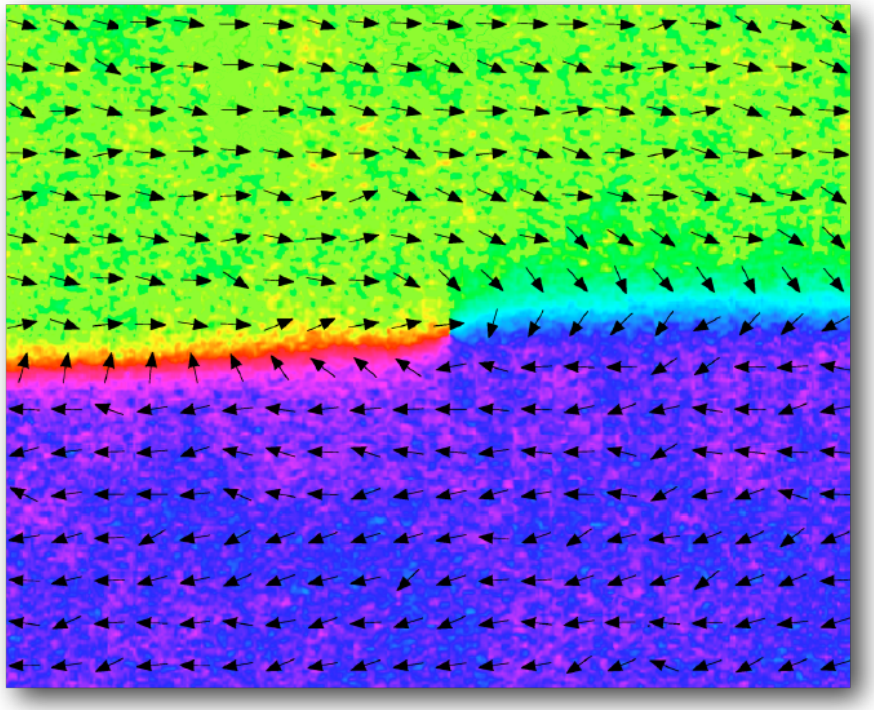


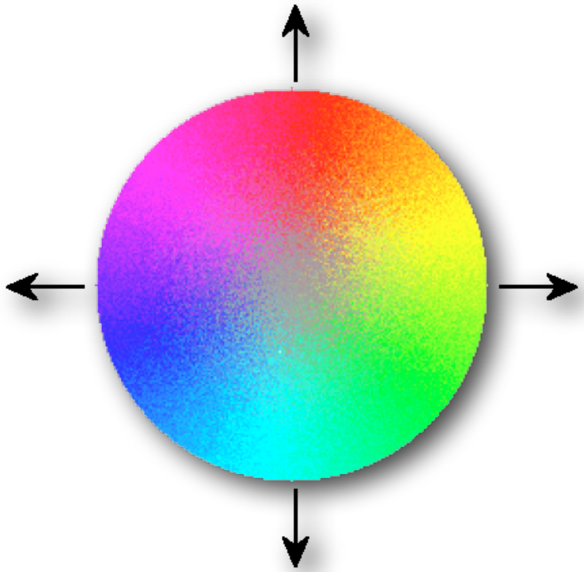
4. Magneto-optical Kerr-Microscopy

Quantitative Kerr microscopy

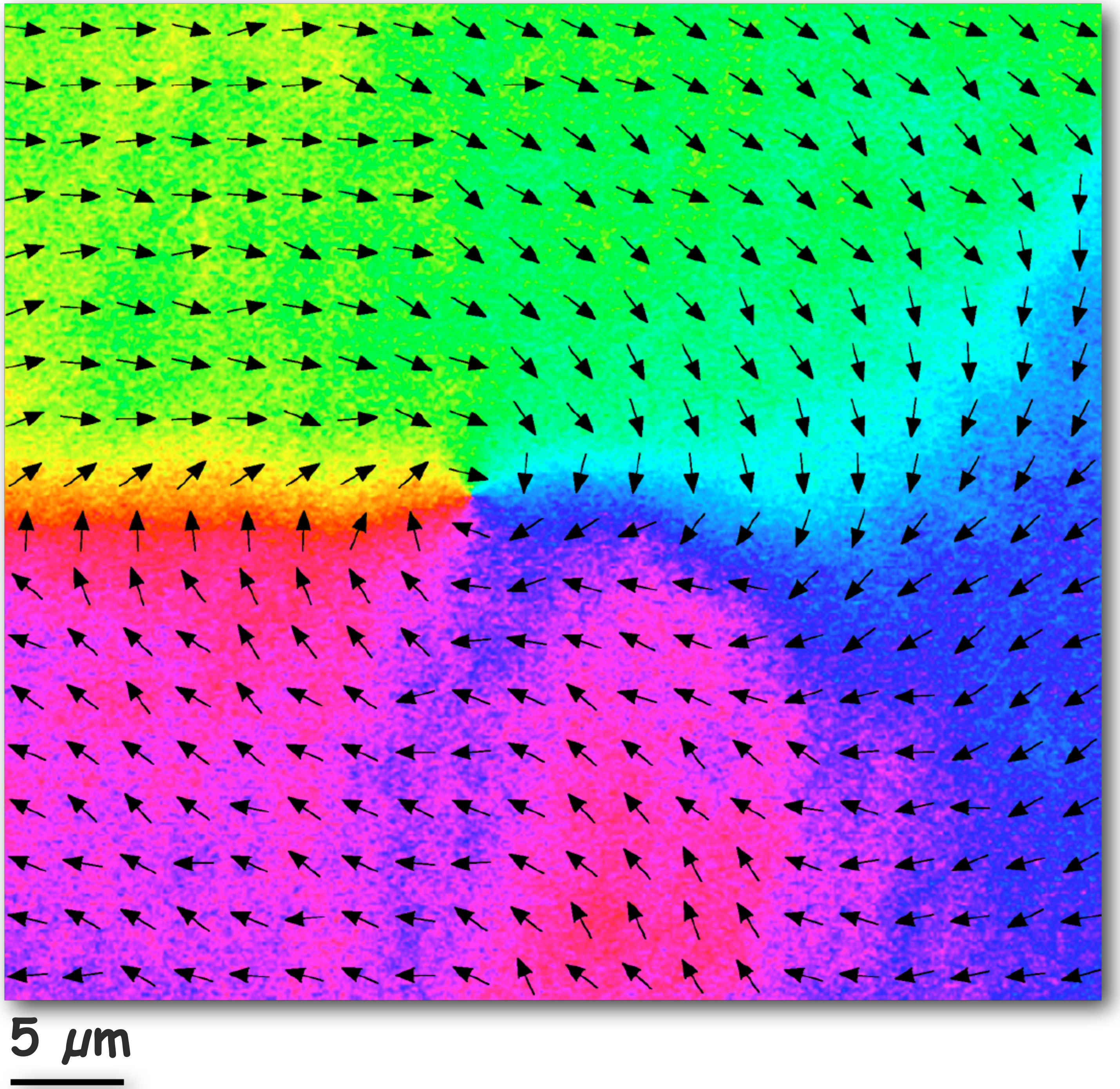
Domains in magnetostriction-free amorphous ribbon

as-quenched state





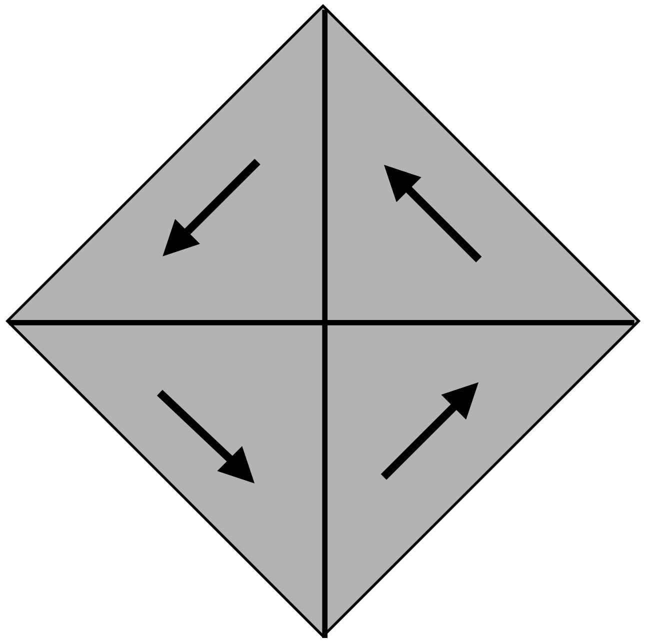
after annealing in rotating field



4. Magneto-optical Kerr-Microscopy

MOKE-Magnetometry and domains

Permalloy (NiFe) film
207 nm thick

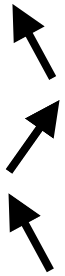


Magnetic field H

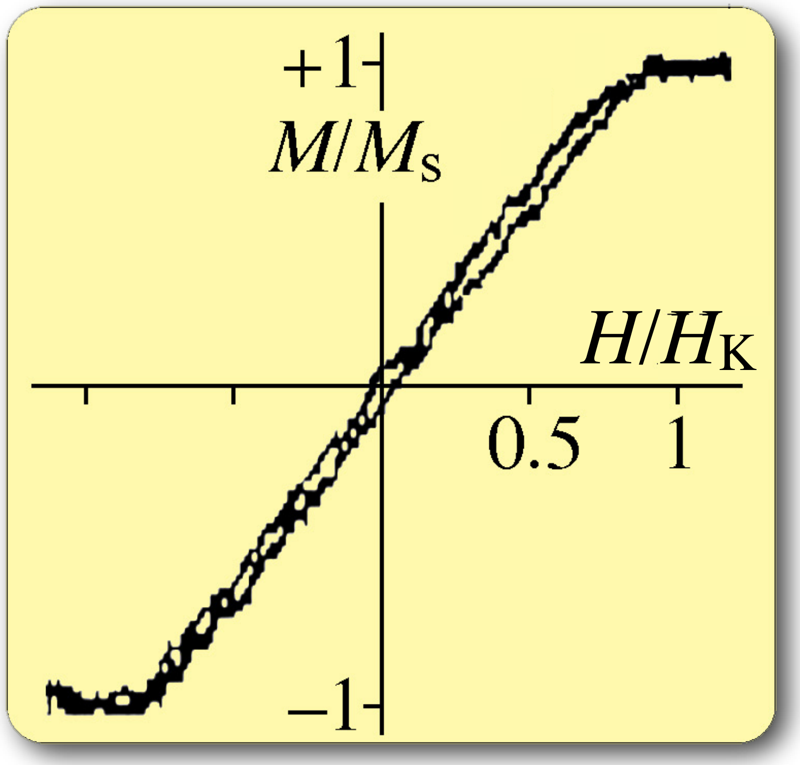
↕

57 µm

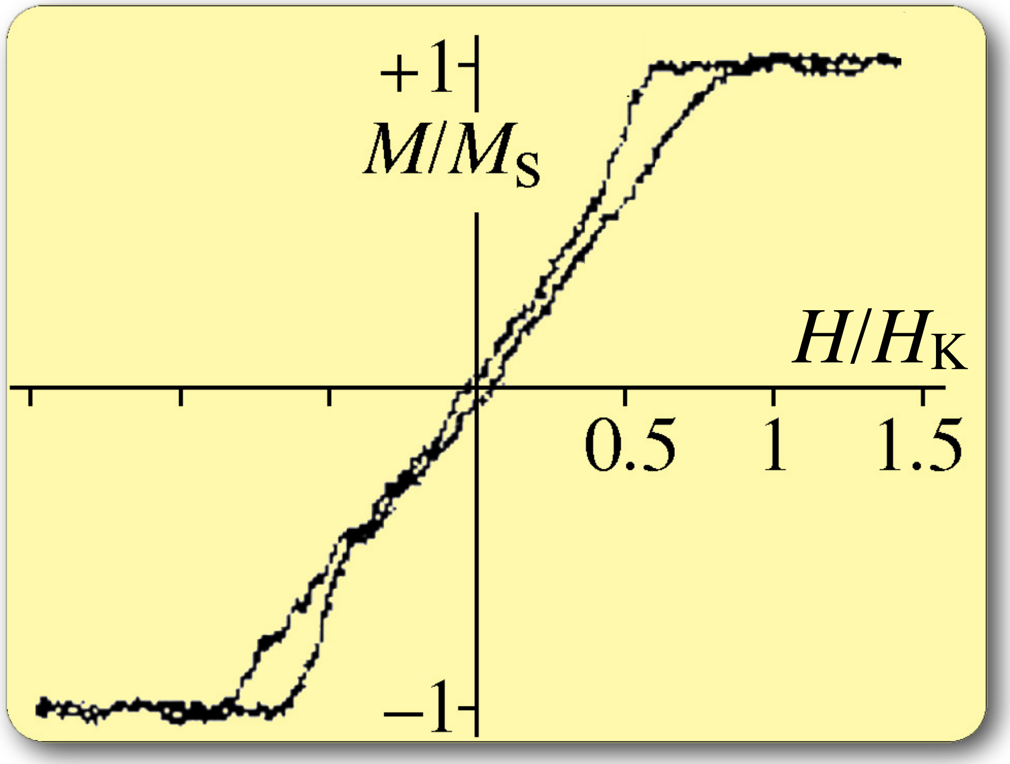
Concertina



$H_{\mathrm{max}} = H_{\mathrm{K}}$



$H_{\mathrm{max}} > H_{\mathrm{K}}$

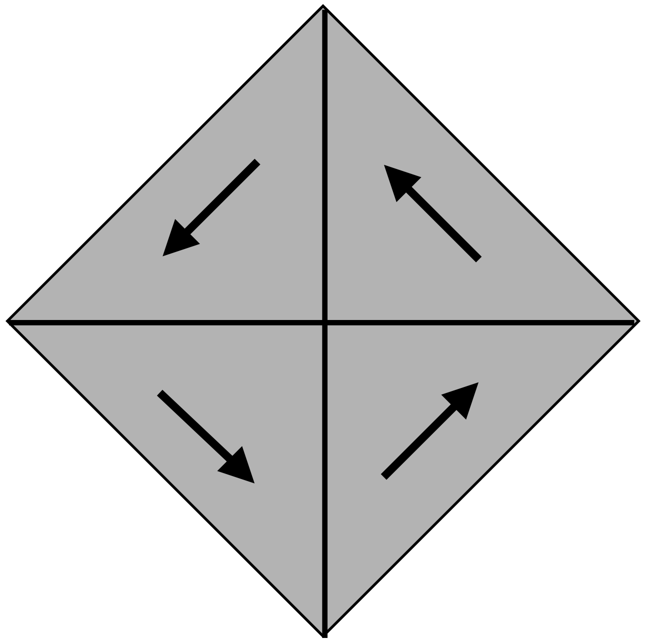


MOKE
M(H) loops

4. Magneto-optical Kerr-Microscopy

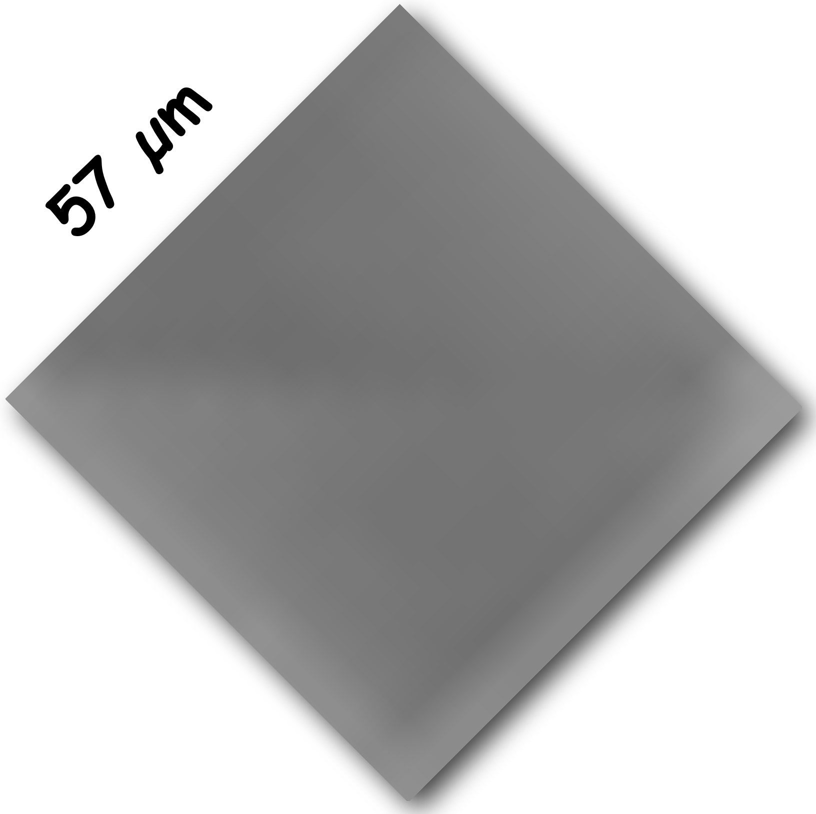
MOKE-Magnetometry and domains

Permalloy (NiFe) film
207 nm thick



Magnetic field H

MOKE
M(H) loops



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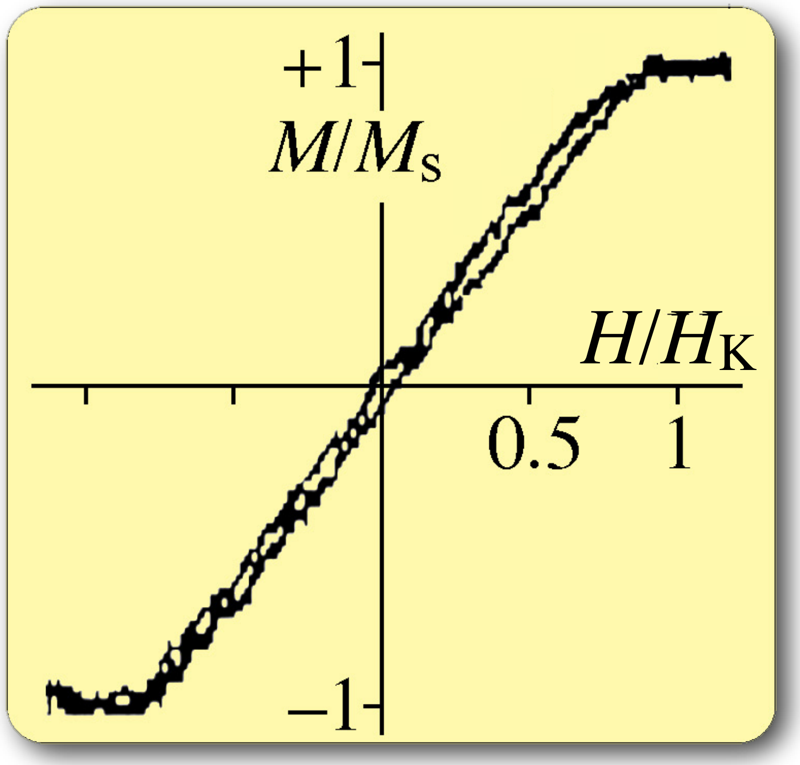
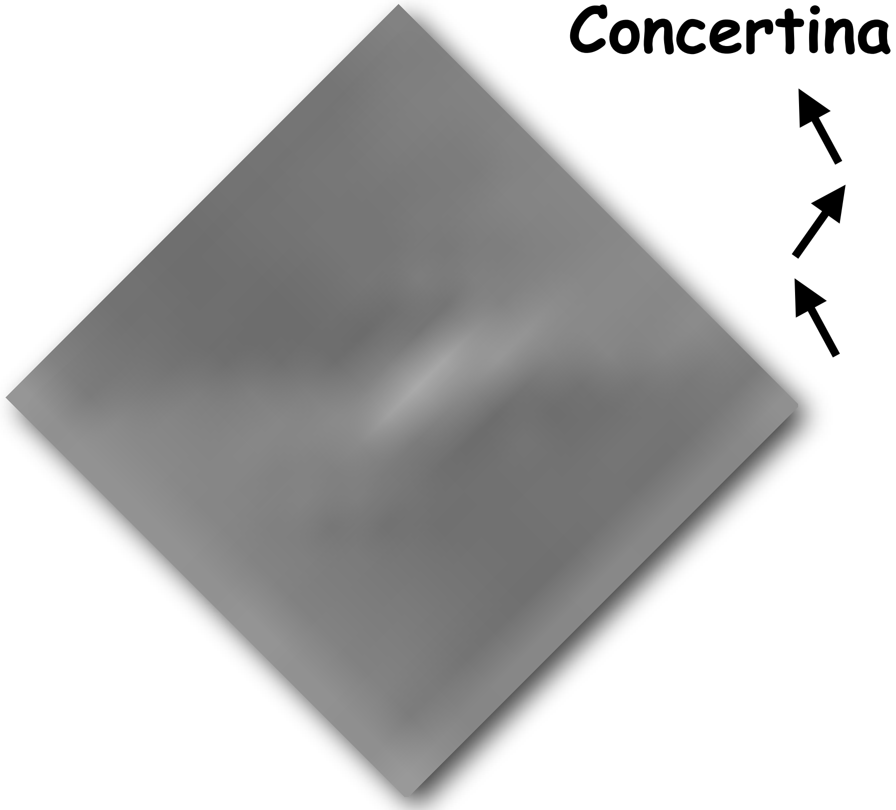
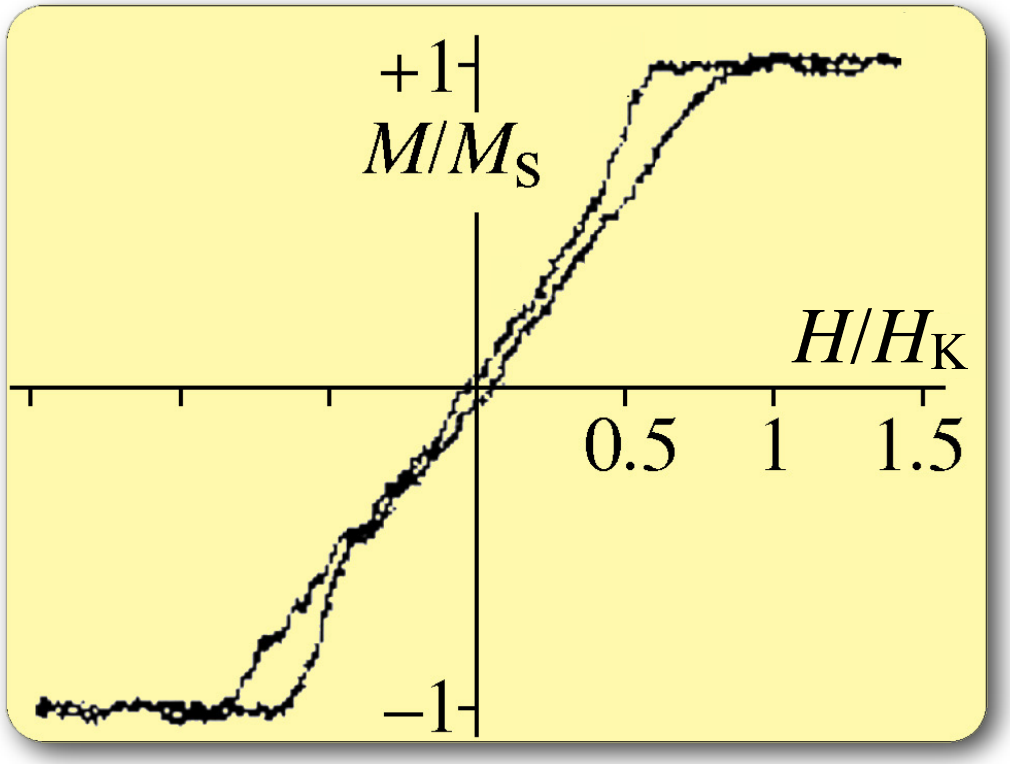




$H_{\mathrm{max}} > H_{\mathrm{K}}$



4. Magneto-optical Kerr-Microscopy

MOKE-Magnetometry and domains

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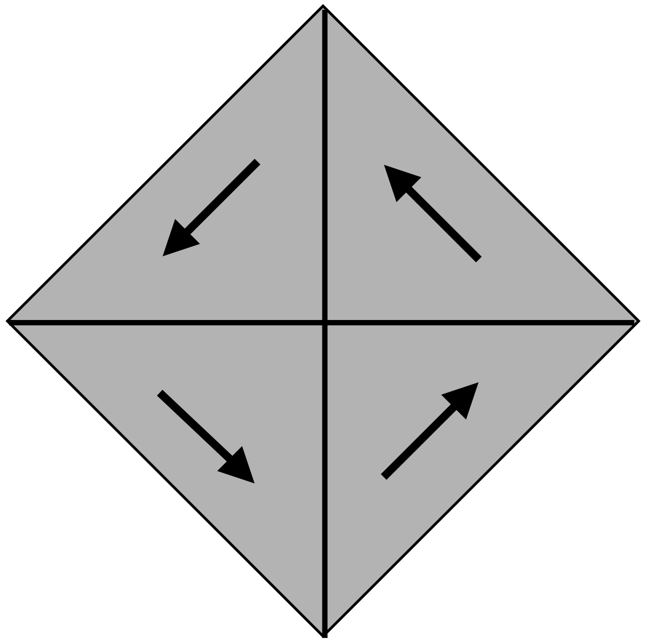
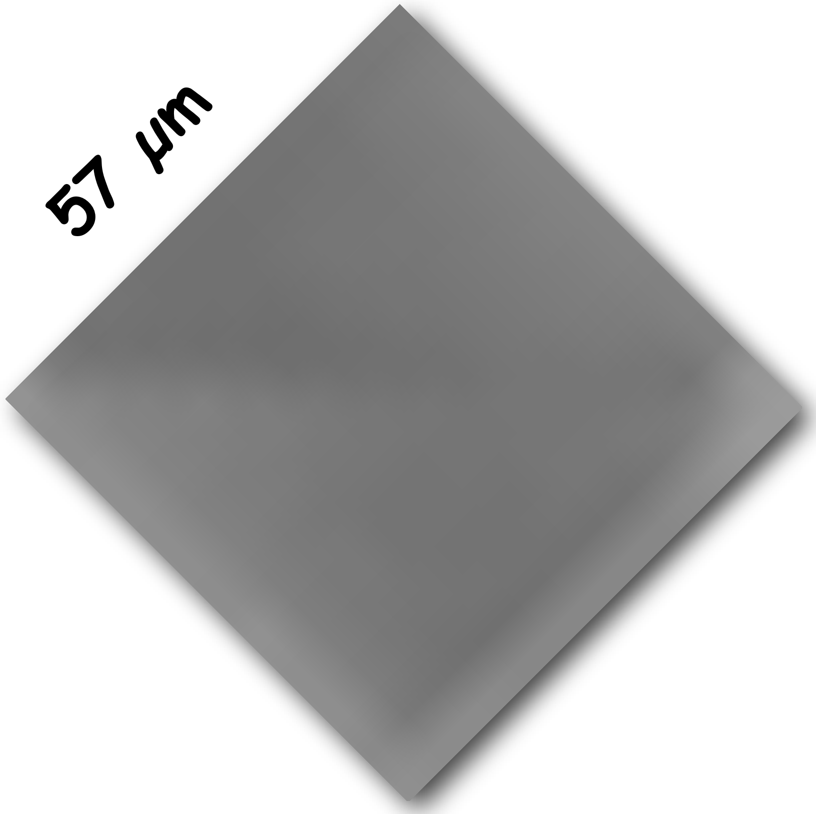
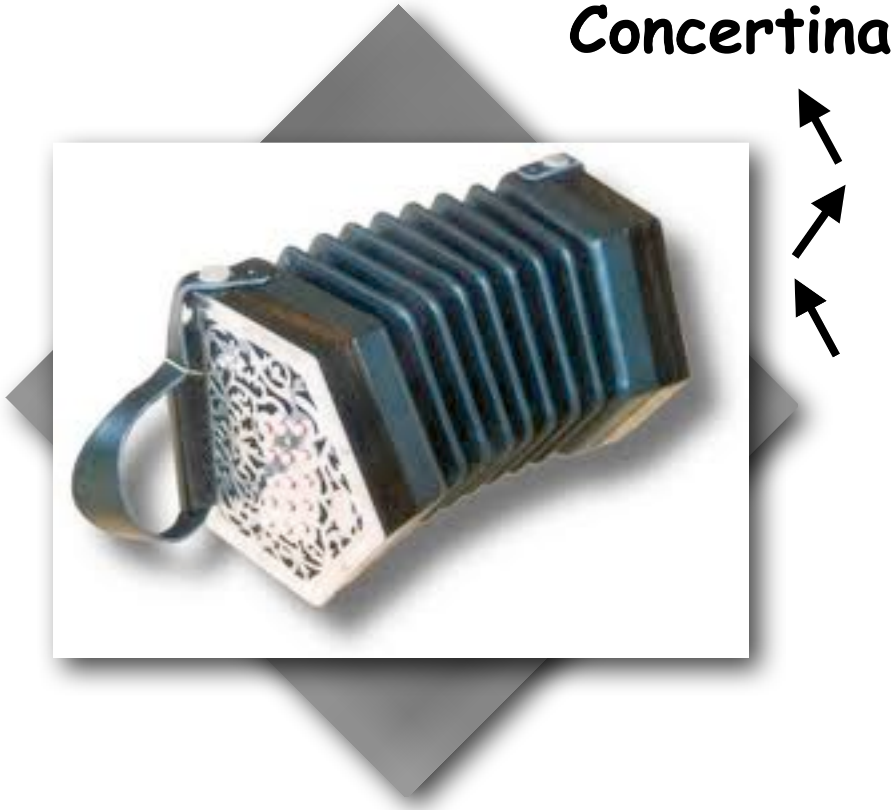


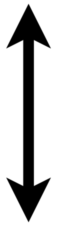


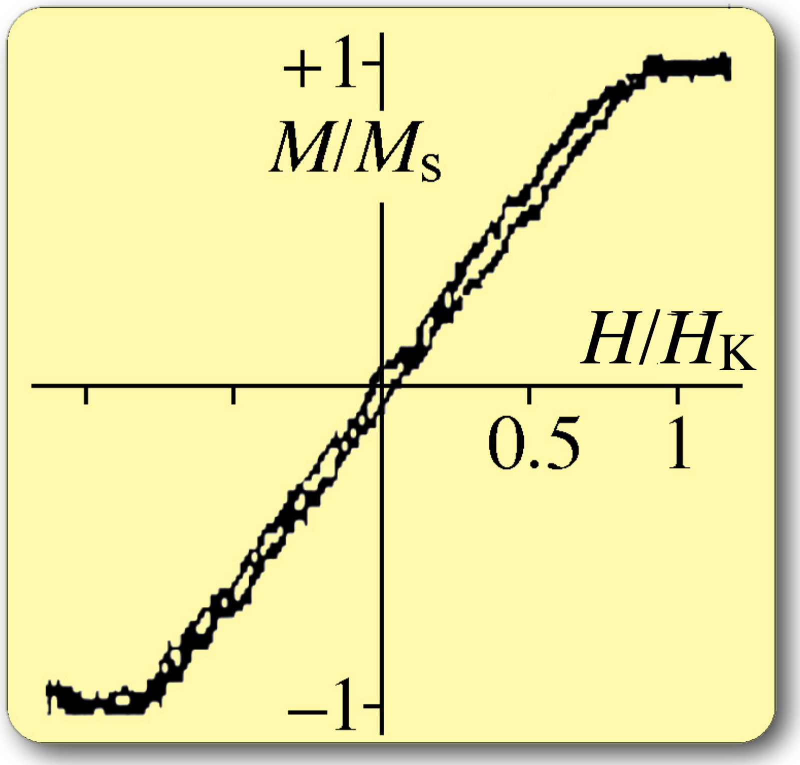


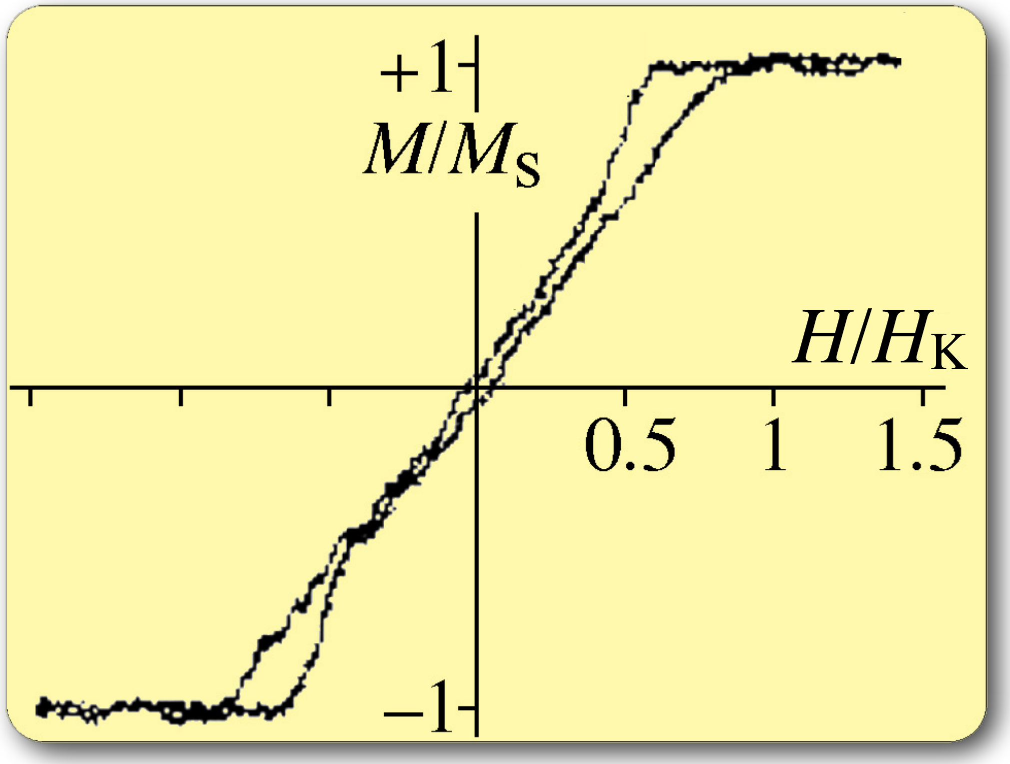
Magnetic field H

$H_{\max} = H_K$

$H_{\max} > H_K$







MOKE
M(H) loops

4. Magneto-optical Kerr-Microscopy

MOKE-Magnetometry and domains

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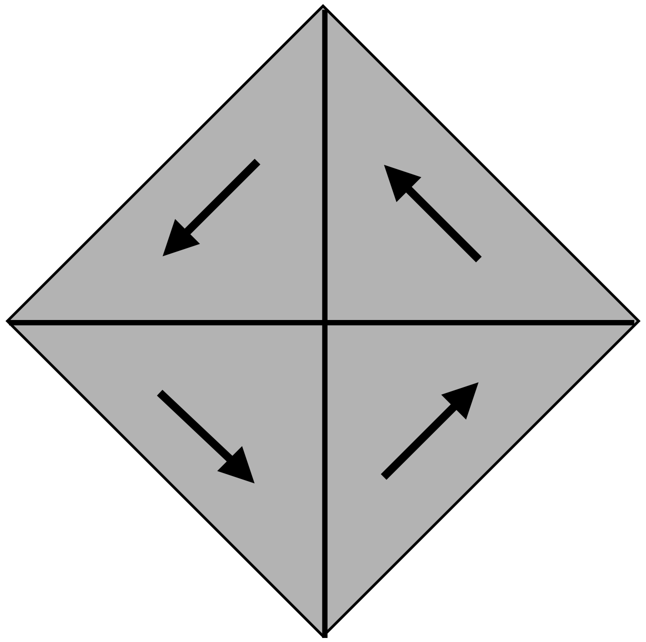
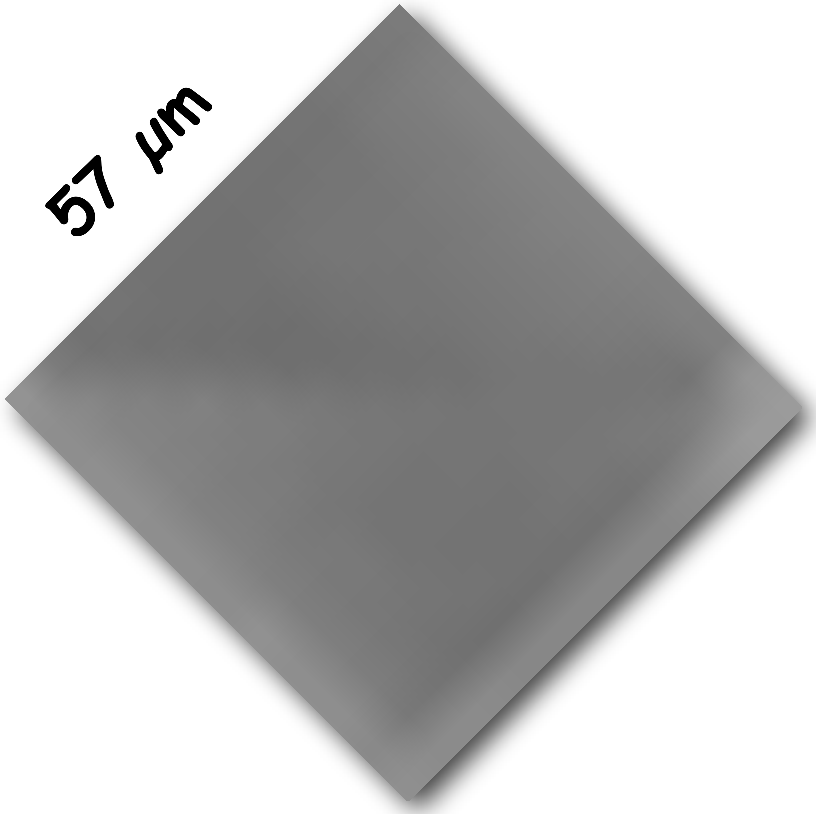
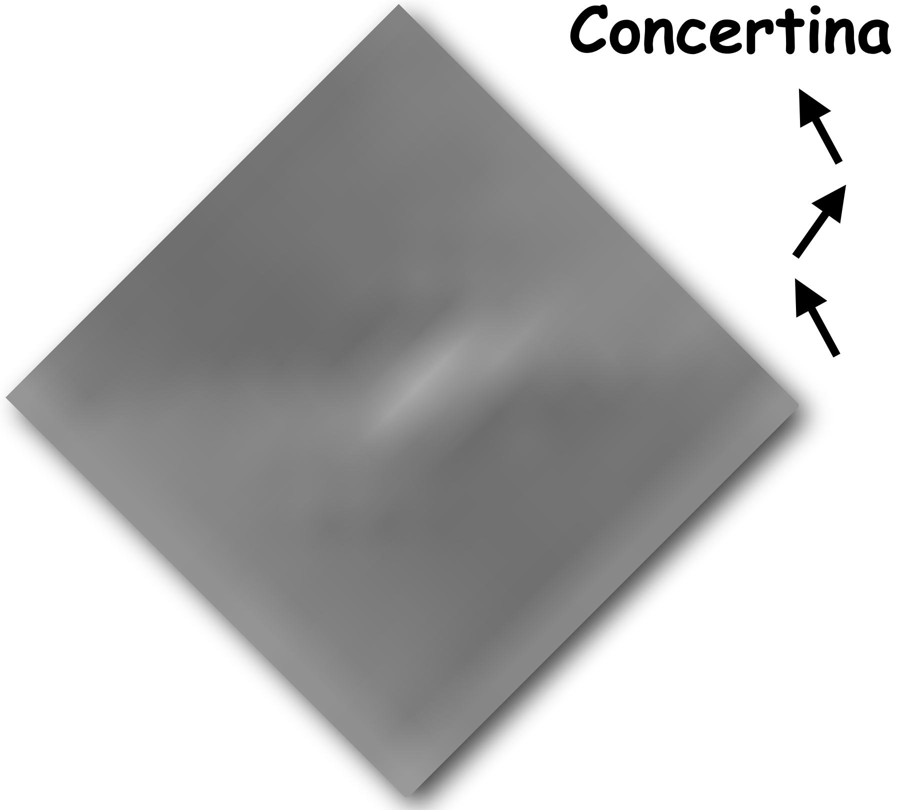


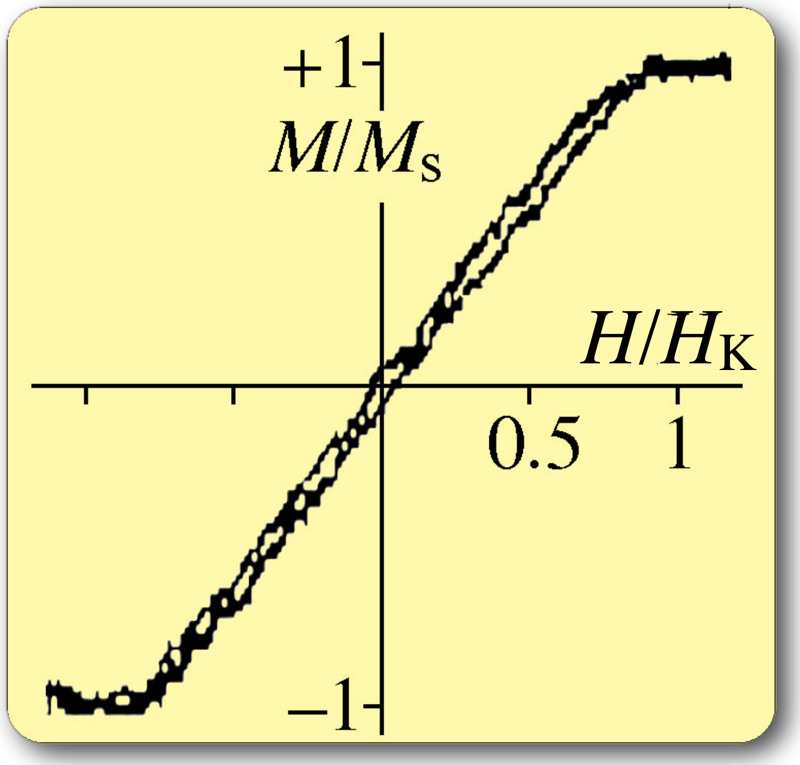
Magnetic field H

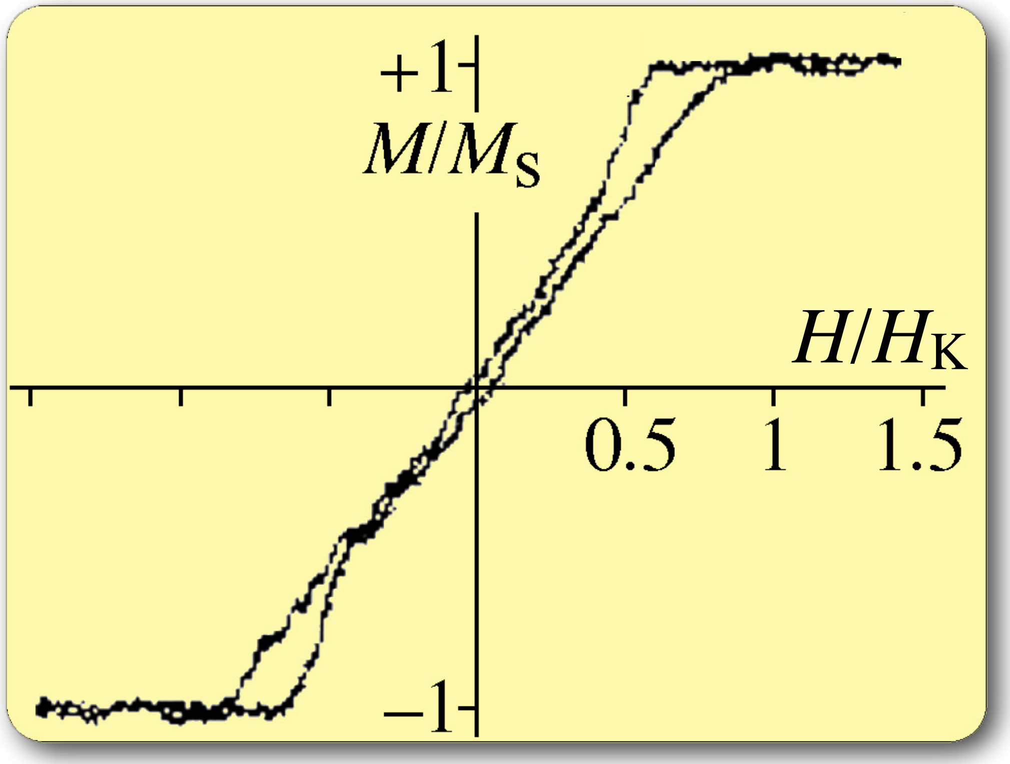


$H_{\max} = H_K$



$H_{\max} > H_K$





MOKE
M(H) loops

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MOKE-Magnetometry and domains

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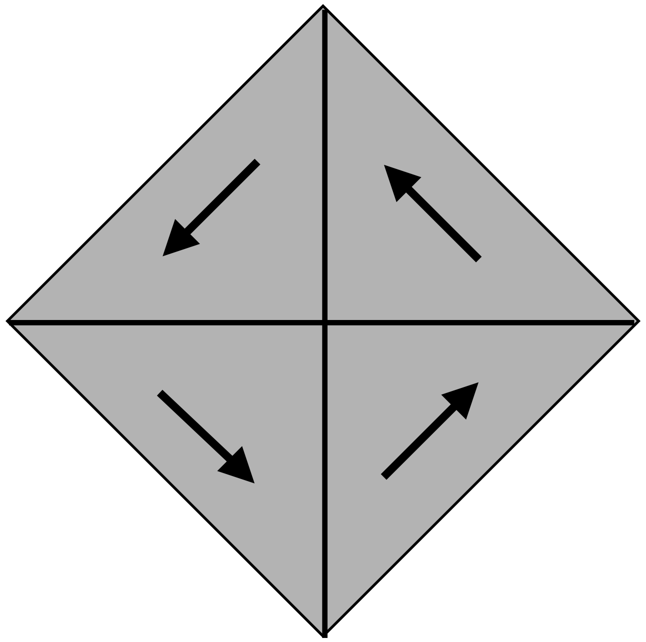
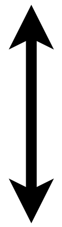
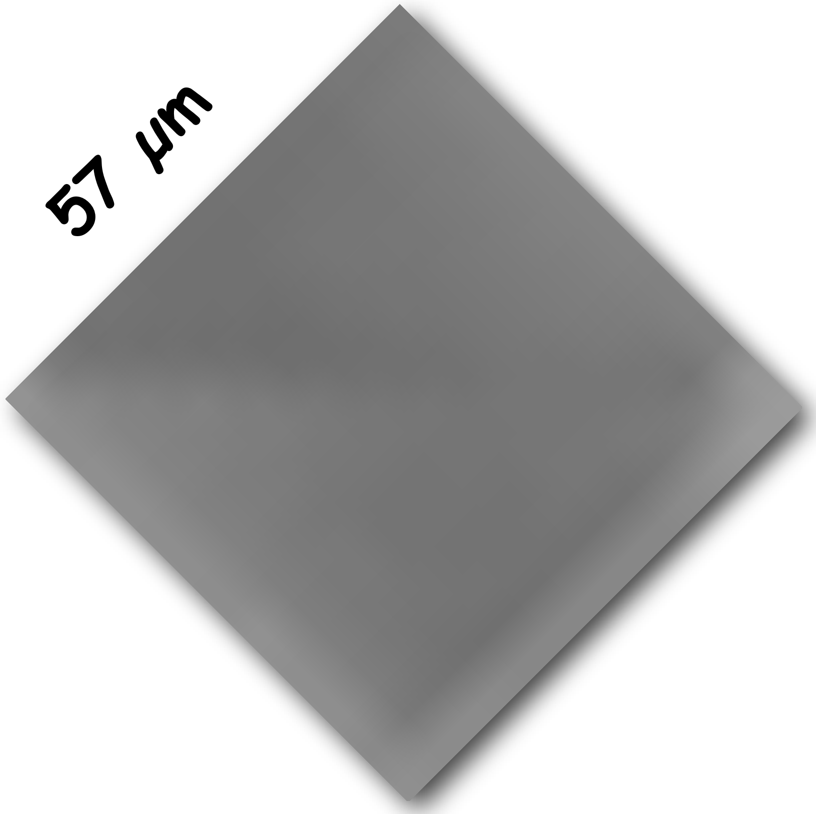
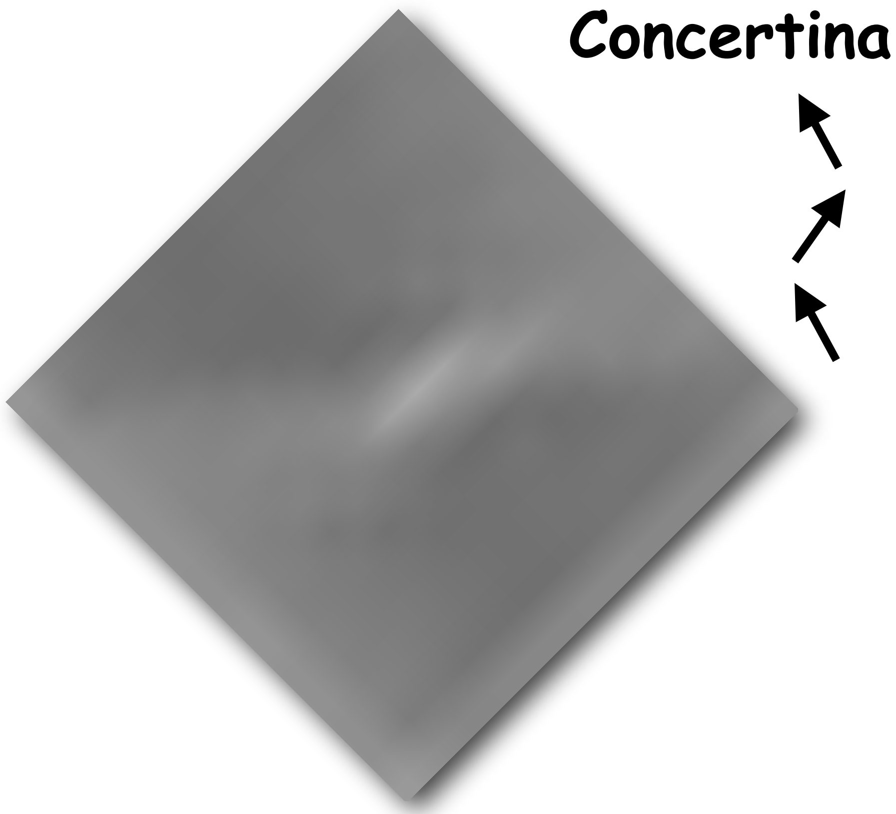


Magnetic field H



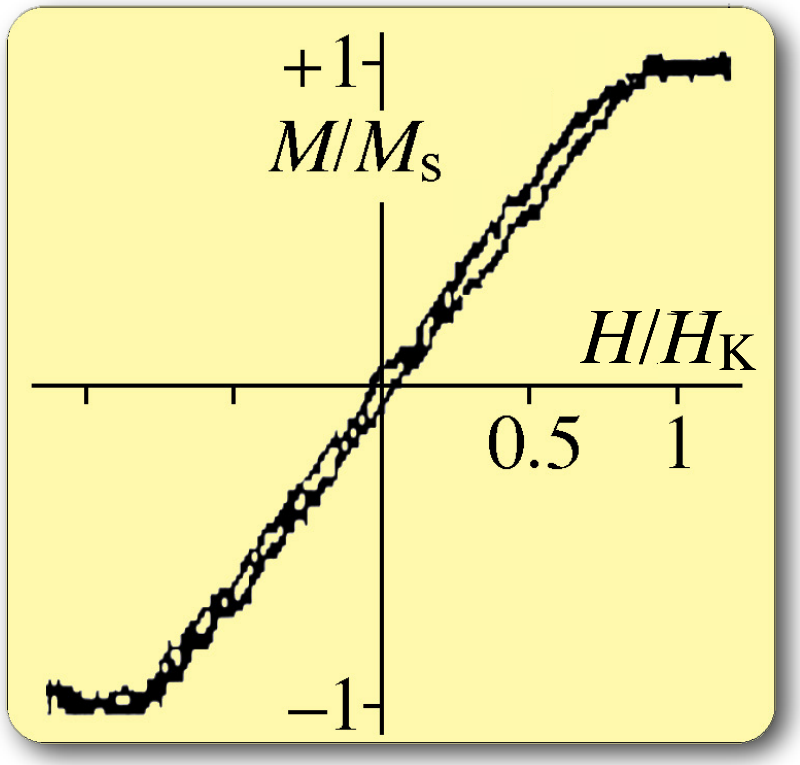


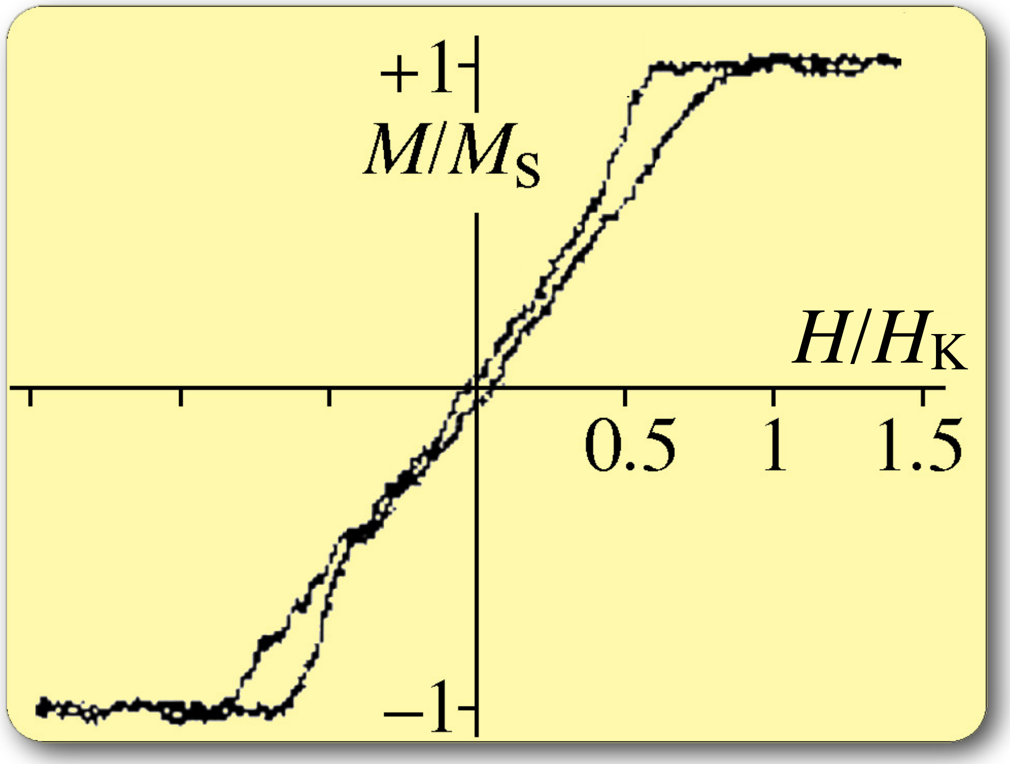
$H_{\mathrm{max}} = H_{\mathrm{K}}$



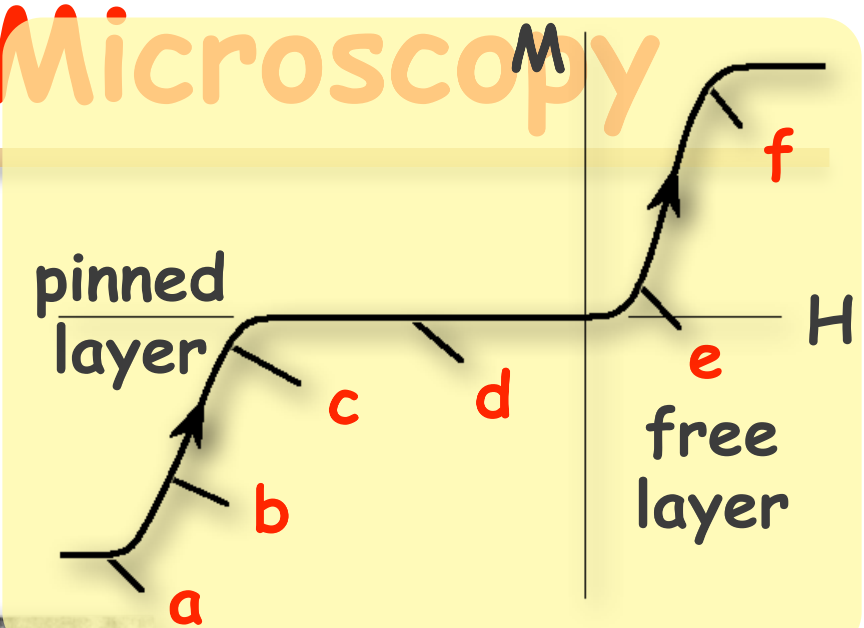
$H_{\mathrm{max}} > H_{\mathrm{K}}$

MOKE
M(H) loops



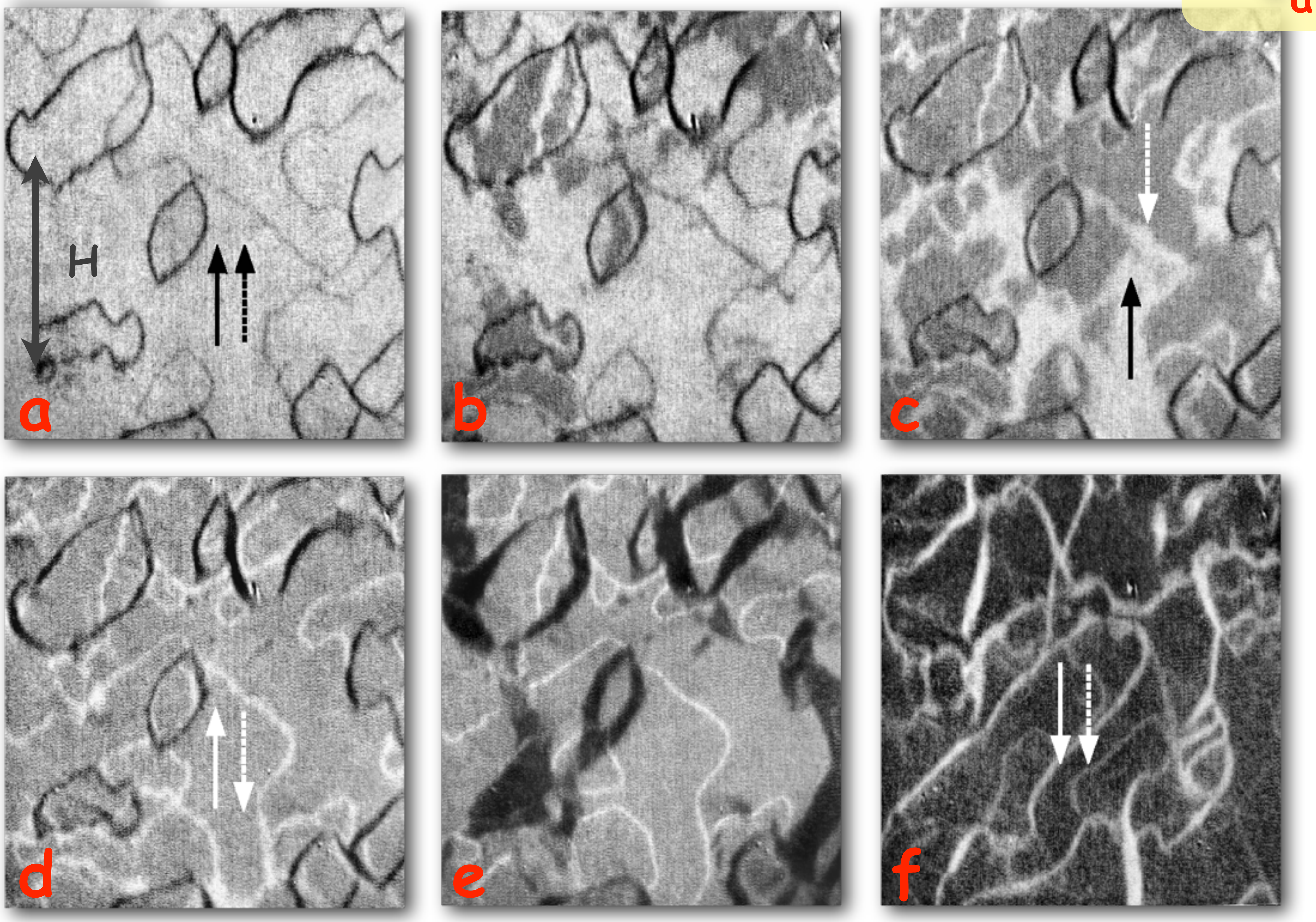


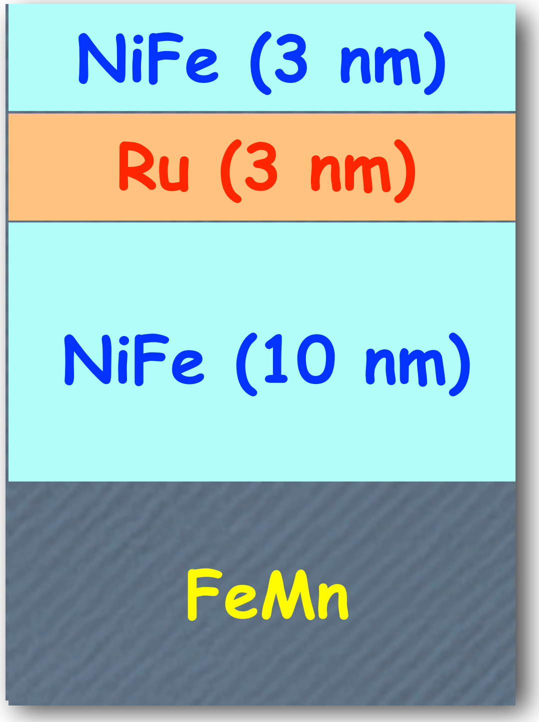
4. Magneto-optical Kerr-Microscopy



Depth sensitivity of Kerr microscopy

Information depth in metals: ~20 nm



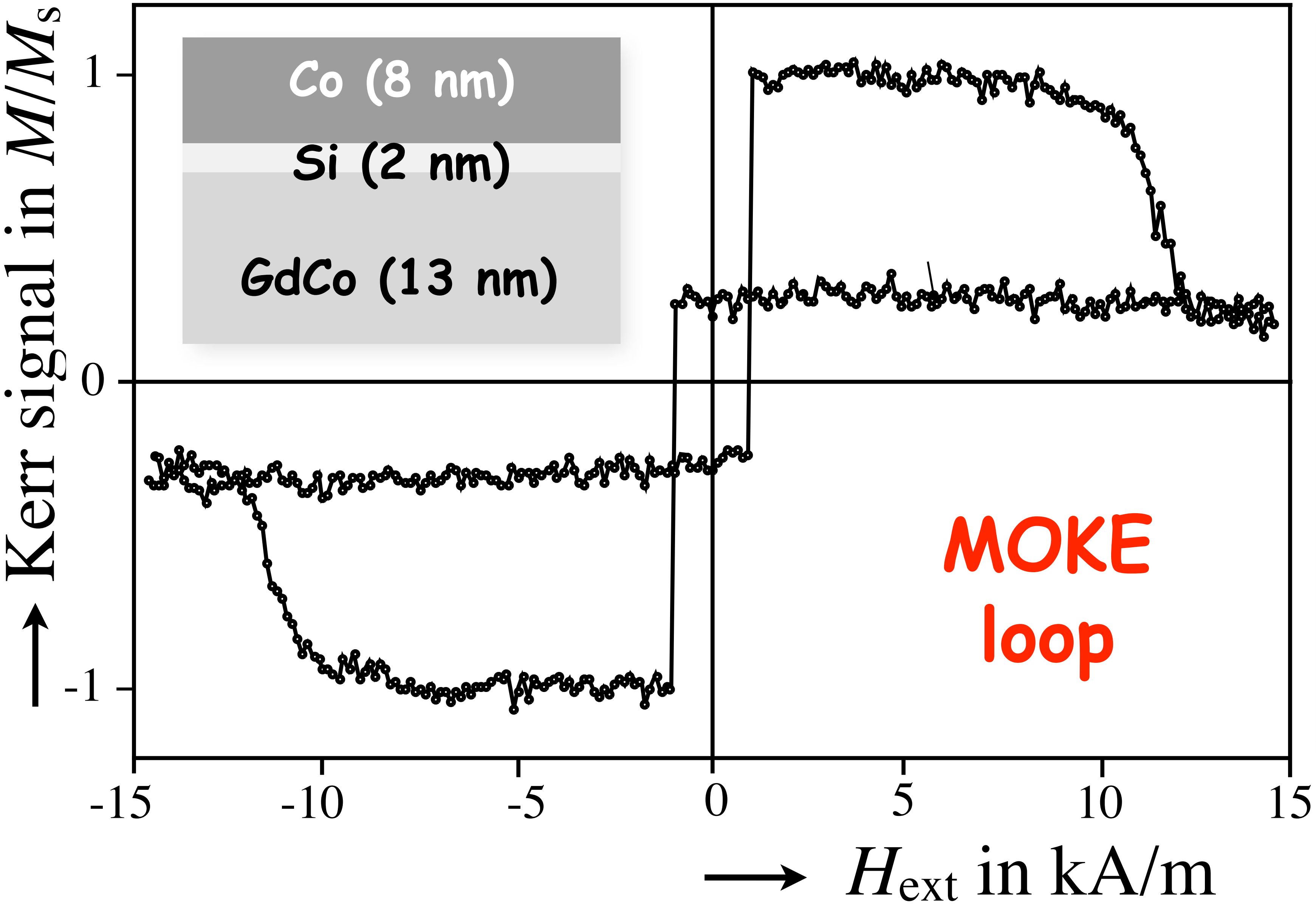


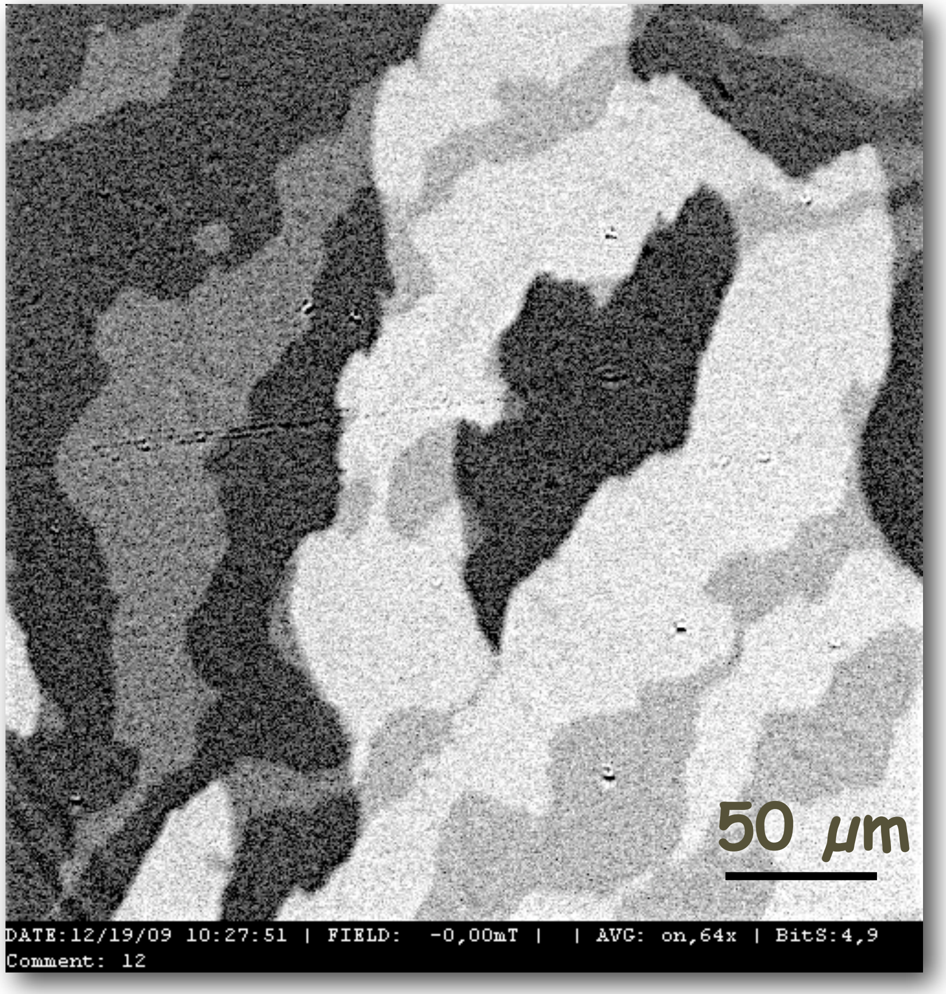
Sample:
S. Parkin, IBM

4. Magneto-optical Kerr-Microscopy

Depth sensitivity of Kerr microscopy

Co/Si/GdCo trilayer



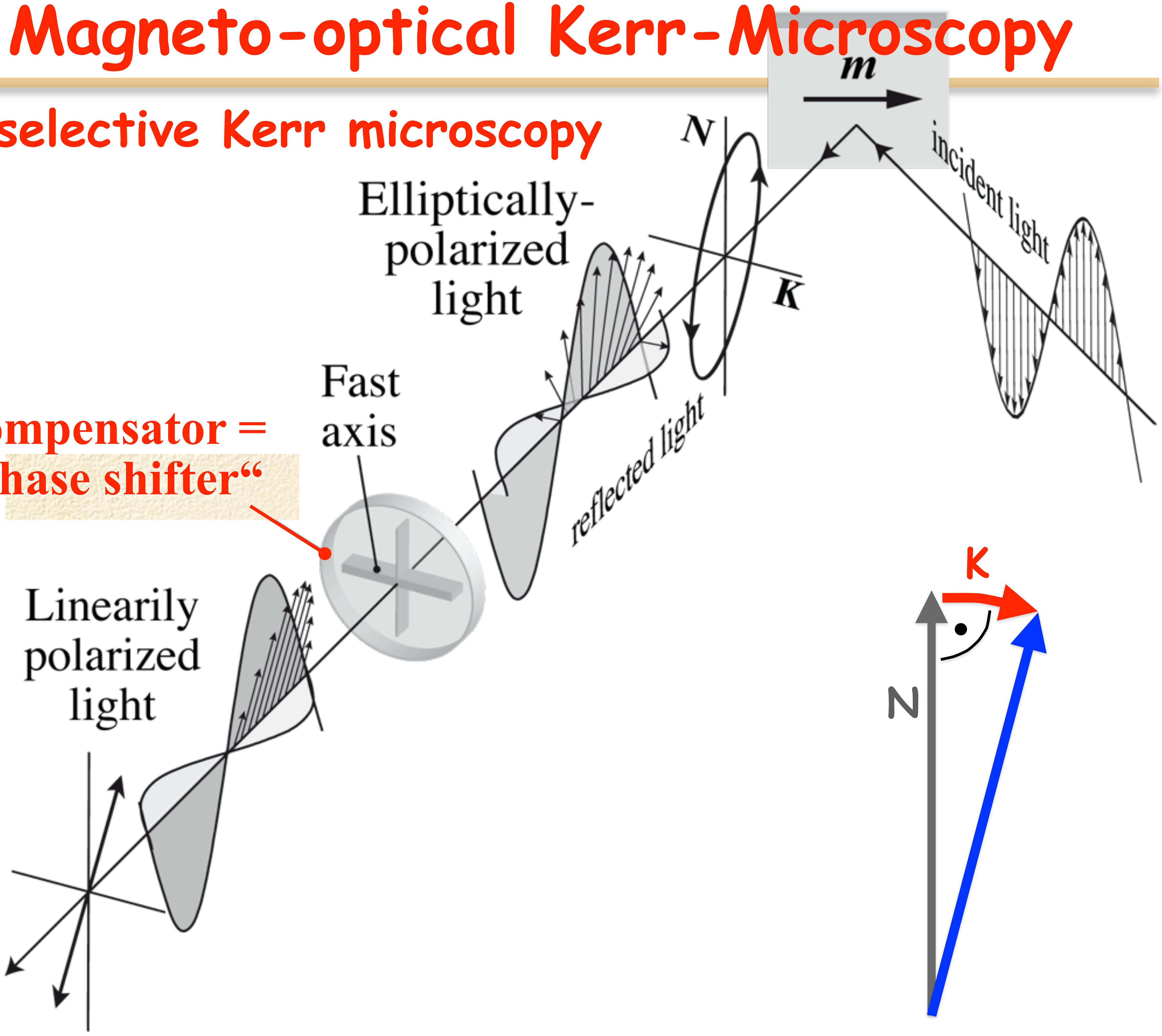


Sample: A. Svalov and G. Kurlyandskaya, Ekaterinburg

Imaging: together with L. Lokamani, Dresden (unpublished)

4. Magneto-optical Kerr-Microscopy

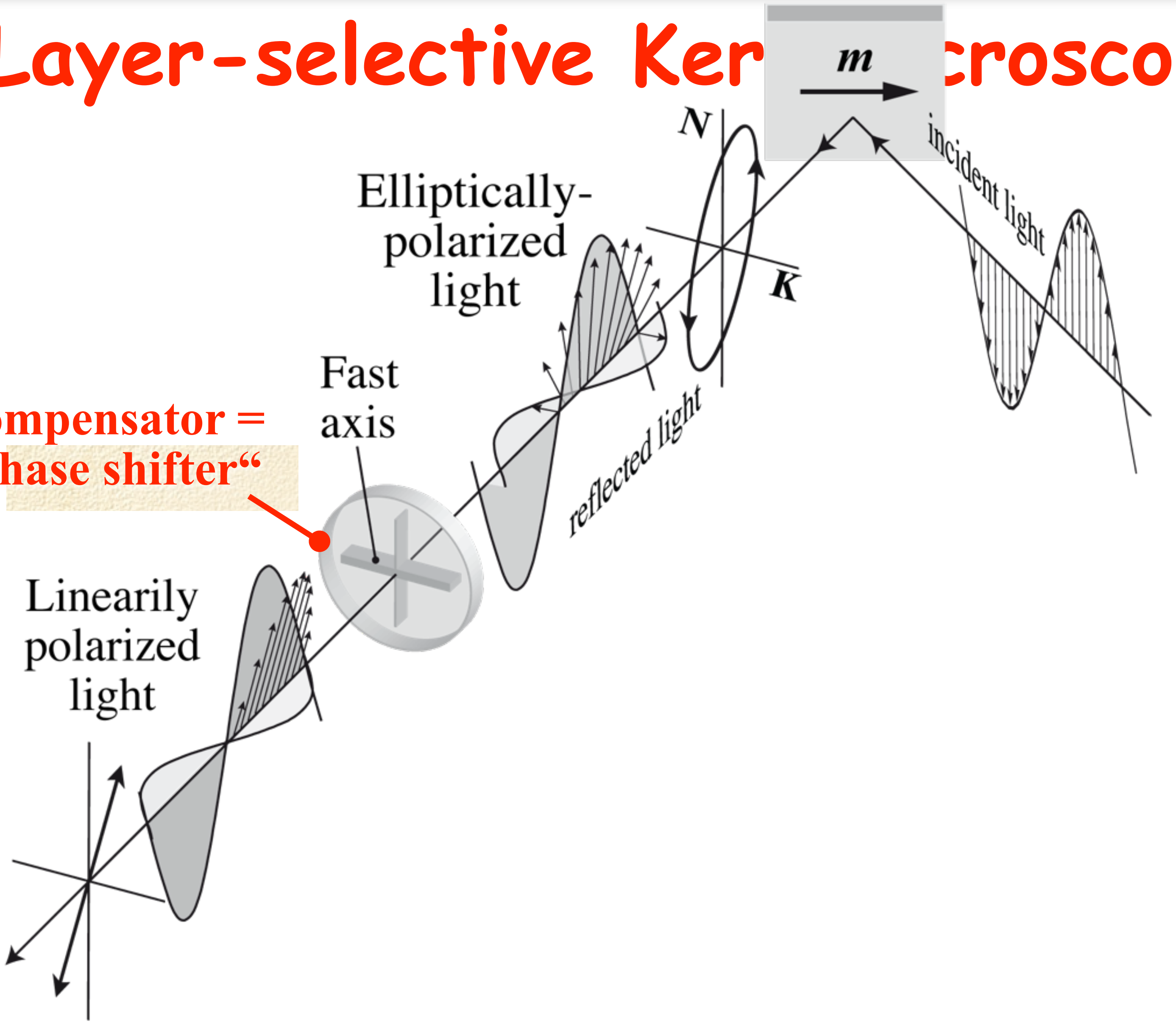
Layer-selective Kerr microscopy



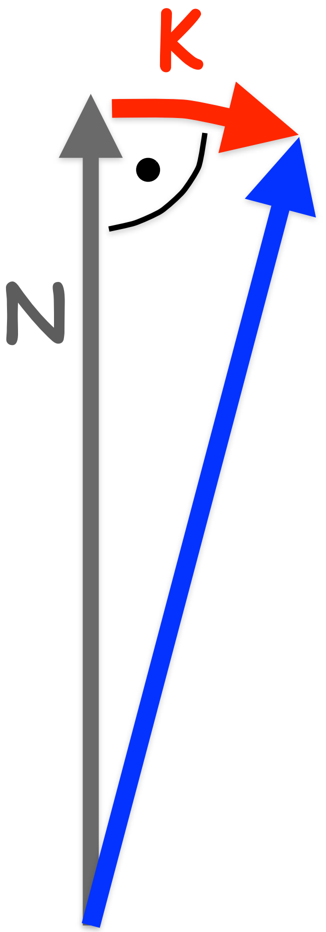
Compensator = „phase shifter“

4. Magneto-optical Kerr-Microscopy

Layer-selective Ker croscopy

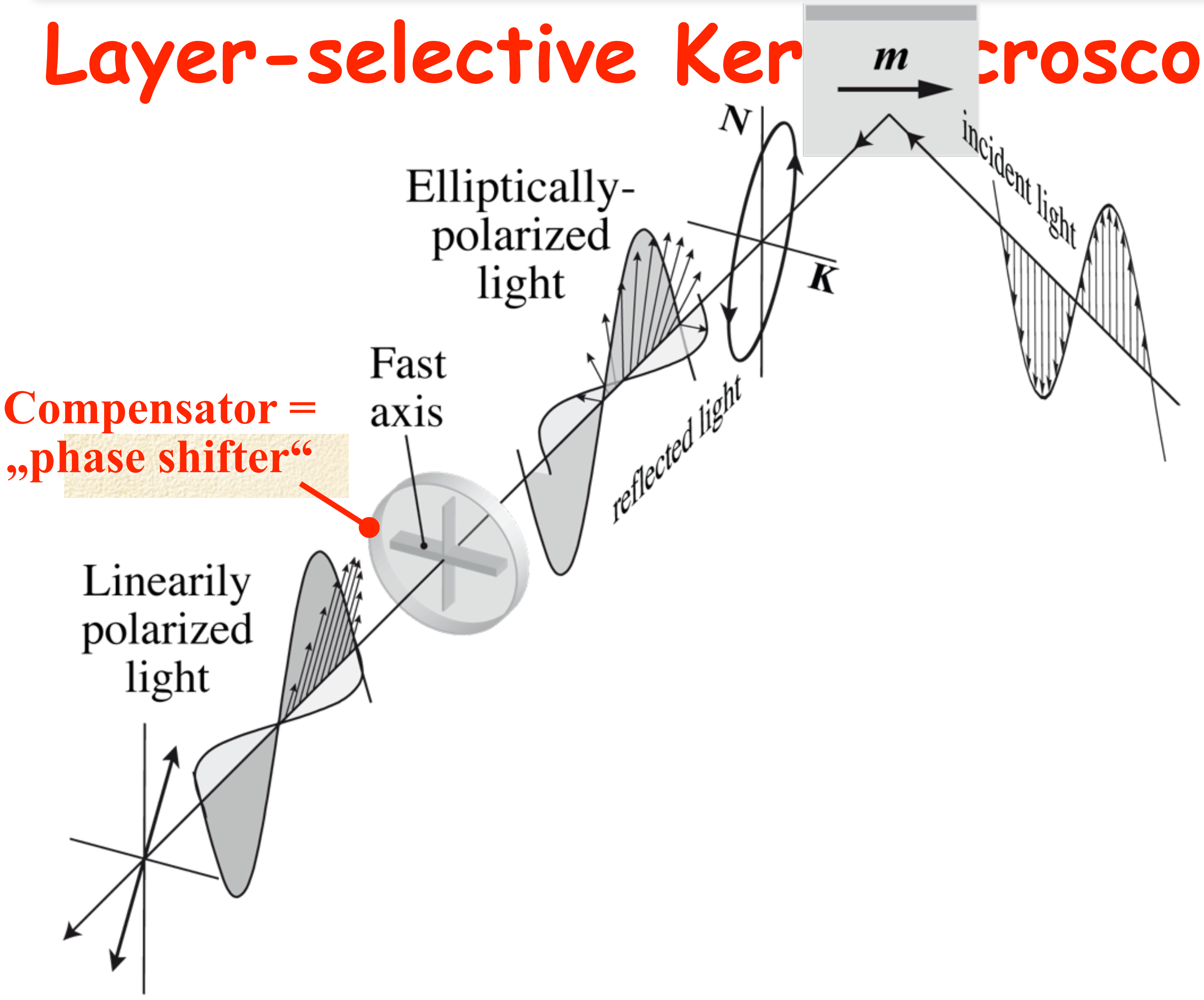


Compensator = „phase shifter“

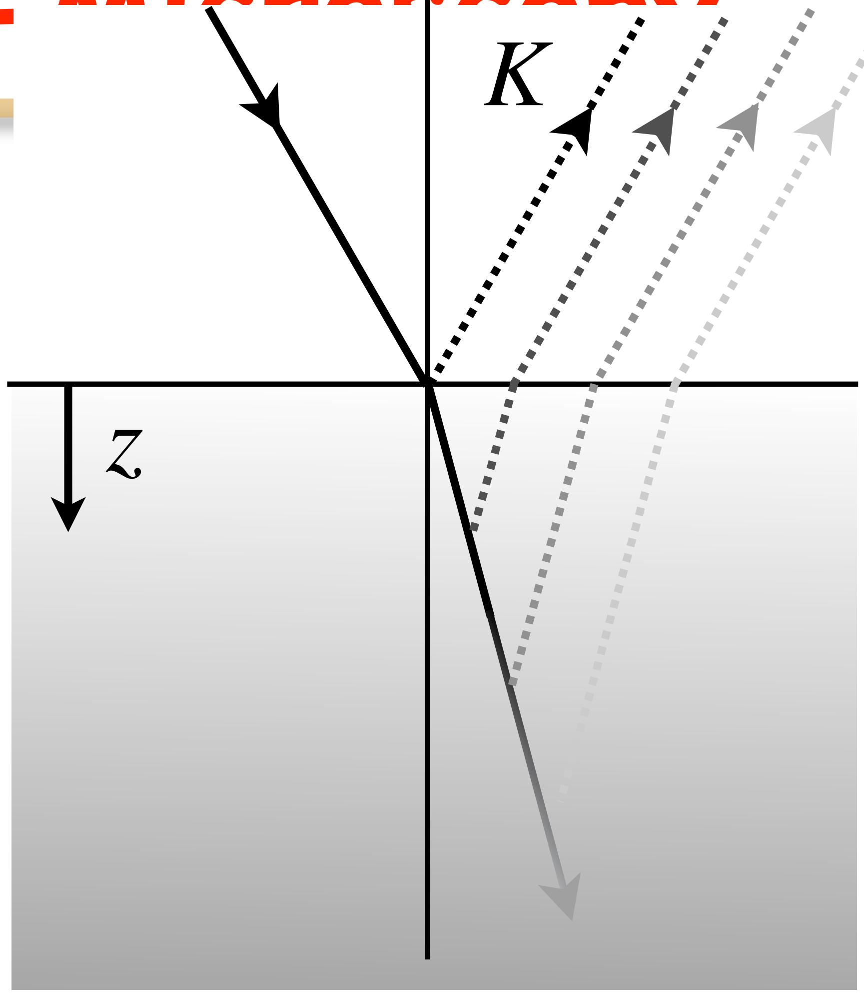


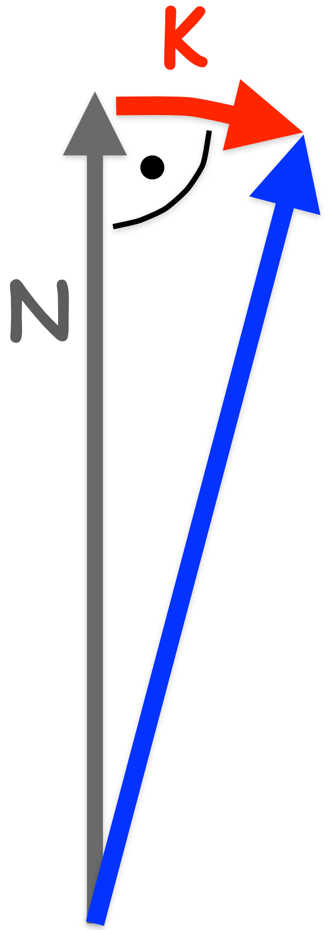
4. Magneto-optical Kerr-

Layer-selective Ker croscopy



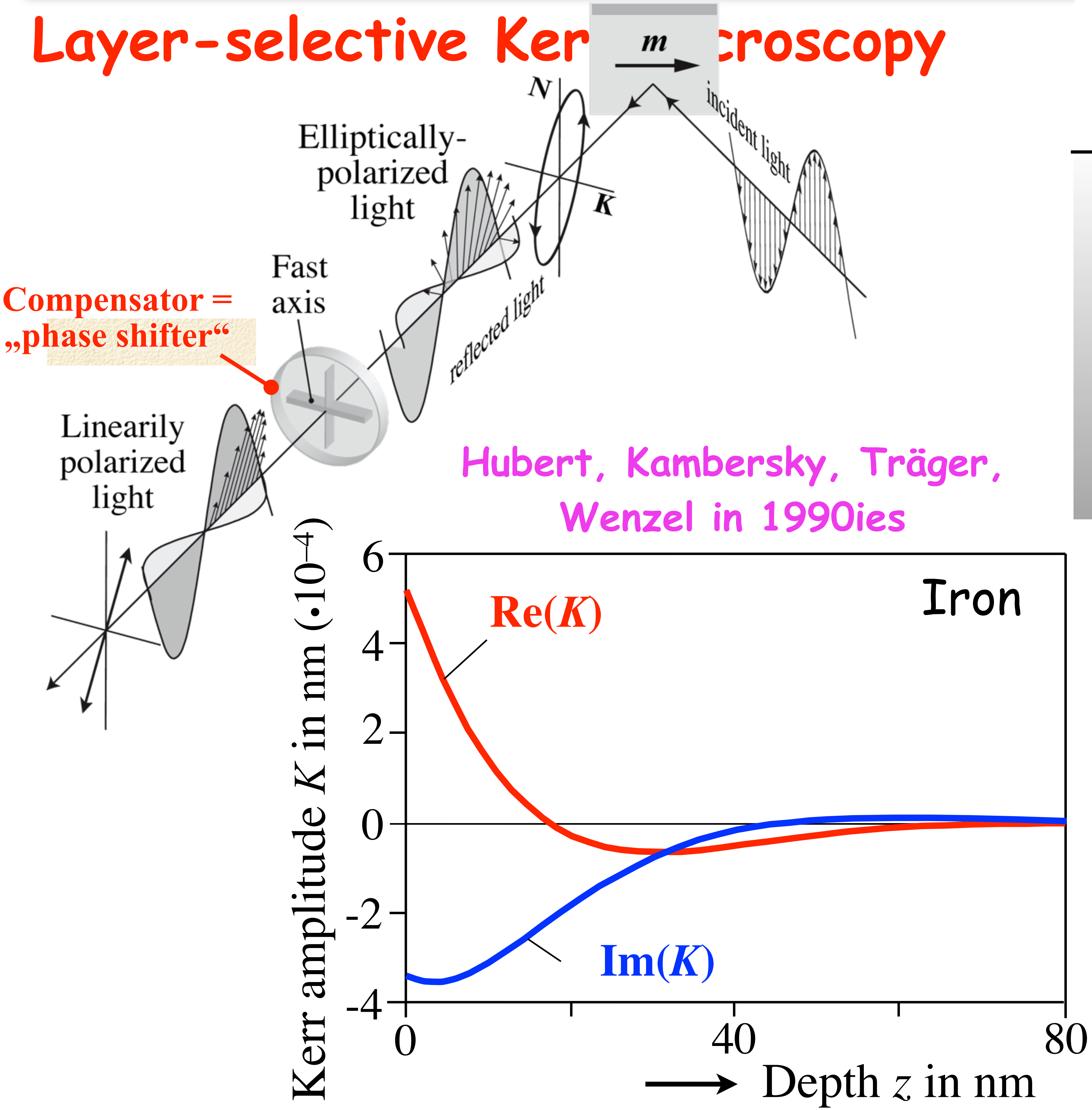
Compensator = „phase shifter“



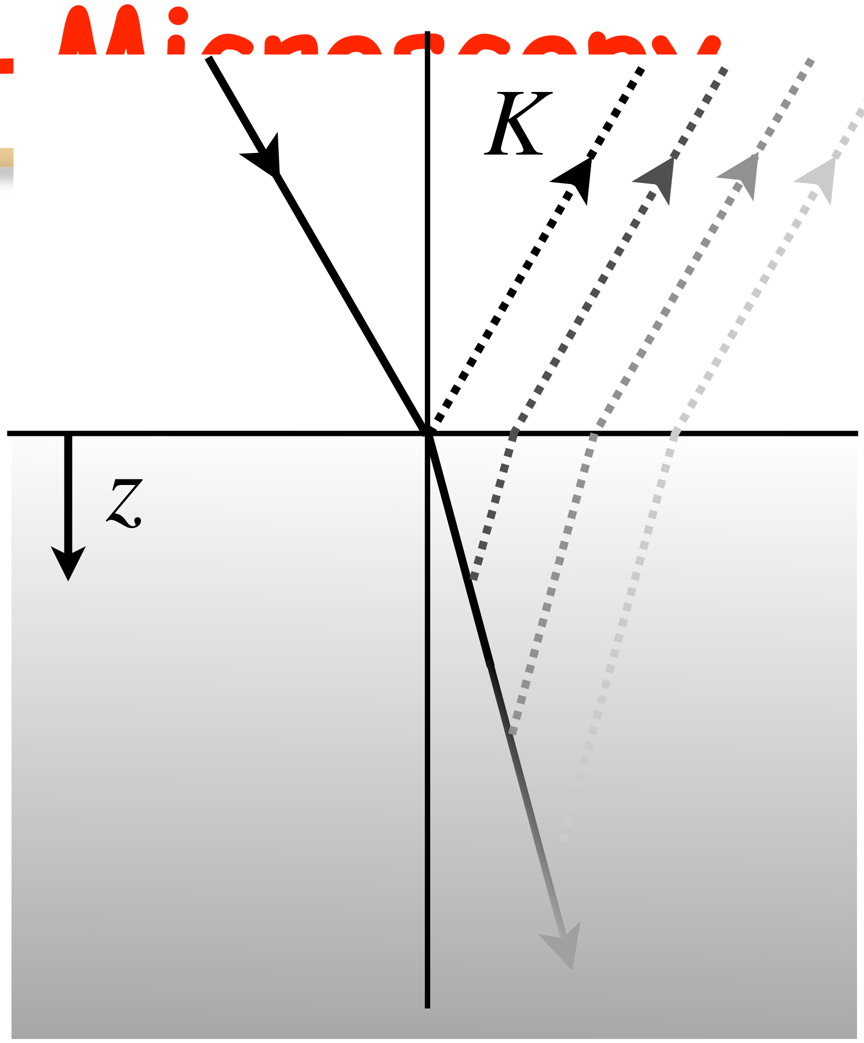


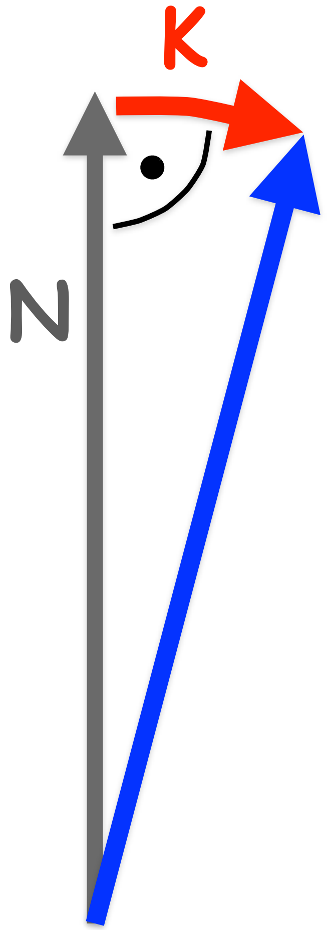
4. Magneto-optical Kerr-Microscopy

Layer-selective Ker croscopy



Hubert, Kambersky, Träger, Wenzel in 1990ies





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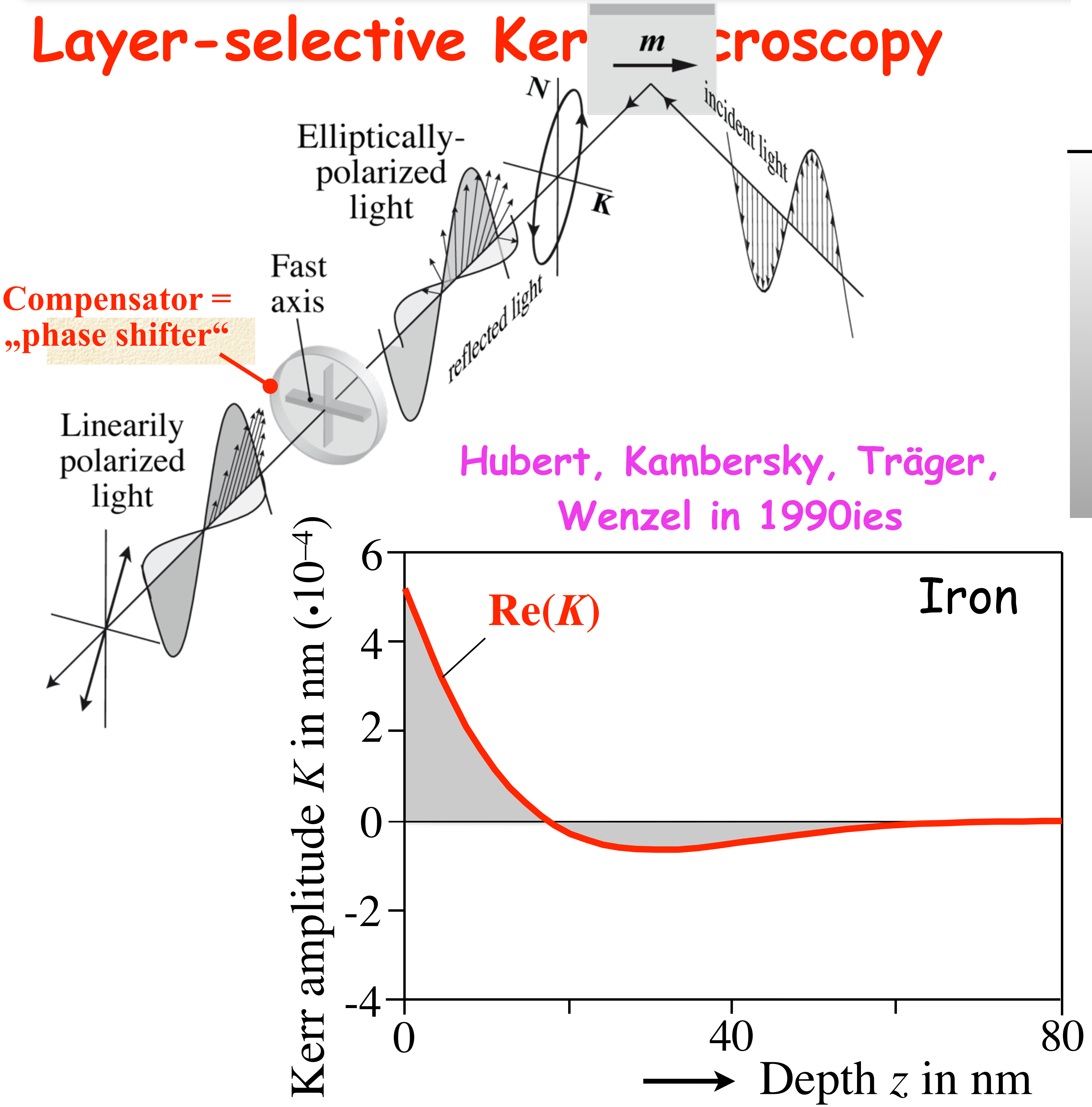
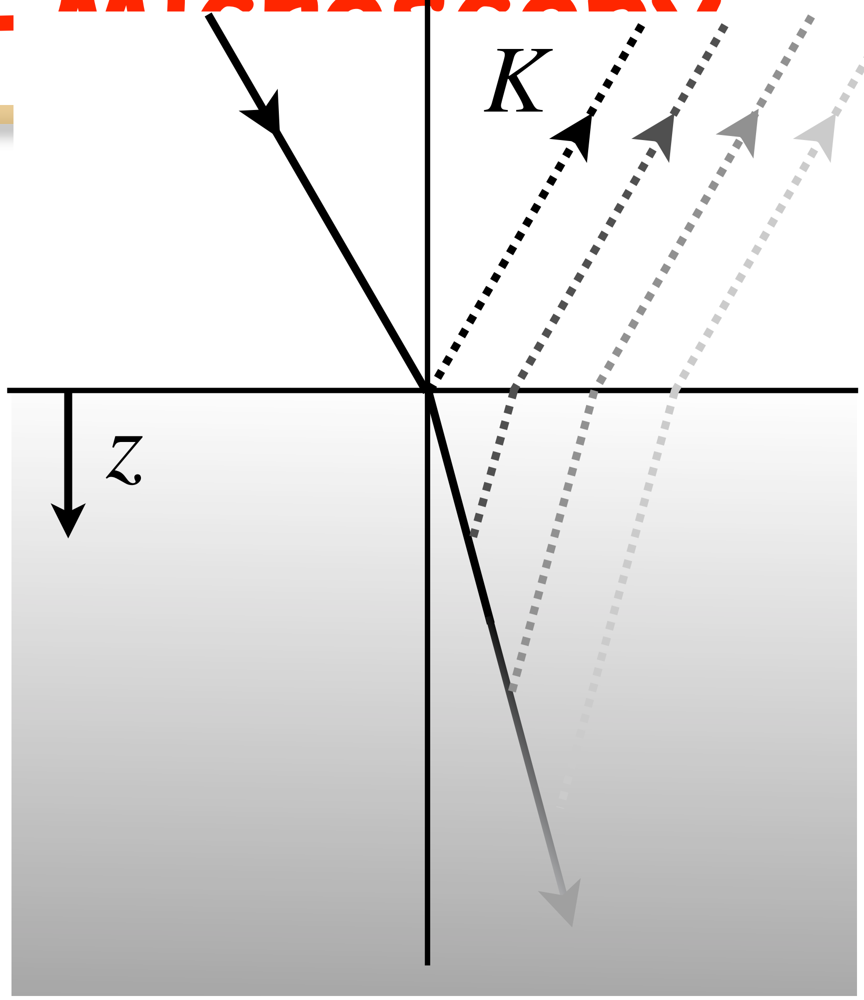
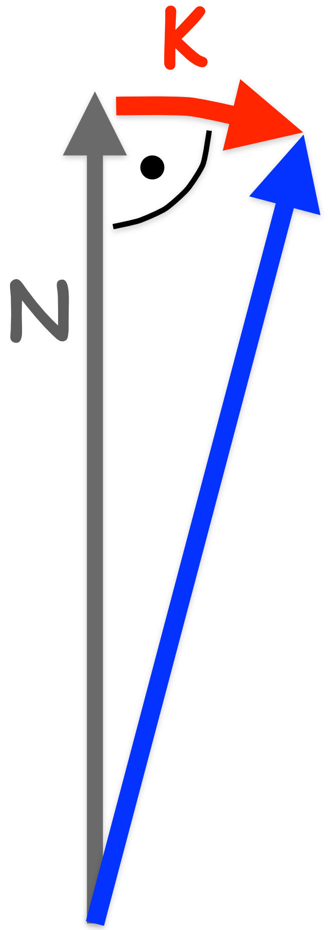
Layer-selective Ker croscopy



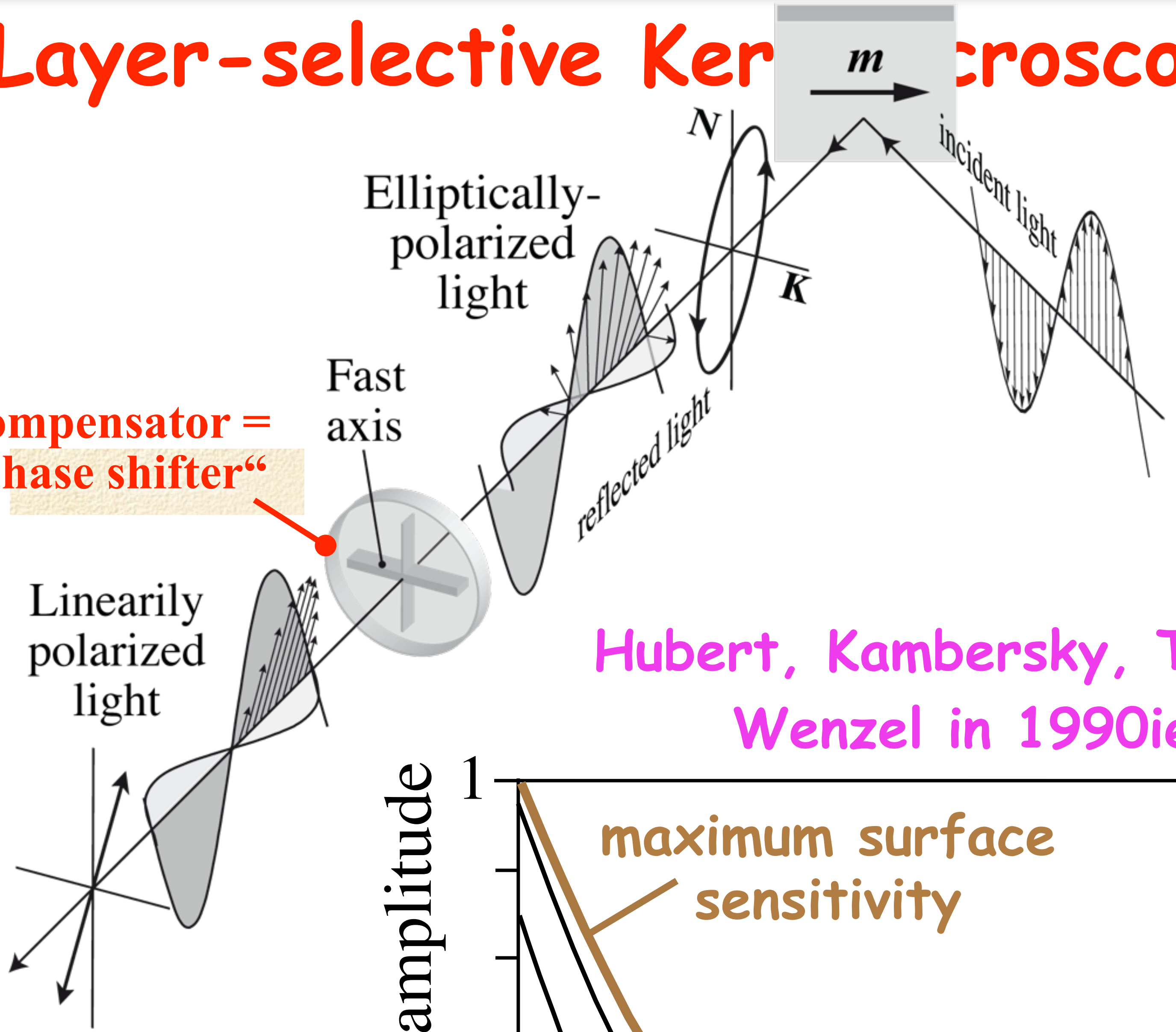
Hubert, Kambersky, Träger, Wenzel in 1990ies





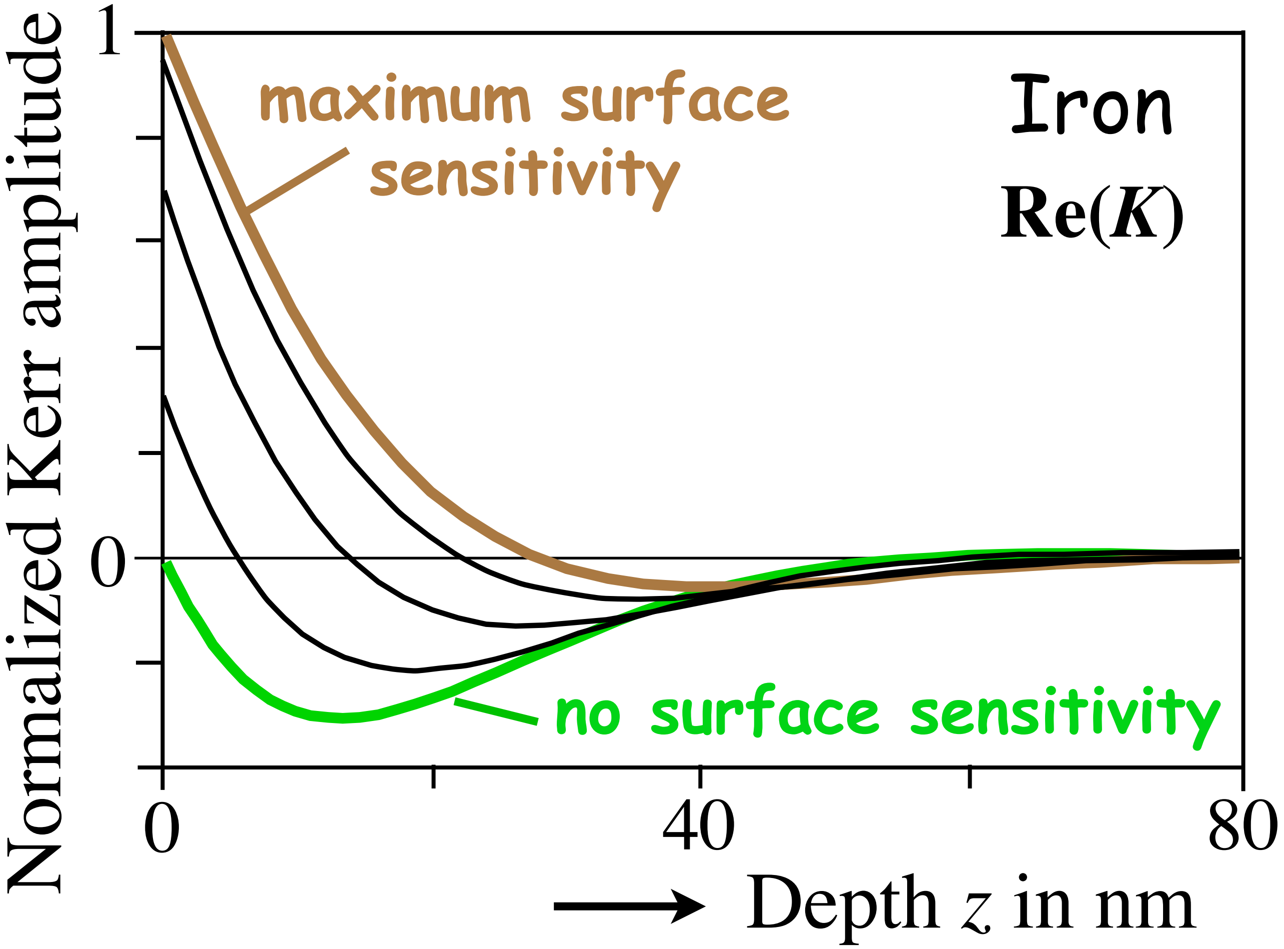
4. Magneto-optical Kerr-Microscopy

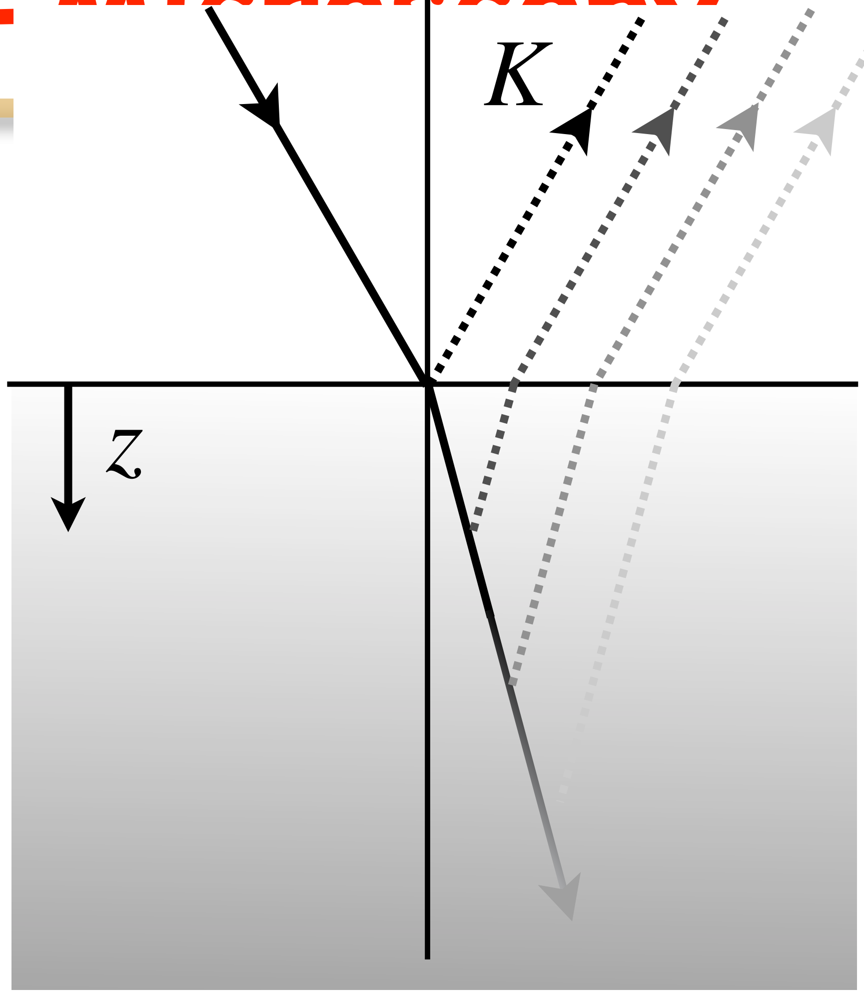
Layer-selective Ker croscopy

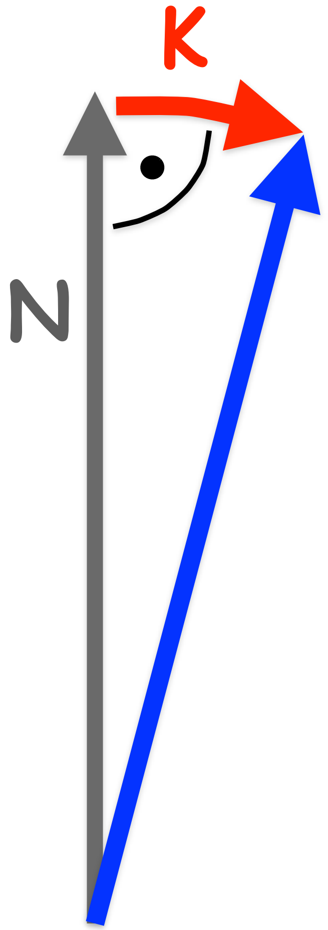


Compensator = „phase shifter“

Hubert, Kambersky, Träger, Wenzel in 1990ies

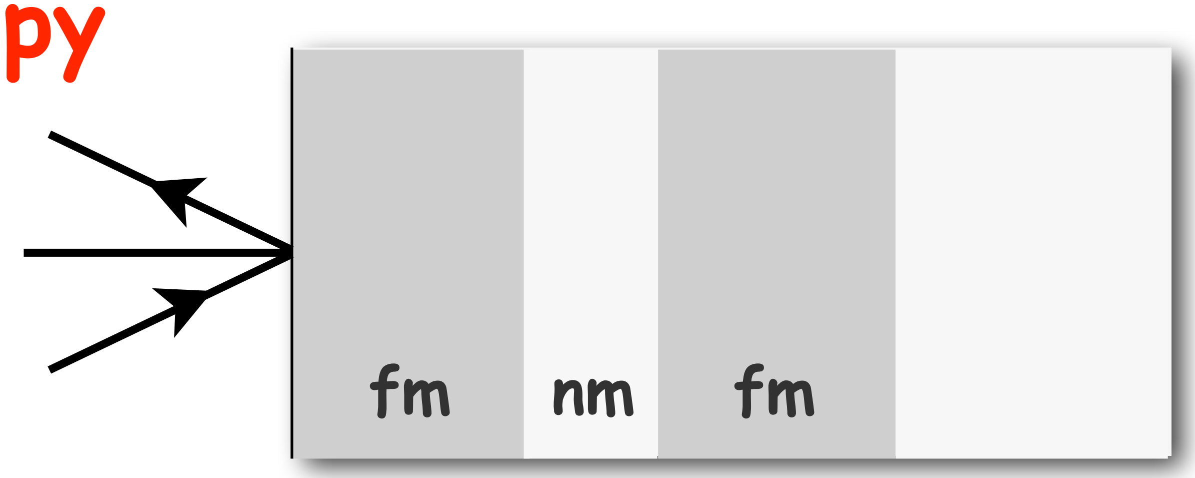


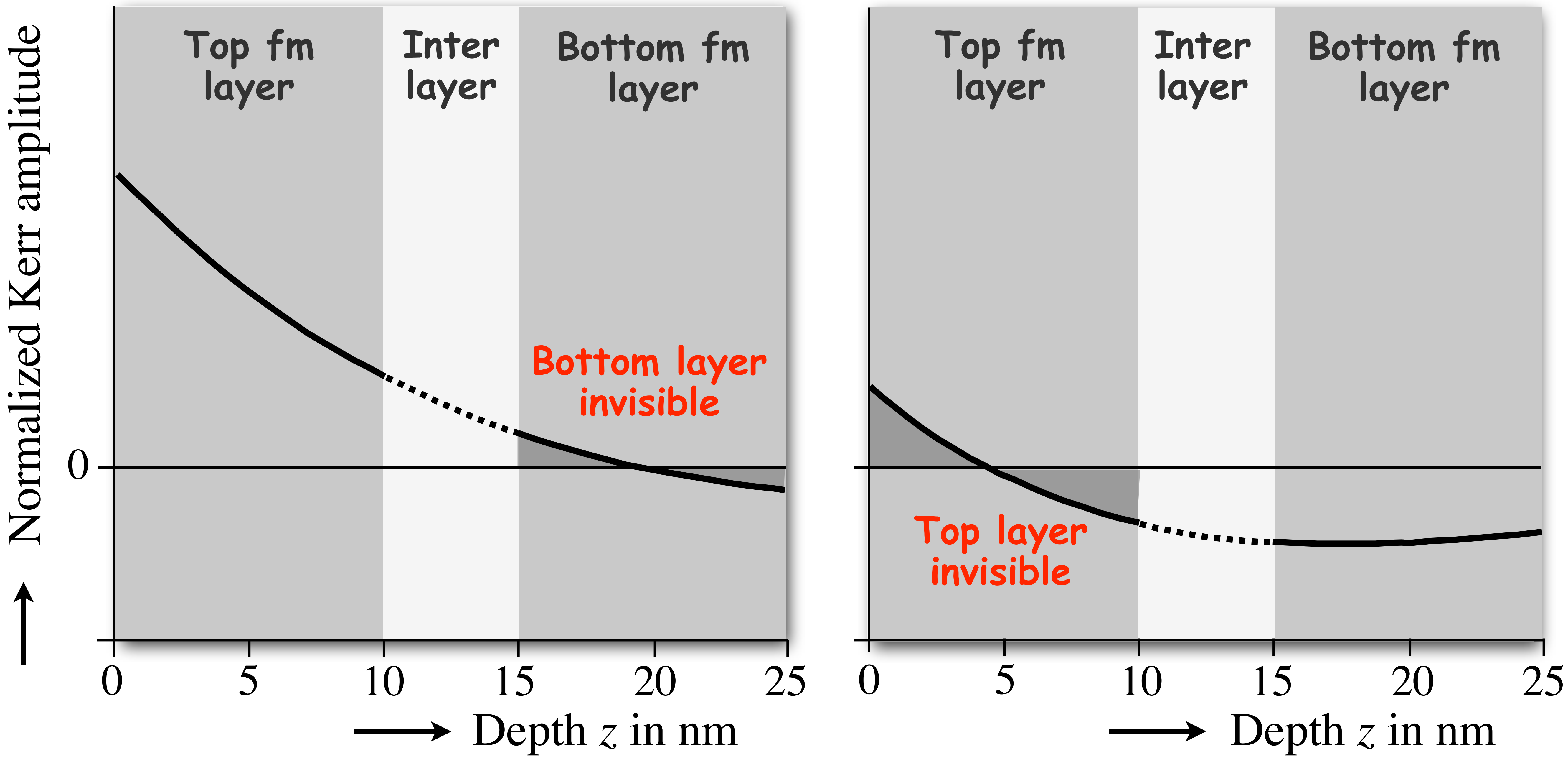


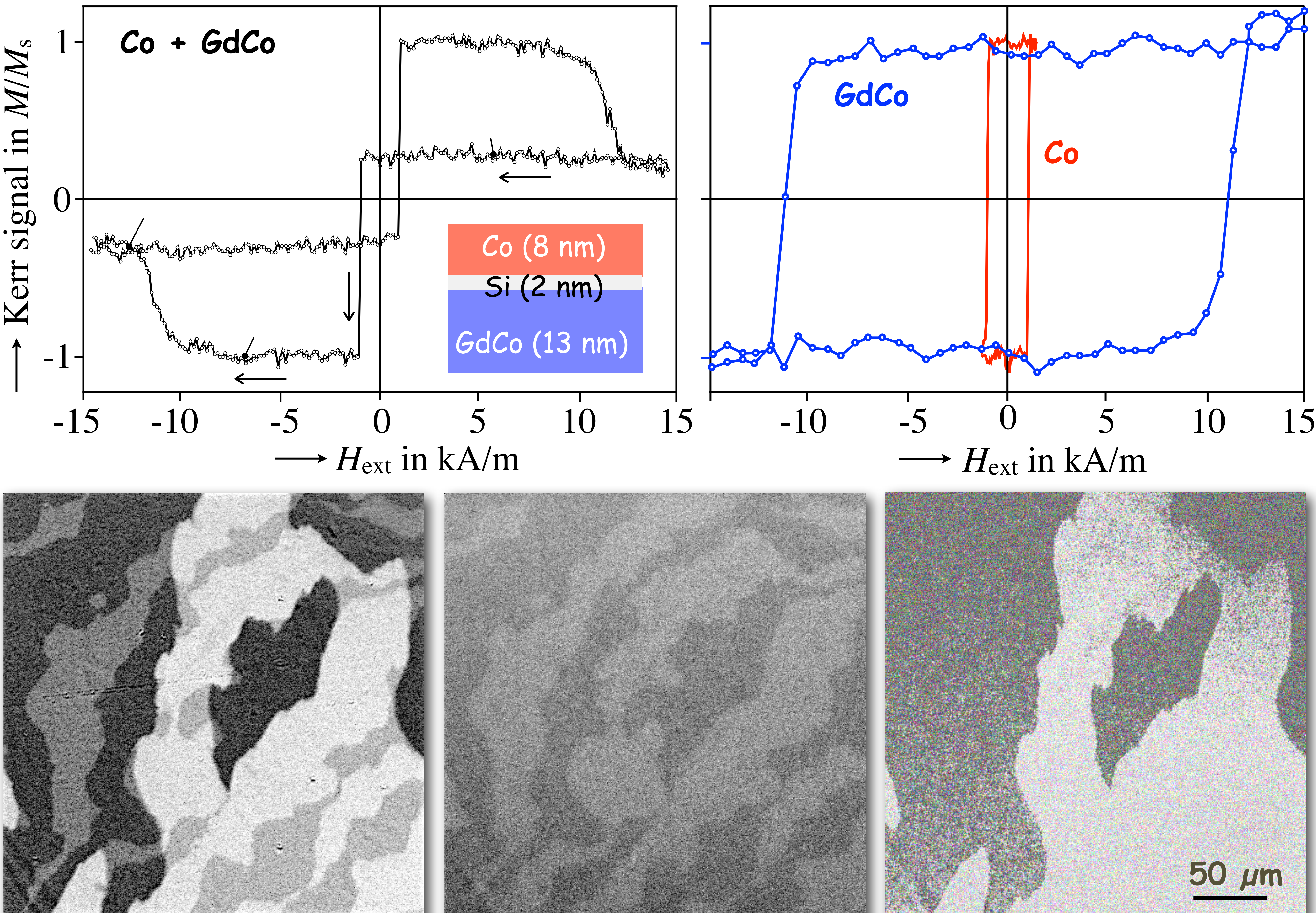


4. Magneto-optical Kerr-Microscopy

Layer-selective Kerr microscopy



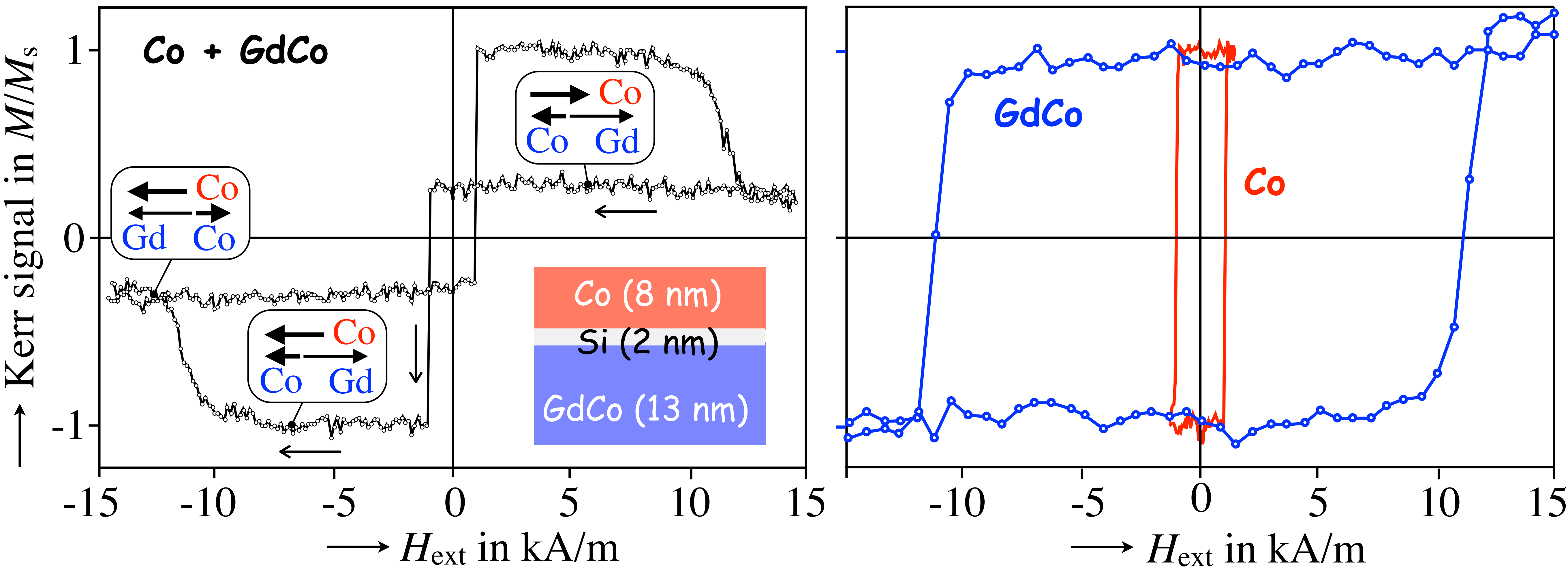


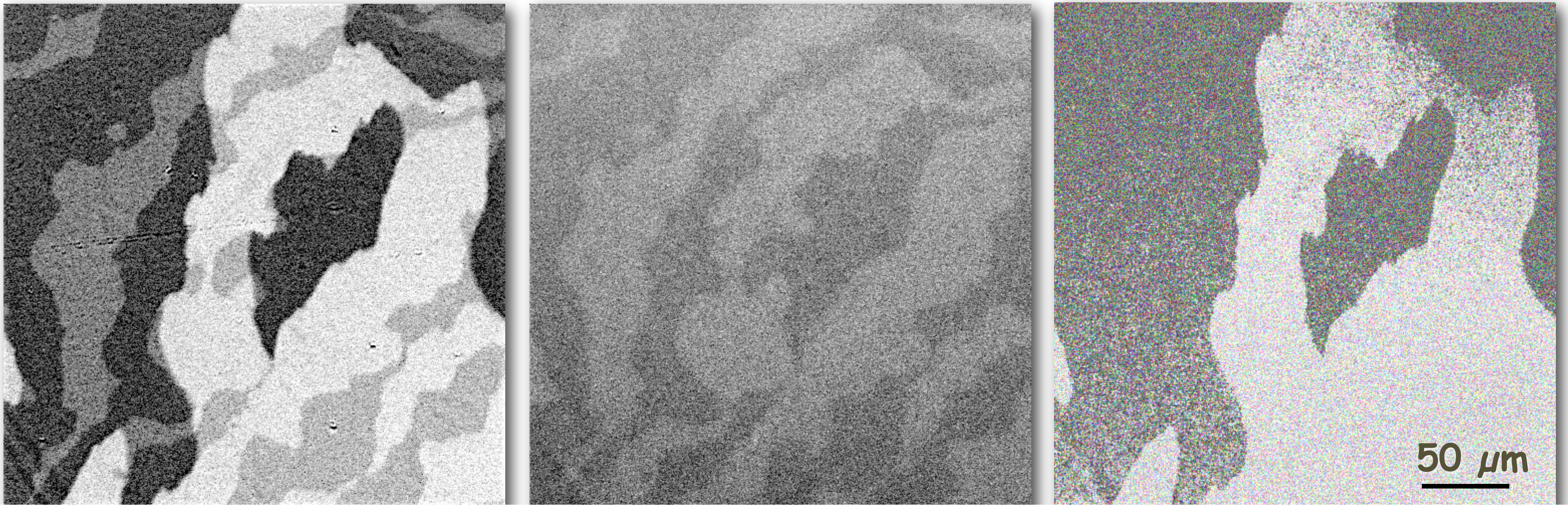


Mixed Kerr signal

GdCo layer

Co layer

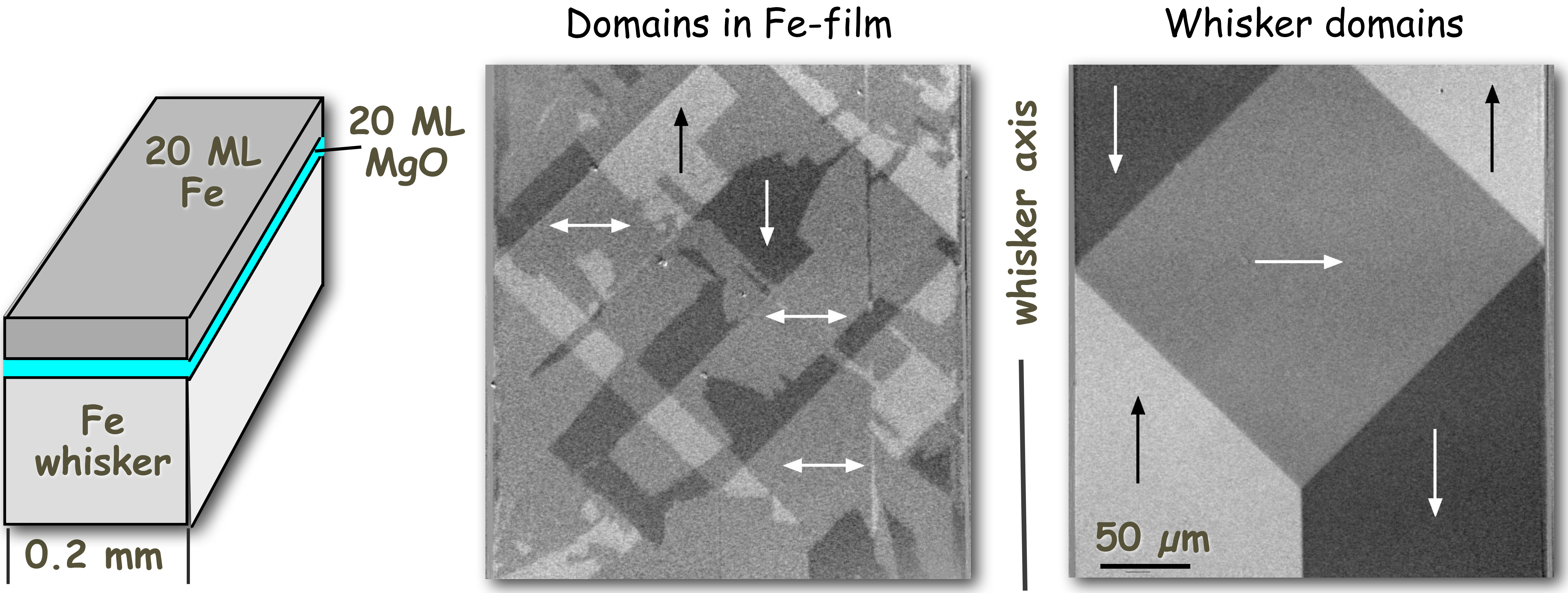




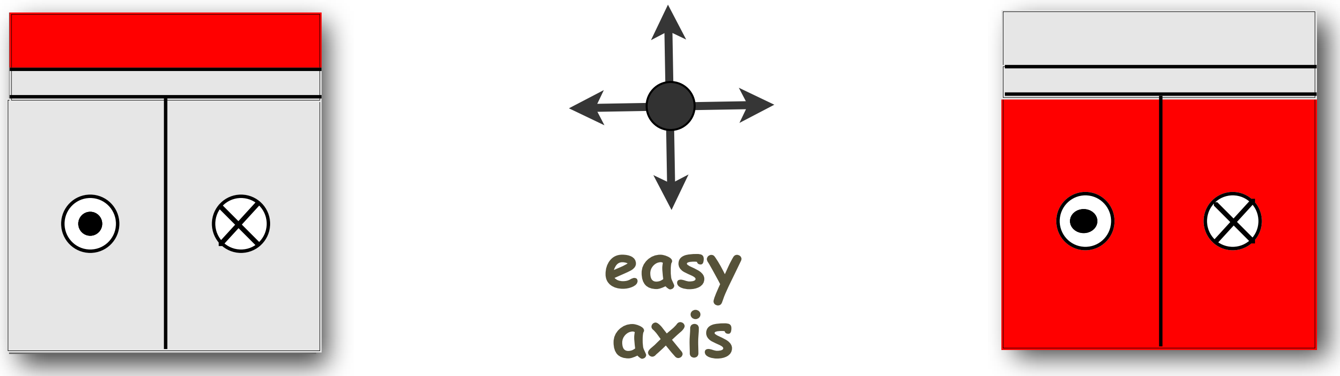
Mixed Kerr signal **GdCo layer** **Co layer**

4. Magneto-optical Kerr-Microscopy

Layer-selective Kerr microscopy



Depth-selectivity by tuning phase of Kerr amplitude with compensator



R. S., R. Urban, D. Ullmann, H. L. Meyerheim, B. Heinrich, L. Schultz, J. Kirschner, PRB 65, 144405 (2003)

4. Magneto-optical Kerr-Microscopy

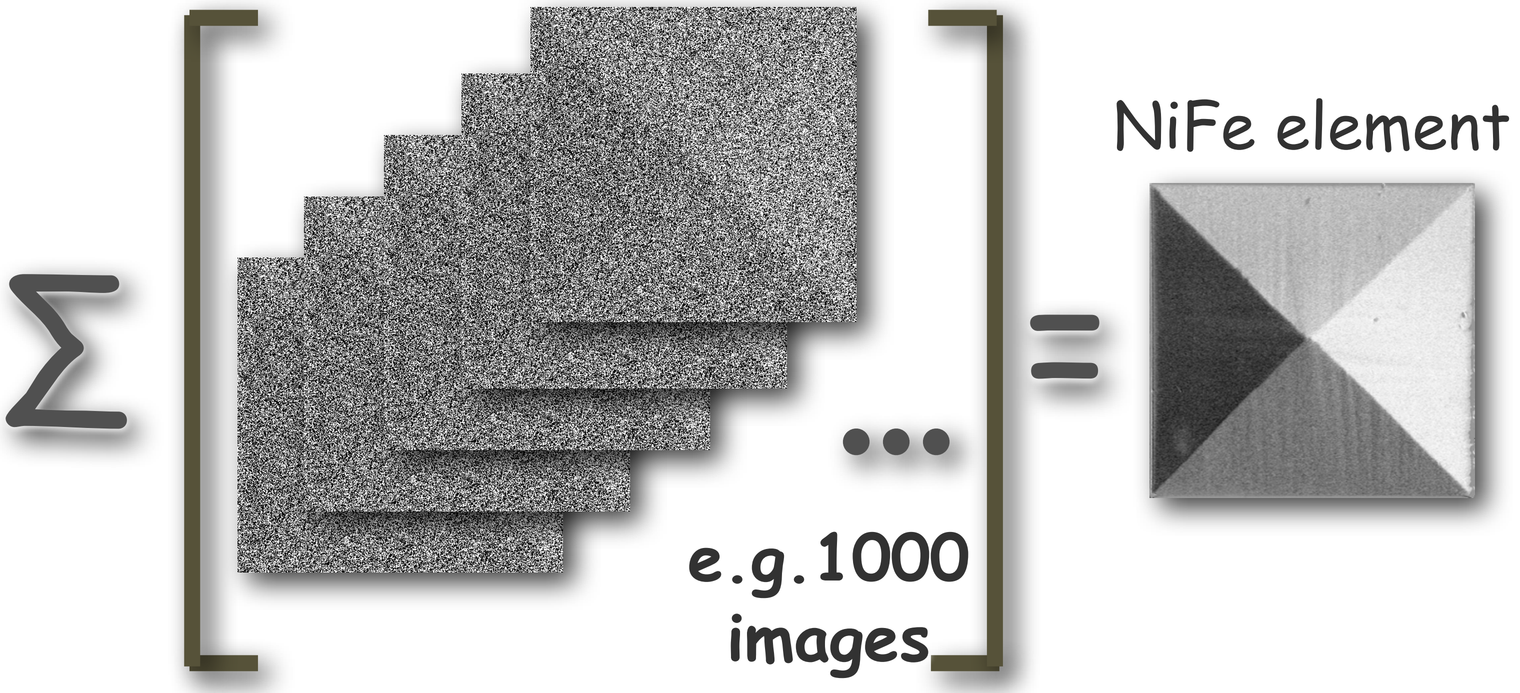
Time-resolved (stroboscopic) imaging

illumination intensity and repetition rate are limited

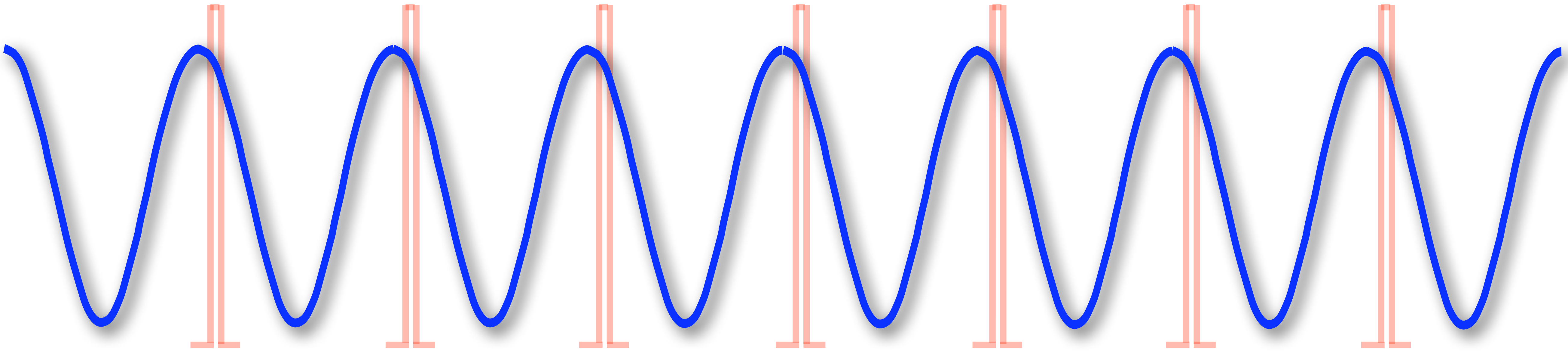
→ no single-shot imaging possible

→ accumulation of large number of independent events necessary (at fixed time delay)

→ requires repetitive magnetization processes !!



probing with defined time delay



periodic magnetic field excitation

4. Magneto-optical Kerr-Microscopy

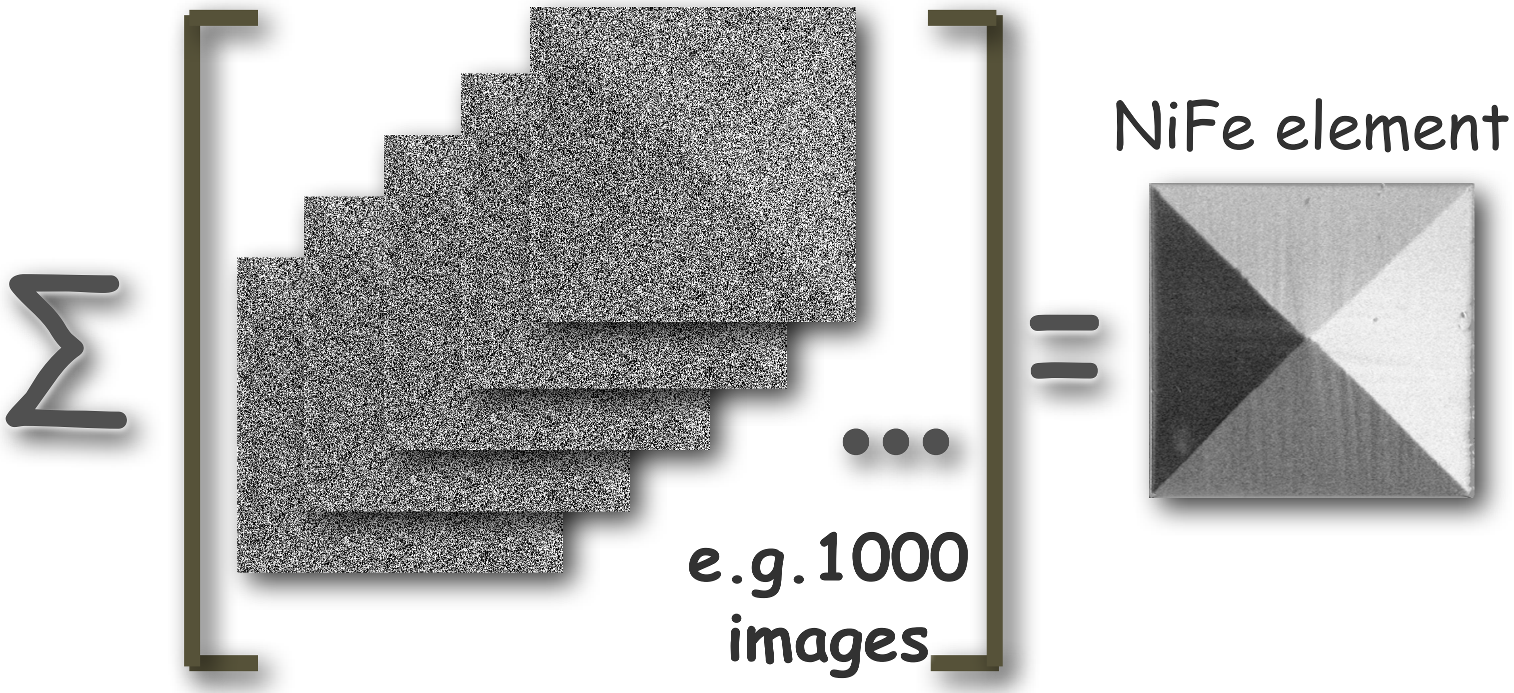
Time-resolved (stroboscopic) imaging

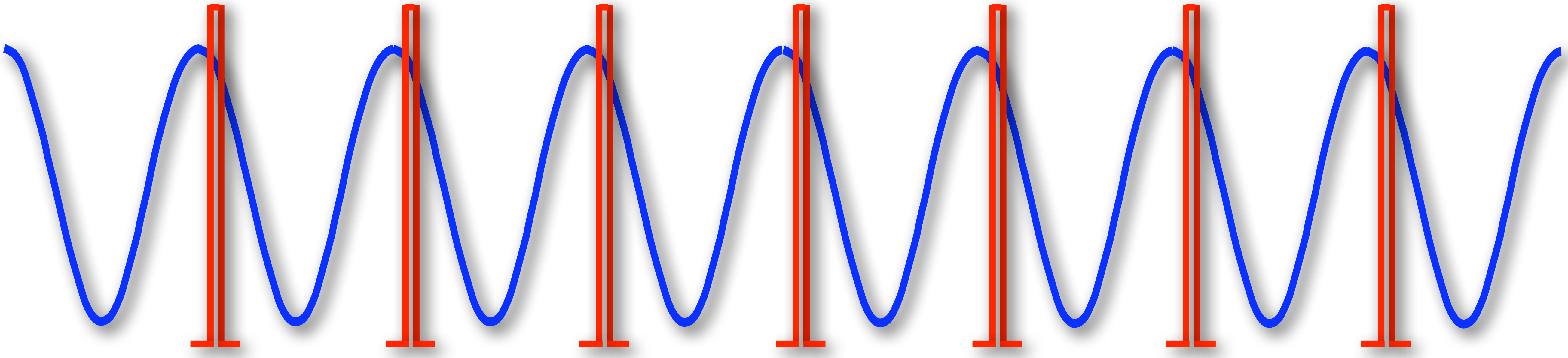
illumination intensity and repetition rate are limited

→ no single-shot imaging possible

→ accumulation of large number of independent events necessary (at fixed time delay)

→ requires repetitive magnetization processes !!





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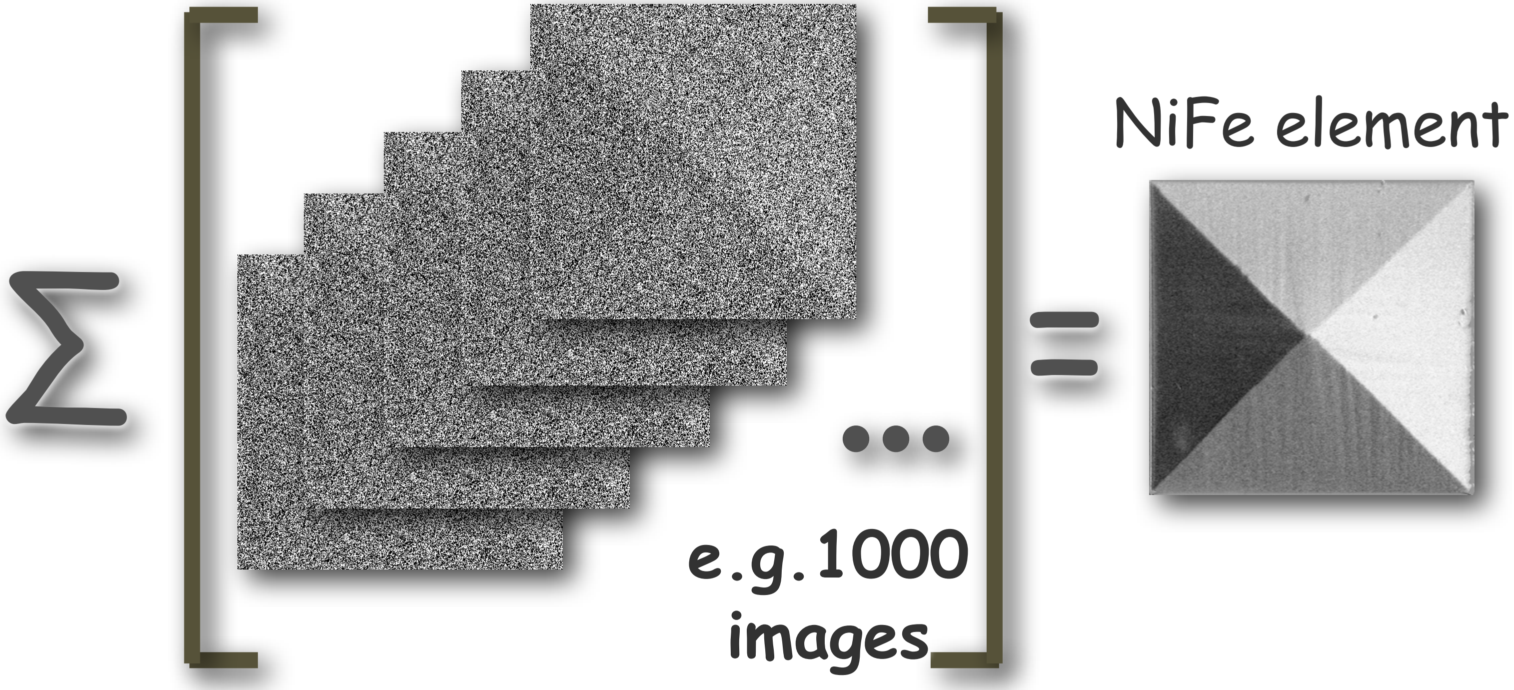
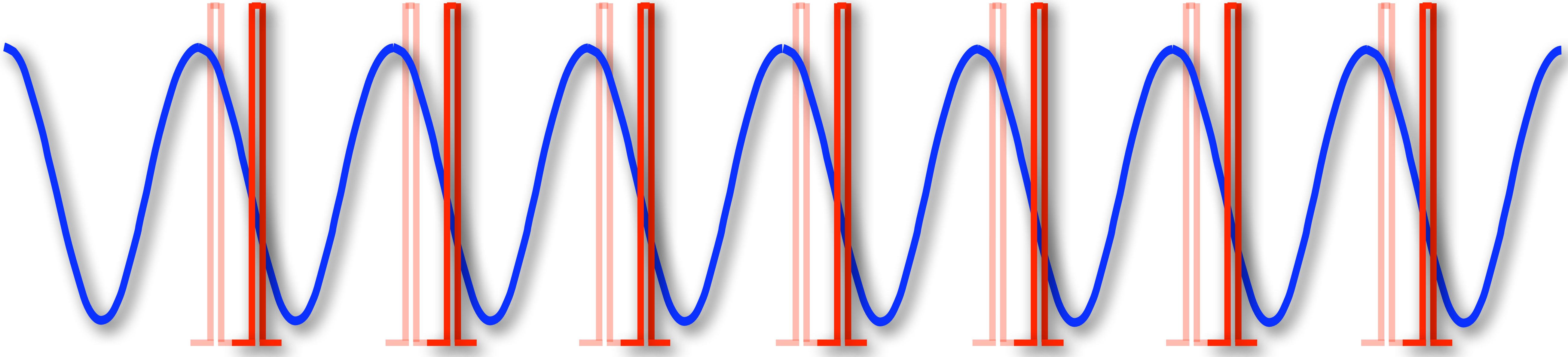


probing with defined time delay

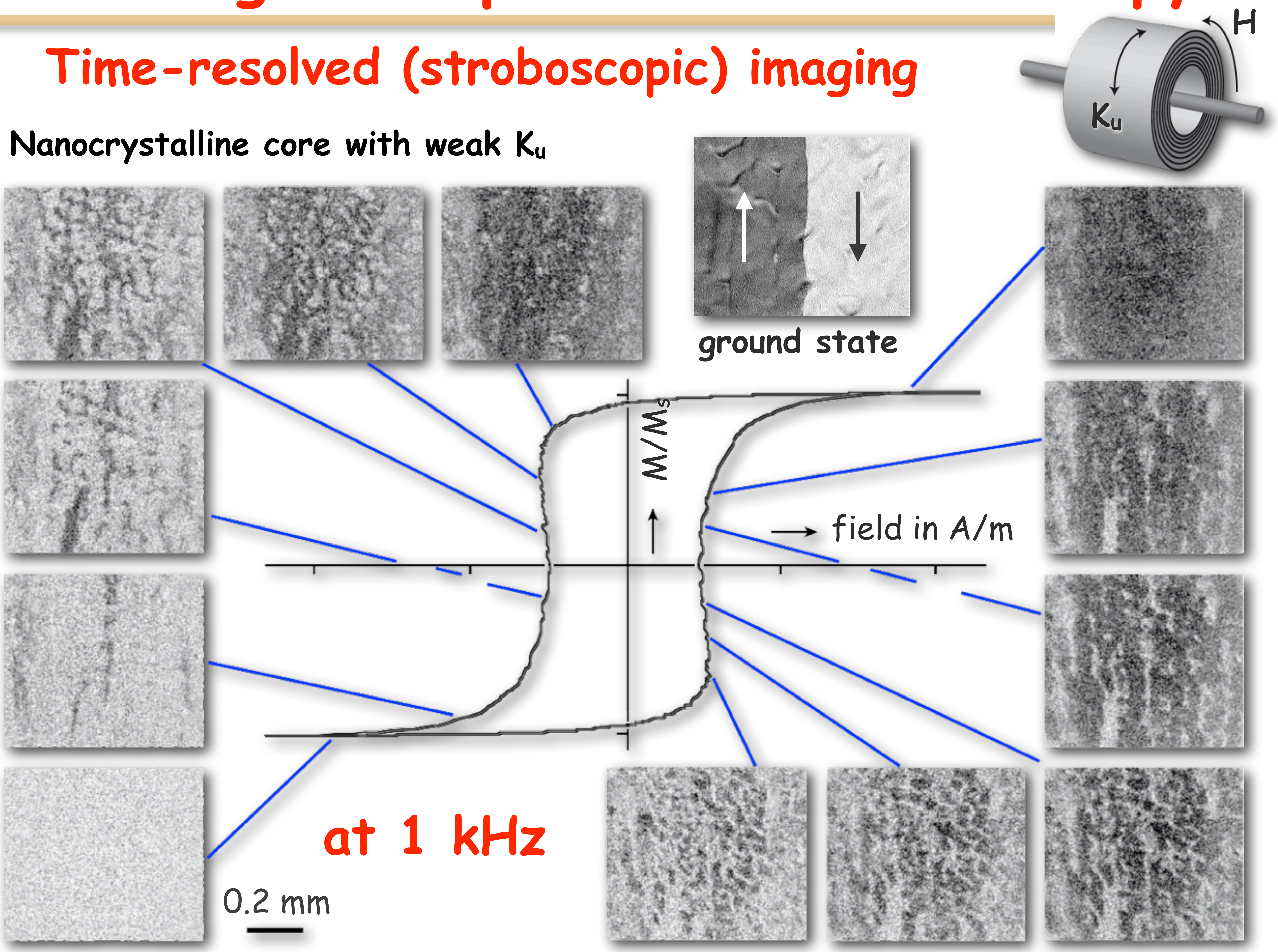


periodic magnetic field excitation

4. Magneto-optical Kerr-Microscopy

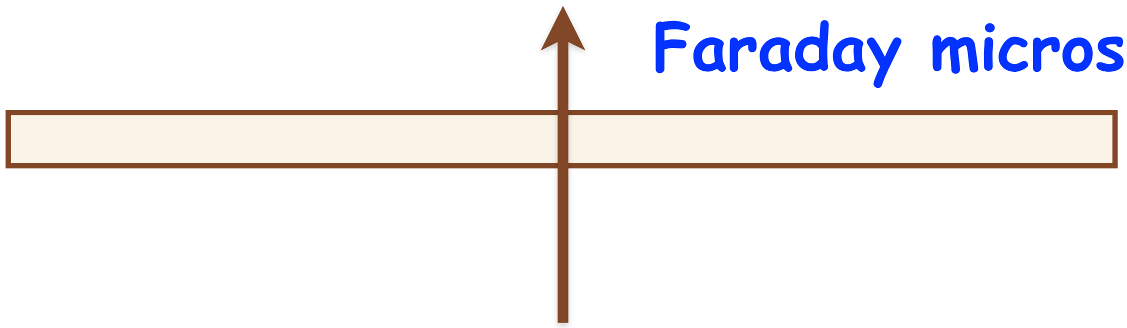
Time-resolved (stroboscopic) imaging

Nanocrystalline core with weak K_u

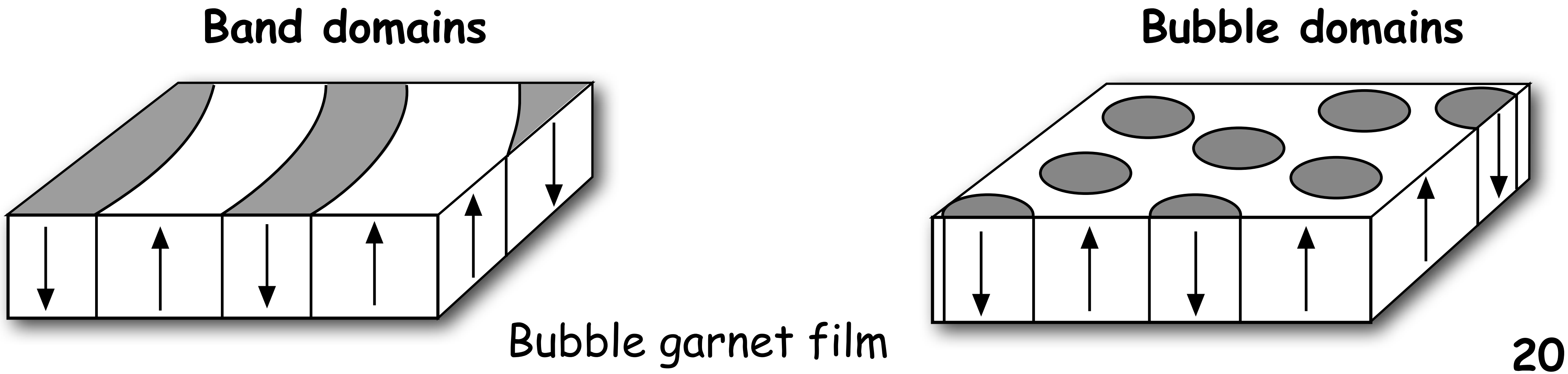


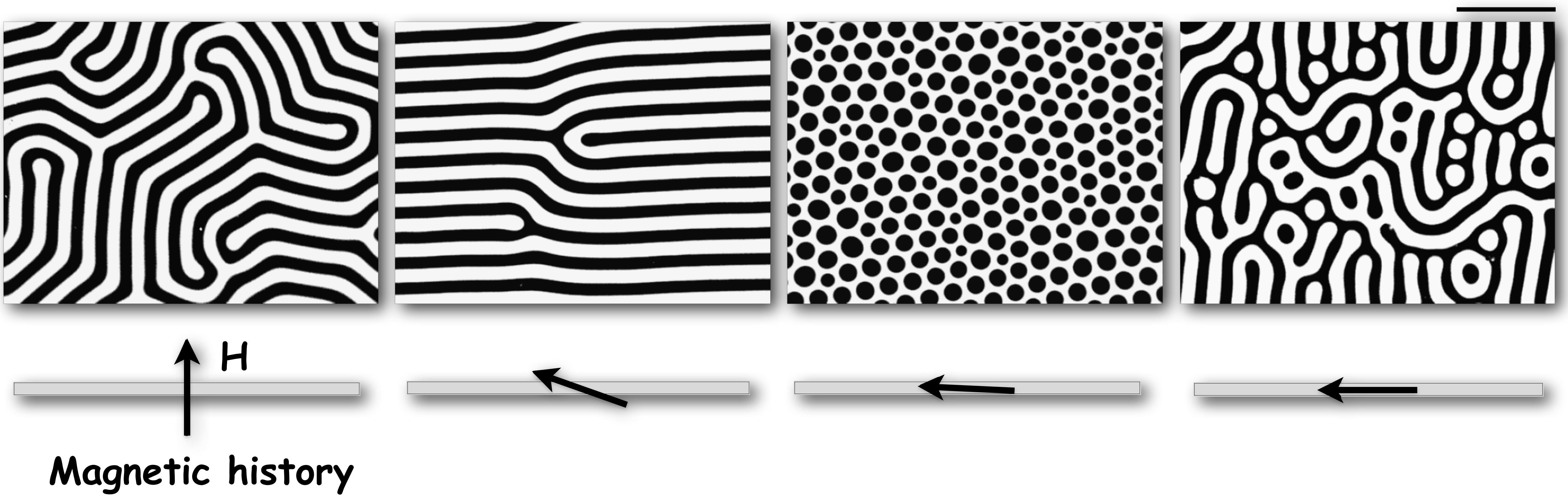
at 1 kHz

4. Magneto-optical Kerr-Microscopy

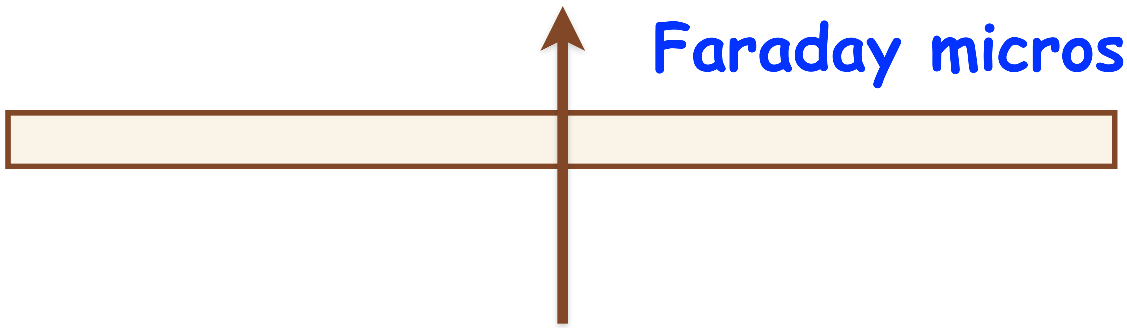


Faraday microscopy (like Kerr, but in transmission)



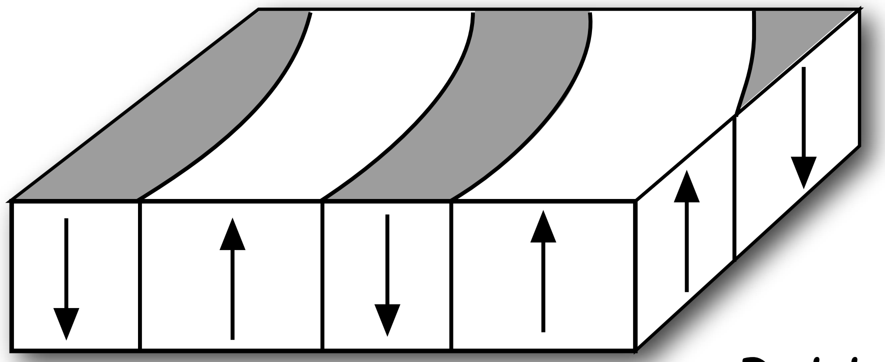


4. Magneto-optical Faraday-Microscopy

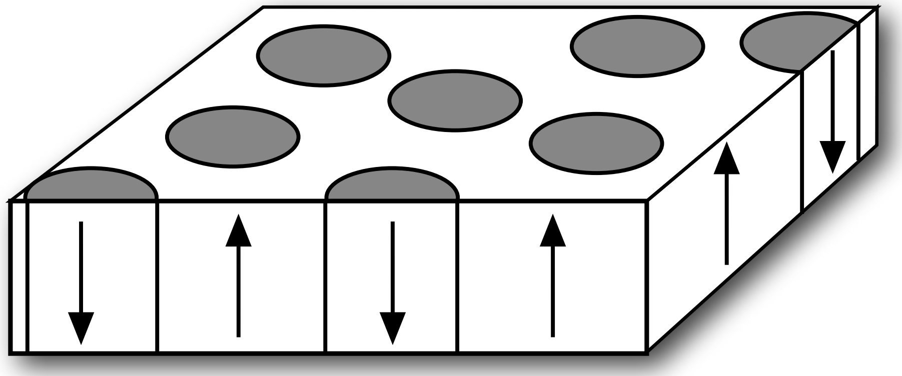


Faraday microscopy (like Kerr, but in transmission)

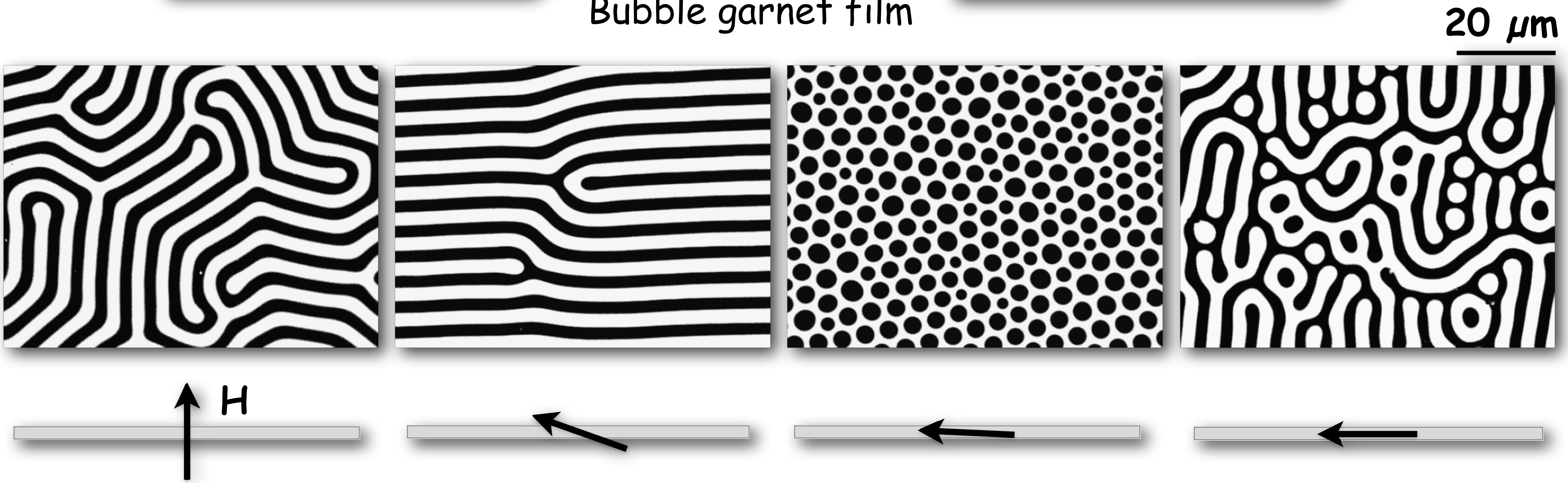
Band domains



Bubble domains



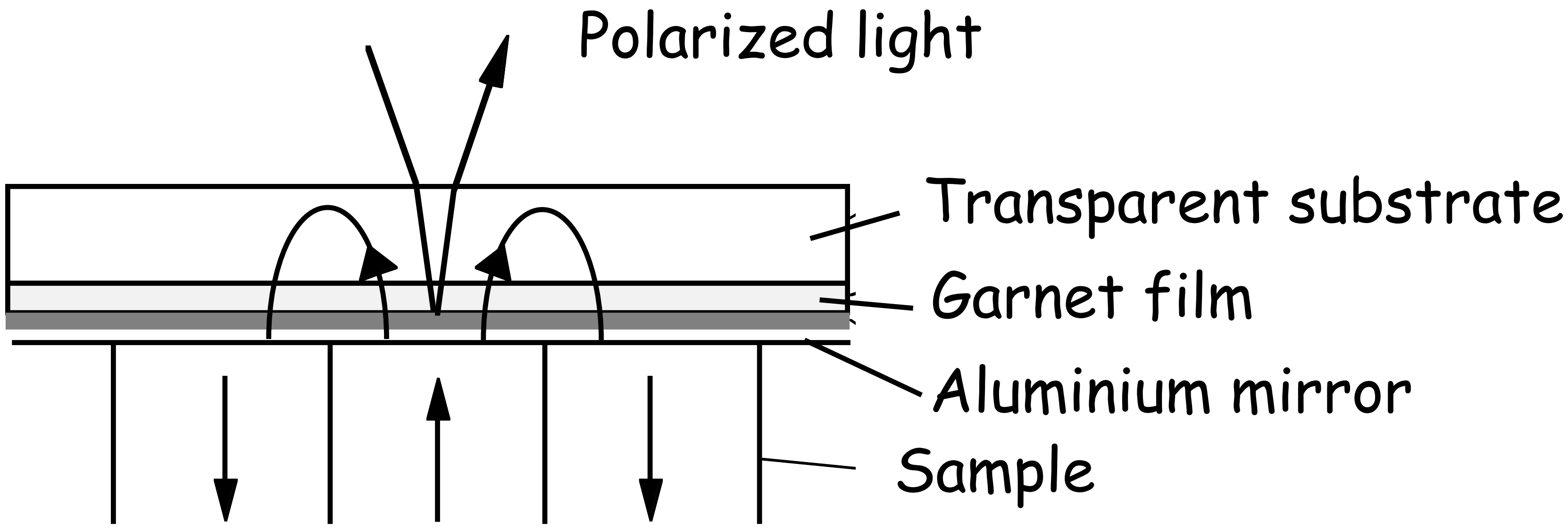
Bubble garnet film

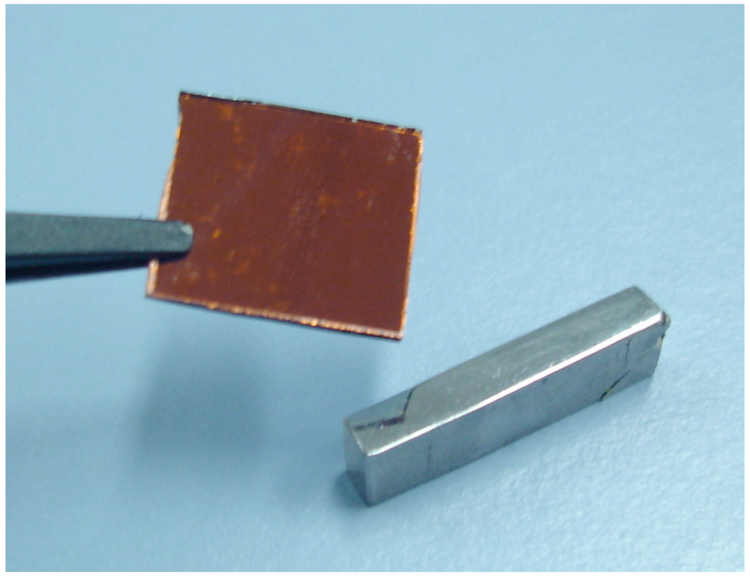


Magnetic history

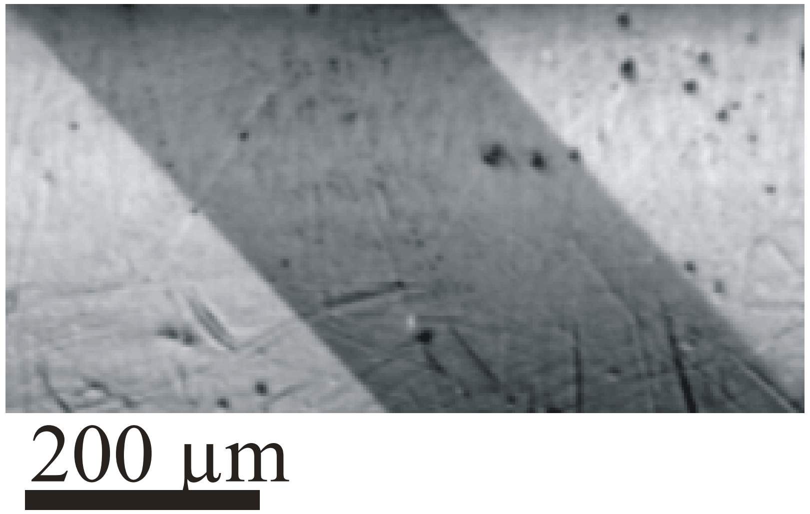
4. Magneto-optical Faraday-Microscopy

Indicator film technique: stray field imaging

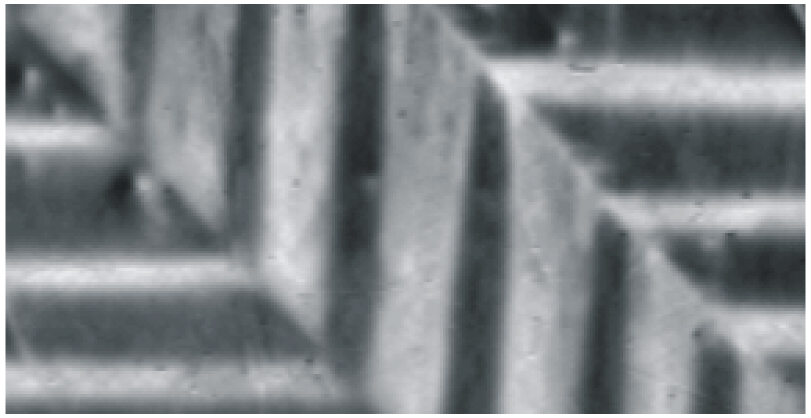


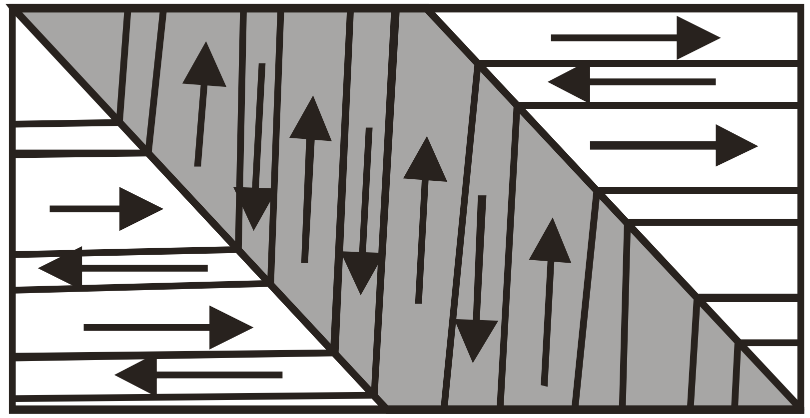


Metallographic contrast



With indicator film





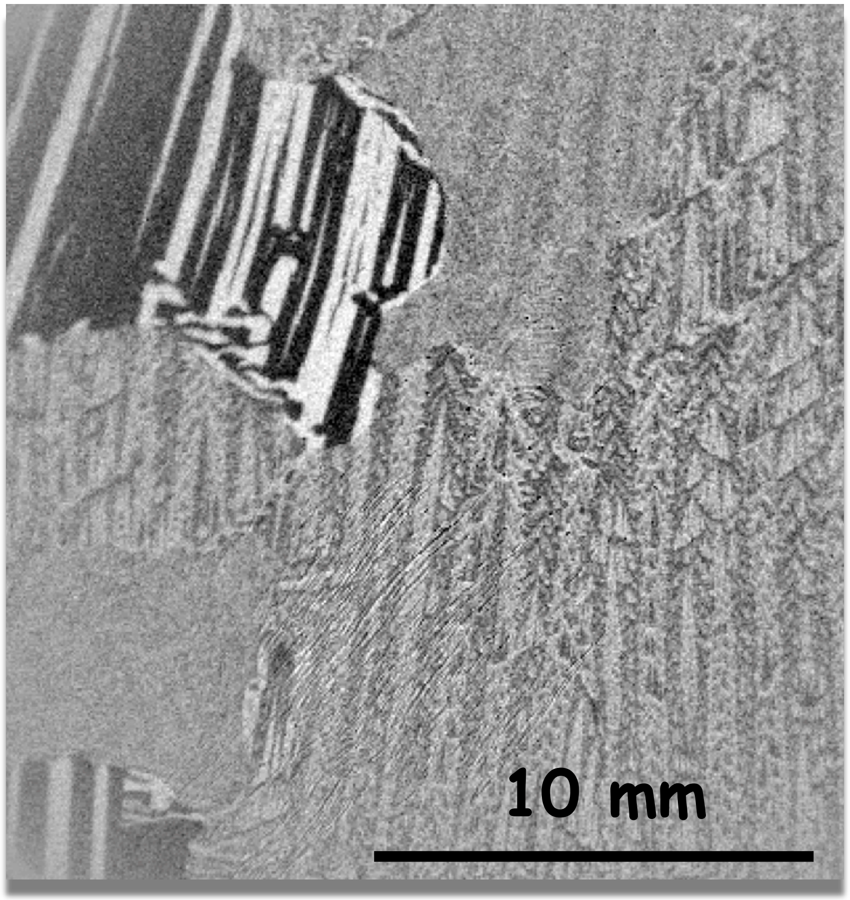
Bulk NiMnGa single crystal

(shape memory material, does not show direct Kerr contrast)

4. Magneto-optical Faraday-Microscopy

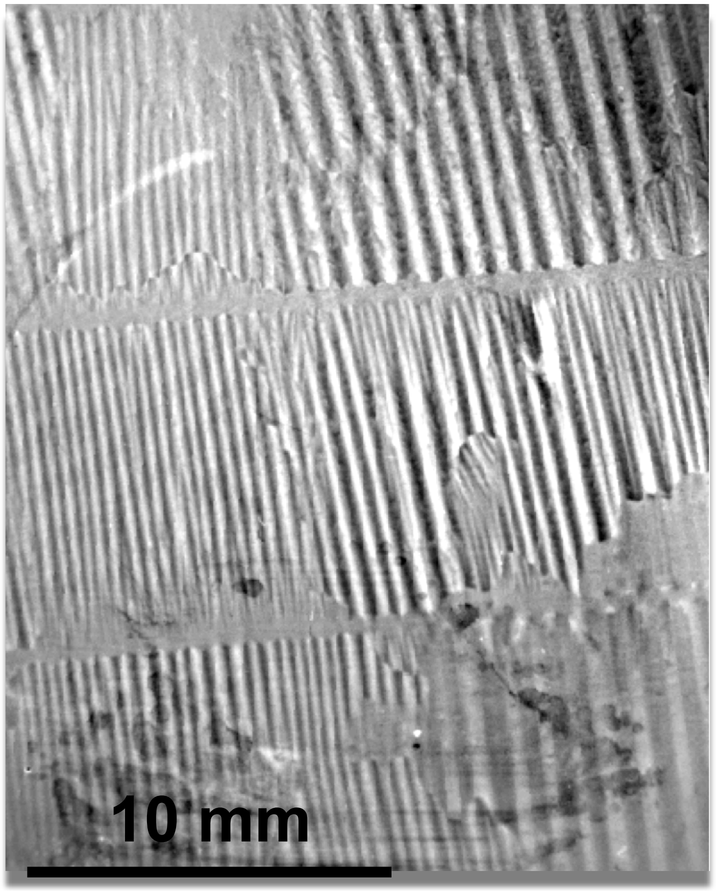
Indicator film technique: stray field imaging

Grain oriented electrical steel:



Direct Kerr contrast
(sample polished)

MOIF contrast
(sample polished)



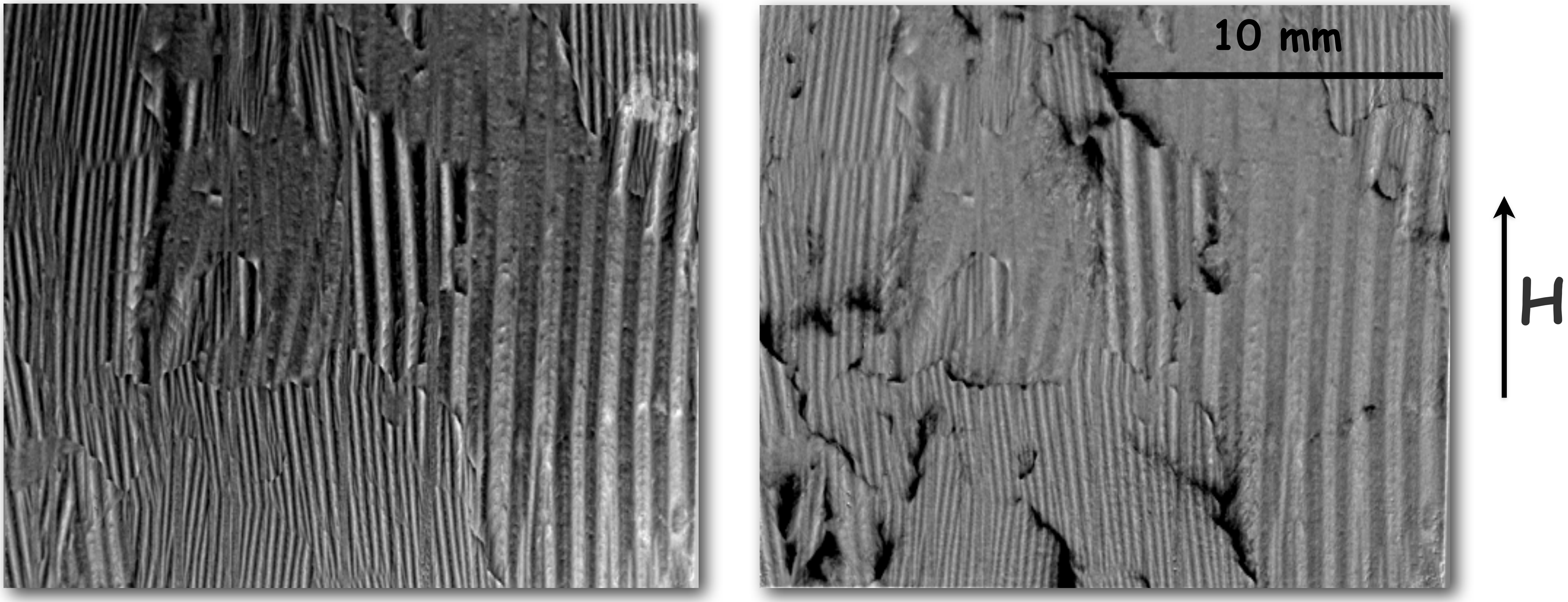
MOIF contrast
(with insulation coating)

Domain contrast even through coating

4. Magneto-optical Faraday-Microscopy

Indicator film technique: stray field imaging

Grain oriented electrical steel:



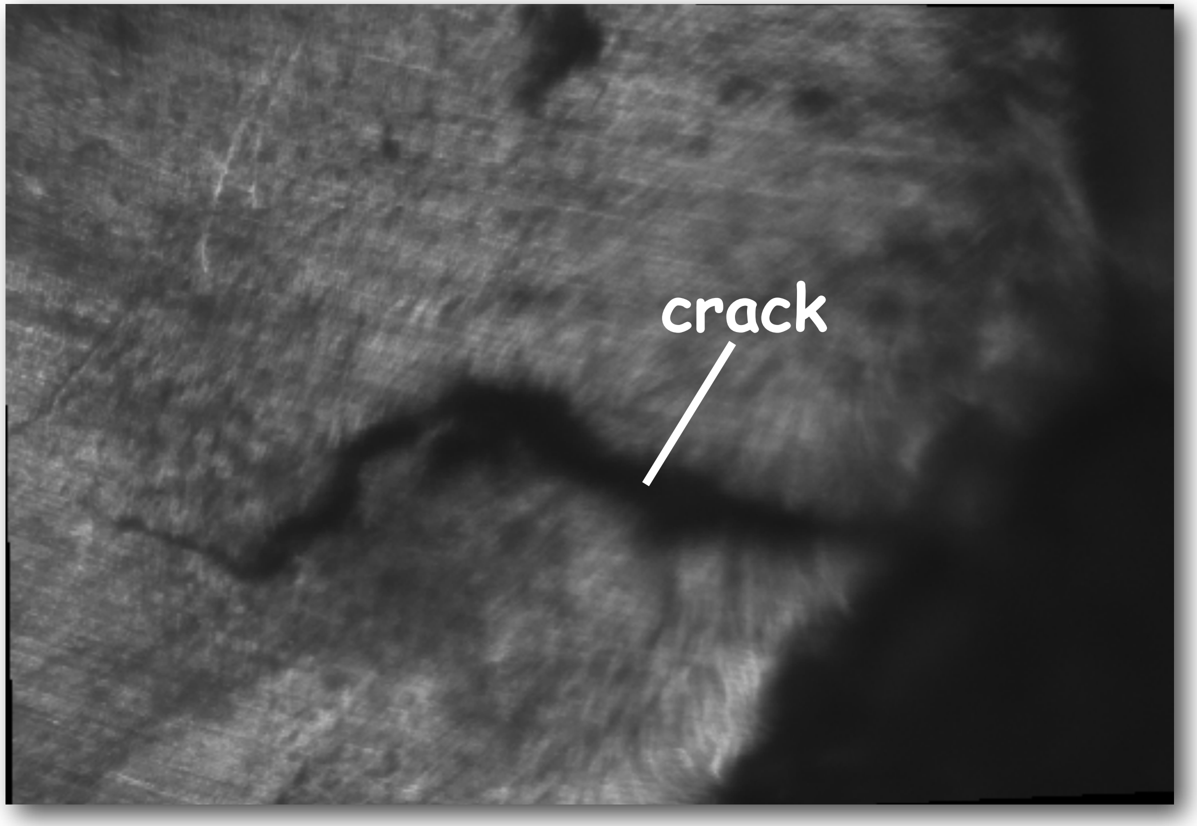
MOIF contrast (with insulation coating)

Magnetic Pole contrast at grain boundaries

4. Magneto-optical Faraday-Microscopy

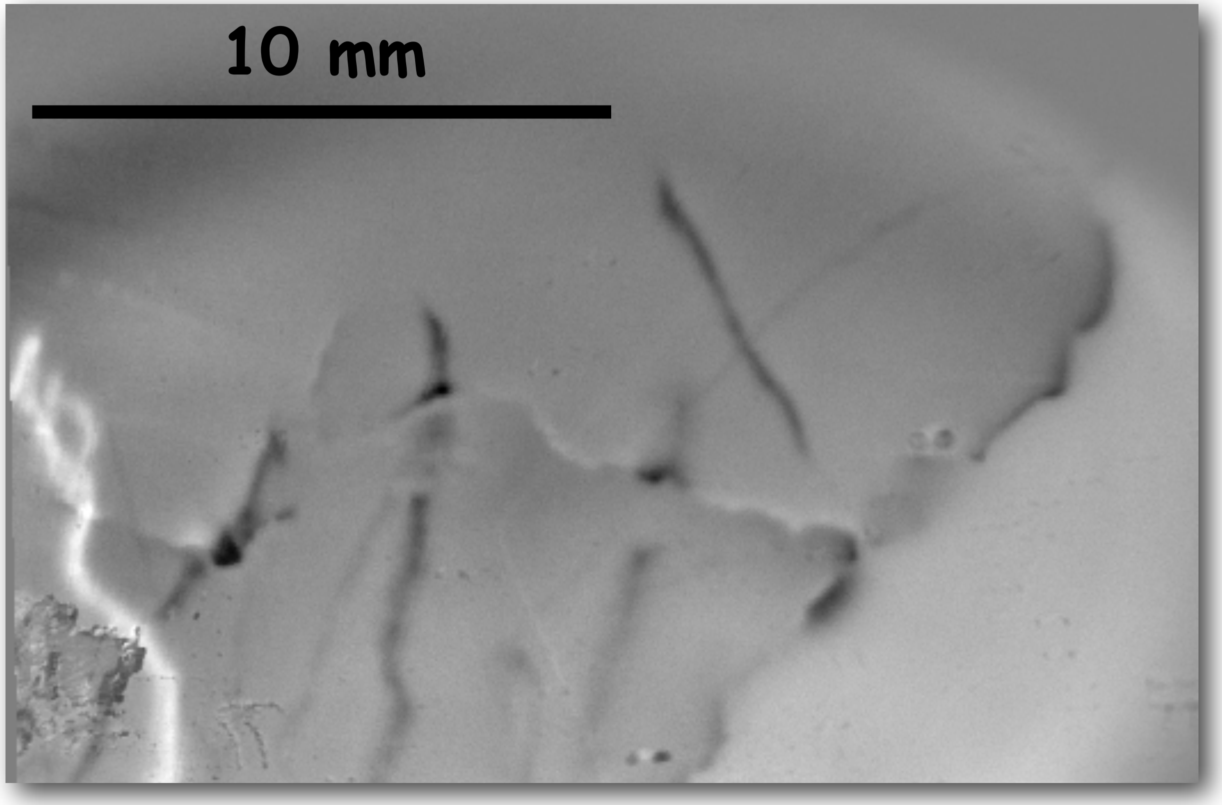
Indicator film technique: stray field imaging

Direct image of surface



Rail track with crack

MOIF image



Imaging of defects for non-destructive testing

Together with G. Y. Tian, Chengdu

5. X-Ray Spectroscopy

X-Ray Magnetic Circular Dichroism (XMCD)

XMCD: Absorption of circularly polarized X-rays depends on orientation of magnetization ***M*** with respect to helicity of the X-rays, change of sign by reversing ***M***

5. X-Ray Spectroscopy

X-Ray Magnetic Circular Dichroism (XMCD)

XMCD: Absorption of circularly polarized X-rays depends on orientation of magnetization ***M*** with respect to helicity of the X-rays, change of sign by reversing ***M***

Physical origin: If energy of absorbed photon exceeds binding energy of an inner core level (e.g. $p_{1/2}$ and $p_{3/2}$ states, separated by spin-orbit coupling)

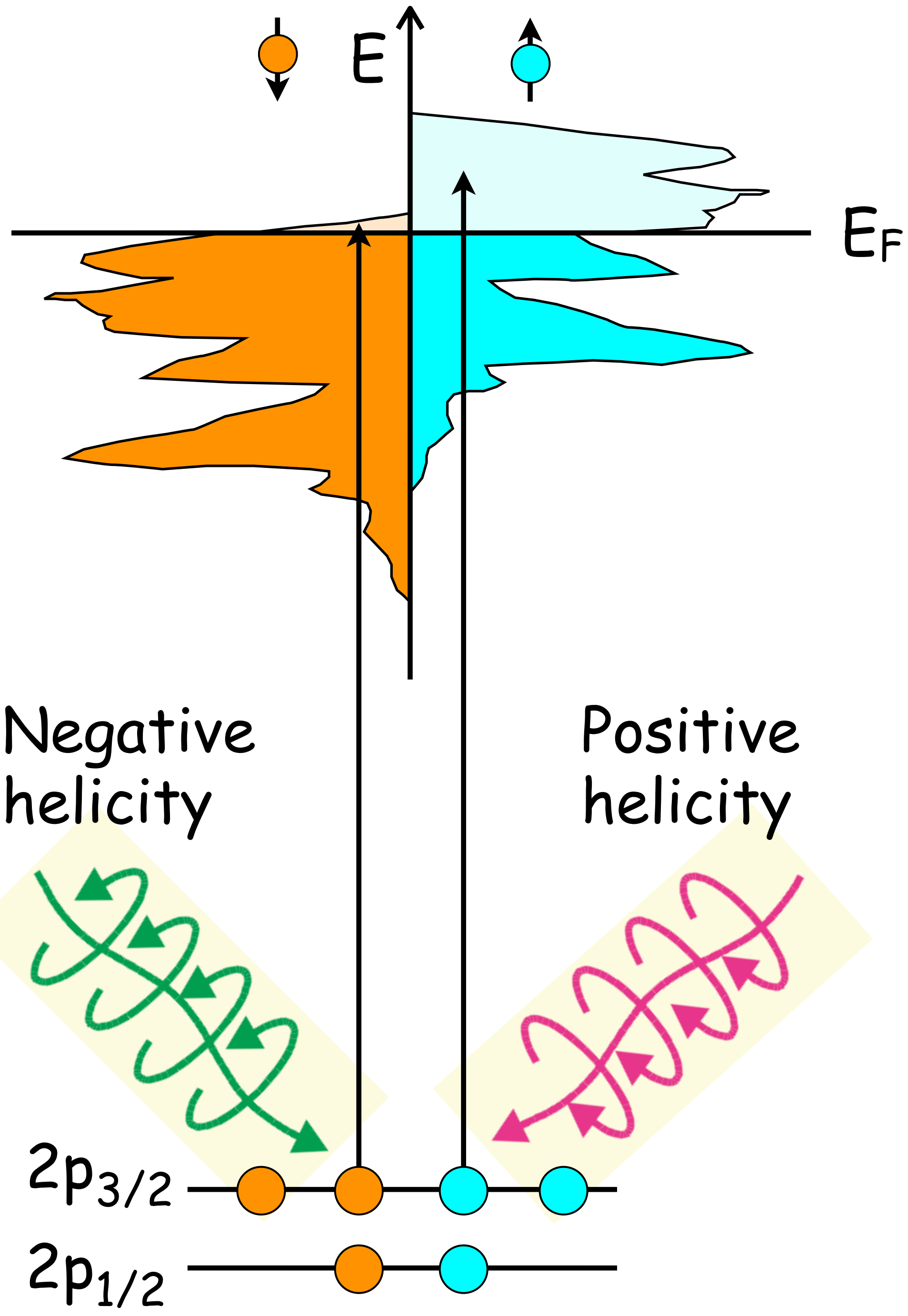
→ Transition into unoccupied spin-split states above Fermi level (e.g. into 3d band)

Initial states are well defined inner-core levels

→ XMCD is element selective

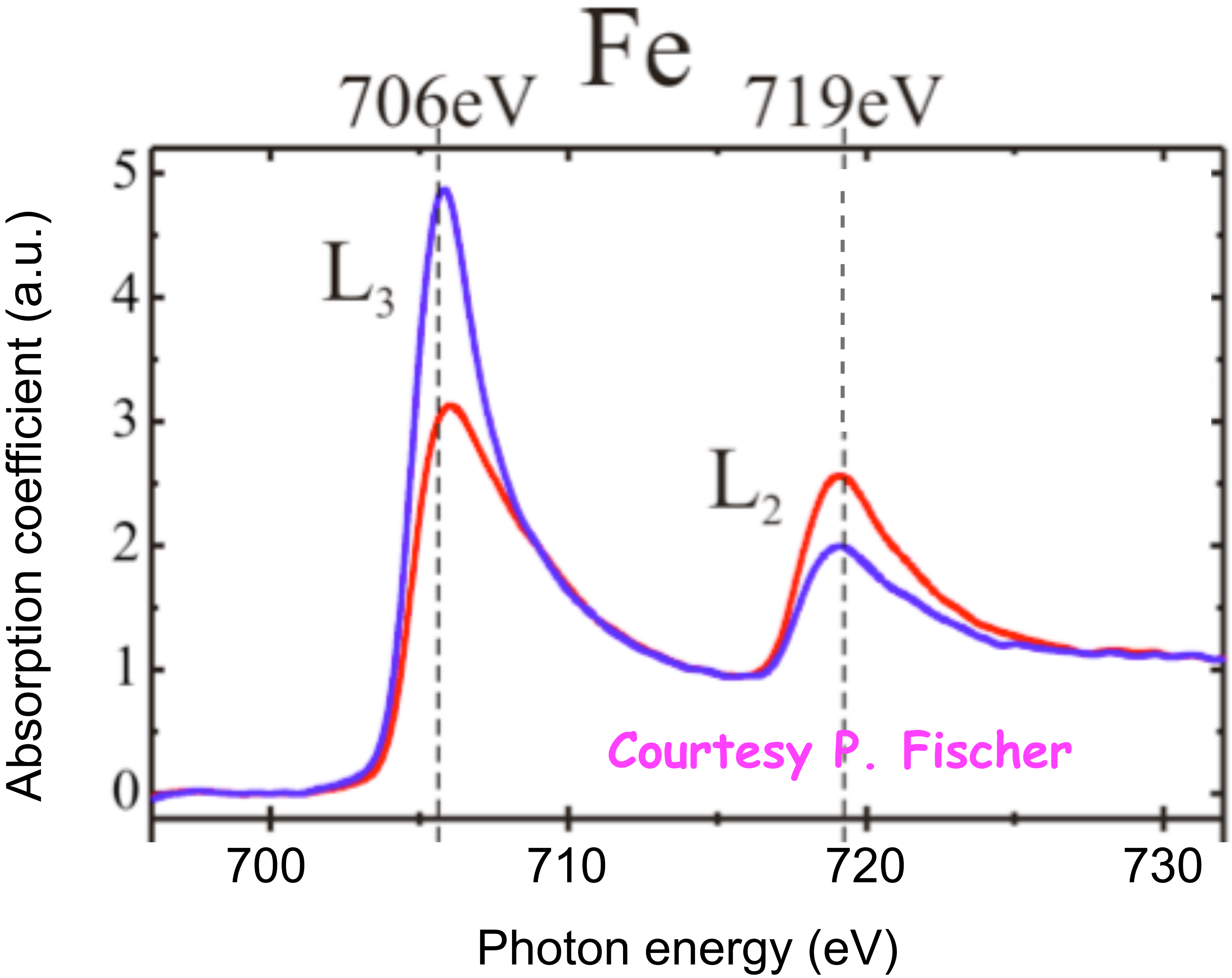
Fermi's golden rule: transition probability of absorption process is related to density of unoccupied states, which are different for minority and majority bands due to exchange interaction

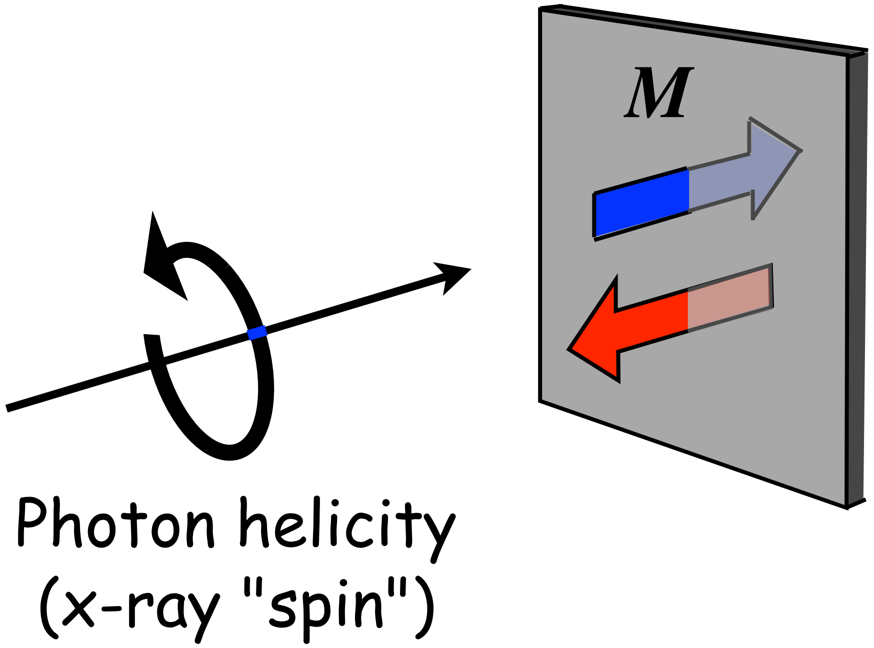
→ X-MCD signal is proportional to magnetic moment of absorbing atom → Sensing of magnetization of sample



5. X-Ray Spectroscopy

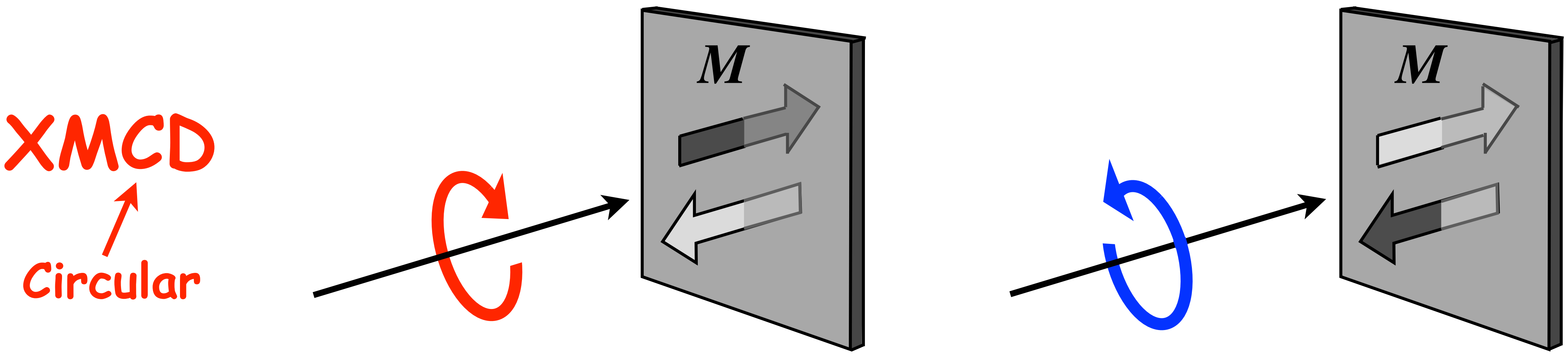
X-Ray Magnetic Circular Dichroism (XMCD)





XMCD effect is localized around L_2 (transition from $2p_{1/2}$ core level to unoccupied 3d states) and L_3 (transition from $2p_{3/2}$ core level) absorption edges

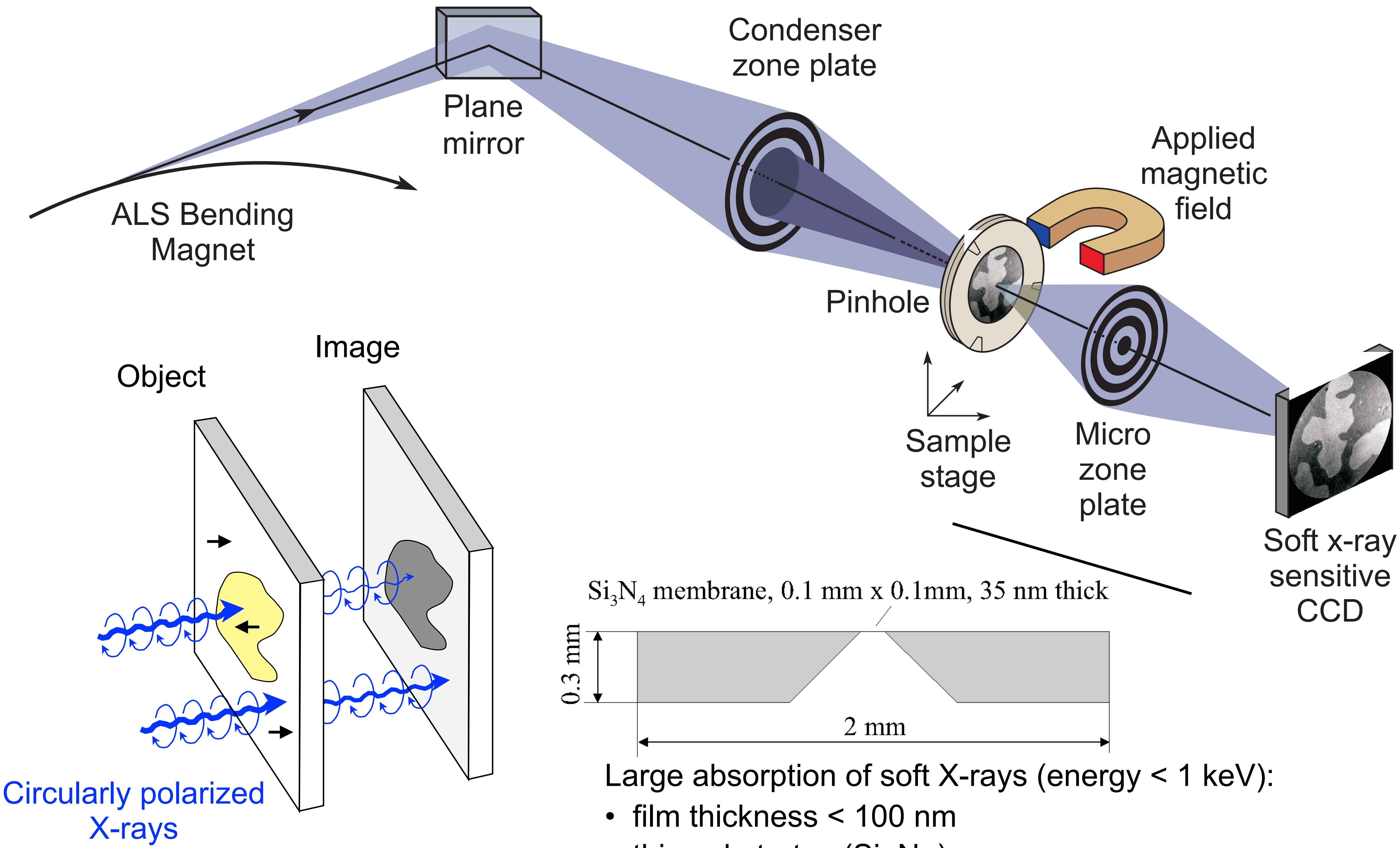
5. X-Ray Spectroscopy



XMCD detects the difference in absorption for the projection of the sample's magnetization onto the propagation direction of circularly polarized X rays.
XMCD distinguishes between magnetization parallel and antiparallel to the light **propagation** direction (in-plane magnetization at perpendicular incidence: no XMCD)

5. X-Ray Spectroscopy

5.1 Transmission X-Ray microscopy



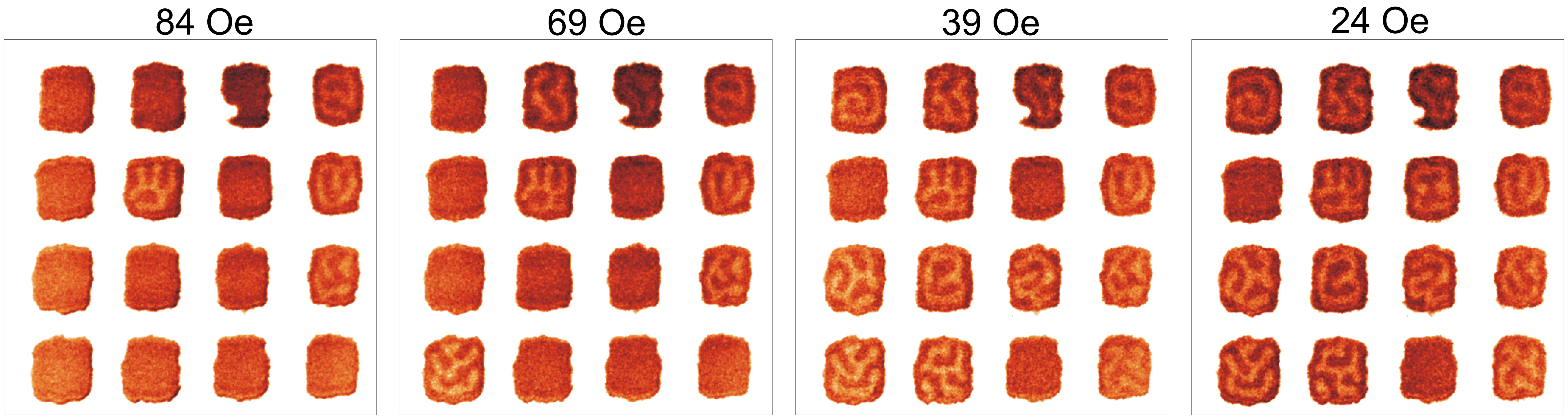
Large absorption of soft X-rays (energy < 1 keV):

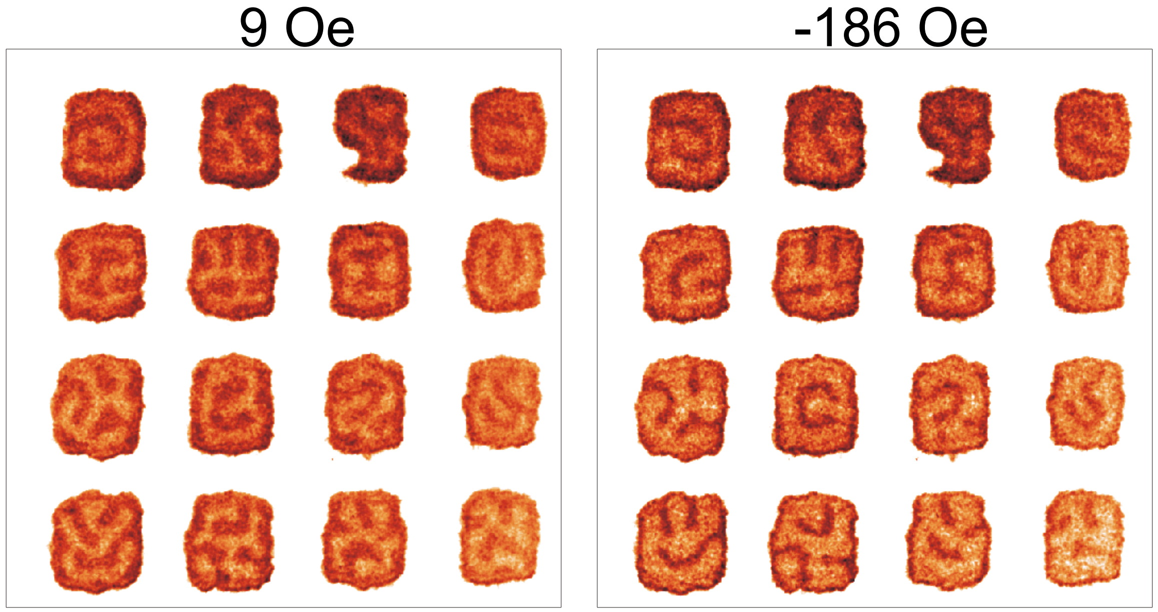
- film thickness < 100 nm
- thin substrates (Si_3N_4)

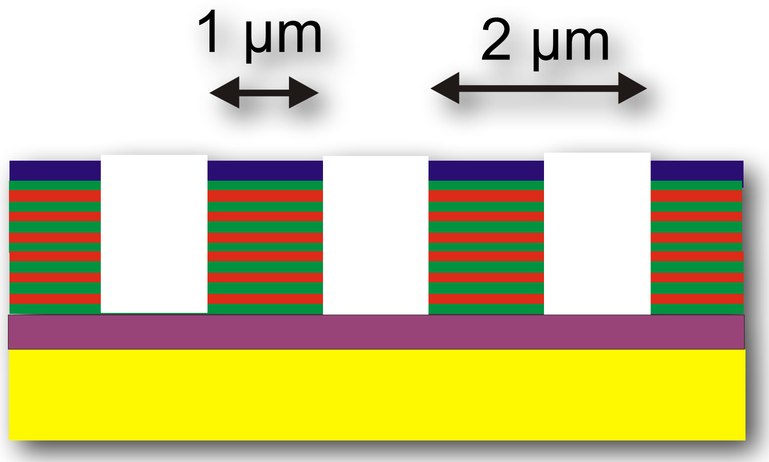
5. X-Ray Spectroscopy

5.1 Transmission X-Ray microscopy

Switching of Fe/Gd multilayered dots





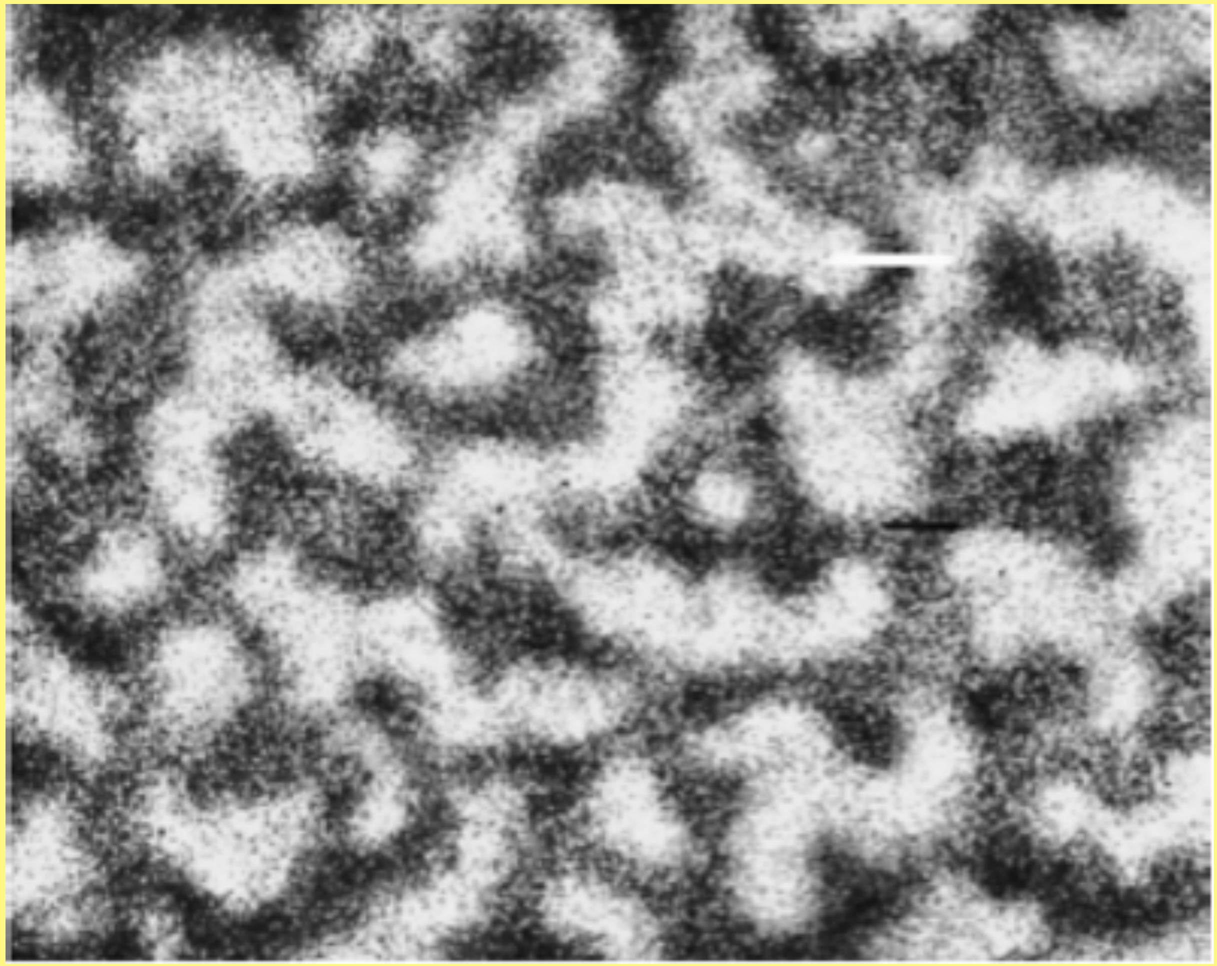


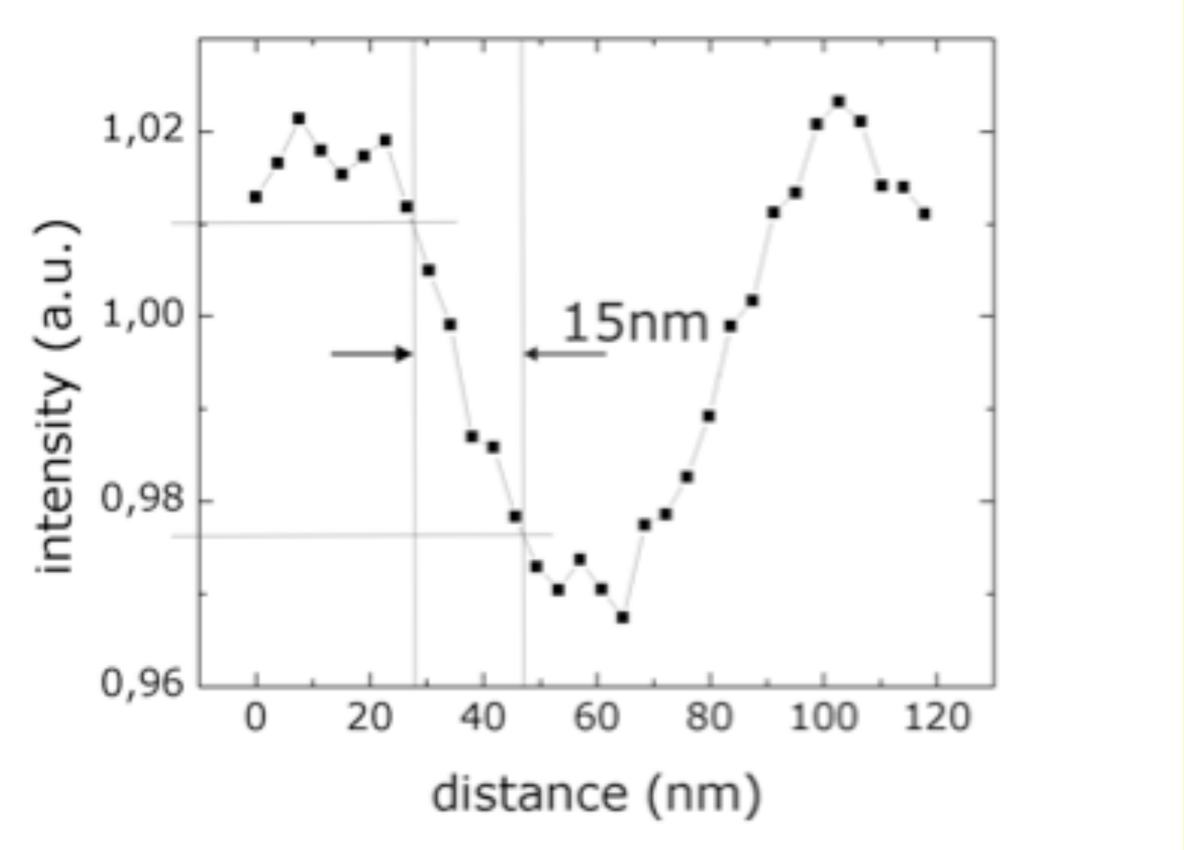
Courtesy P. Fischer, Th. Eimüller

5. X-Ray Spectroscopy

5.1 Transmission X-Ray microscopy

Intensity profile across the boundary of a magnetic domain in an amorphous GdFe layer showing a lateral resolution of less than 15 nm



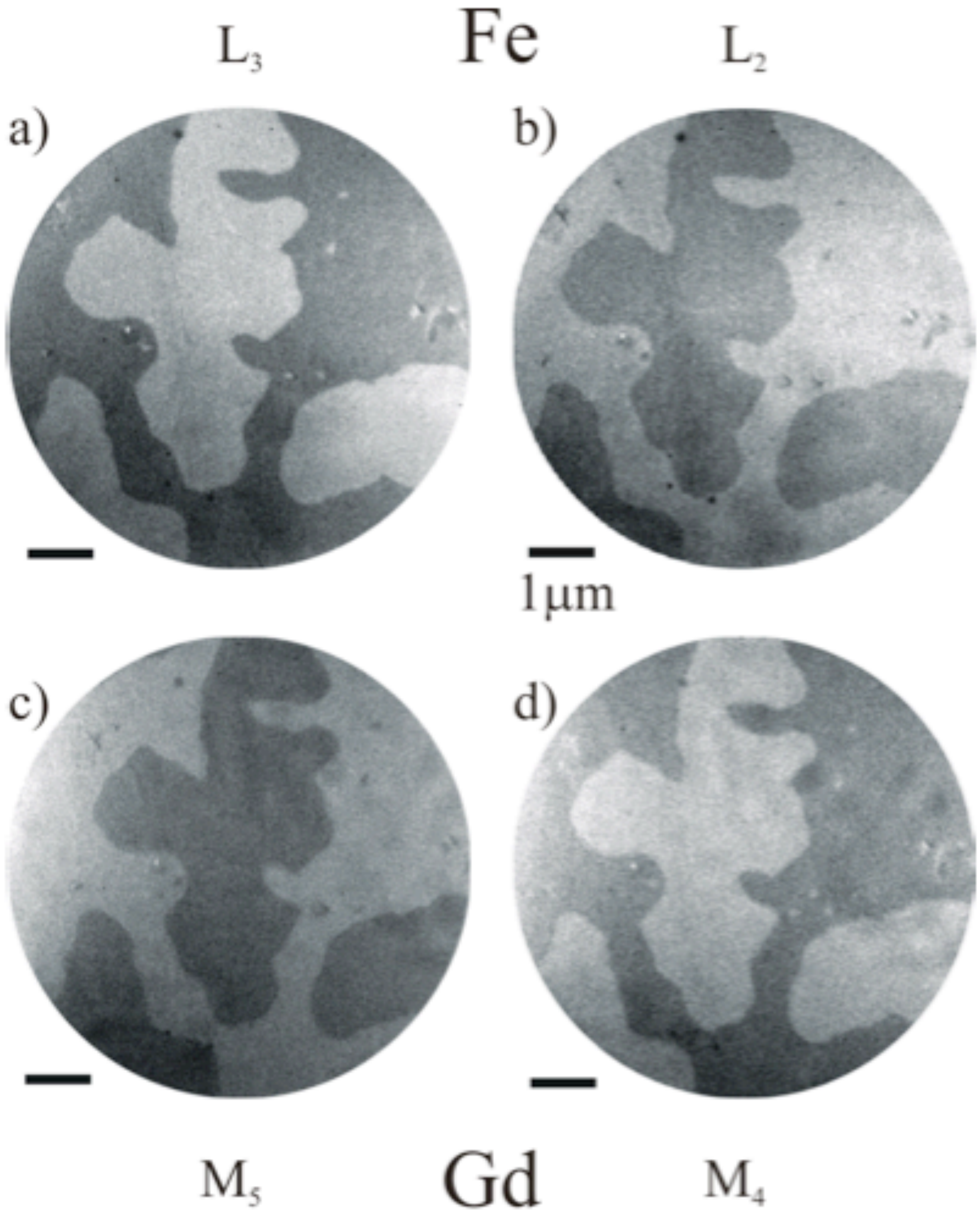


Courtesy P. Fischer

Courtesy P. Fischer, Th. Eimüller

5. X-Ray Spectroscopy

5.1 Transmission X-Ray microscopy



- MTXM images of an amorphous $Gd_{25}Fe_{75}$ layer recorded at the spin- orbit-coupled Fe L_3 (a) and L_2 (b) as well as at the Gd M_5 (c) and M_4 (d) absorption edges
- Contrast inversion: magnetic moments on Cd and Fe couple antiparallel

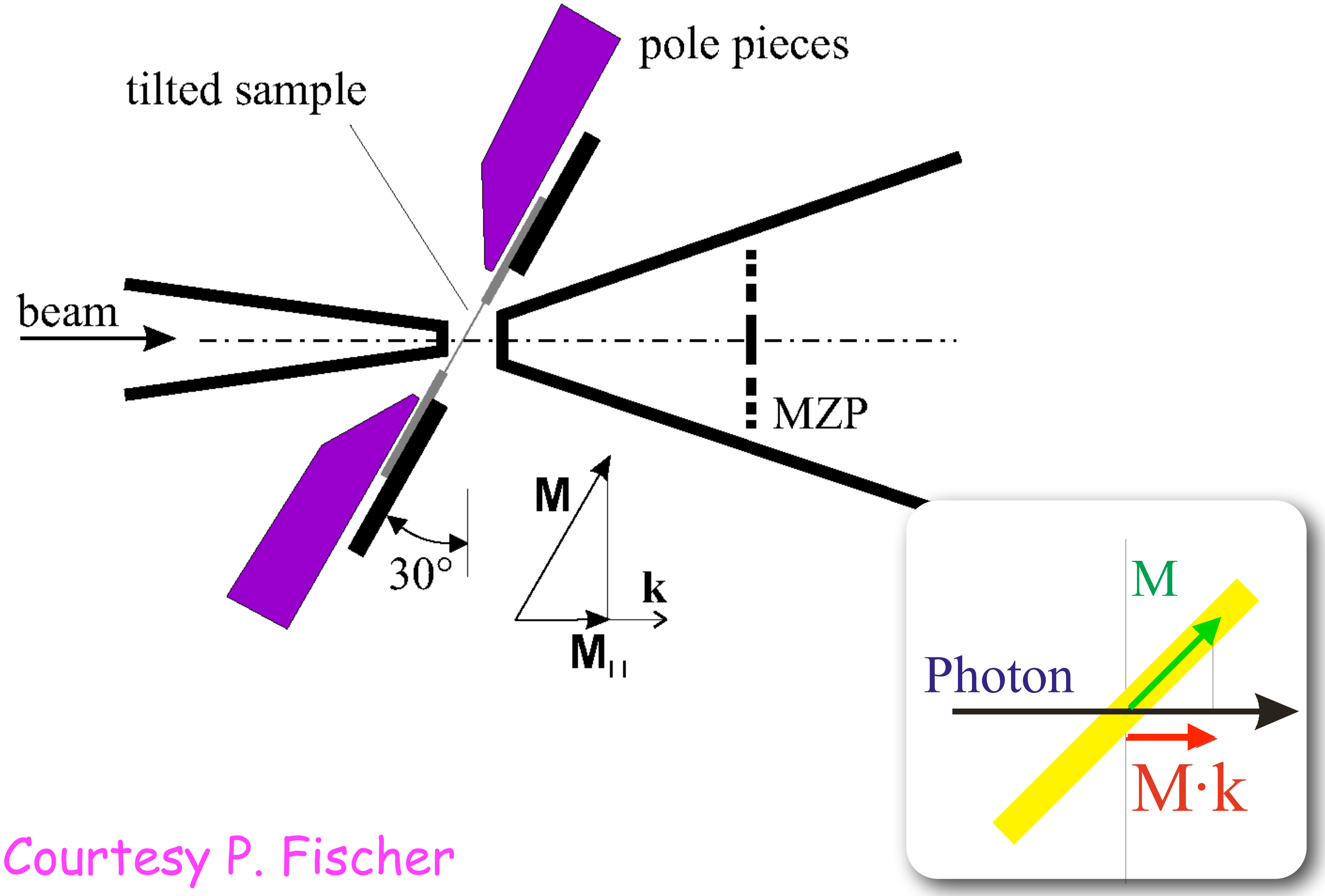
Courtesy P. Fischer

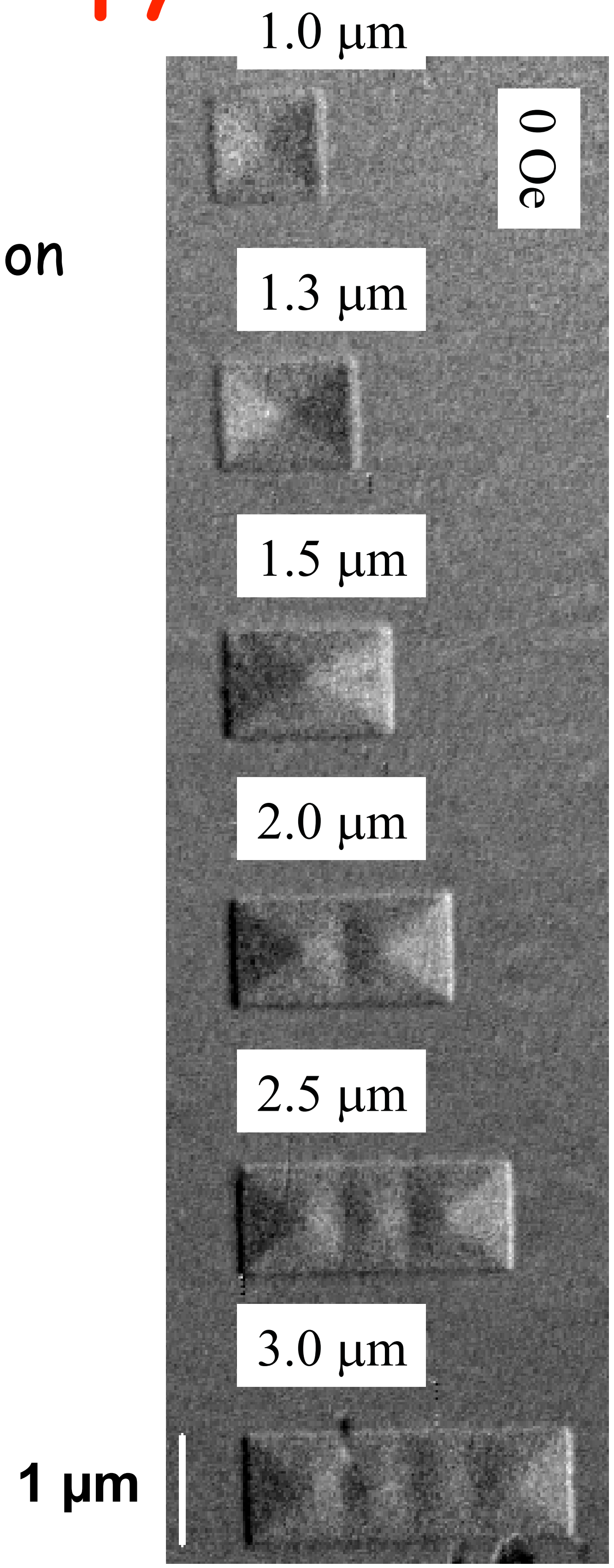
5. X-Ray Spectroscopy

5.1 Transmission X-Ray microscopy

In-plane imaging

Dichroic contrast: given by projection of M on photon propagation direction. In-plane imaging by tilting the sample (30°):



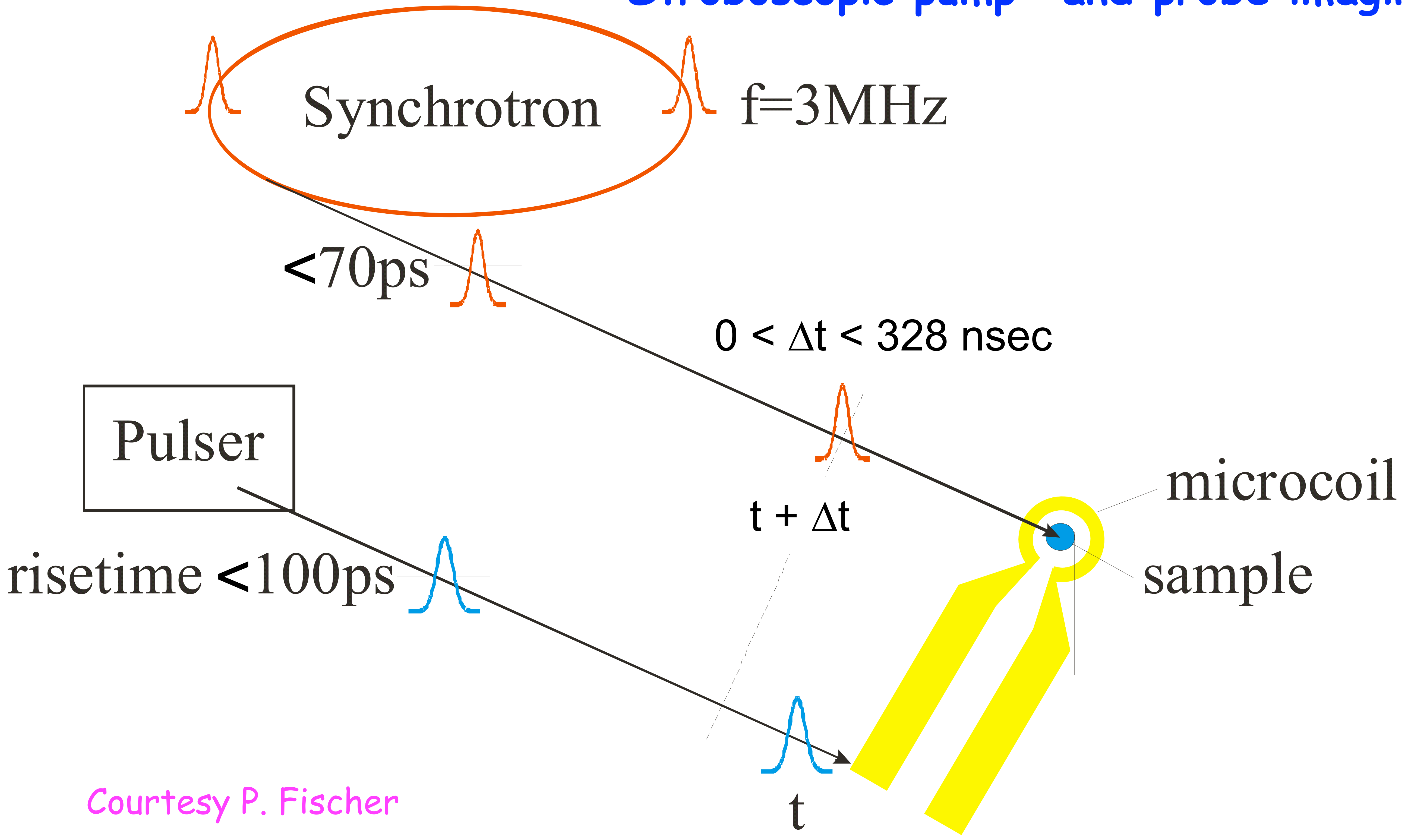


Courtesy P. Fischer

5. X-Ray Spectroscopy

5.1 Transmission X-Ray microscopy

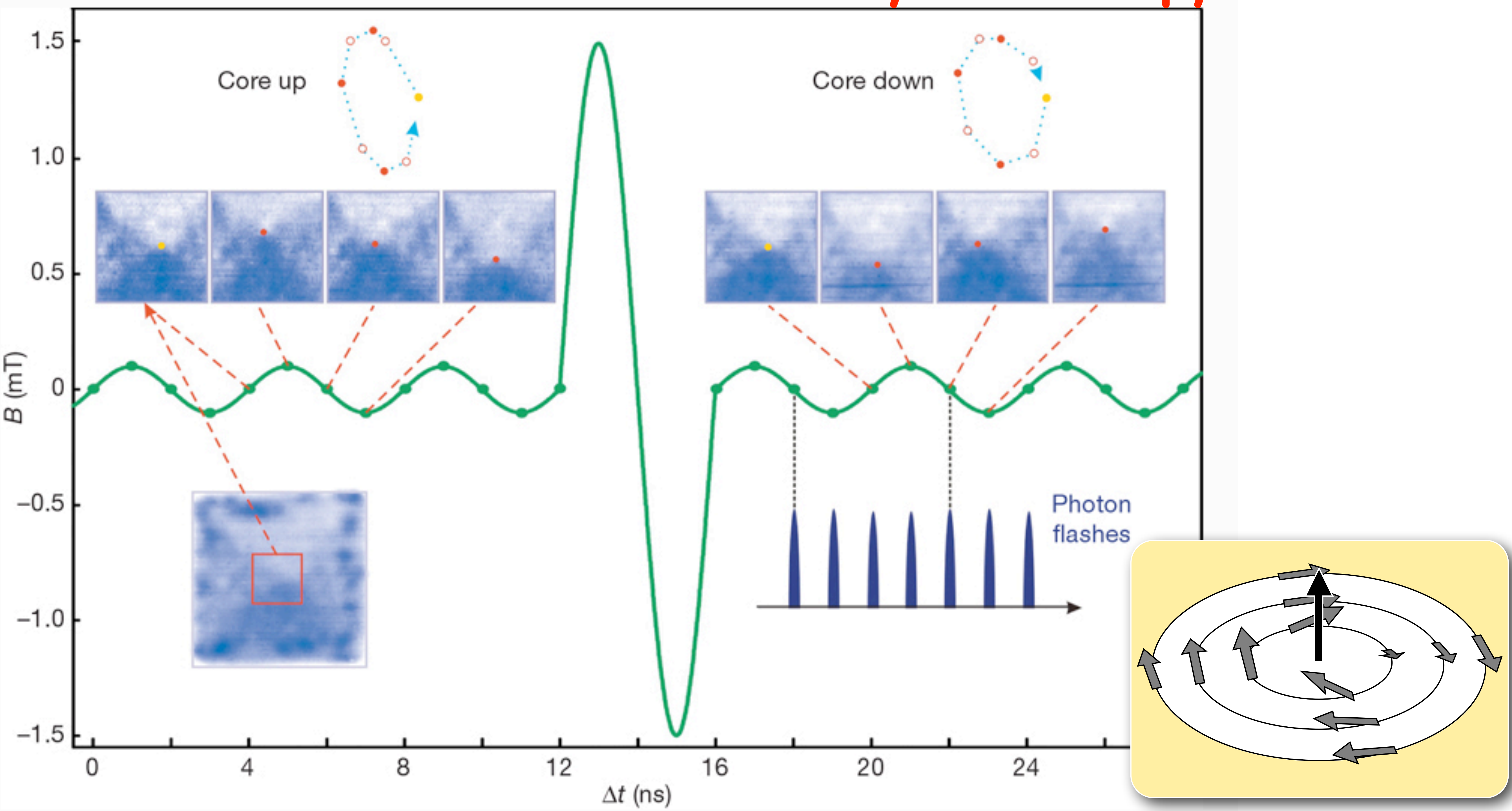
Stroboscopic pump- and probe imaging



Courtesy P. Fischer

5. X-Ray Spectroscopy

5.1 Transmission X-Ray microscopy

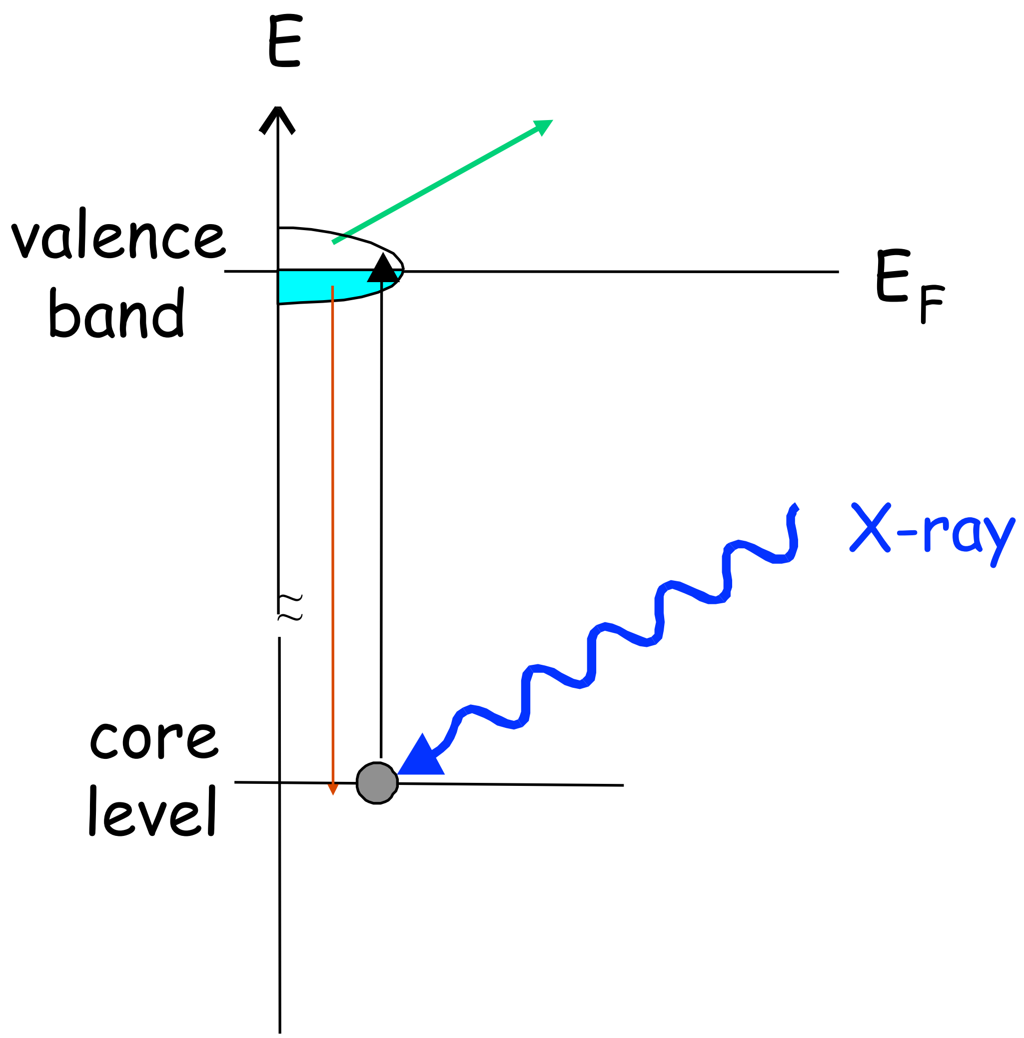


Excitation of vortex structure in ac field (frequency 250 MHz, amplitude 0.1 mT). After a 4 ns 'single period' burst (amplitude 1.5 mT) the vortex core polarity is inverted

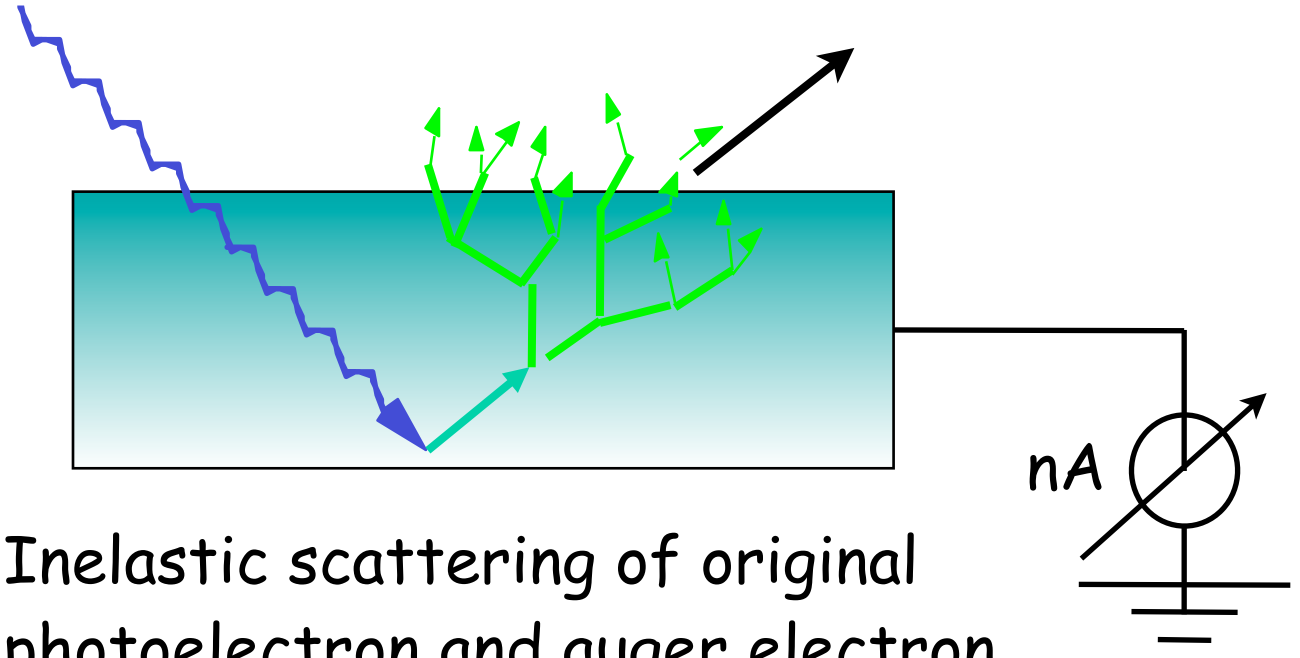
B. Van Waeyenberge et al., Nature **444**, 461 (2006)

5. X-Ray Spectroscopy

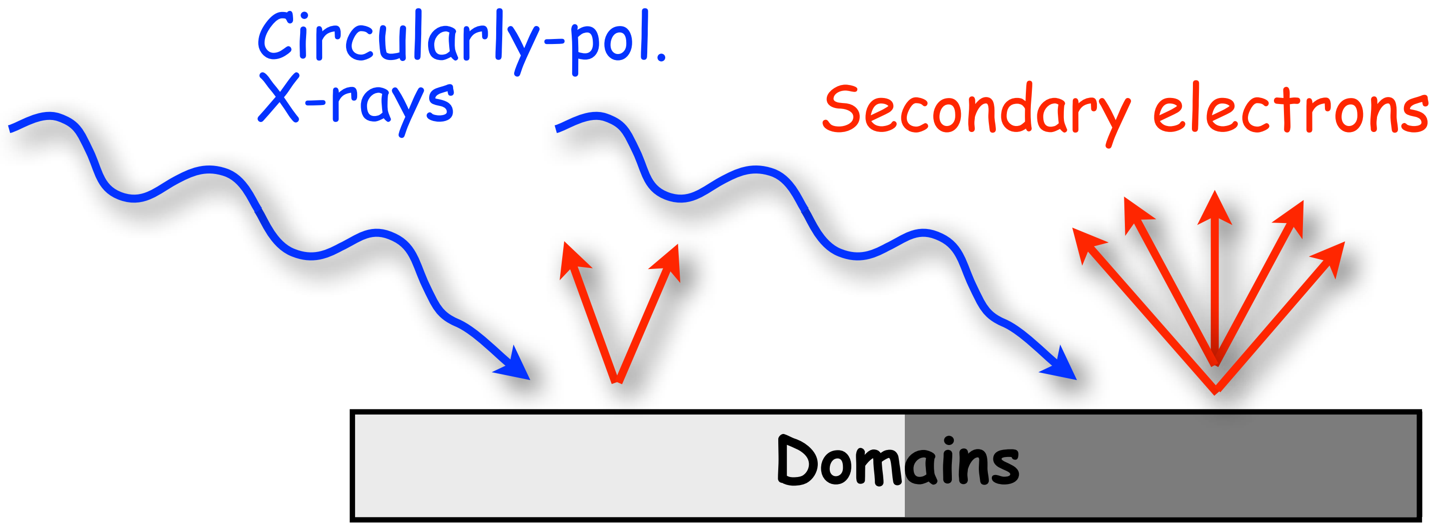
5.2 X-ray Photoemission Electron Microscopy (X-PEEM)



- Excitation of core electron into empty valence state by incoming X-ray
- Recombination (e.g.) by Auger decay



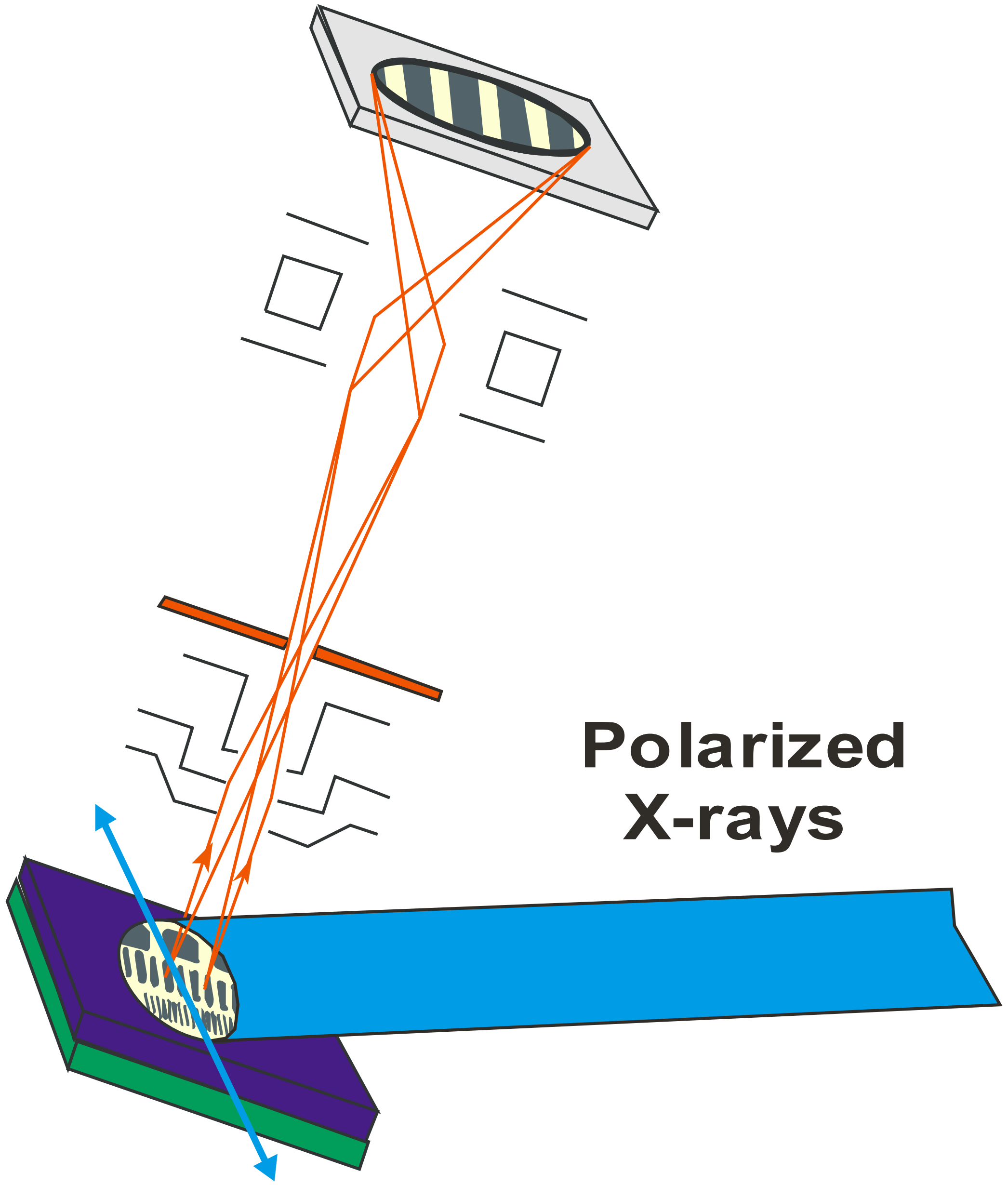
- Inelastic scattering of original photoelectron and auger electron leads to emission of secondary electrons
- Electron yield ~ X-ray absorption coefficient
- Probing depth ~ electron escape length $\exp(-\lambda\, t)$ with λ **~ 2 nm**



Courtesy H.Ohldag

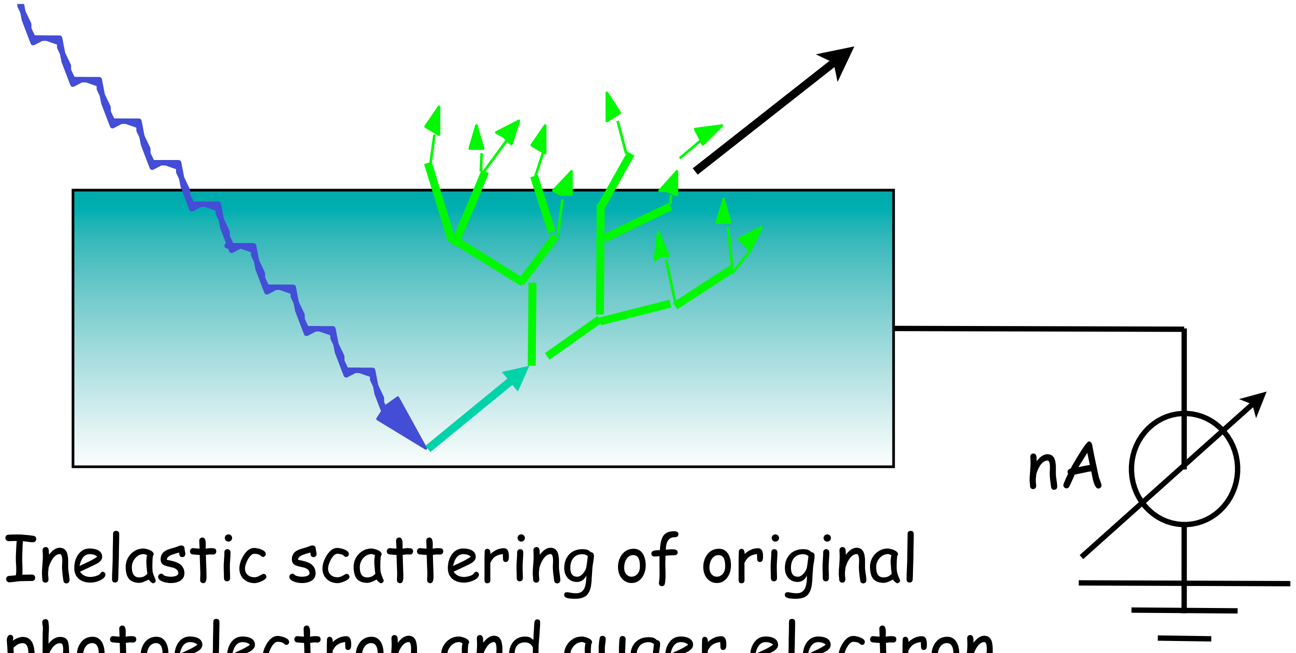
5. X-Ray Spectroscopy

5.2 X-ray Photoemission Electron Microscopy (X-PEEM)



- Full Field Imaging
- 20 – 50 nm Resolution
- Linear and circular polarization

Courtesy H.Ohldag



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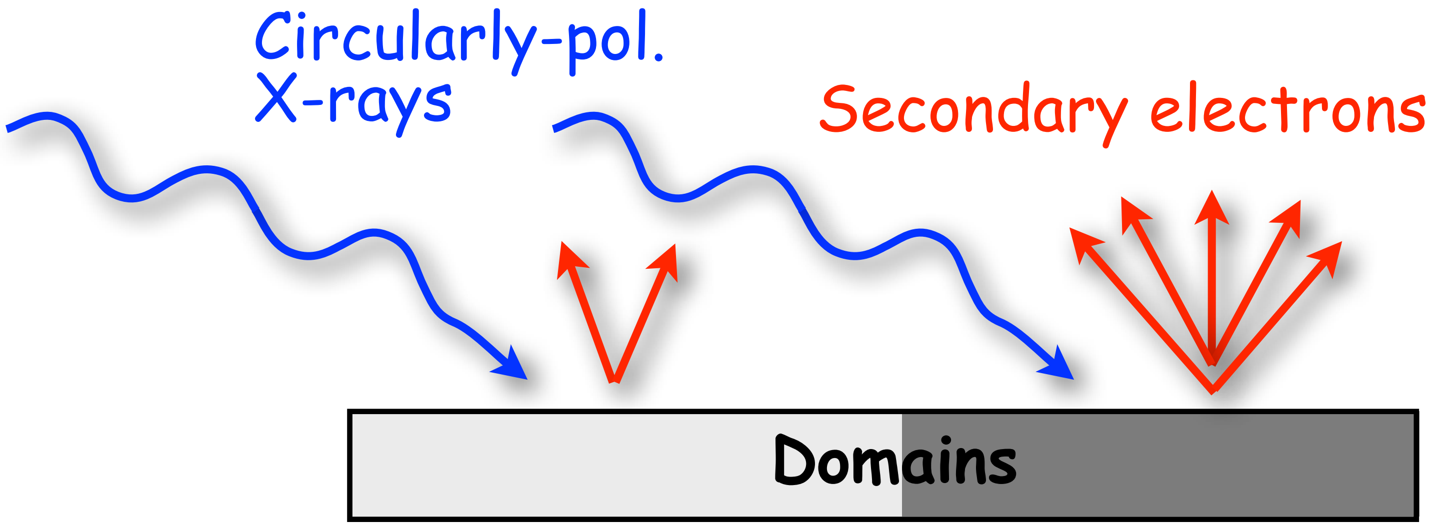

5. X-Ray Spectroscopy

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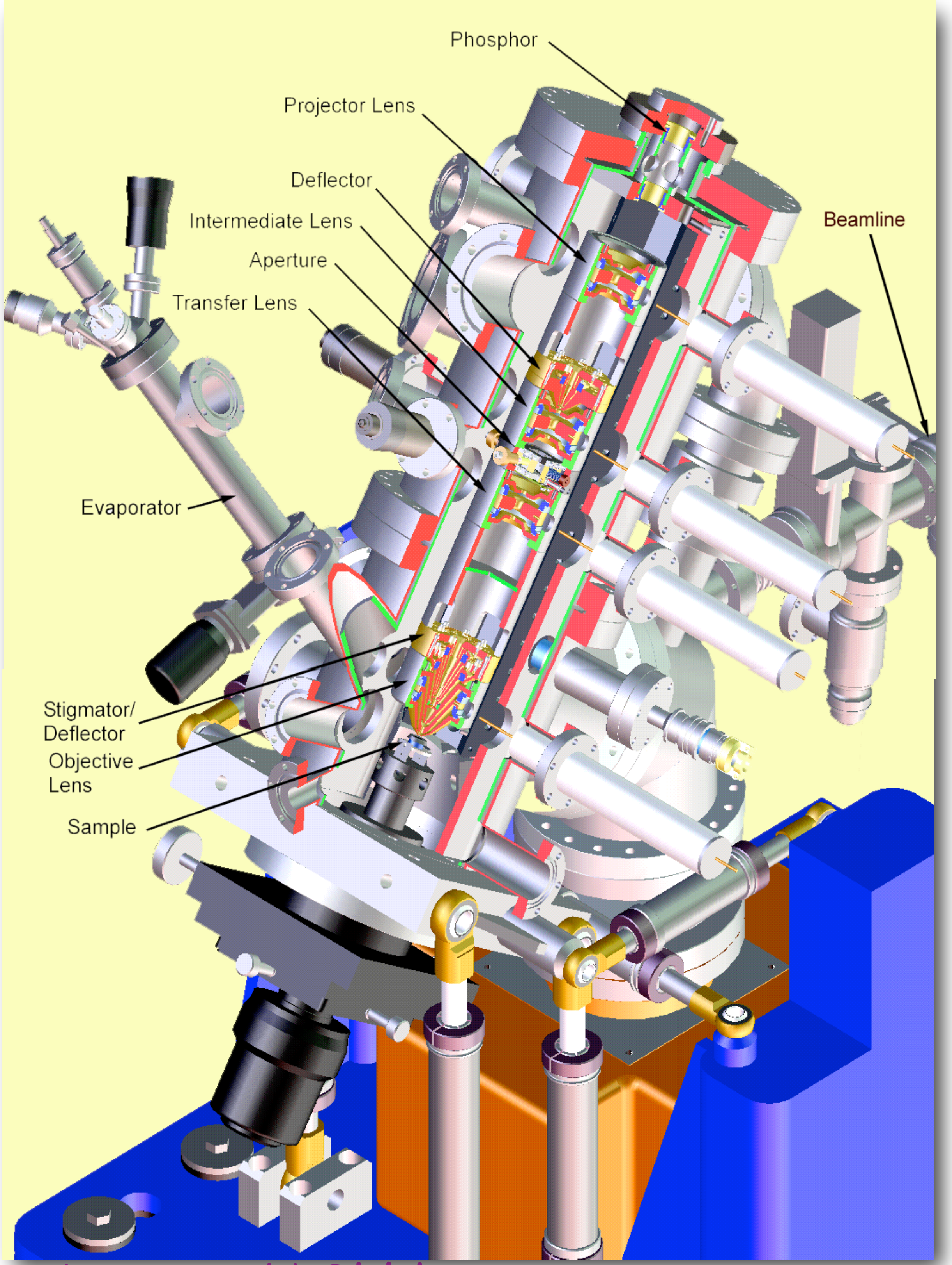
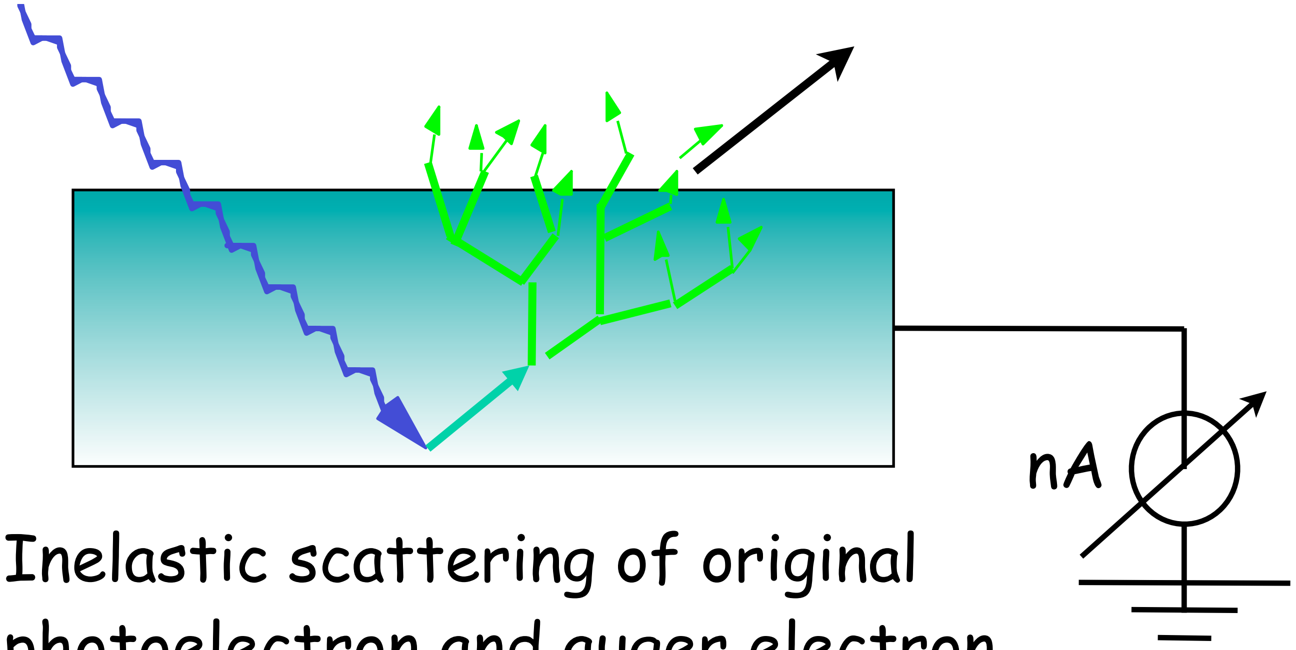


Courtesy H.Ohldag



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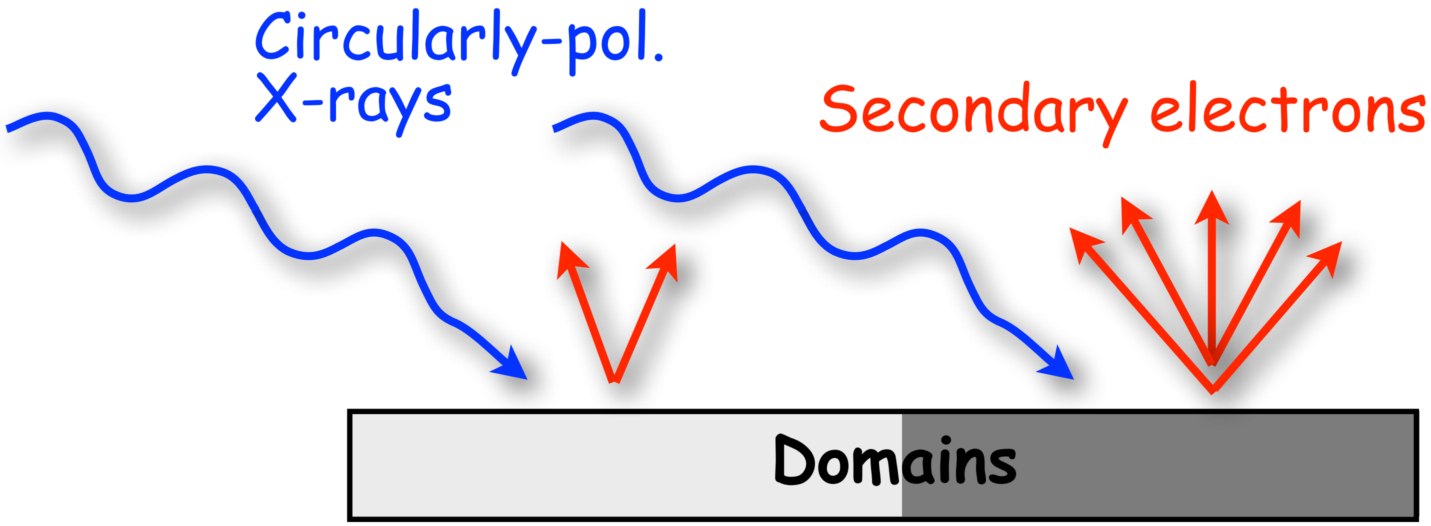
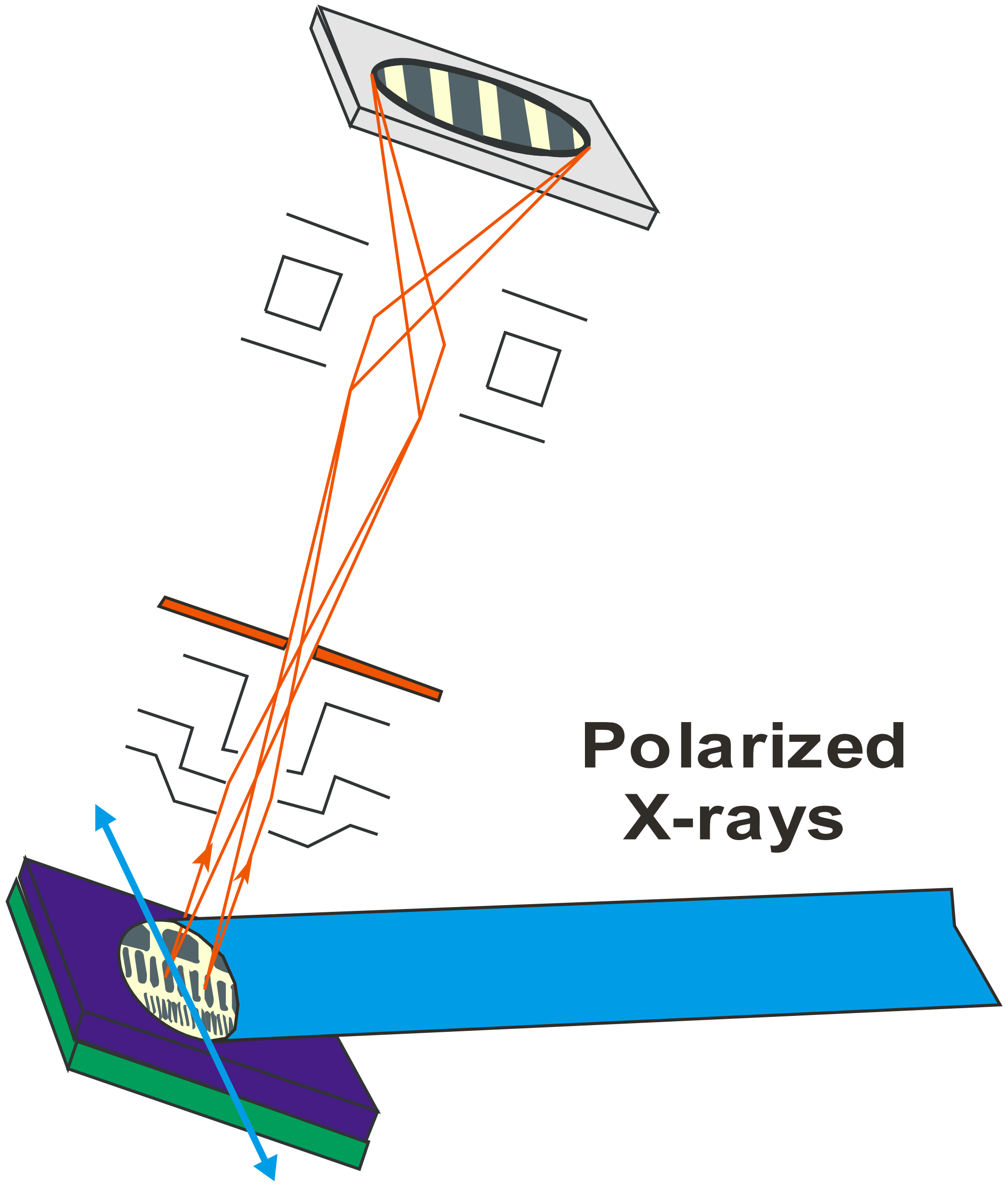


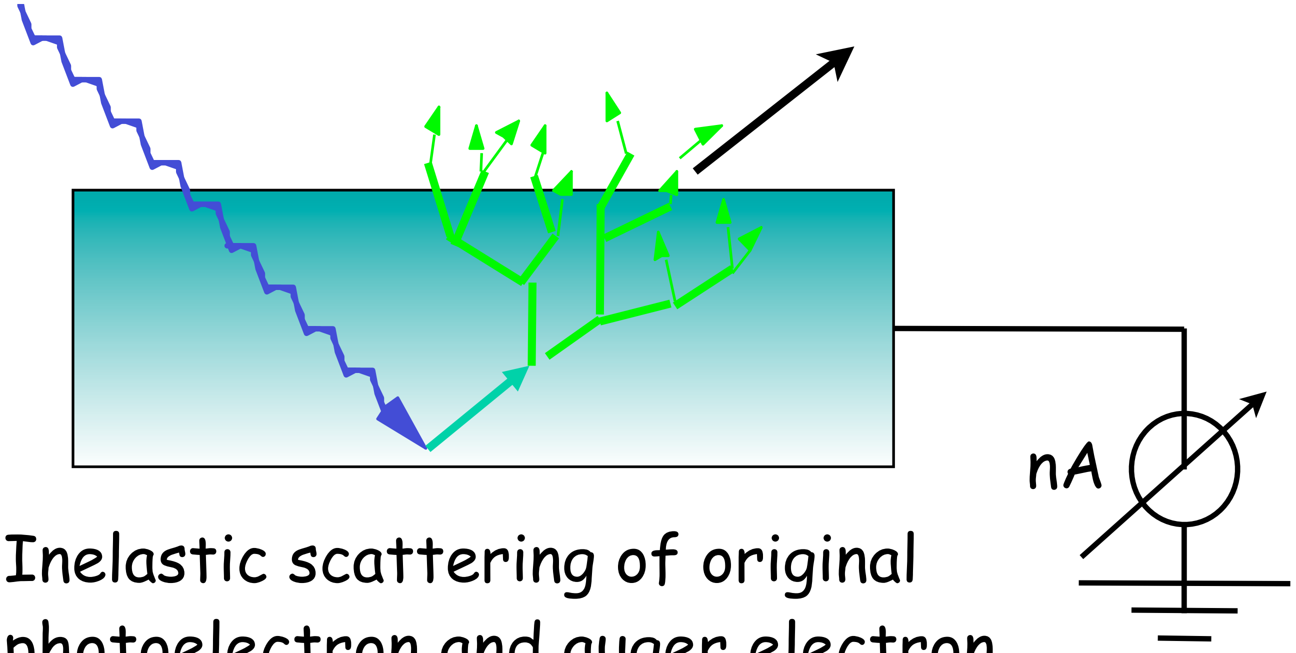
5. X-Ray Spectroscopy

5.2 X-ray Photoemission Electron Microscopy (X-PEEM)



- Full Field Imaging
- 20 – 50 nm Resolution
- Linear and circular polarization

Courtesy H.Ohldag



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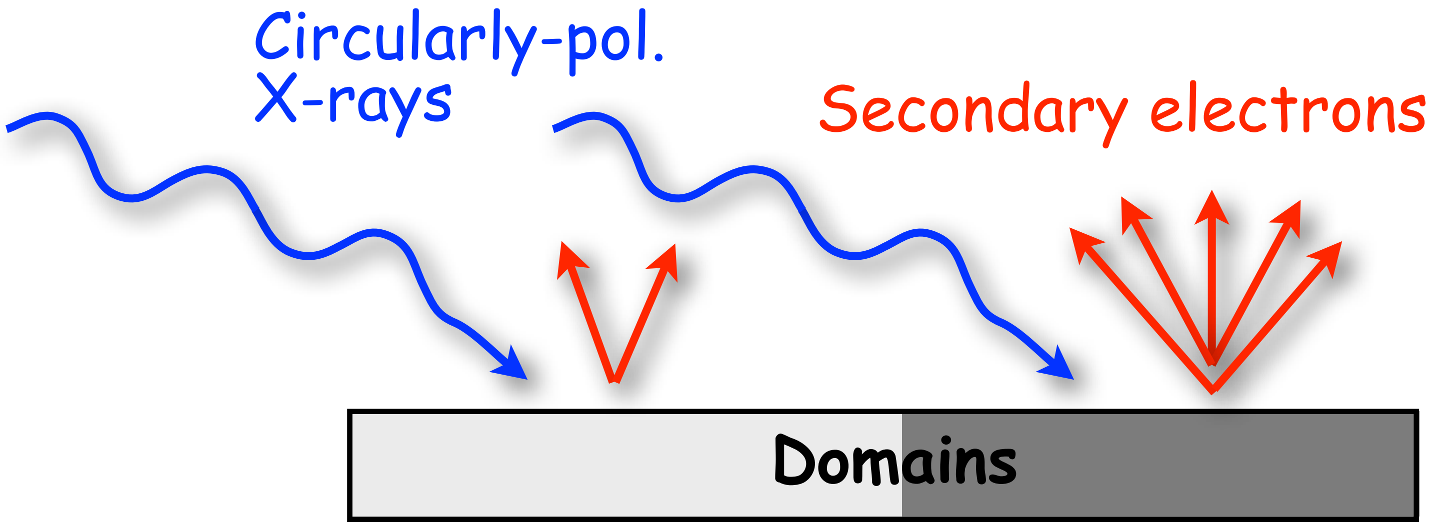
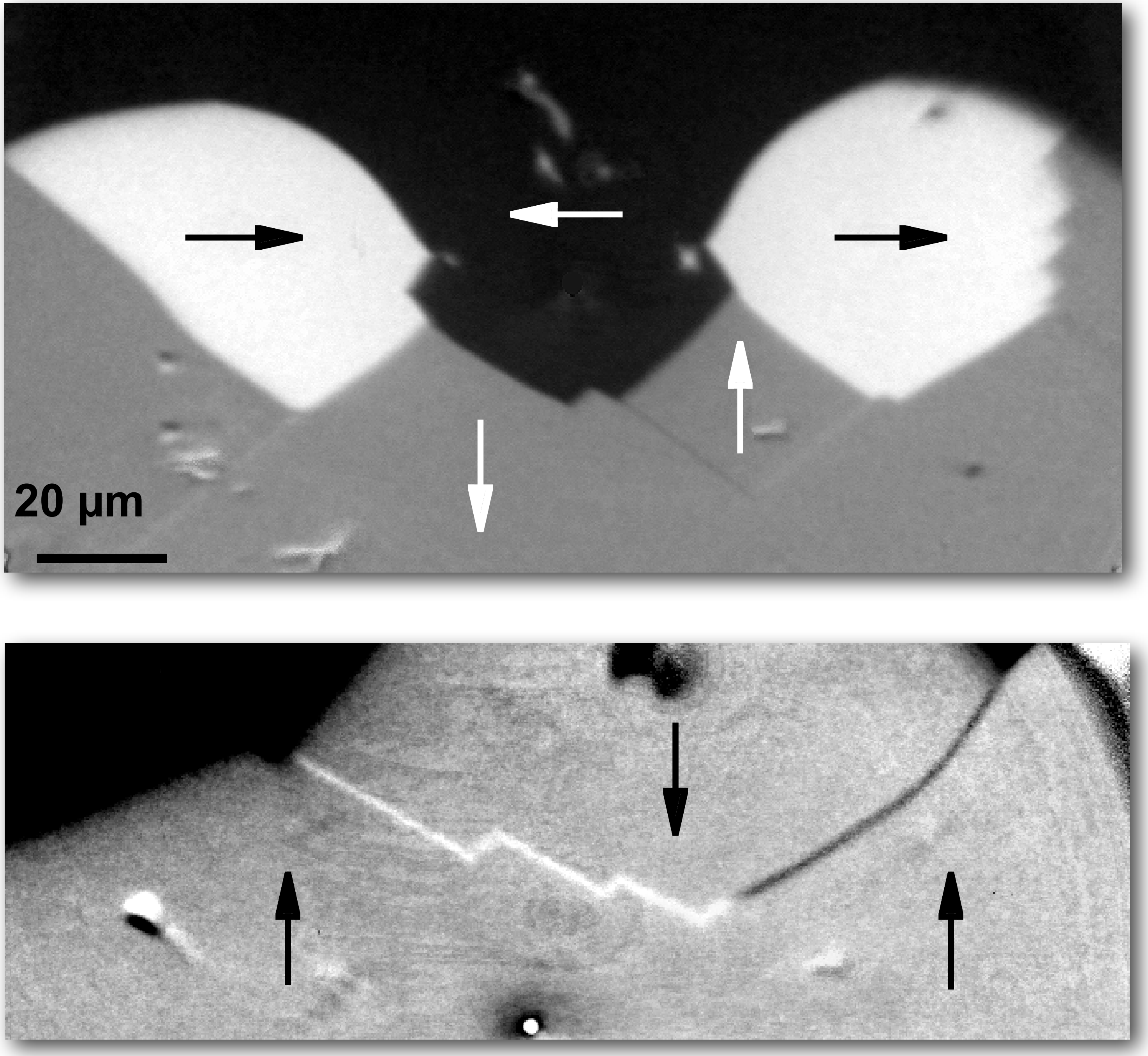


5. X-Ray Spectroscopy

5.2 X-ray Photoemission Electron Microscopy (X-PEEM)

Iron whisker



Excitation by circularly-polarized electrons (MXCD)

⟷

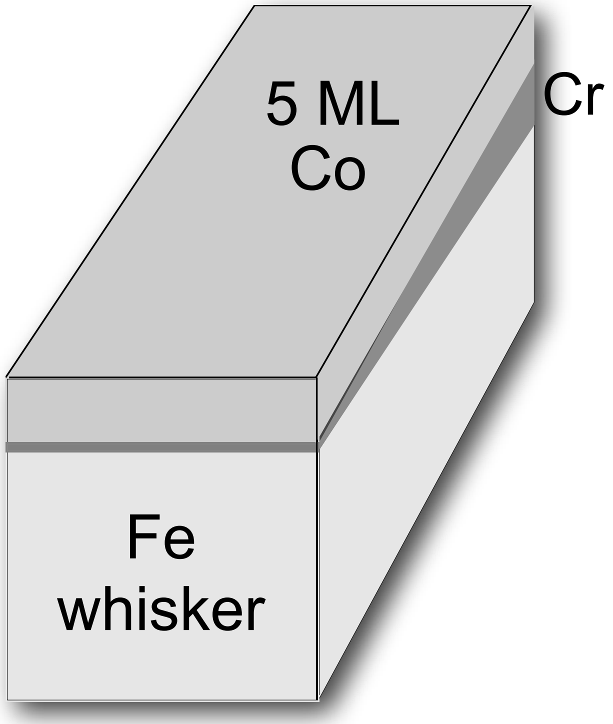
Sensitivity

Courtesy C.M. Schneider

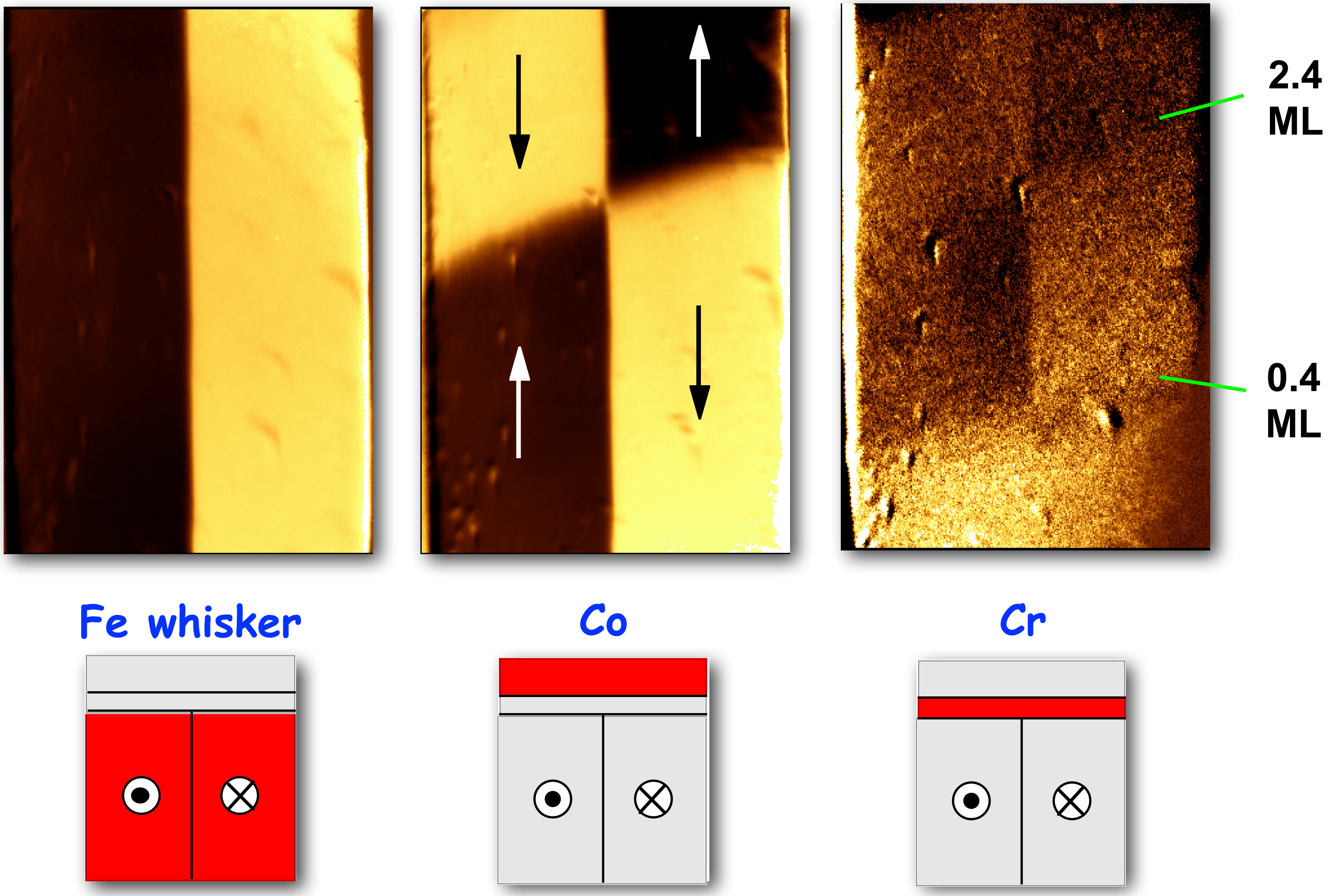
5. X-Ray Spectroscopy

5.2 X-ray Photoemission Electron Microscopy (X-PEEM)

Depth-selectivity by element-specific PEEM imaging

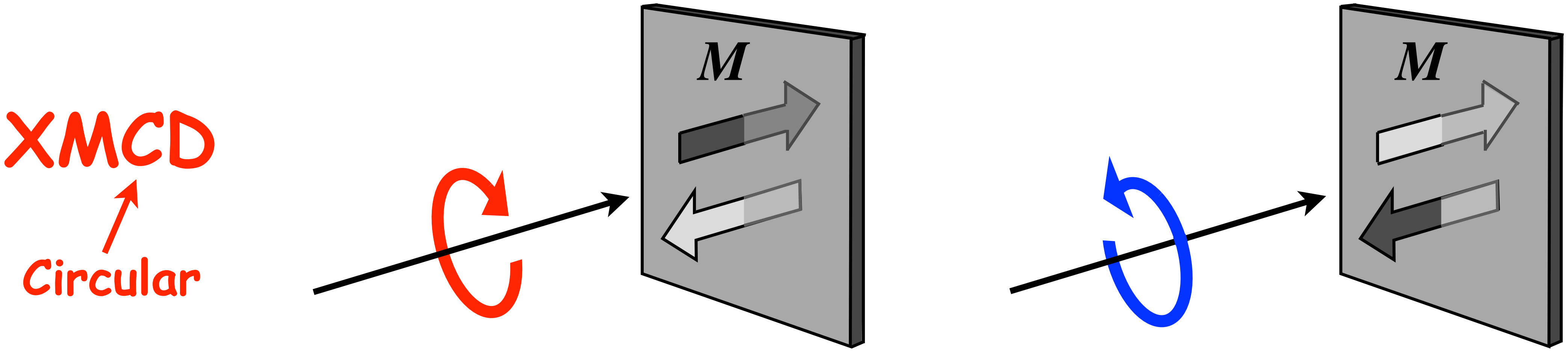


Fe-Cr-Co layer system

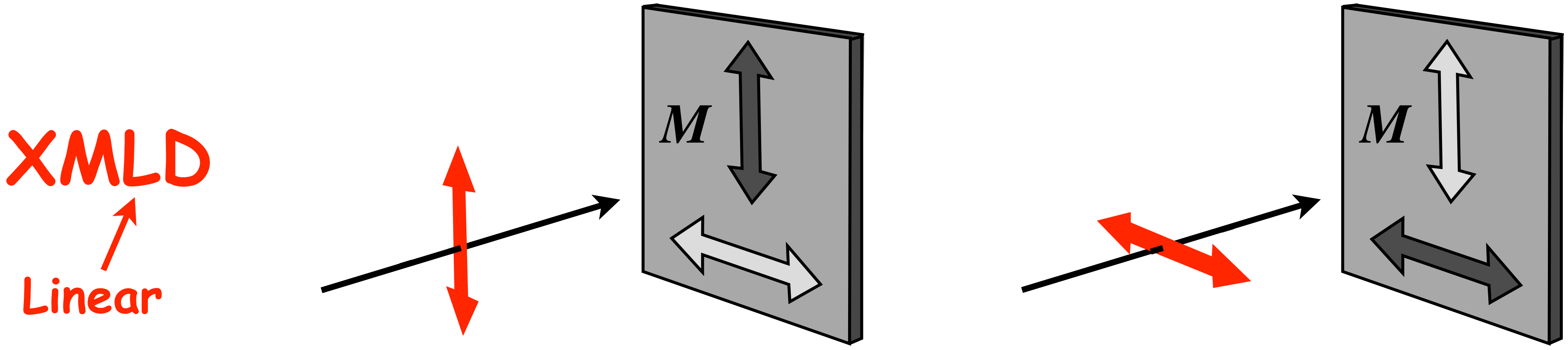


courtesy
C.M. Schneider

5. X-Ray Spectroscopy



XMCD detects the difference in absorption for the projection of the sample's magnetization onto the propagation direction of circularly polarized X rays.
XMCD distinguishes between magnetization parallel and antiparallel to the light **propagation** direction (in-plane magnetization at perpendicular incidence: no XMCD)



XMLD detects the difference in absorption of the axis of magnetization aligned parallel or perpendicular to the E-field of the X-rays.
XMLD distinguishes between the sample being magnetized parallel or perpendicular to the light **polarization** direction

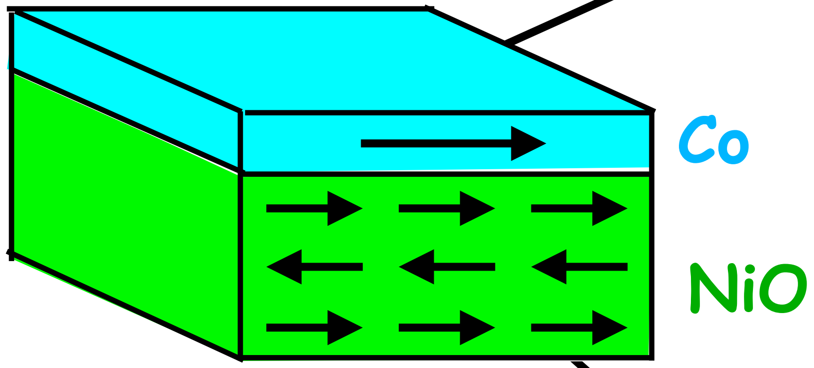
5. X-Ray Spectroscopy

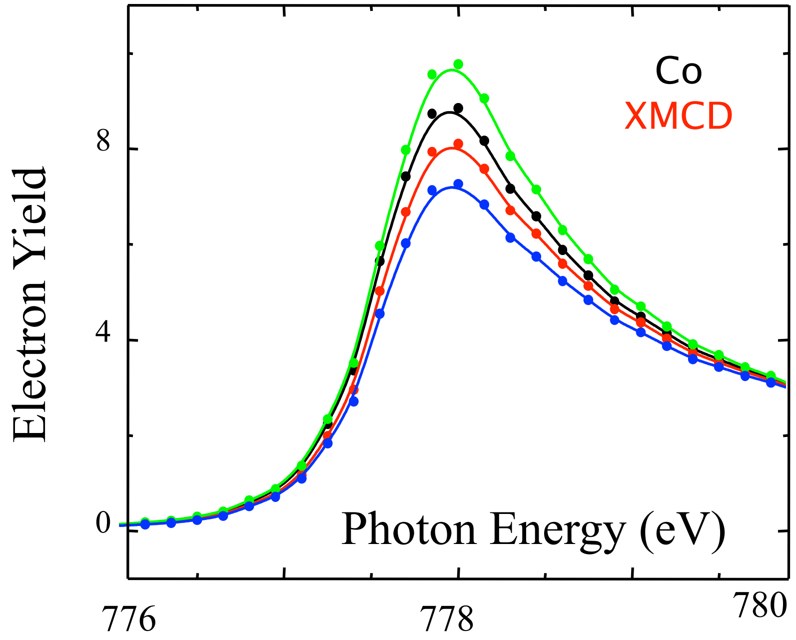
5.2 X-ray Photoemission Electron Microscopy (X-PEEM)

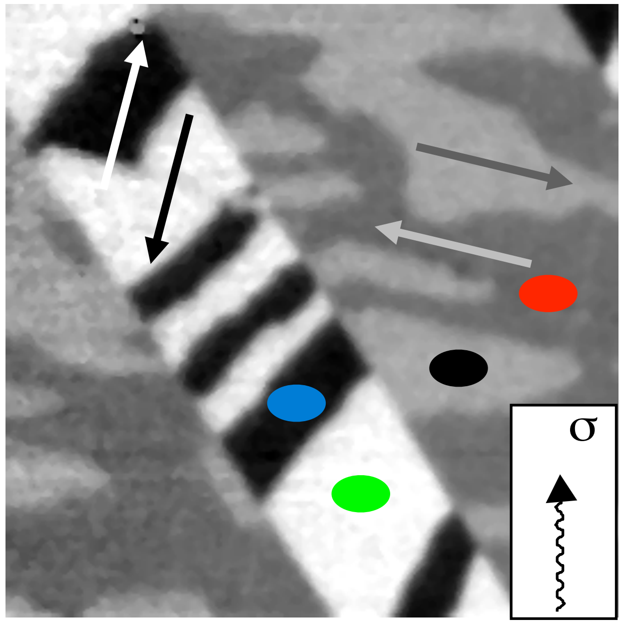
Layer-selective imaging

Tune to Co edge, use **circular** polarization: ferromagnetic domains

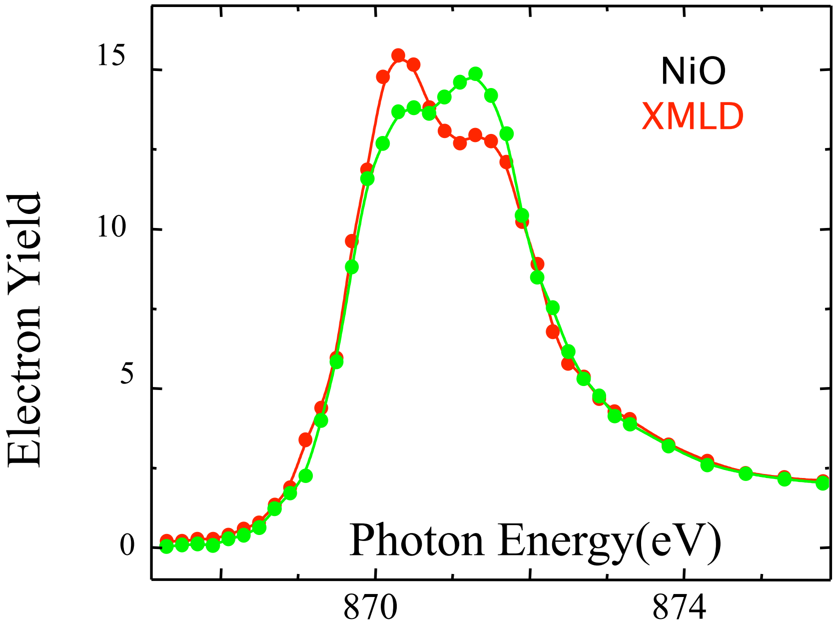
Circular polarization:
Sensitive to direction of
magnetic moment

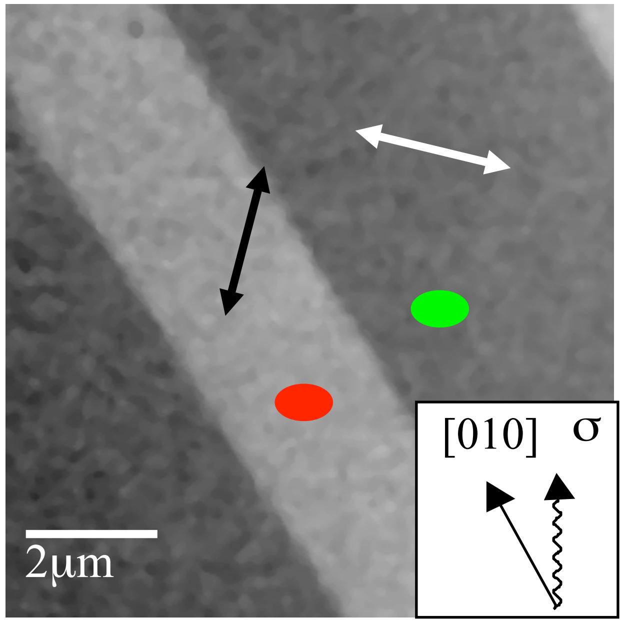






Tune to Ni edge, use linear polarization: antiferromagnetic domains



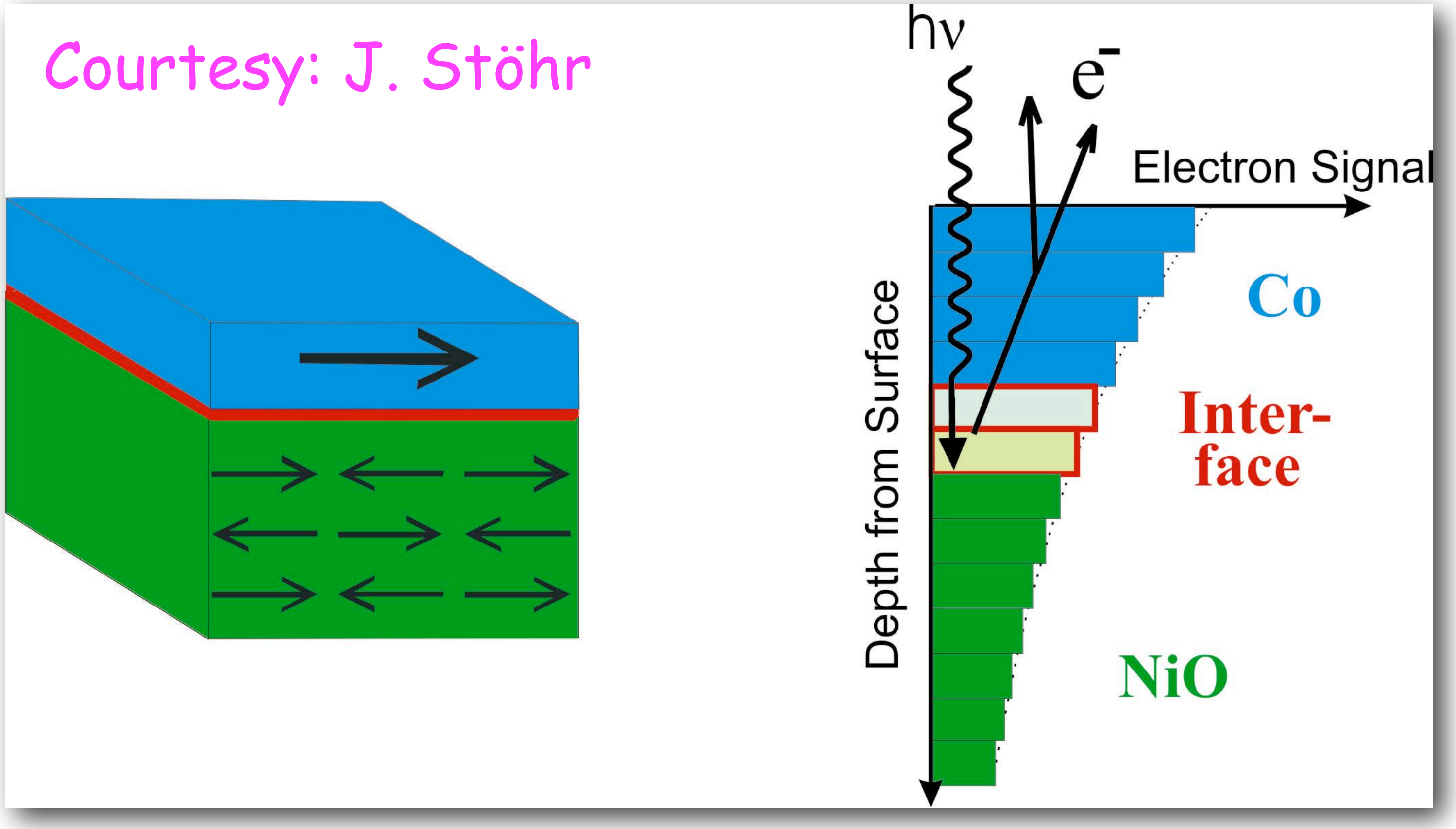


Linear polarization:
Sensitive to axis of
magnetic moment

H. Ohldag et al.,
PRL 86, 2878 (2001)

5. X-Ray Spectroscopy

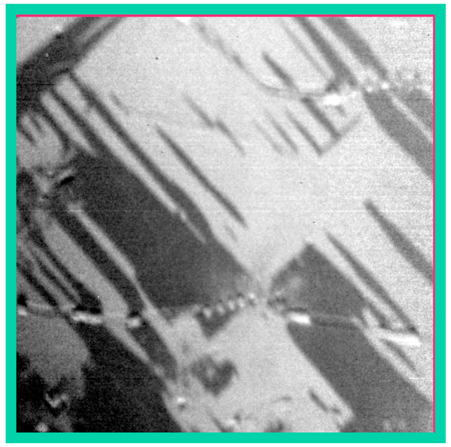
Interface studies of Ferromagnets on Antiferromagnets


Courtesy: J. Stöhr

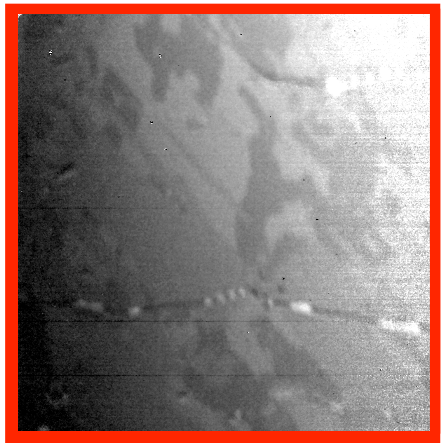
FM Co – tune to Co edge – circular polarization

AFM NiO – tune to Ni edge – linear polarization

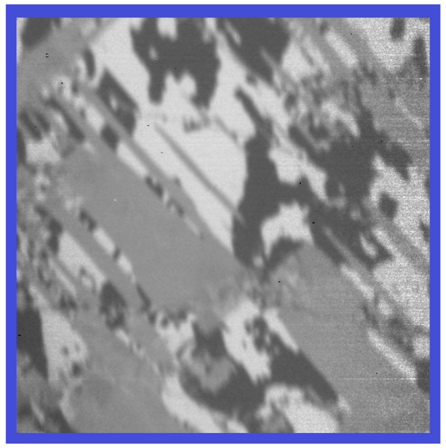
FM Ni(O) – tune to Ni edge – circular polarization


AFM: NiO

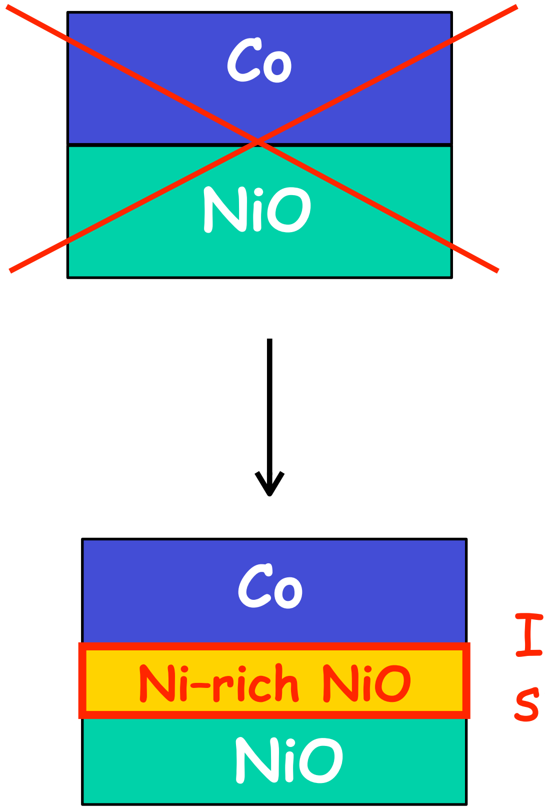
Linear pol.
Ni edge


**FM:
Ni-rich NiO**

Circular pol.
Ni edge


FM: Co

Circular pol.
Co edge

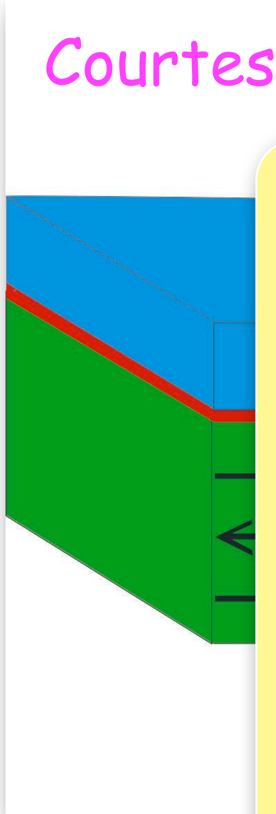


Interfacial spins

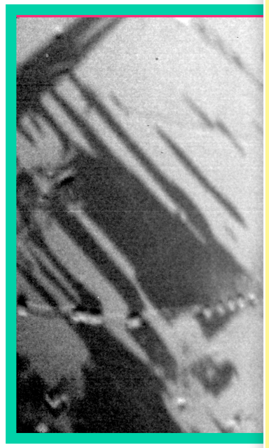
5. X-Ray Spectroscopy

Interface studies of Ferromagnets on Antiferromagnets

Courtesy: J. Stöhr



FM Co – tune to Co edge –
circular polarization



Chemically induced
interfacial Ni spins
provide the magnetic link

AFM: NiO

Linear pol.
Ni edge

**FM:
Ni-rich NiO**

Circular pol.
Ni edge

FM: Co

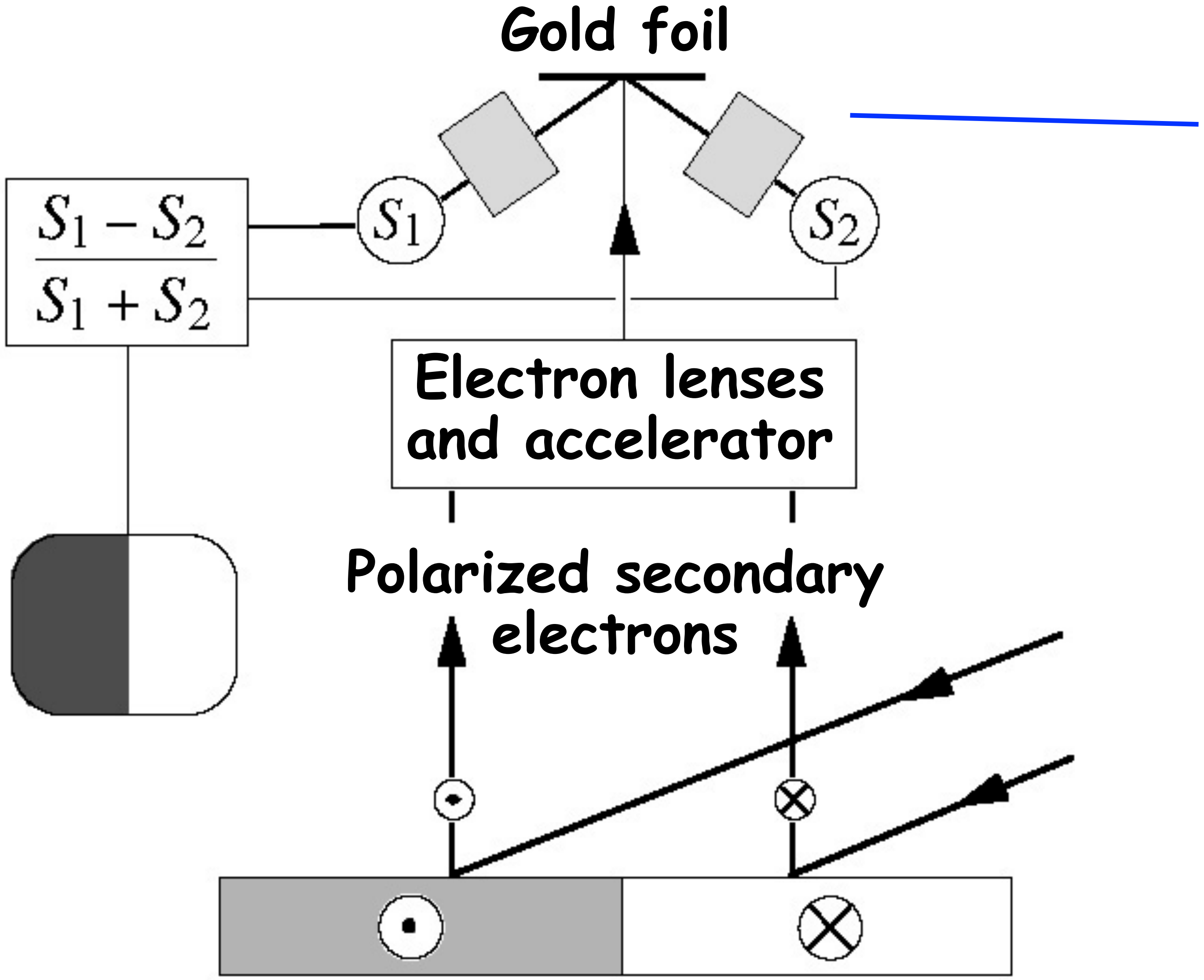
Circular pol.
Co edge

Co
Ni-rich NiO
NiO

Interfacial
spins

6. Electron Polarization Analysis

6.1 Scanning Electron Microscopy with Polarization Analysis (SEMPA)



Mott detector:

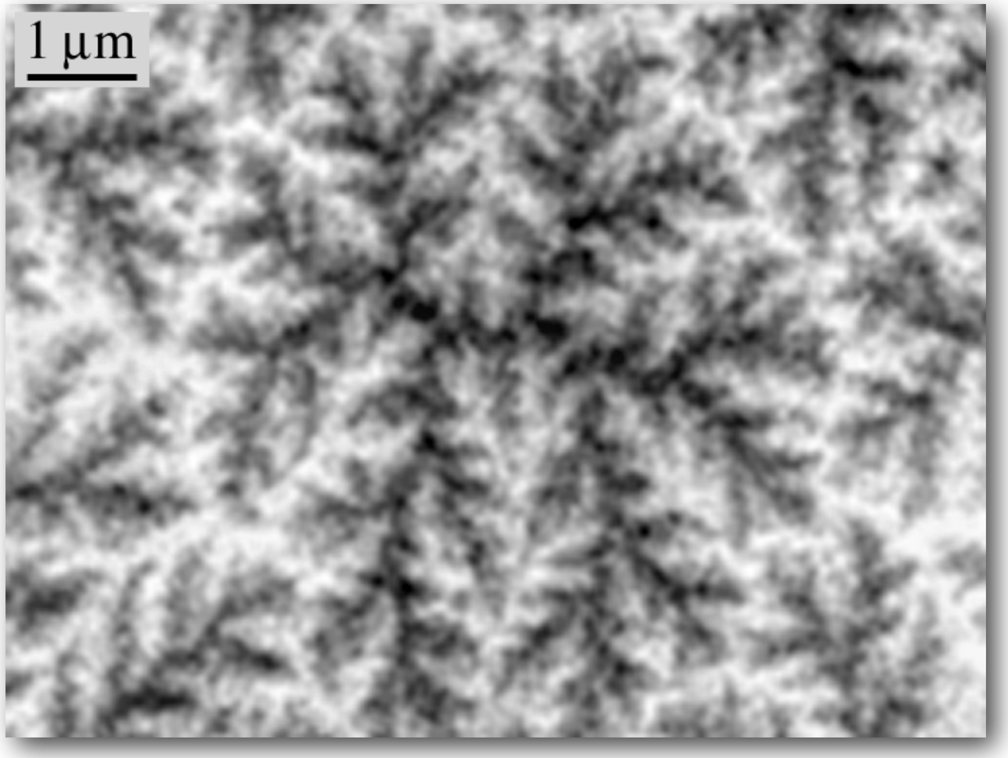
Scattering of polarized electrons by gold foil is asymmetric (spin-orbit coupling effects)

- Secondary electrons are spin polarized, moment along magnetization direction
- Surface sensitive (secondary electrons emerge from top nanometer)
- Quantitative (independent measurement of 3 magnetization components)
- Resolution in 10 nm range

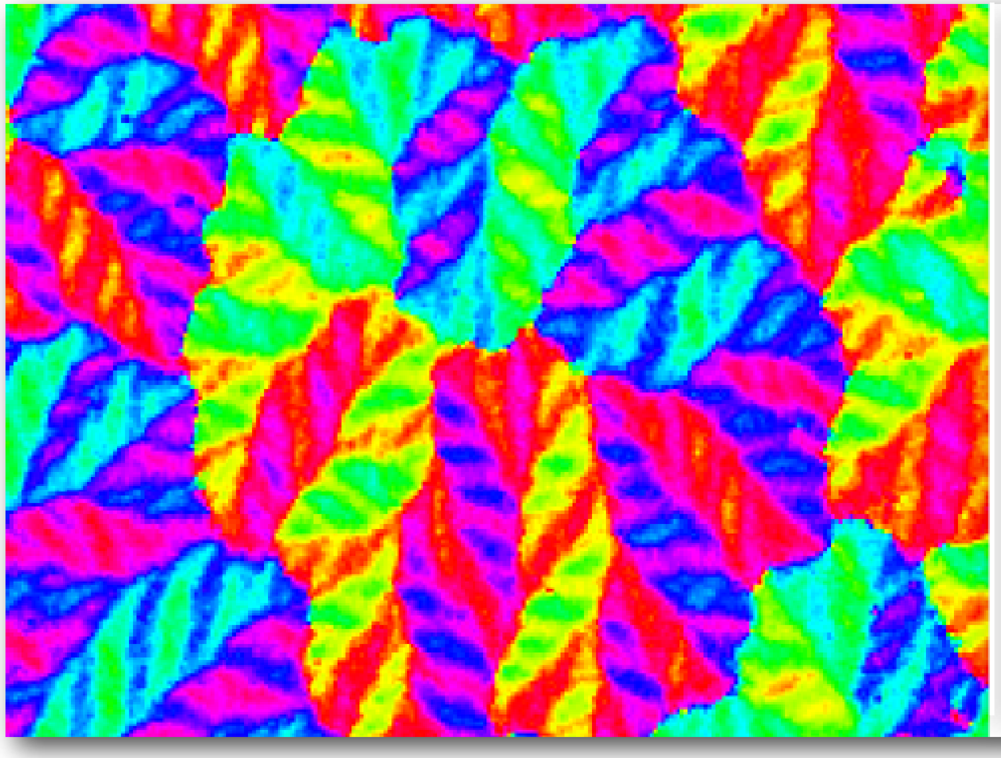
6. Electron Polarization Analysis

6.1 Scanning Electron Microscopy with Polarization Analysis

Basal plane of
Co crystal

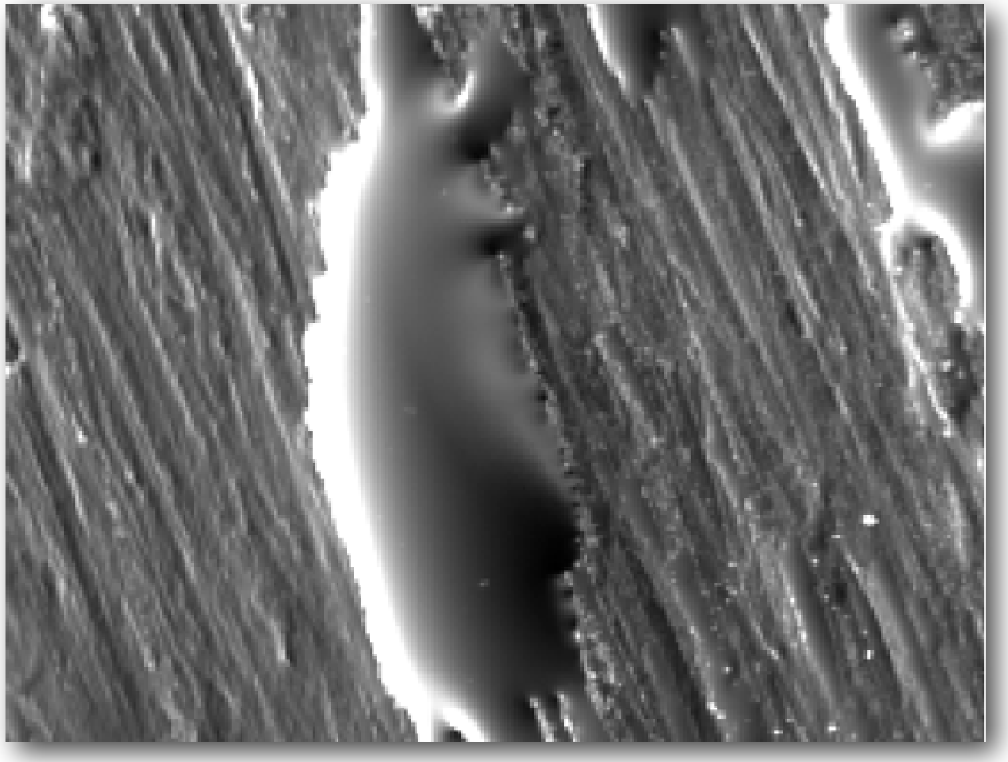


Polar components

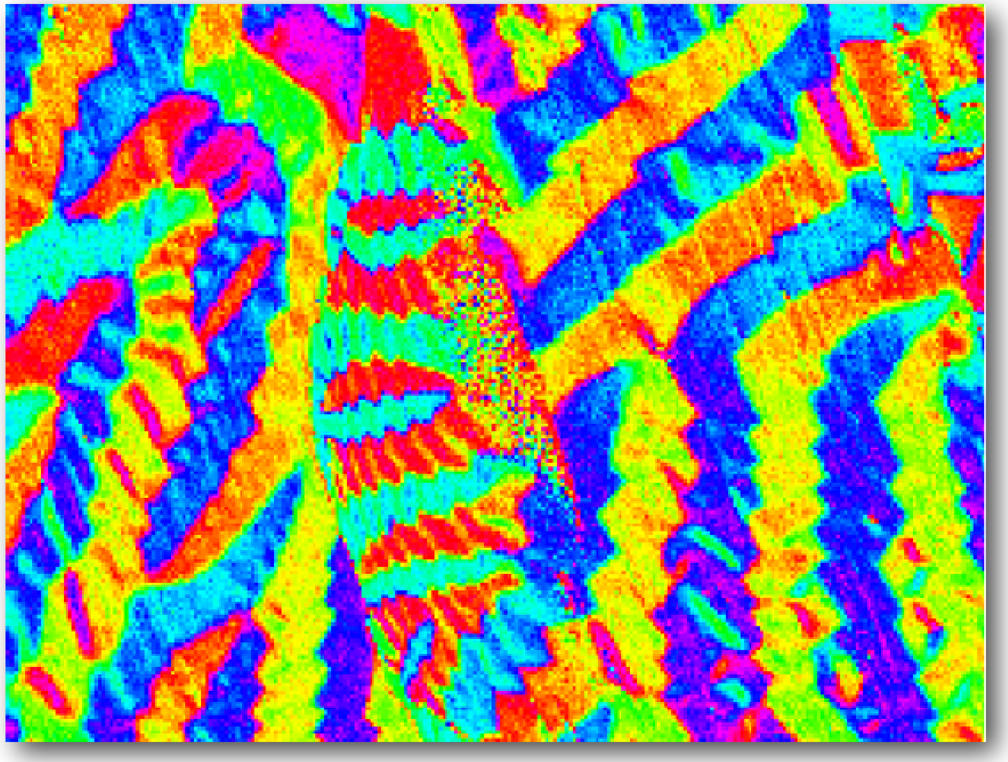


In-plane components

“Wheel side” of
amorphous ribbon



Topography



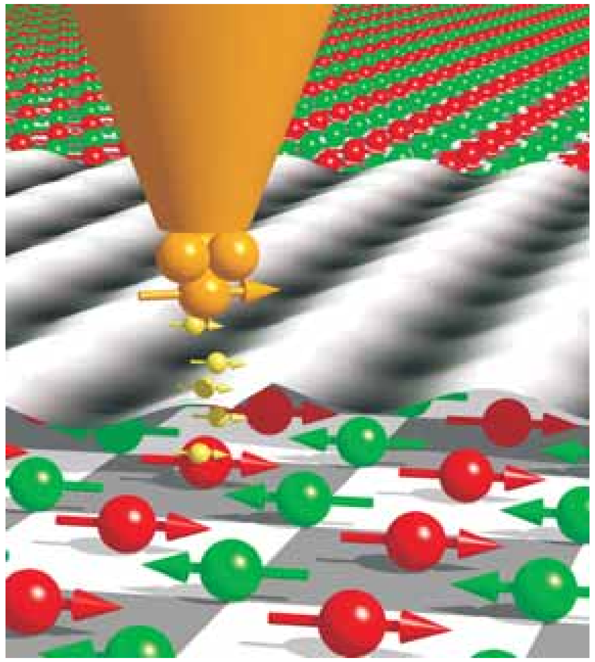
In-plane components

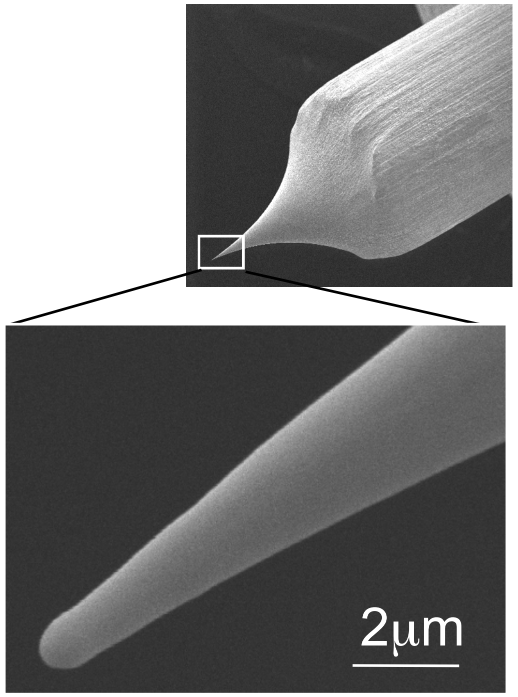
courtesy
J. Unguris

6. Electron Polarization Analysis

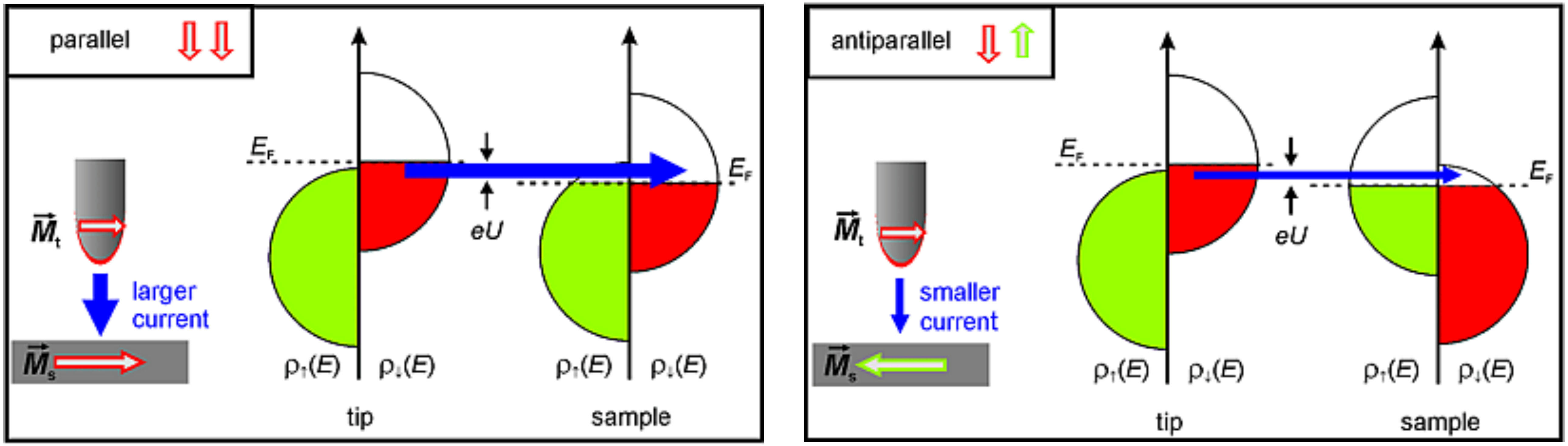
6.2 Spin-polarized Tunneling Microscopy

- Tunneling of spin-polarized current between tip and sample surface
- Tunneling resistance depends on relative orientation of current polarity and domain magnetization
- Extreme resolution





M. Julliere, Phys. Lett. 54A, 225 (1975):



from Wiesendanger homepage

http://www.nanoscience.de/nanojoom/index.php/en/methods/sp-stm.html

6. Electron Polarization Analysis

6.2 Spin-polarized Tunneling Microscopy

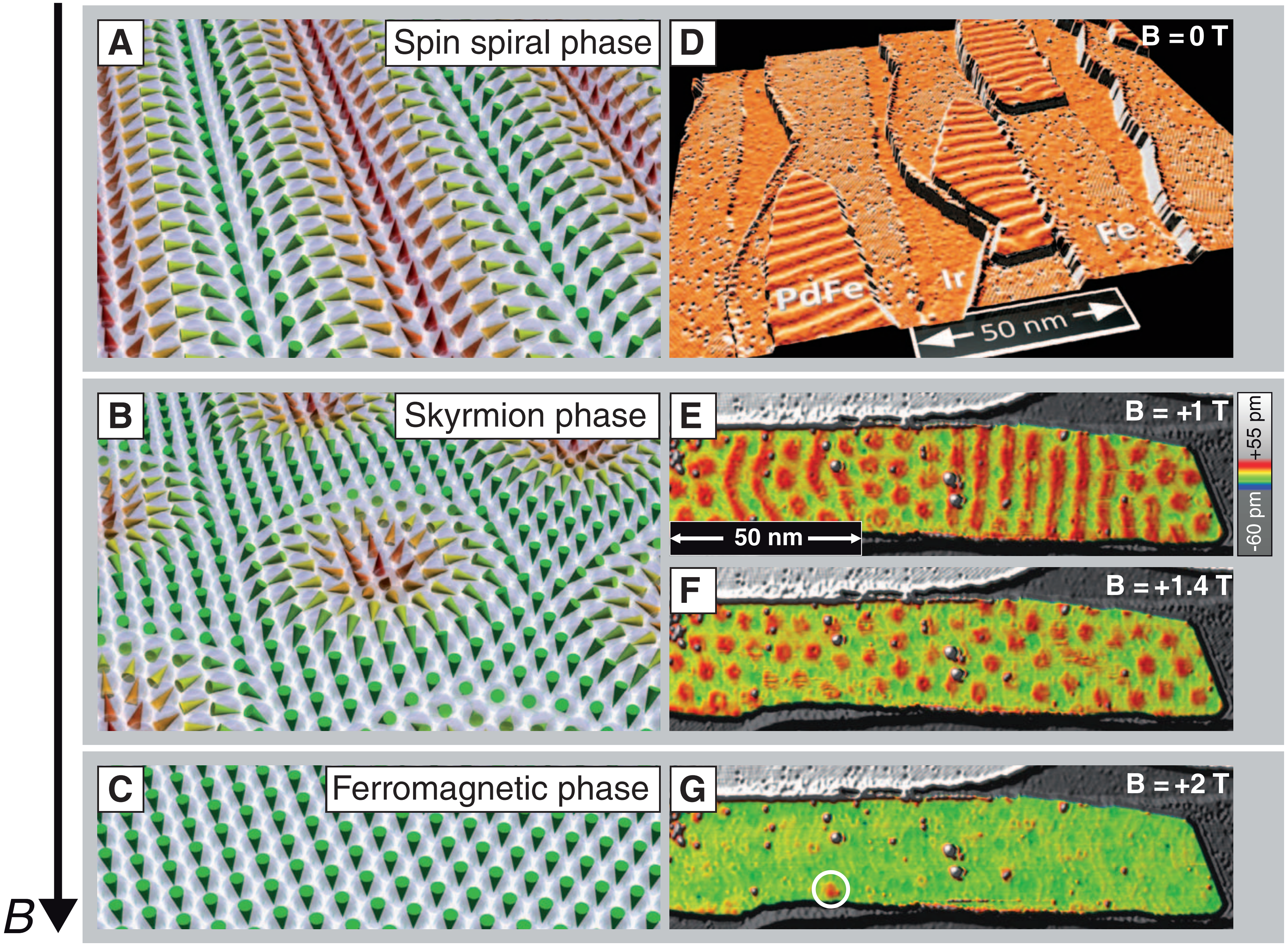


Fig. 1. Magnetic field dependence of the PdFe bilayer on the Ir(111) surface at $T = 8$ K. (**A** to **C**) Perspective sketches of the magnetic phases. (**D**) Overview SP-STM image, perspective view of constant-current image colorized with its derivative. (**E** to **G**) PdFe bilayer at different magnetic fields ($U = +50$ mV, $I = 0.2$ nA, magnetically out-of-plane sensitive tip). (E) Coexistence of spin spiral and skyrmion phase. (F) Pure skyrmion phase. (G) Ferromagnetic phase. A remaining skyrmion is marked by the white circle.

Science
AAAS

Writing and Deleting Single Magnetic Skyrmions.
Niklas Romming et al.
Science 341, 636 (2013);
DOI: 10.1126/science.1240573

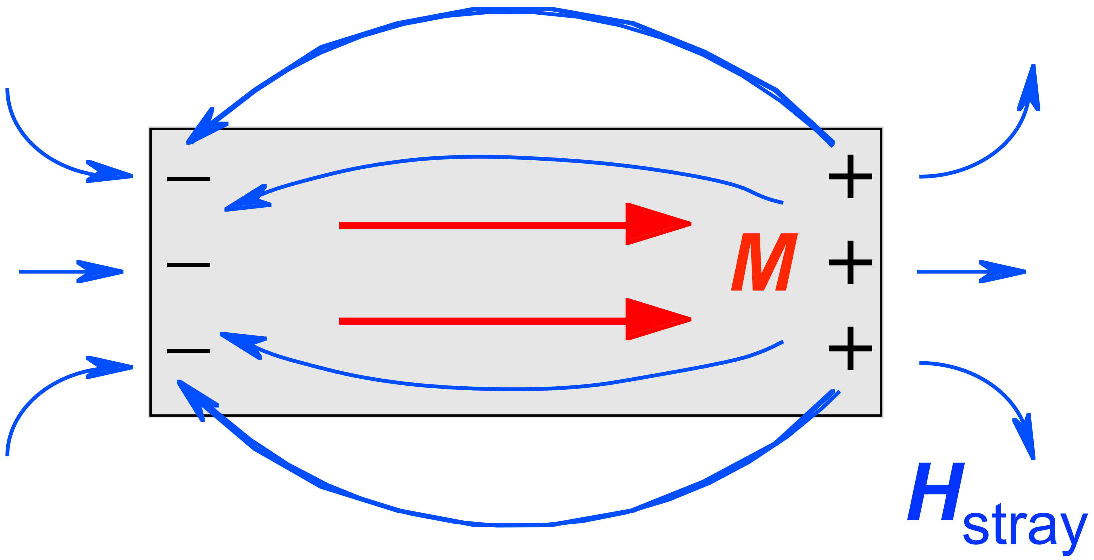
Sensitivity of imaging methods

$\boldsymbol{B} = \mu_0 (\boldsymbol{H} + \boldsymbol{M})$ $(\boldsymbol{H} = \boldsymbol{H}_{ext} + \boldsymbol{H}_{stray})$

$\mathrm{div}\, \boldsymbol{B} = 0$

$\downarrow$

$\mathrm{div}\, \boldsymbol{H}_{stray} = - \mathrm{div}\, \boldsymbol{M}$



- Sensitive to $\boldsymbol{H}_{stray}$
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  2. Magnetic force microscopy
  3. Hall probe microscopy
- Sensitive to $\boldsymbol{M}$
  4. Magneto-optical microscopy
  5. X-ray spectroscopy
  6. Polarized electrons (SEMPA, SPT)
- Sensitive to $\boldsymbol{B}$
  - Transmission electron microscopy
- Sensitive to distortions
  - X-ray, neutron scattering

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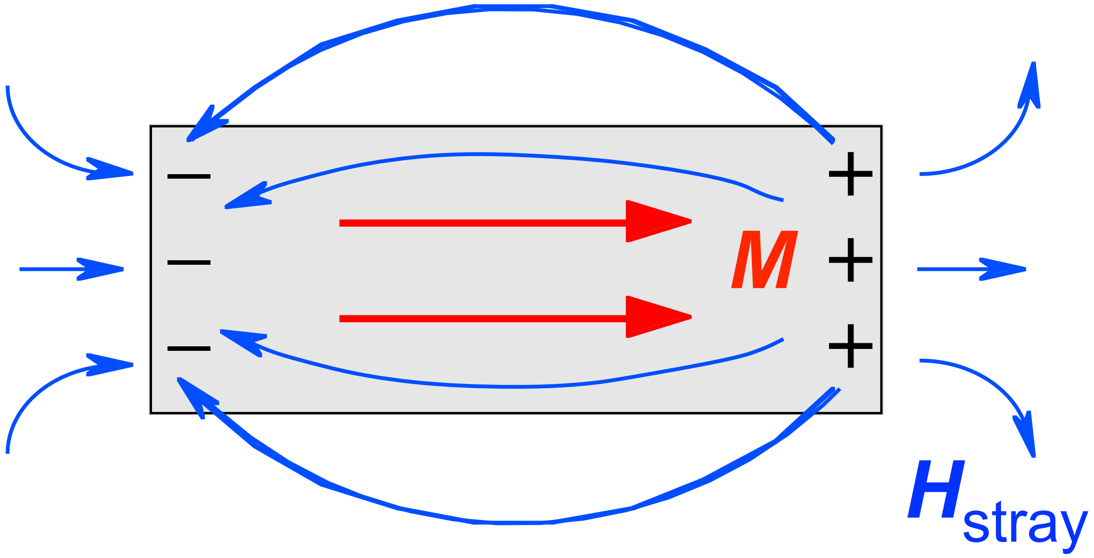

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- Sensitive to $\boldsymbol{B}$
  7. Transmission electron microscopy
- Sensitive to distortions
  - X-ray, neutron scattering

7. Transmission Electron Microscopy

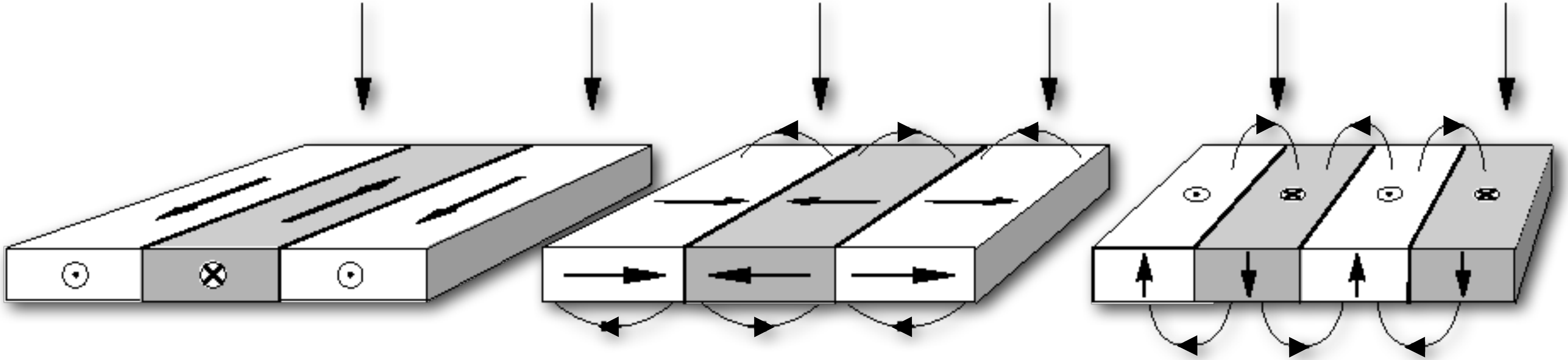
Principle

- **Electrons are deflected by Lorentz force**

$F_L = q_e (v_e \times B)$

q_e: Electron charge
v_e: Electron velocity
B : Magnetic flux density

- Stray fields outside the sample contribute to contrast



Net deflection of electrons

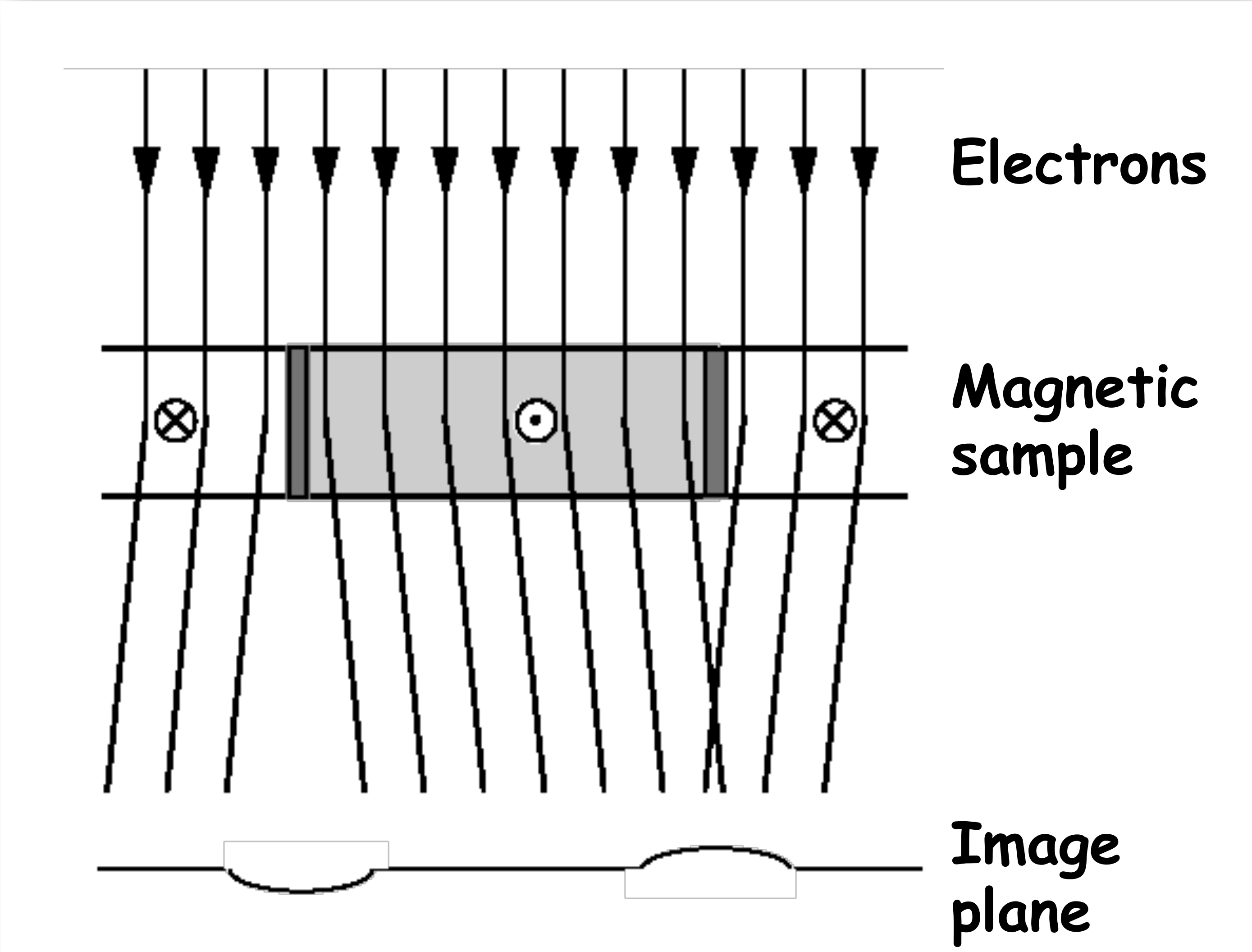
Deflection by magnetization is canceled by deflections due to stray field

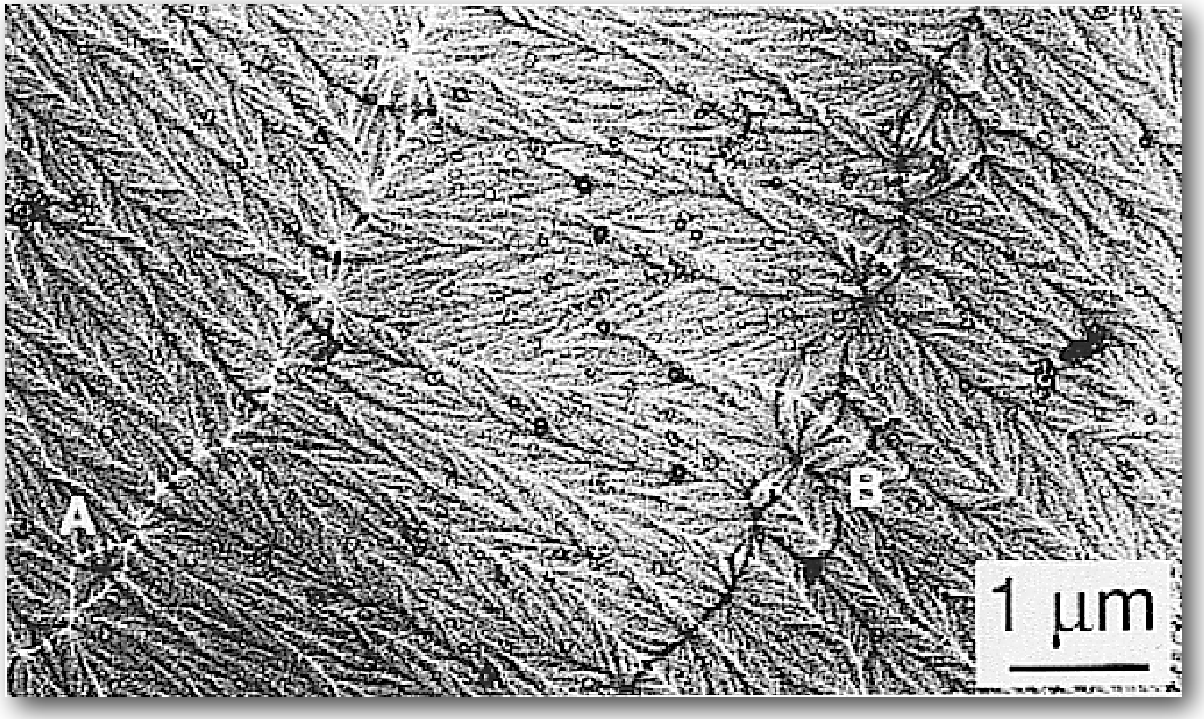
No deflection by magnetization, stray field deflection cancels

- Tilting of sample may be required
- Maximum sample thickness: some 100 nm

7. Transmission Electron Microscopy

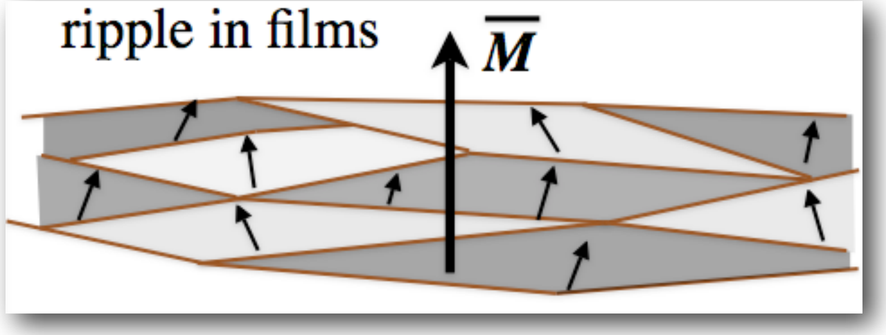
7.1 Fresnel technique (defocused mode imaging)





Metallic glass, partially crystallized
(courtesy J. Chapman)

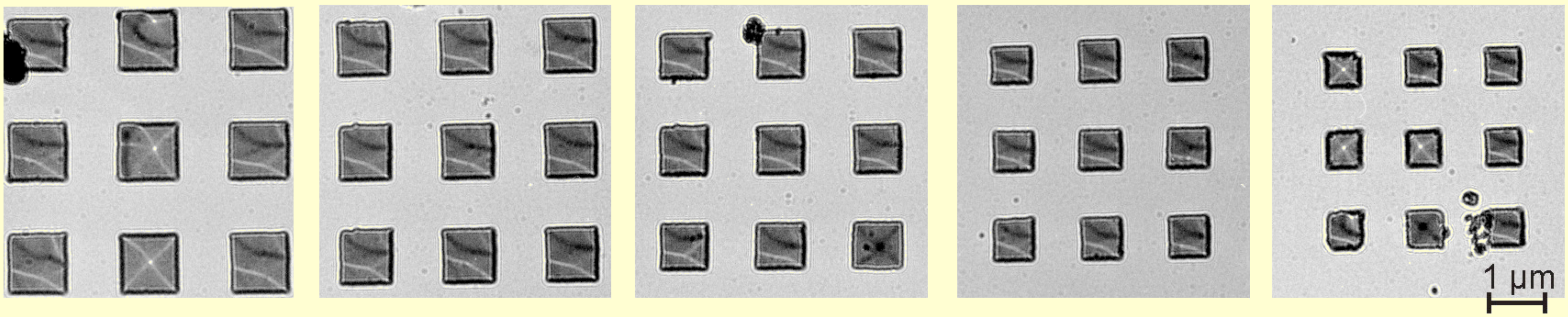
- Out-of-focus: shadow effects delineate domain boundaries
- Magnetization direction can be derived from ripple (if present)



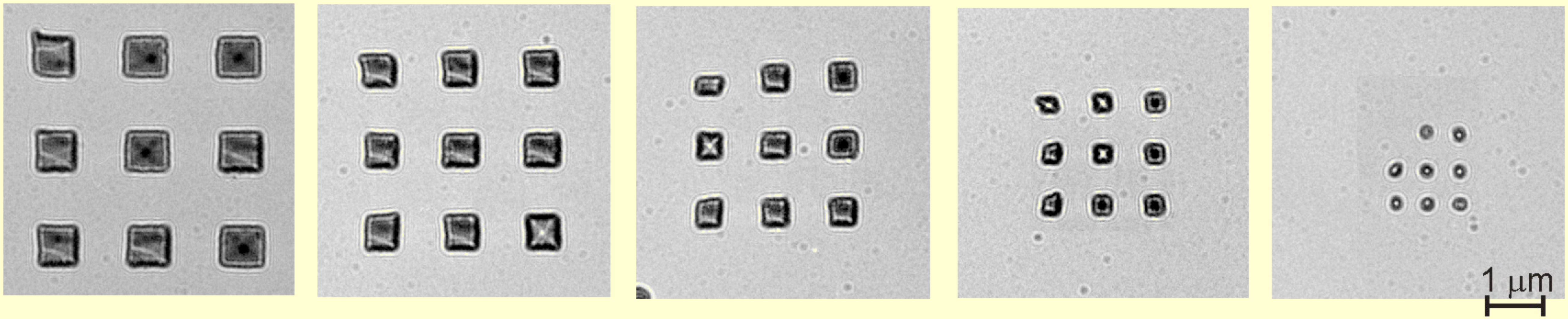
7. Transmission Electron Microscopy

7.1 Fresnel technique (defocused mode imaging)

Fresnel imaging of differently sized magnetic particles (Co, 35 nm thick)



a) 1 x 1 µm^2 b) 900 x 900 nm^2 c) 800 x 800 nm^2 d) 700 x 700 nm^2 e) 600 x 600 nm^2

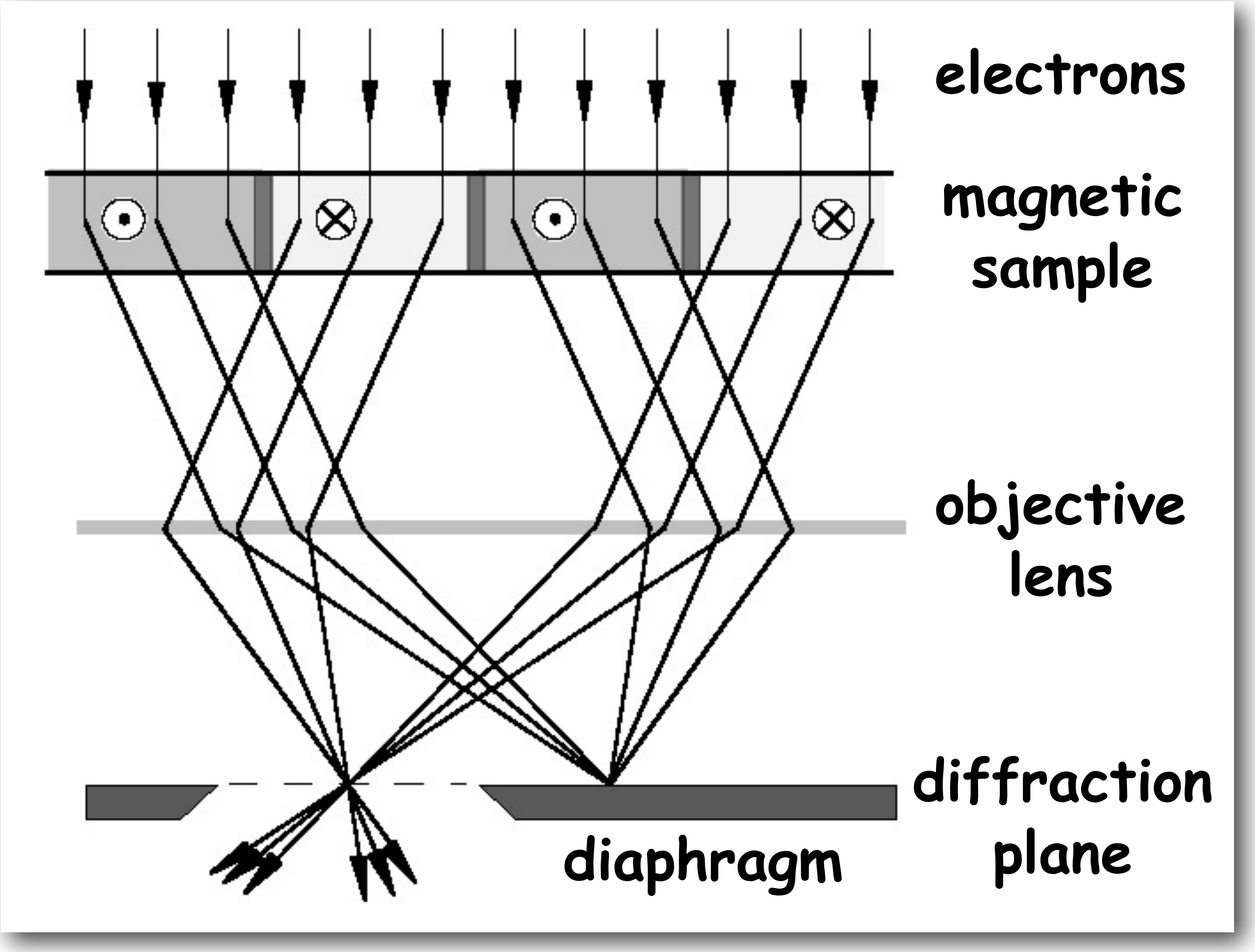


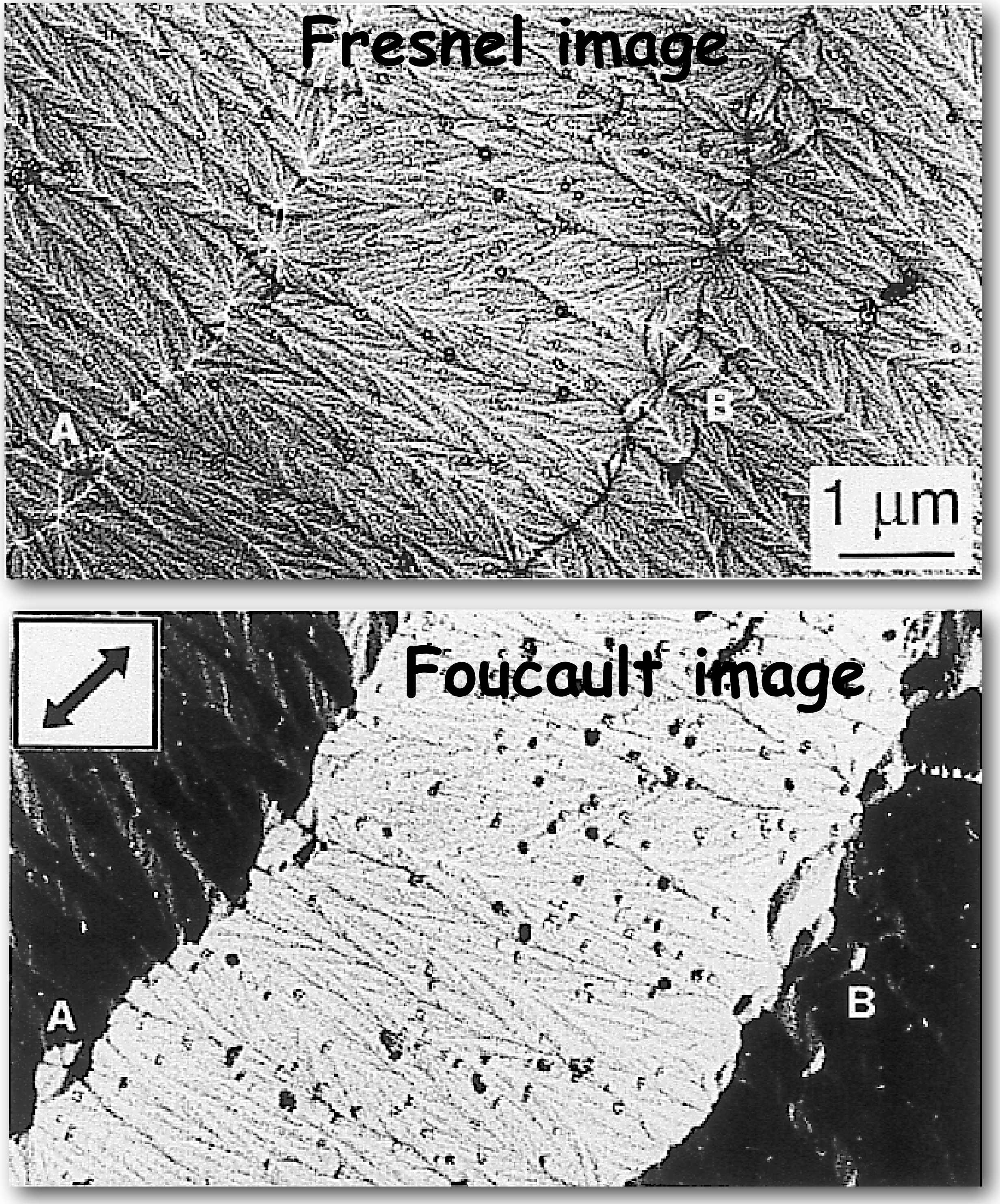
f) 500 x 500 nm^2 g) 400 x 400 nm^2 h) 300 x 300 nm^2 i) 200 x 200 nm^2 j) 100 x 100 nm^2

Courtesy J. Zweck

7. Transmission Electron Microscopy

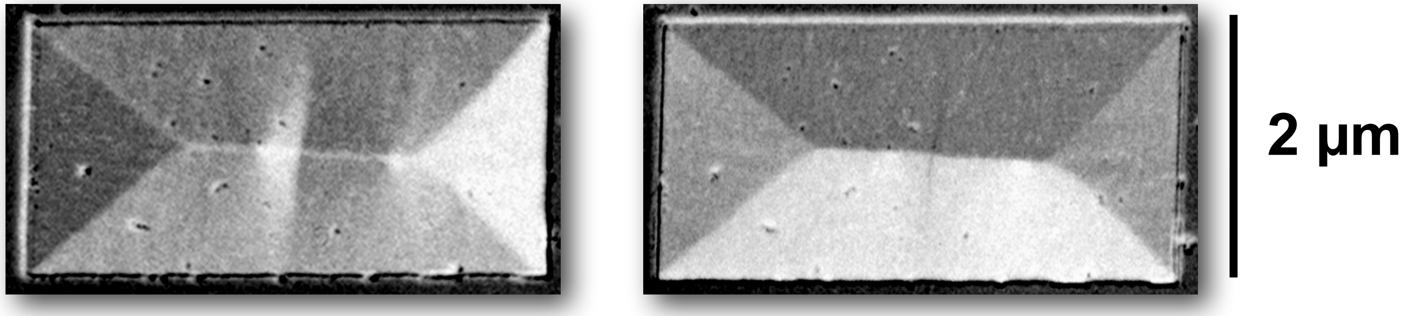
7.2 Foucault technique (in-focus)




Fresnel image

Foucault image

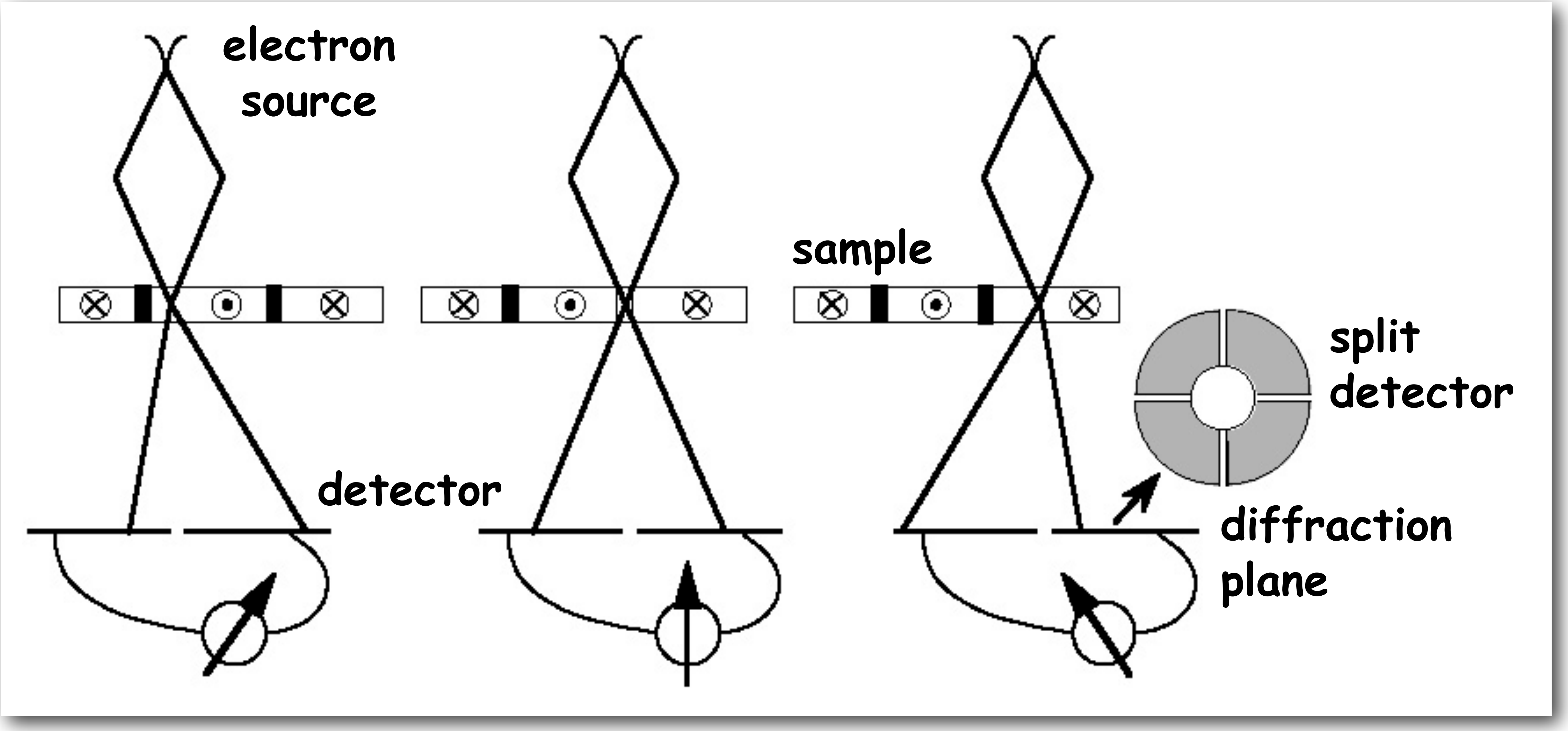
Metallic glass, partially crystallized
(courtesy J. Chapman)



Permalloy, 24 nm thick
(courtesy J. Chapman)

7. Transmission Electron Microscopy

7.3 Differential Phase Contrast (DPC) Microscopy

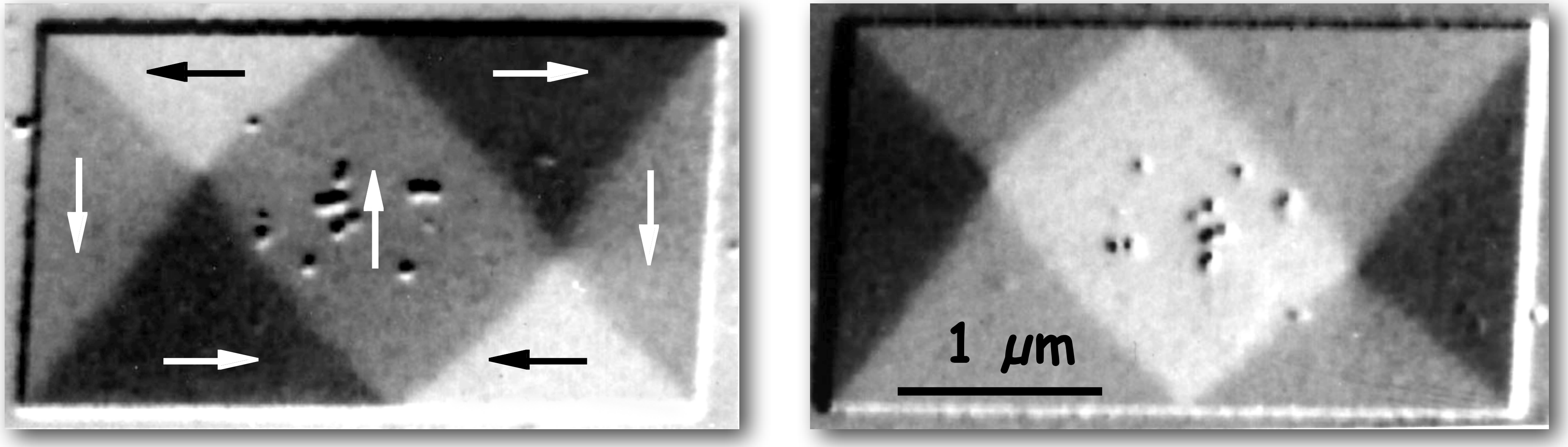


- Domain contrast like in Kerr microscopy
- Resolution better than 10 nm
- Quantitative determination of magnetization direction (by combining signals of a quadrant detector)

7. Transmission Electron Microscopy

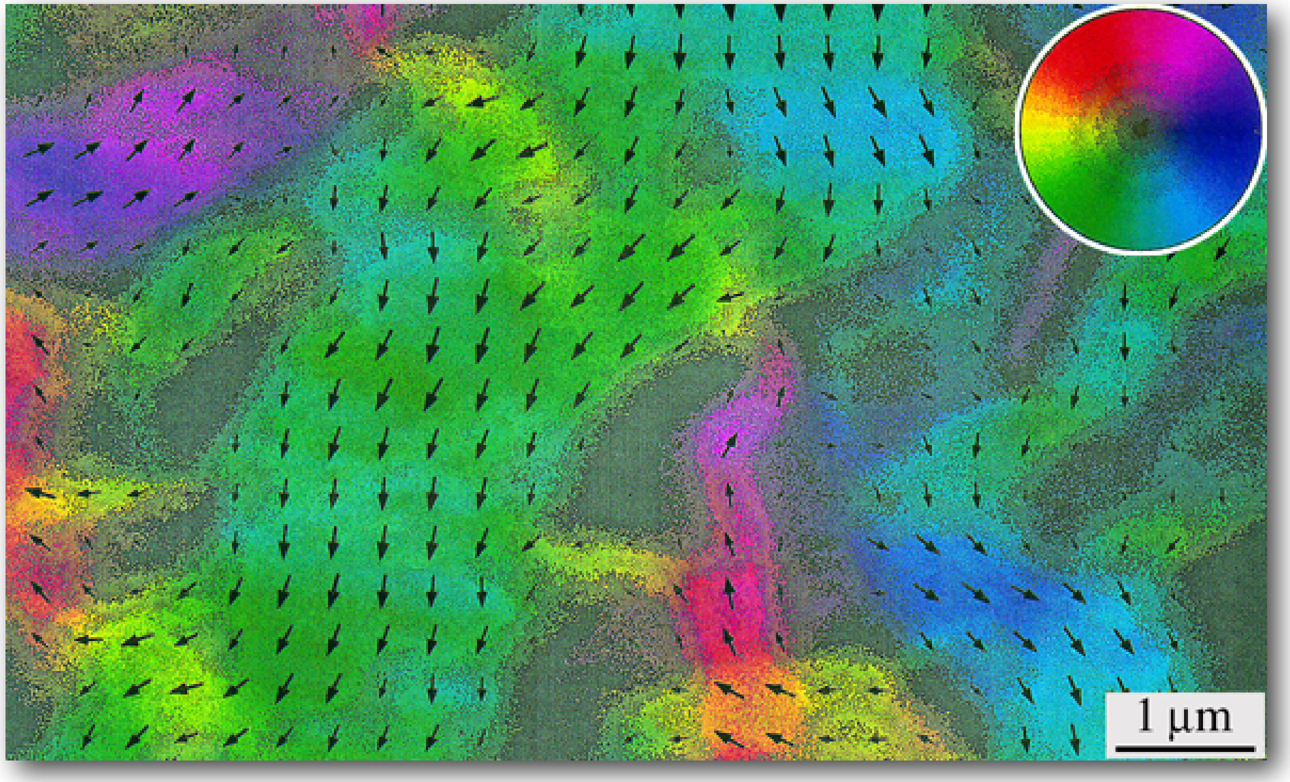
7.3 Differential Phase Contrast (DPC) Microscopy

In a scanning TEM



Permalloy, 60 nm thick
(courtesy J. Chapman)

In a conventional TEM



AFM coupled Co-Cr-Co sandwich
(courtesy J.P. Jakubovics)

Difference between Foucault images, obtained at different angles of incidence

7. Transmission Electron Microscopy

7.4 Electron Holography

Principle

$$\operatorname{grad}\varphi_{\mathrm{e}} = (2\,\pi\; q_{\mathrm{e}}\, D/h)\,\boldsymbol{B}_0 \times \boldsymbol{s}_{\mathrm{b}}$$

phase variation

phase of electron wave

film thickness

magnetic flux density perpendicular to beam direction

beam direction

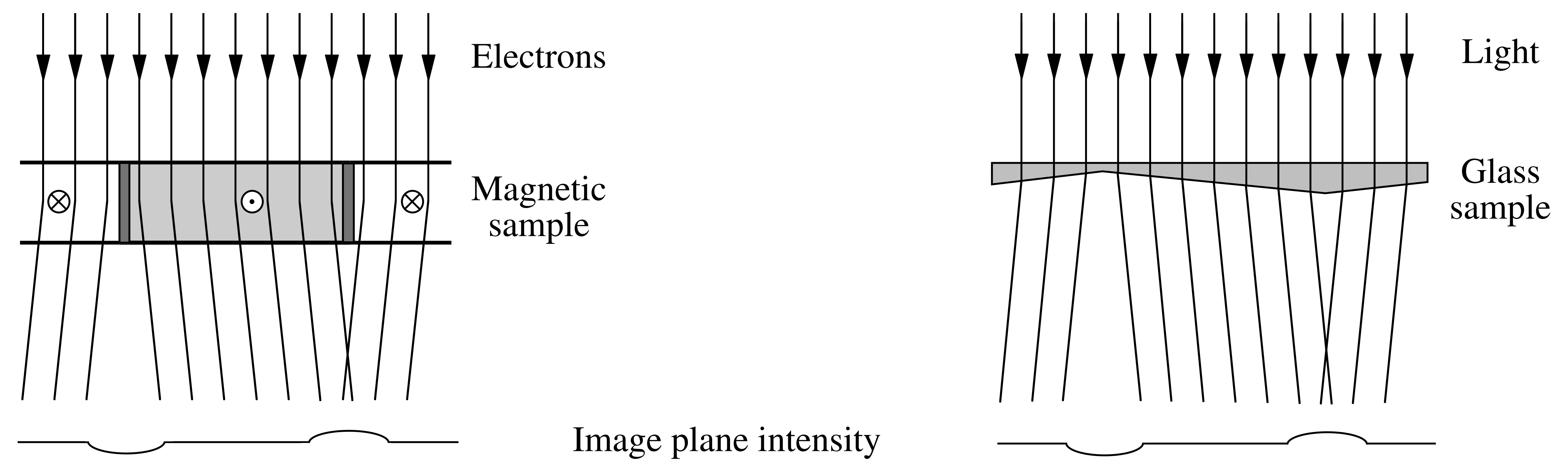
- Magnetization influences phase of electron wave
- Phase gradient is perpendicular to B_0
- Lines of constant phase are parallel to B_0
- Flux between two lines is equal to flux quantum h/q_{e}

Electron Holography:

- Interference pattern of 2 electron waves shifted in phase
- Evaluation in optical interferometer

7. Transmission Electron Microscopy

7.4 Electron Holography



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Electron Holography:

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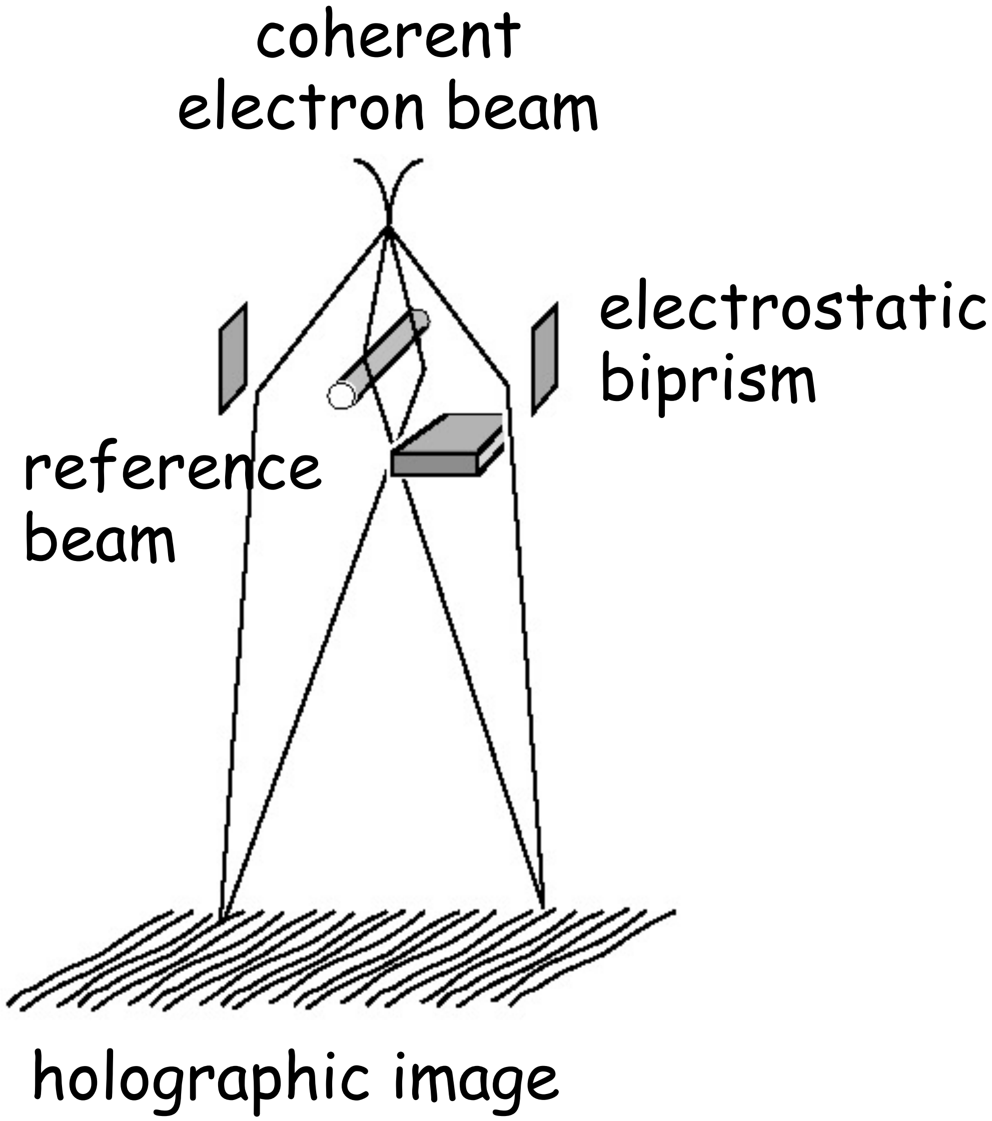
- Evaluation in optical interferometer

7. Transmission Electron Microscopy

7.4 Electron Holography

Off-axis holography (Tonomura et al. 1980)

Generation of hologram



Optical reconstruction

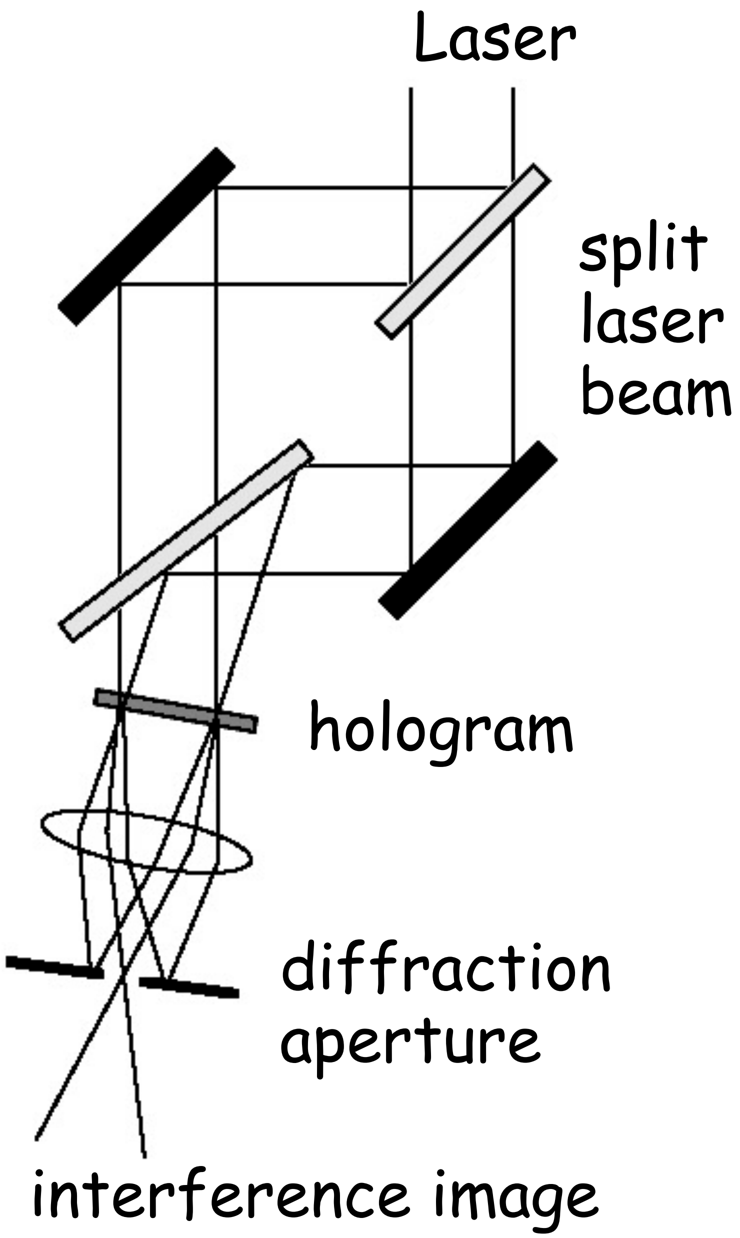
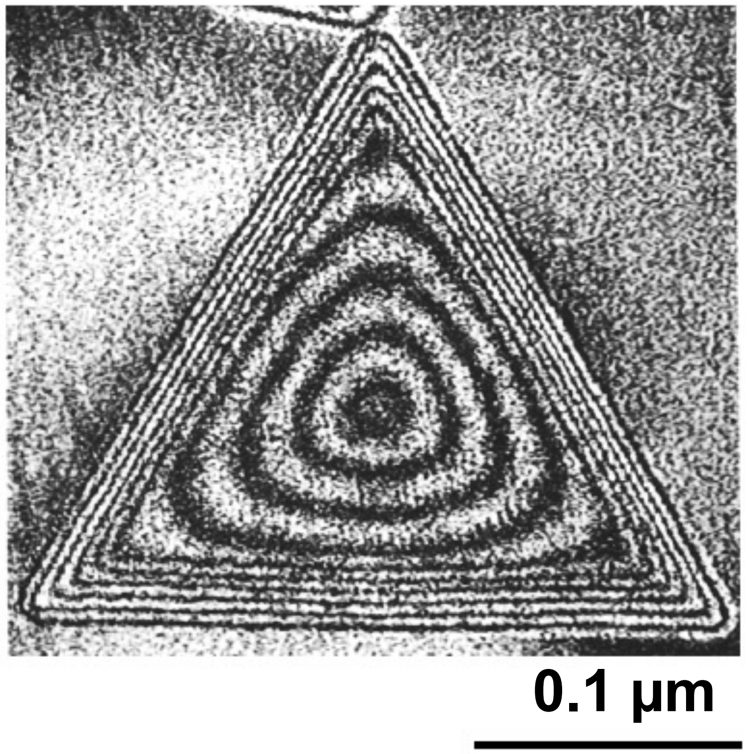
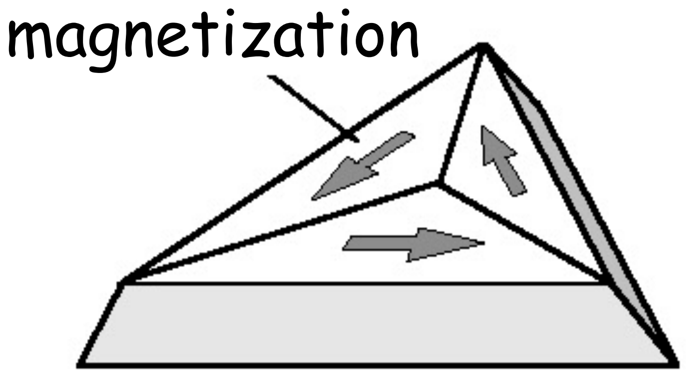


Image shows lines of constant phase



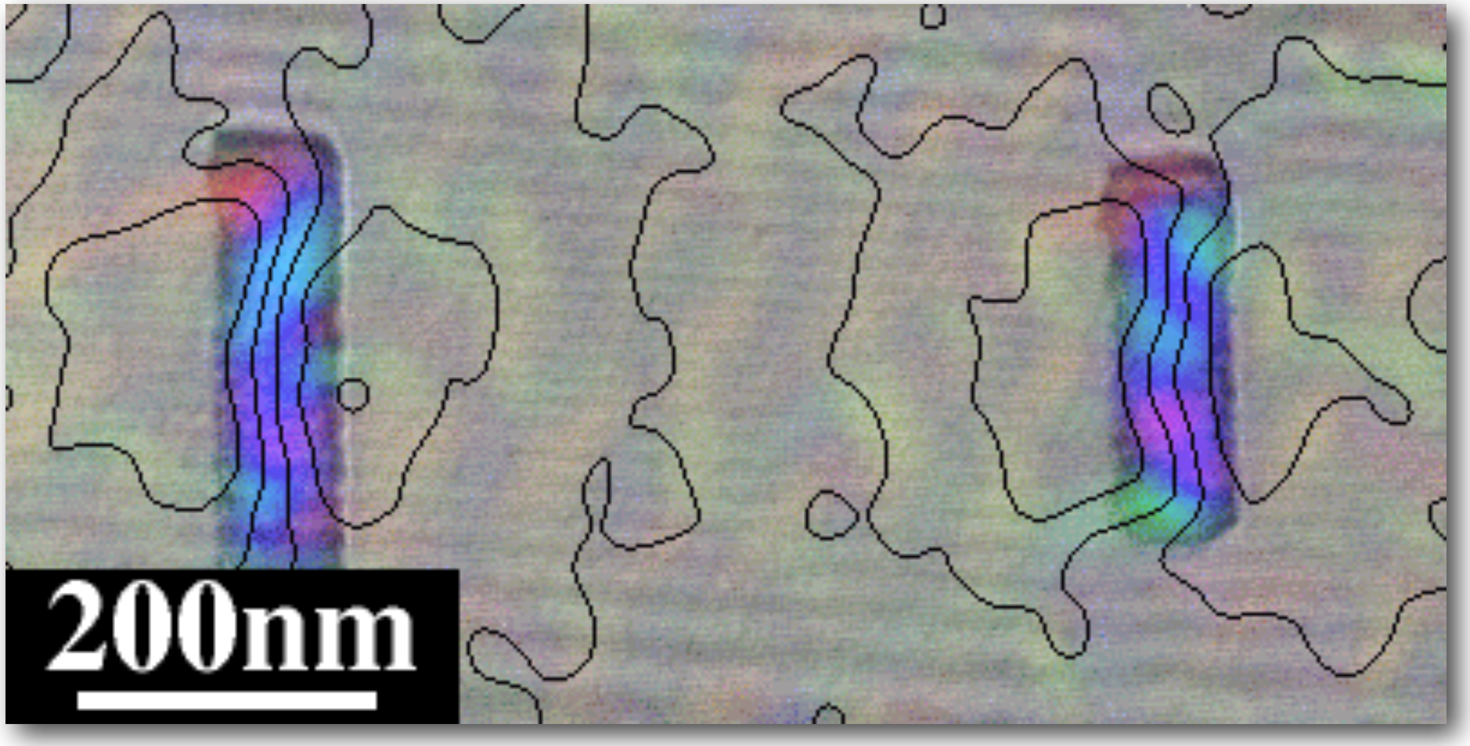


7. Transmission Electron Microscopy

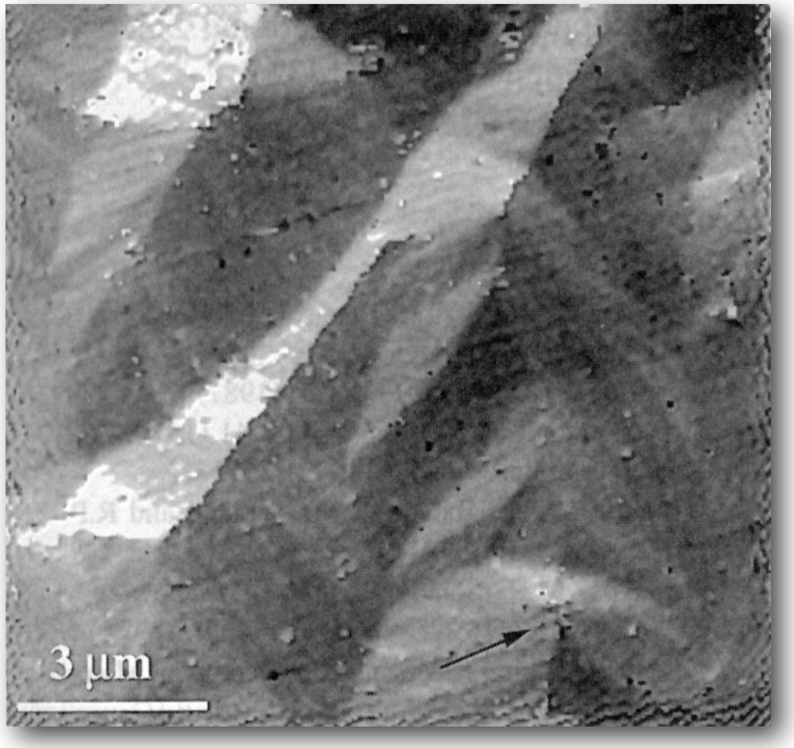
7.4 Electron Holography

Differential Holography (Mankos et al. 1994)

- Both interfering beams pass through sample along slightly different paths (distance: 10 nm)
- Reconstruction contains information about their phase difference
  - ⟶ phase gradient is recorded, which is proportional to magnetization
  - ⟶ "real" domain images like in Kerr microscopy
- Quantitative information about magnetization direction at high resolution



Co/Au/Ni/Al multilayer
(courtesy M. McCartney)



30 nm Co film
(courtesy M. Scheinfein)

Sensitivity of imaging methods

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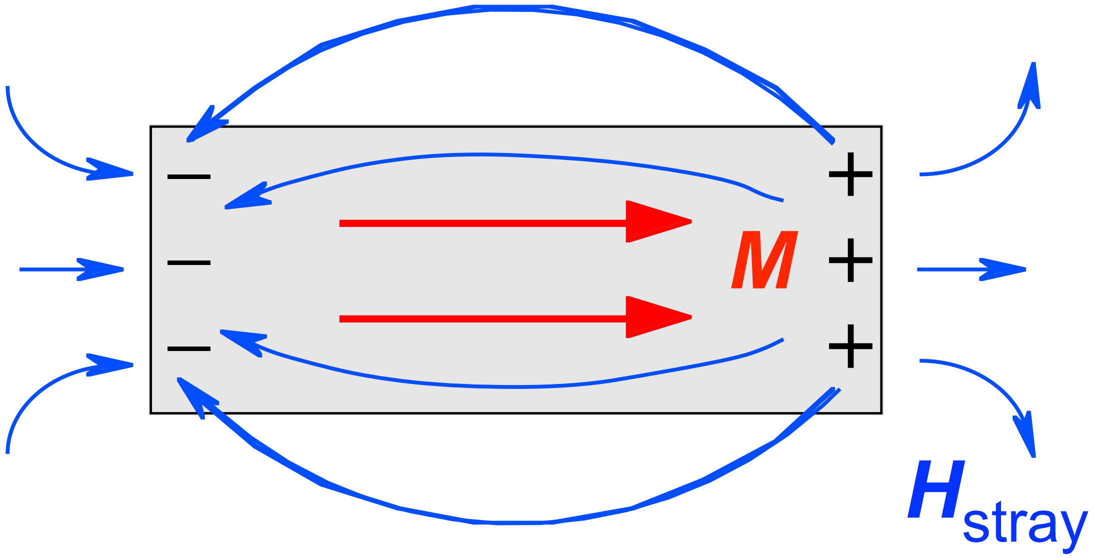
$$\downarrow$$

$\mathrm{div}\, \boldsymbol{H}_{stray} = -\, \mathrm{div}\, \boldsymbol{M}$



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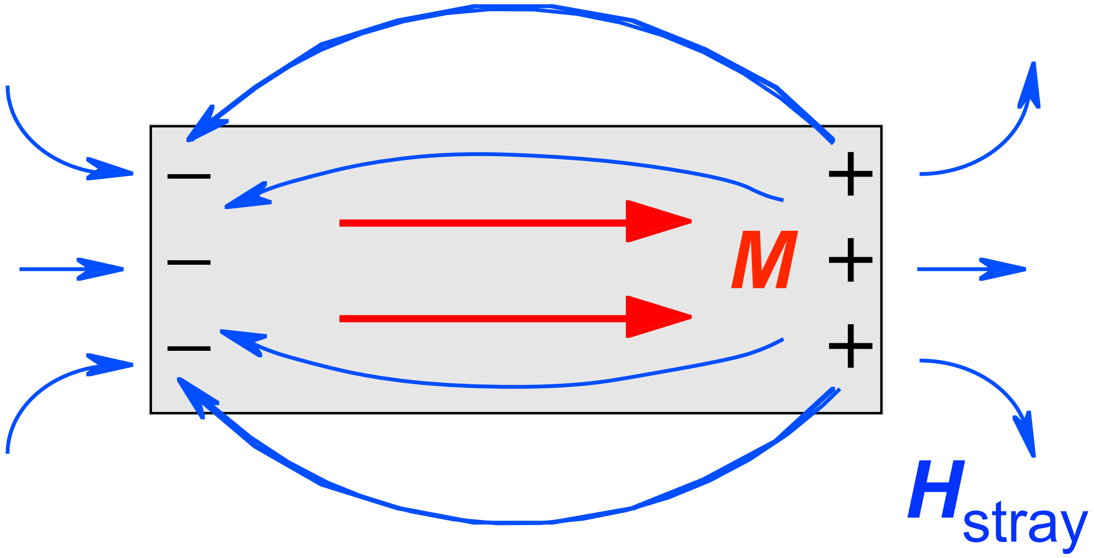

- Sensitive to $\boldsymbol{H}_{\text{stray}}$

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3. Hall probe microscopy

- Sensitive to $\boldsymbol{M}$

4. Magneto-optical microscopy
5. X-ray spectroscopy
6. Polarized electrons (SEMPA, SPT)

- Sensitive to $\boldsymbol{B}$

7. Transmission electron microscopy

- Sensitive to distortions

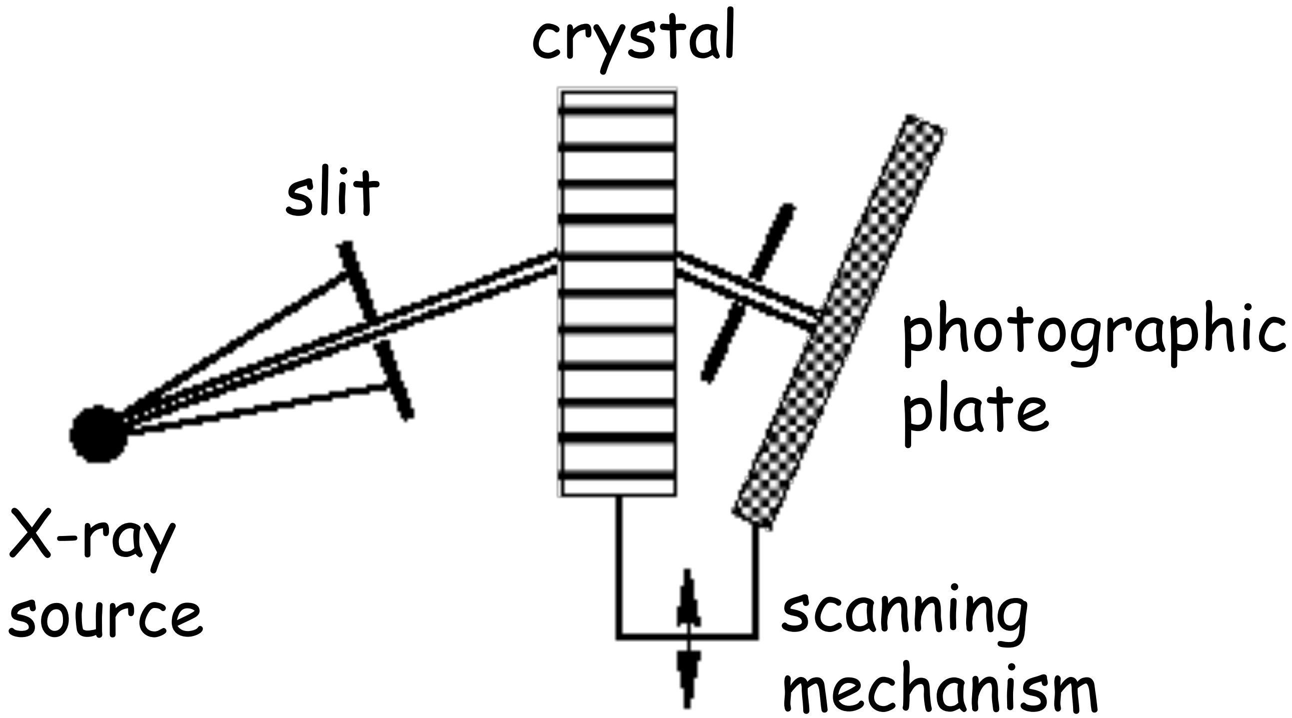
8. X-ray, neutron scattering

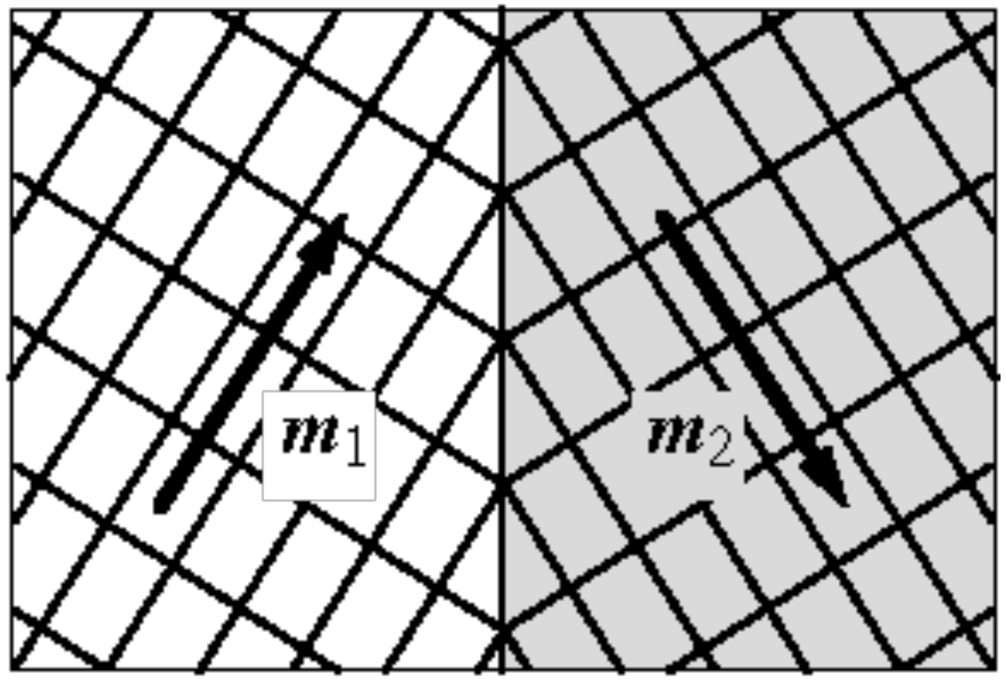
8.1 X-ray Topography

- Plane-parallel X-ray beam, restricted to narrow strip
- Bragg condition fulfilled for some set of lattice planes
- Diffracted beam recorded by photographic plate
- Crystal and plate are advanced synchronously (scanning)

Contrast mechanism:

- Magnetostrictive strains disturb Bragg reflection
- Contrast at those positions, where rotation or spacing of lattice changes

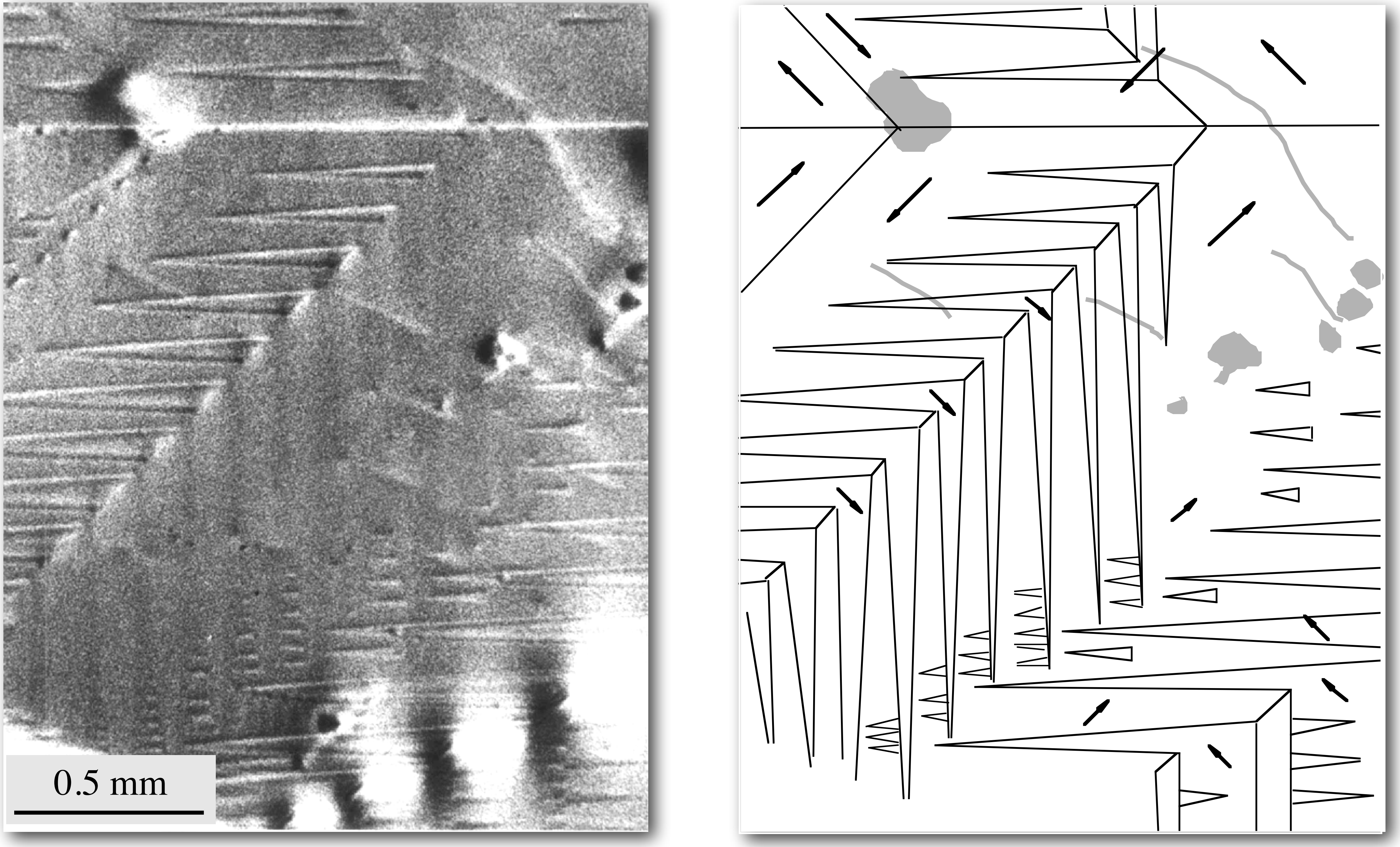




Change of lattice orientation at 90° wall (10 – 5 radian)

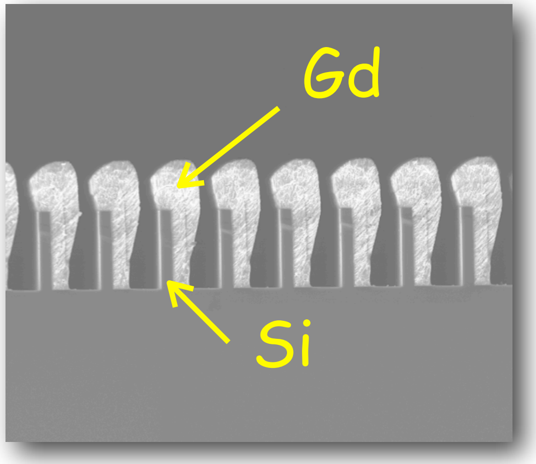
8.1 X-ray Topography

X-ray topogram of fir-tree domains on slightly misoriented (100) FeSi crystal (0.1 mm thick)

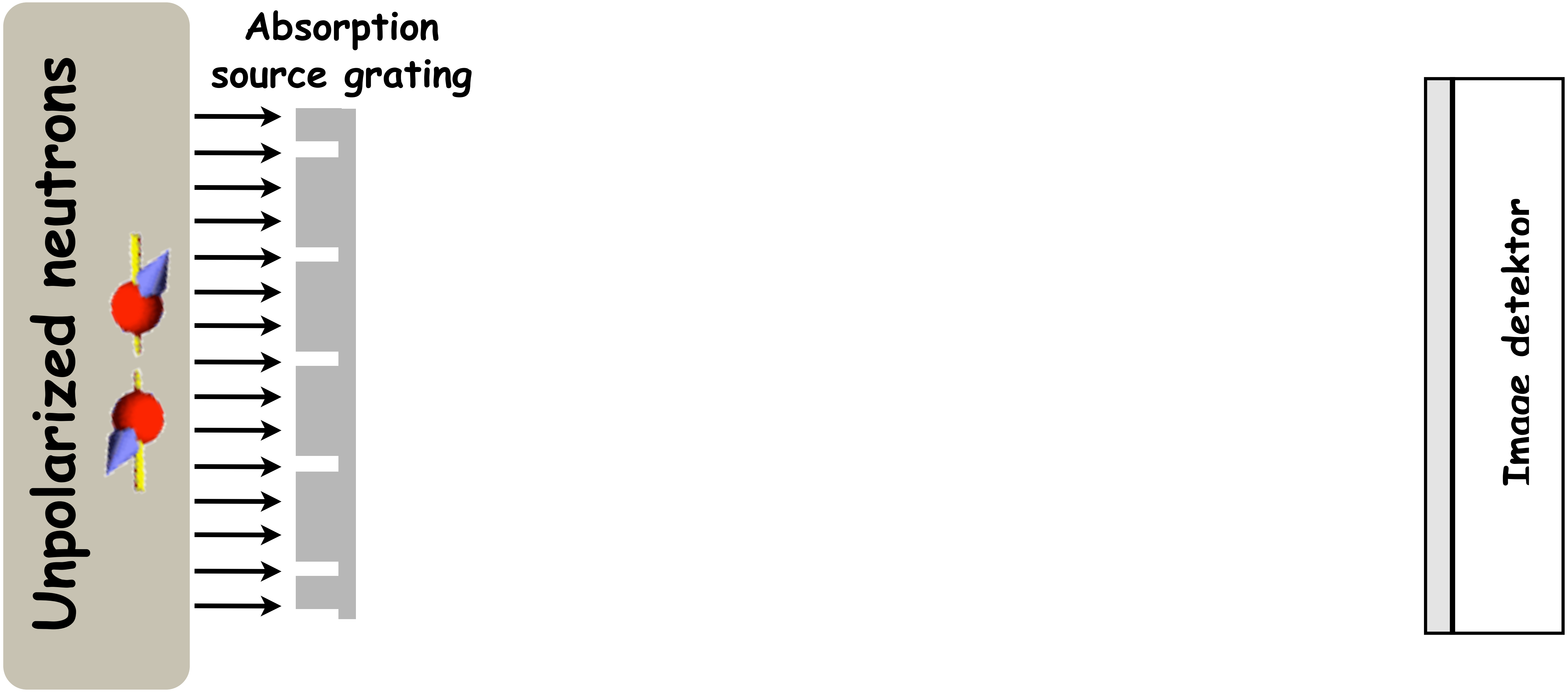


(Courtesy J. Miltat)

8.2 Neutron Dark-Field Microscopy



C. Grünzweig, et al., Phys. Rev. Lett. 101, 025504 (2008)



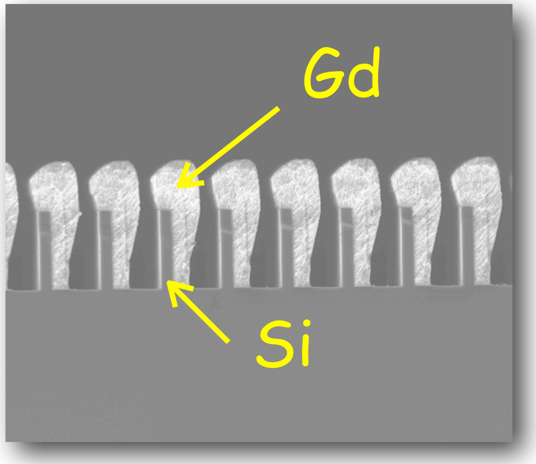
Unpolarized neutrons

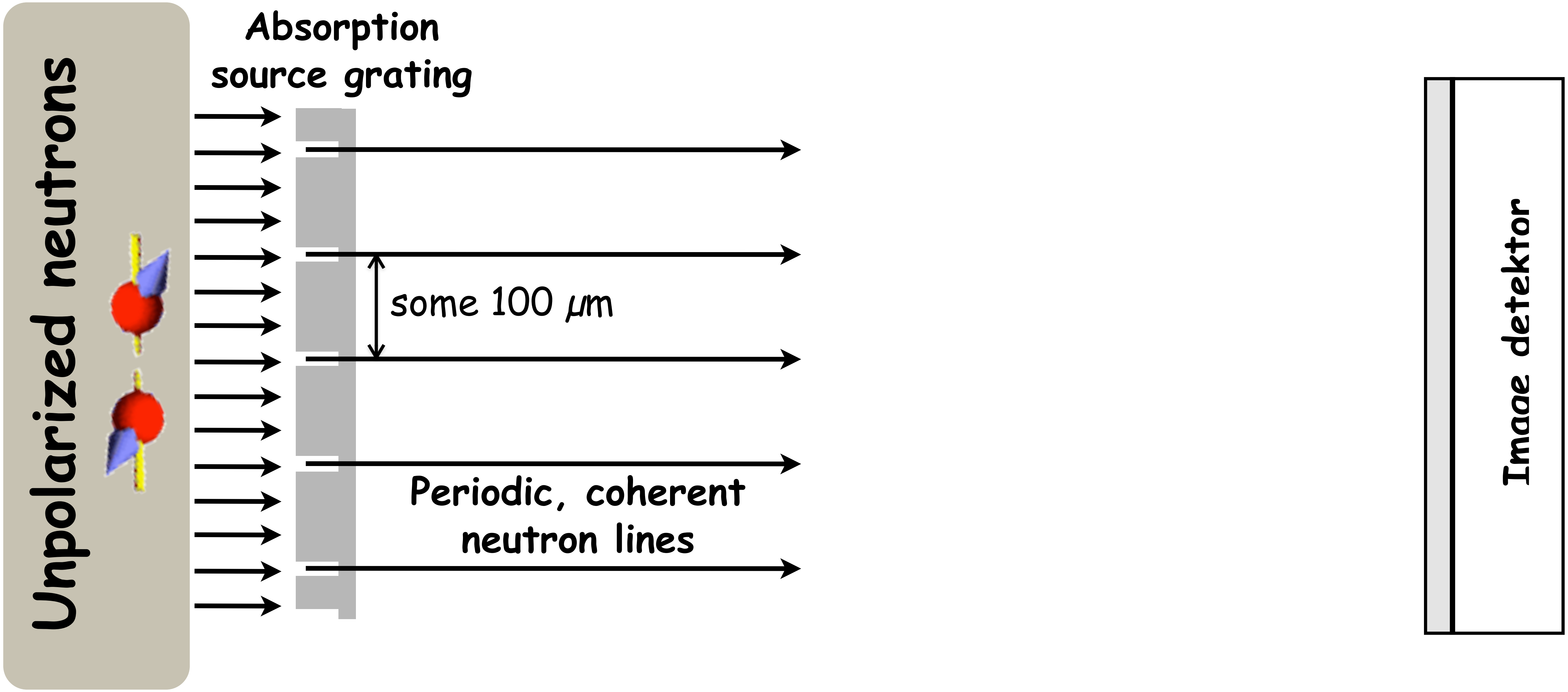
Absorption source grating

Image detektor

8.2 Neutron Dark-Field Microscopy

C. Grünzweig, et al.,
Phys. Rev. Lett. 101,
025504 (2008)





Unpolarized neutrons

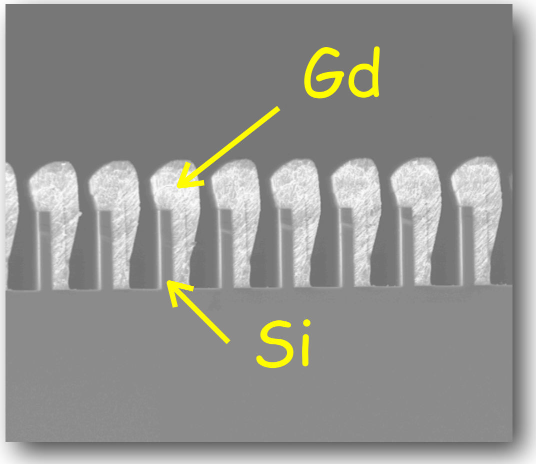
Absorption
source grating

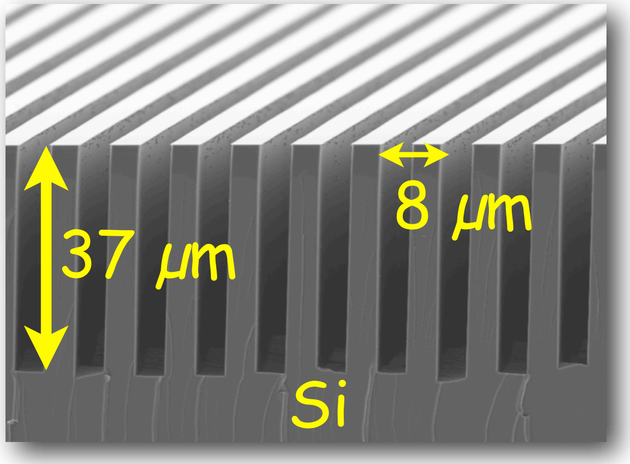
some 100 μm

Periodic, coherent
neutron lines

Image detektor

8.2 Neutron Dark-Field Microscopy





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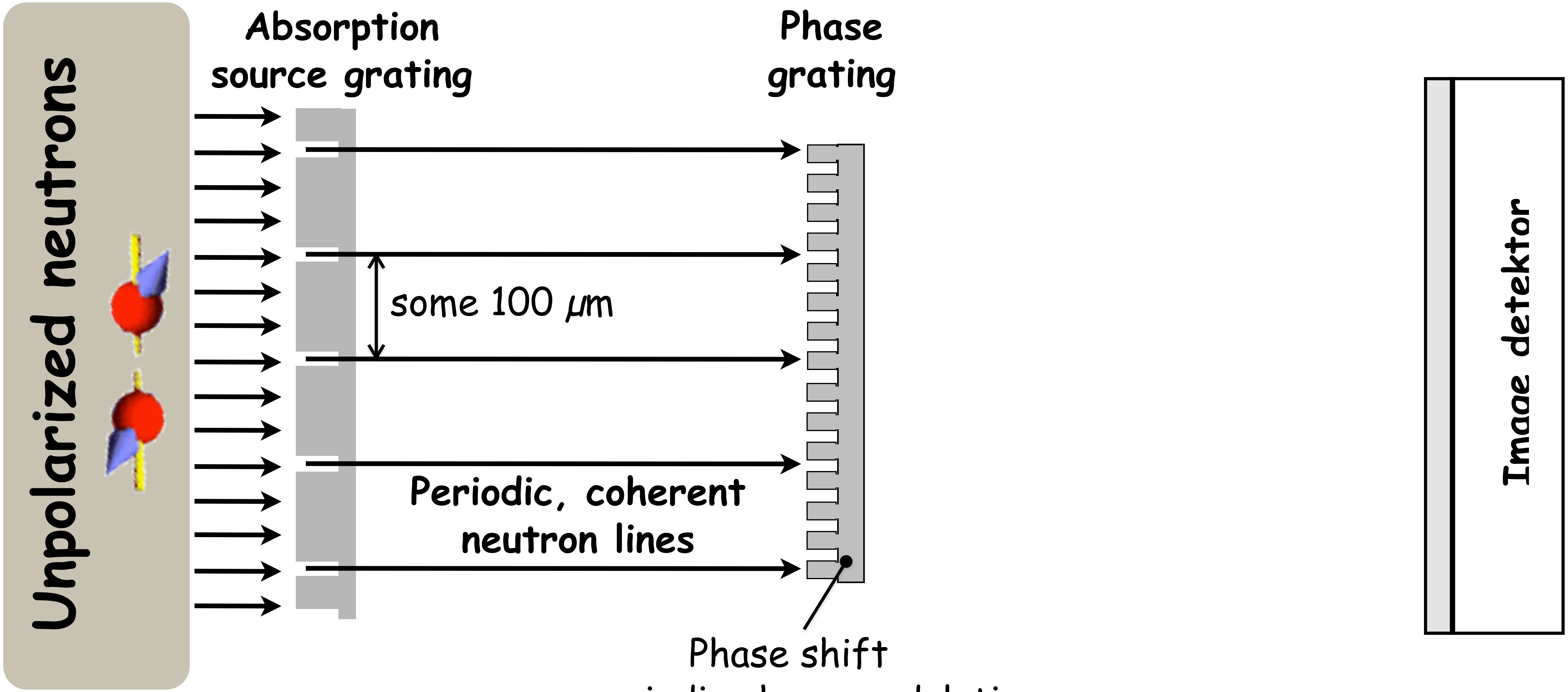


Unpolarized neutrons

Absorption source grating

Phase grating

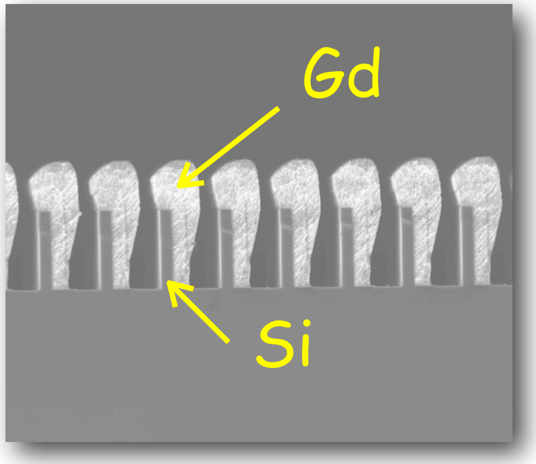
some 100 µm

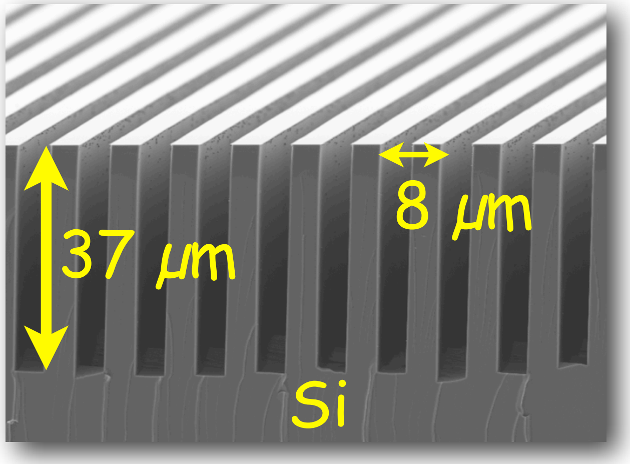
Periodic, coherent neutron lines

Image detektor

Phase shift
→ periodic phase modulation
→ Talbot effect

8.2 Neutron Dark-Field Microscopy





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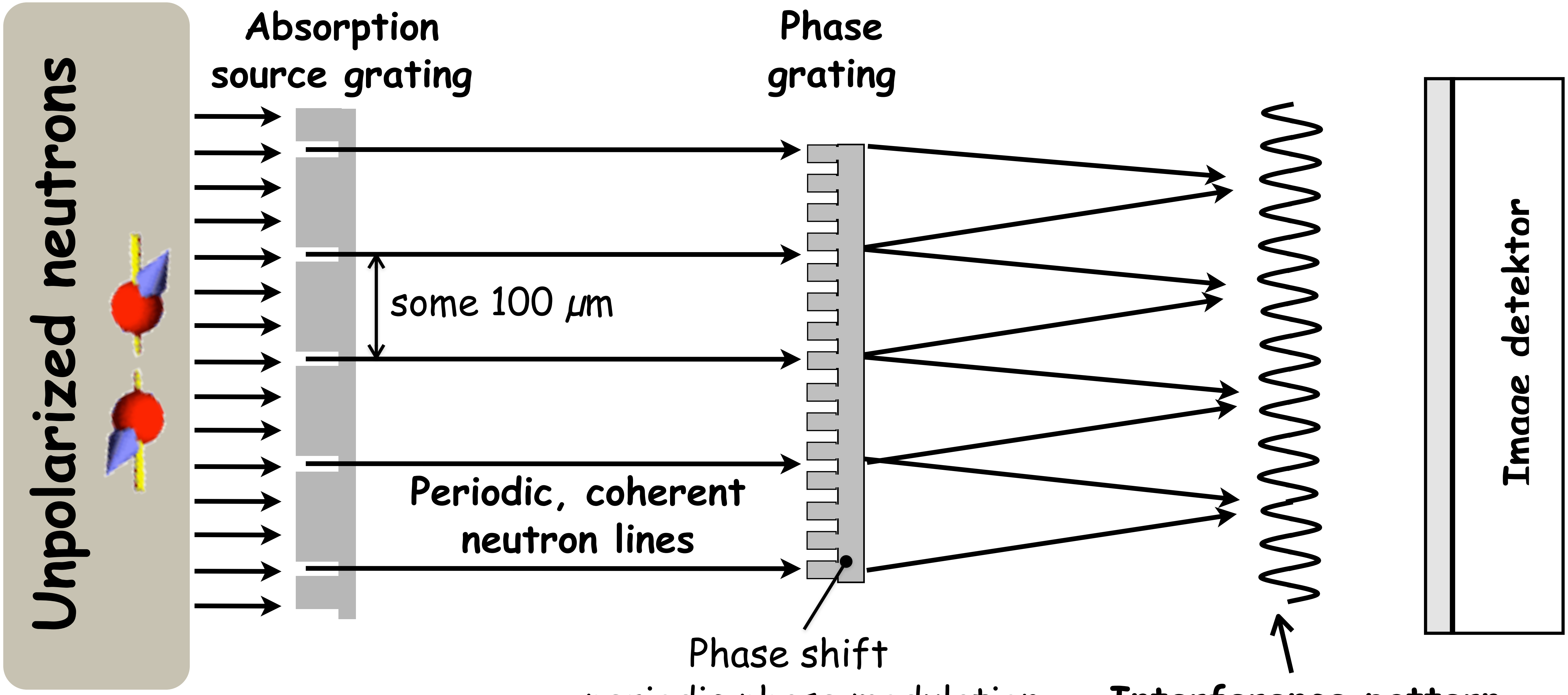
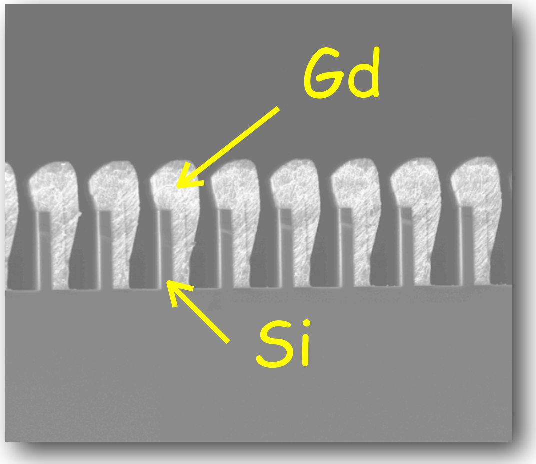


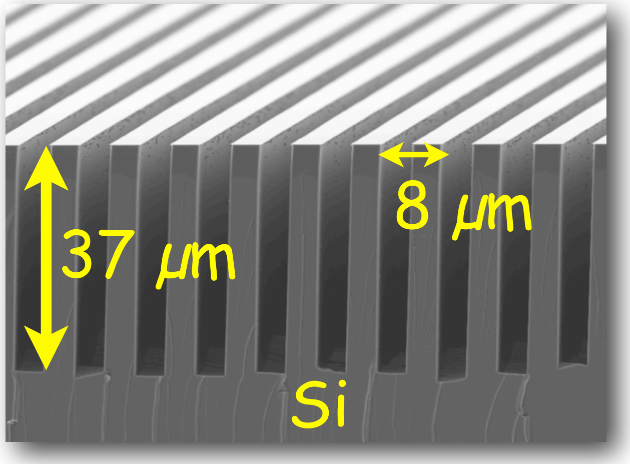
Periodic, coherent neutron lines

Phase shift
→ periodic phase modulation
→ Talbot effect

Interference pattern (period 4 µm)

8.2 Neutron Dark-Field Microscopy





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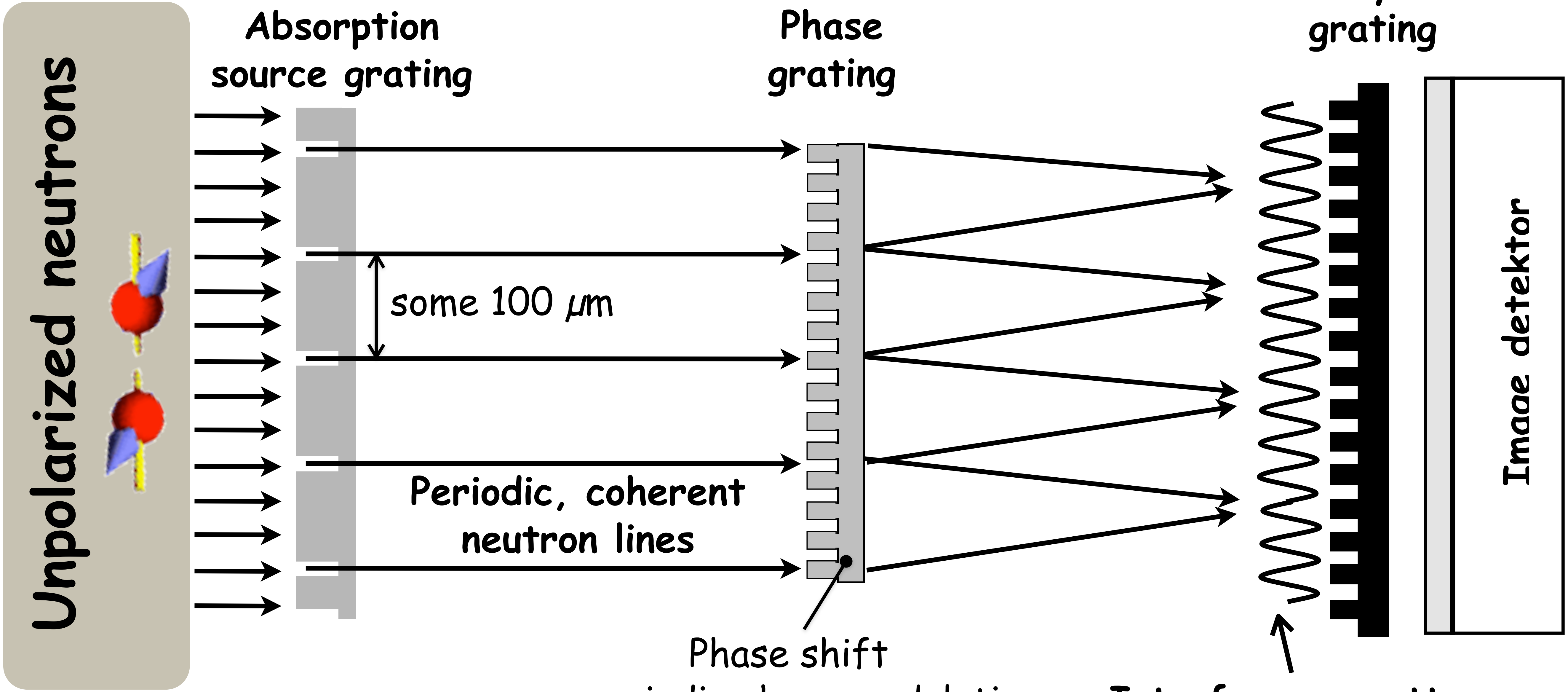

Unpolarized neutrons

Absorption source grating

Phase grating

Absorption analyser grating

Image detektor

some 100 μm

Periodic, coherent neutron lines

Phase shift
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Interference pattern (period 4 μm)

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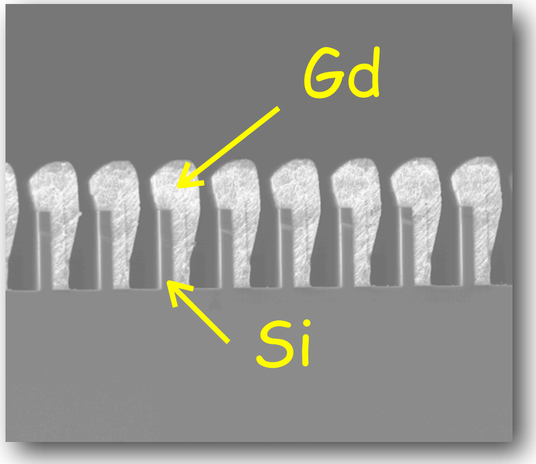

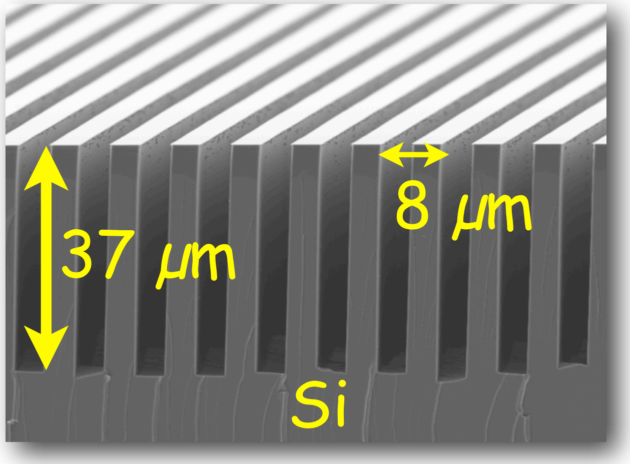

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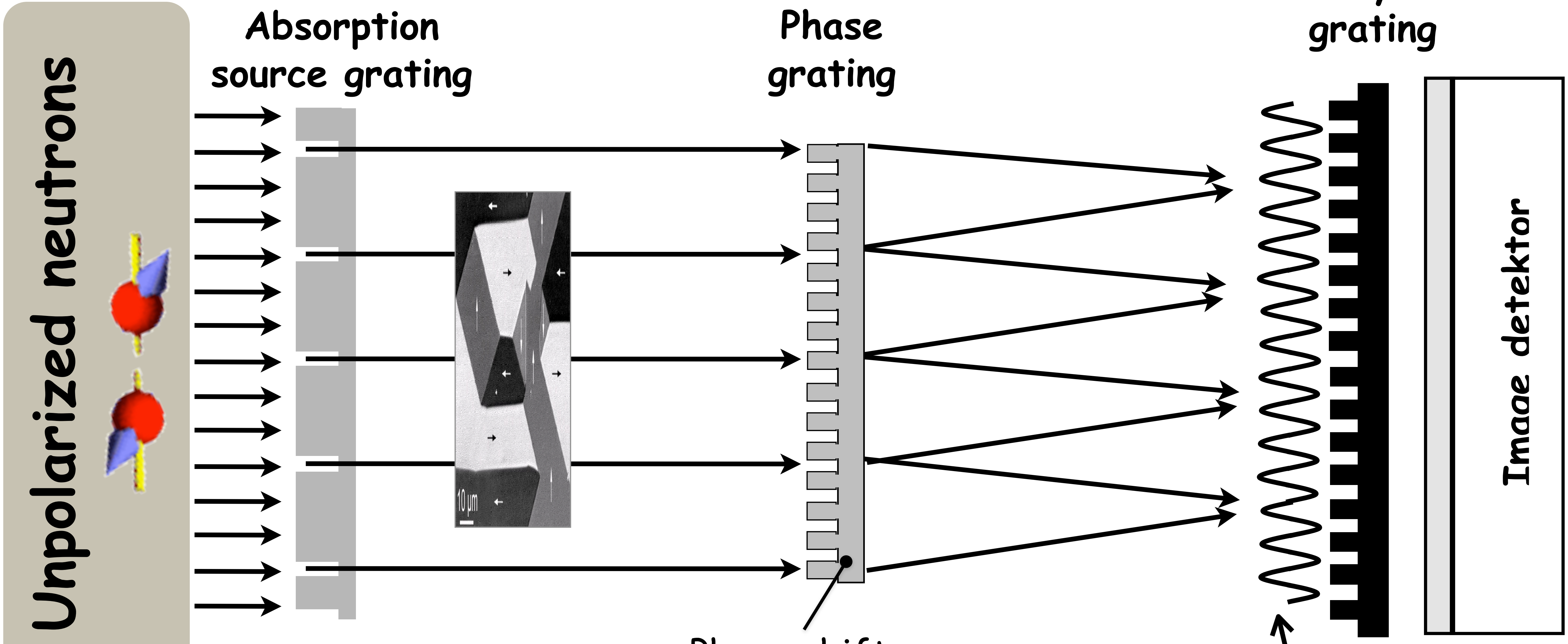

Absorption source grating

Phase grating

Absorption analyser grating

Phase shift
→ periodic phase modulation
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Interference pattern (period 4 μm)

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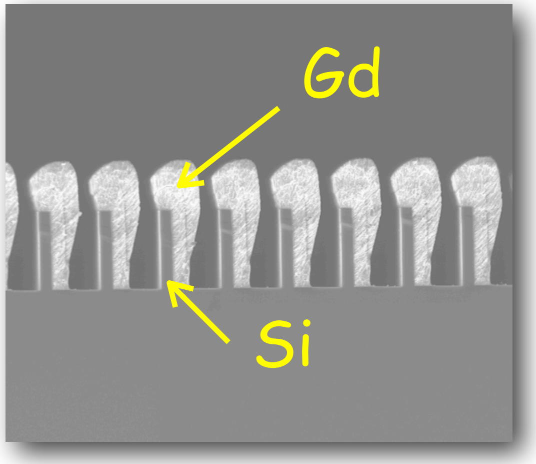

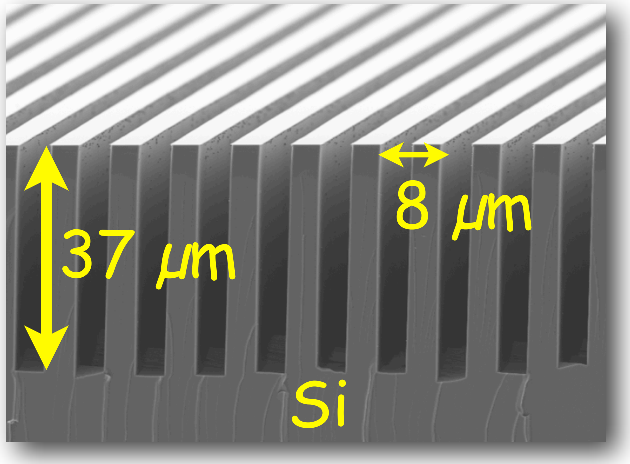

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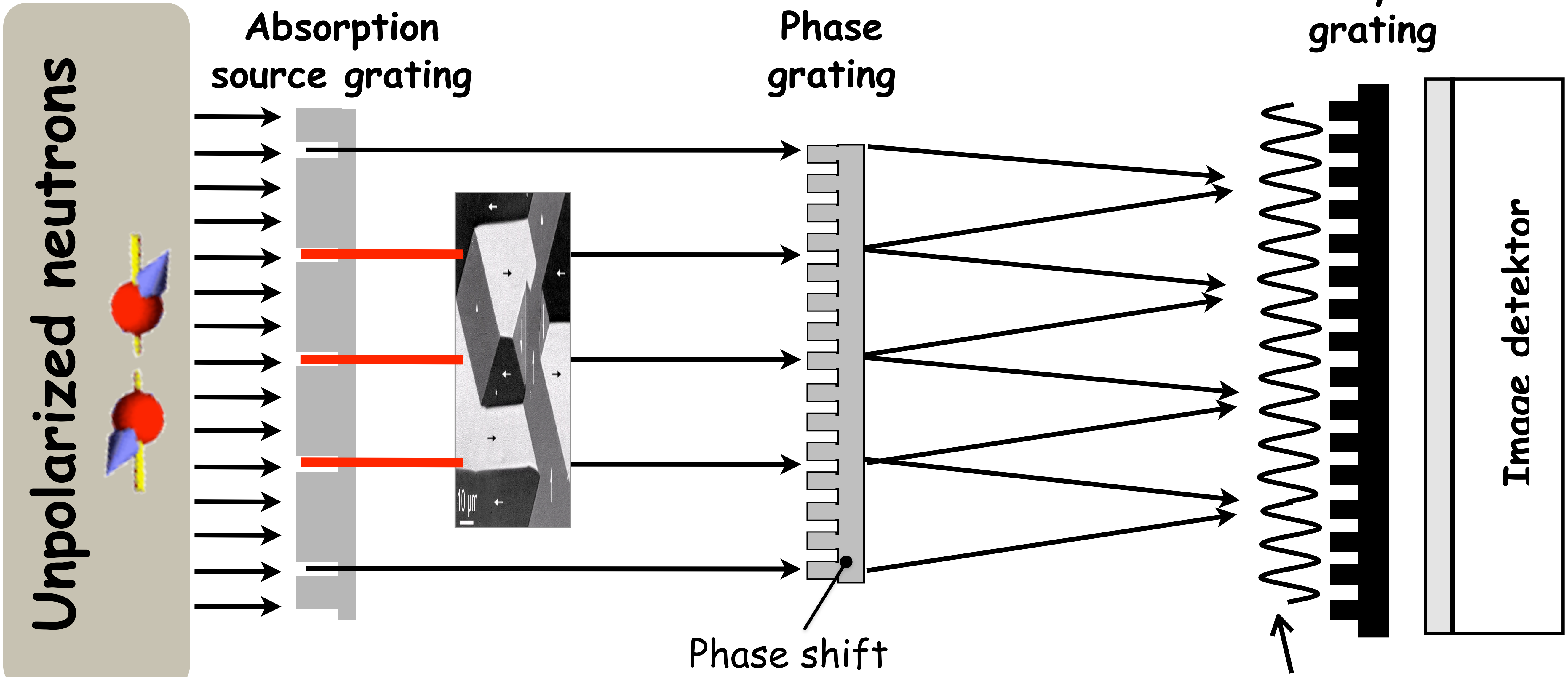

Unpolarized neutrons

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Interference pattern
(period 4 µm)

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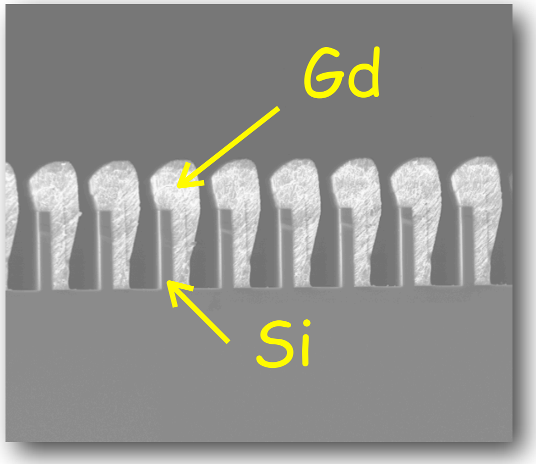

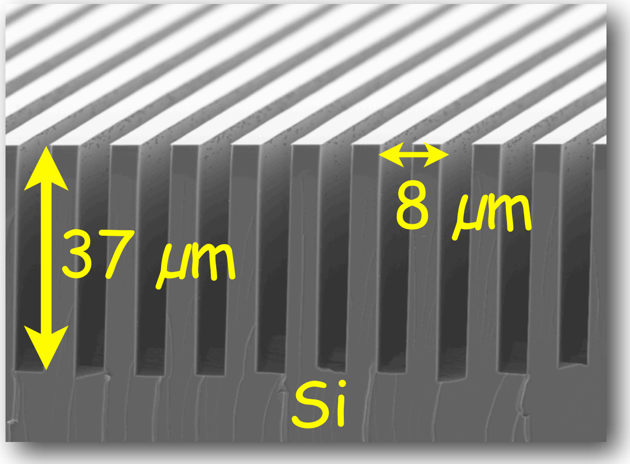

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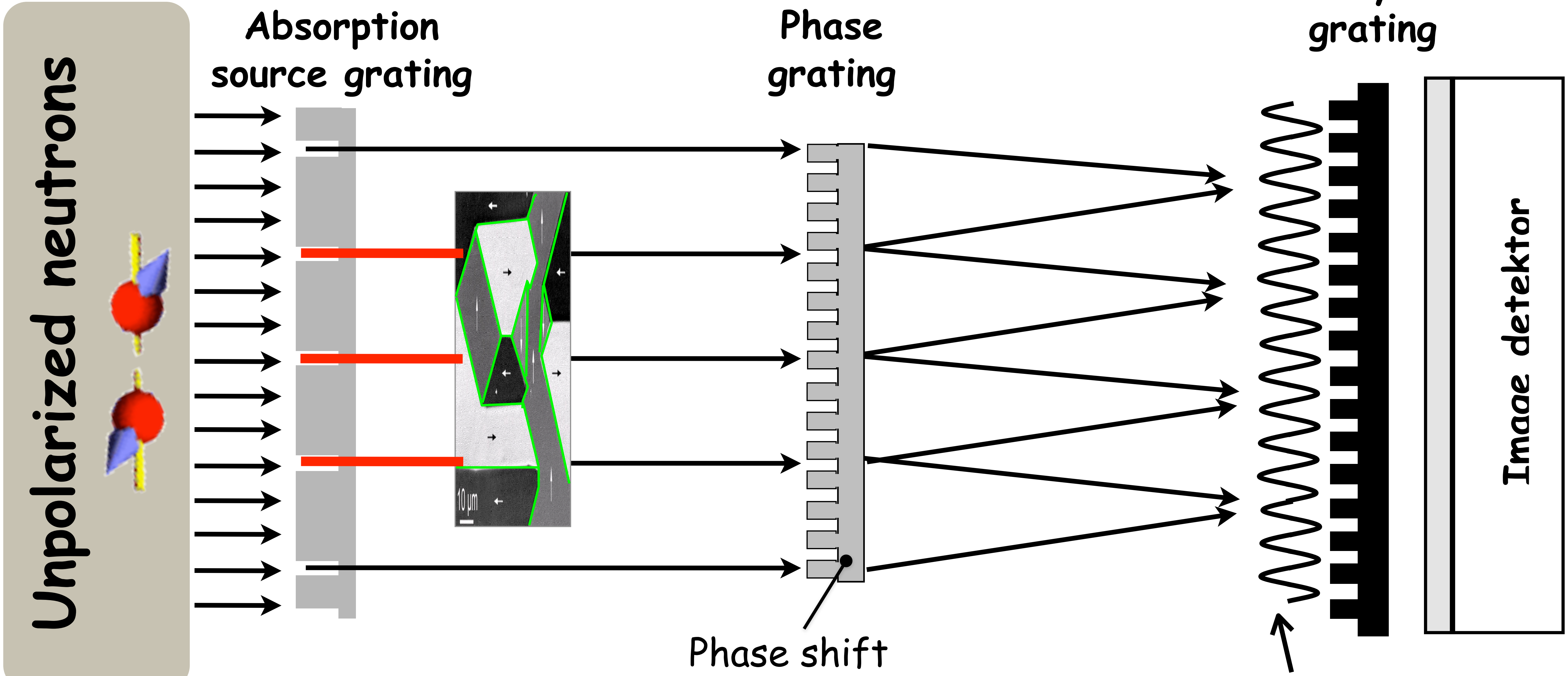

Absorption source grating

Phase grating

Absorption analyser grating

Phase shift
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Interference pattern (period 4 µm)

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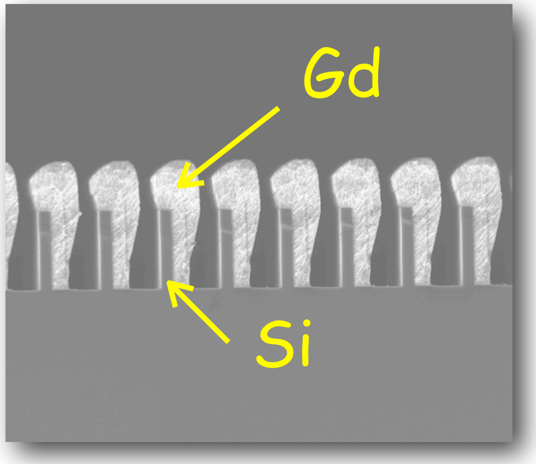

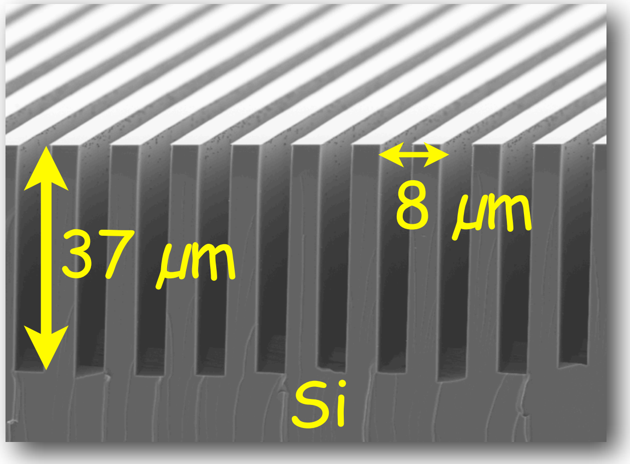

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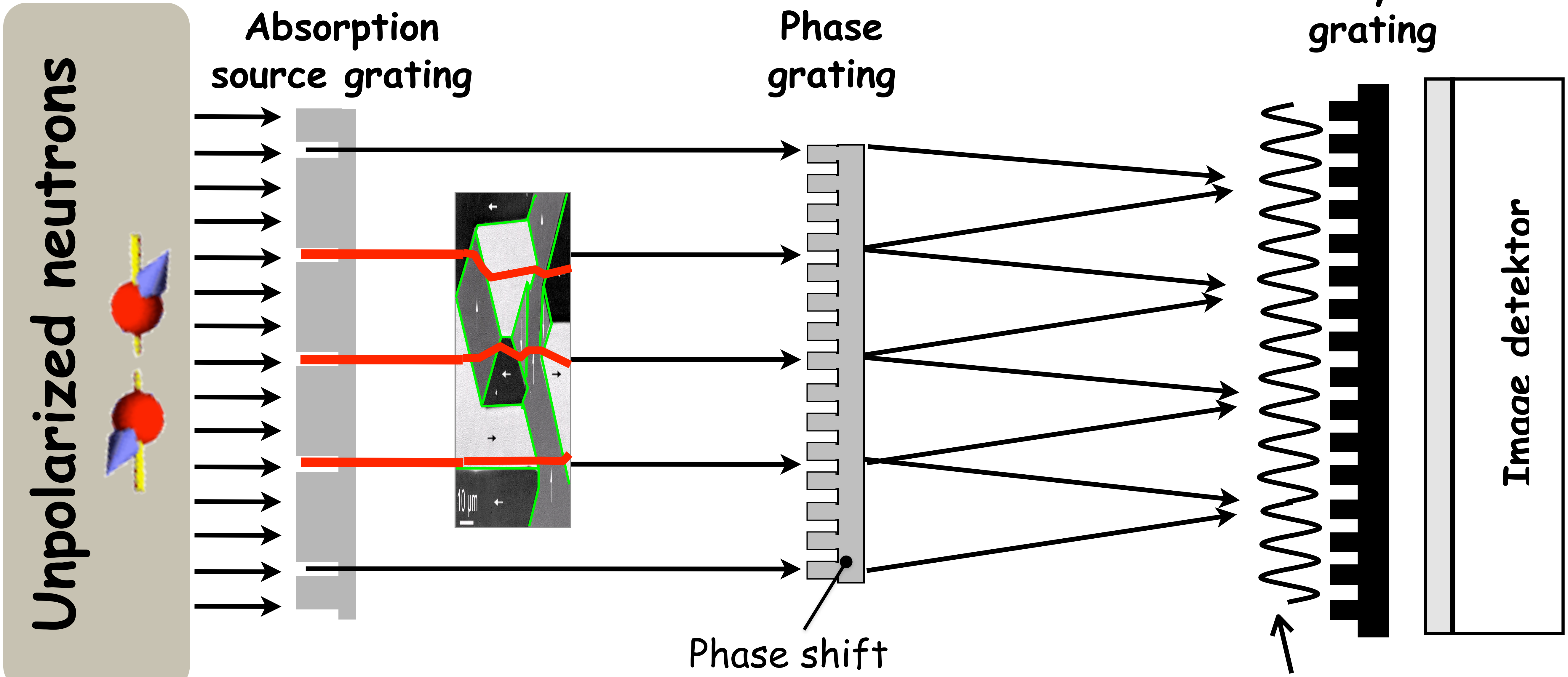

Unpolarized neutrons

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Phase grating

Absorption analyser grating

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Interference pattern (period 4 µm)

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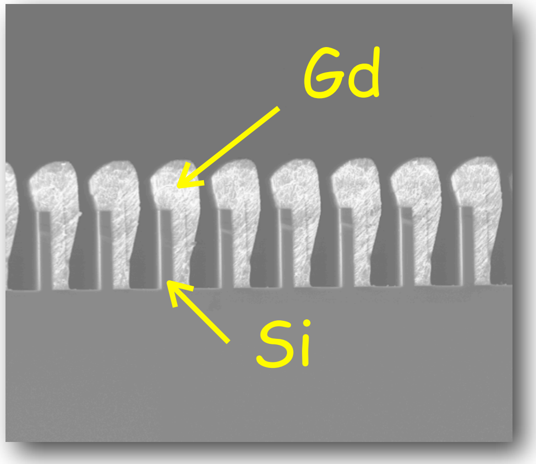

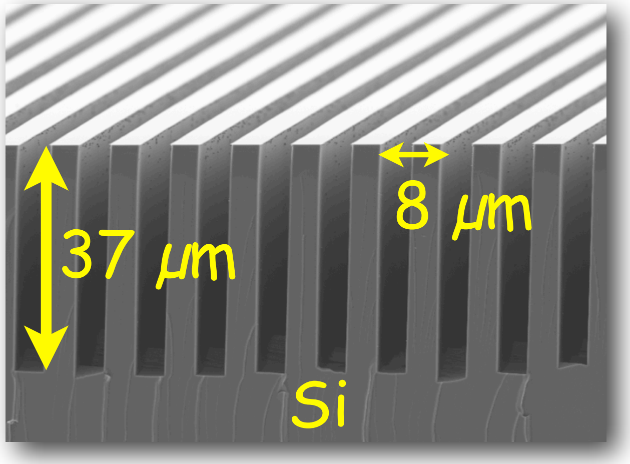

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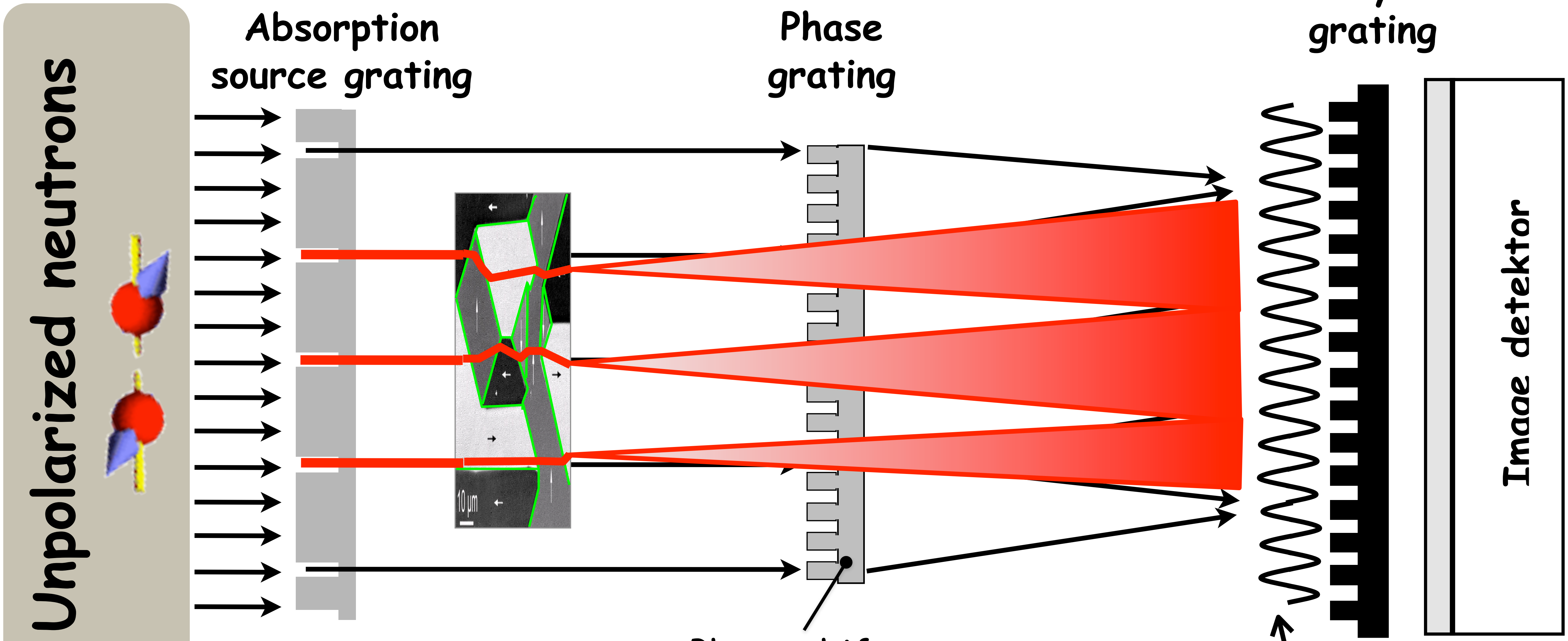

Unpolarized neutrons

Absorption source grating

Phase grating

Absorption analyser grating

Image detektor

Phase shift
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Interference pattern (period 4 µm)

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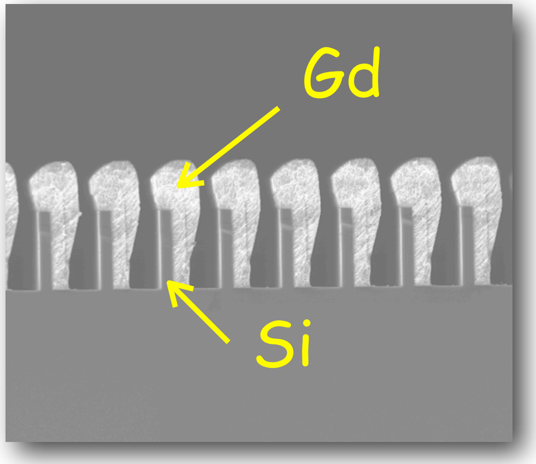

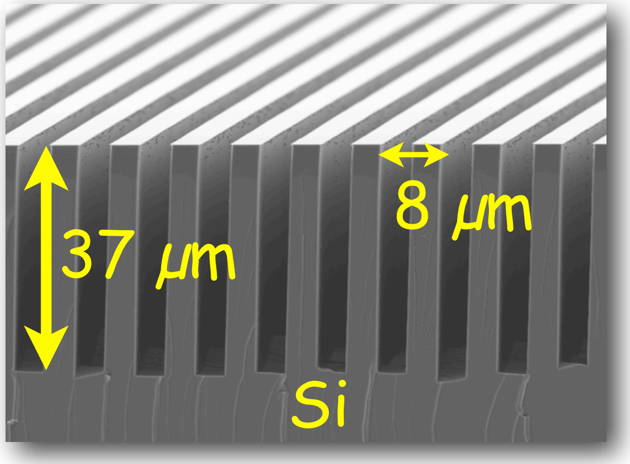

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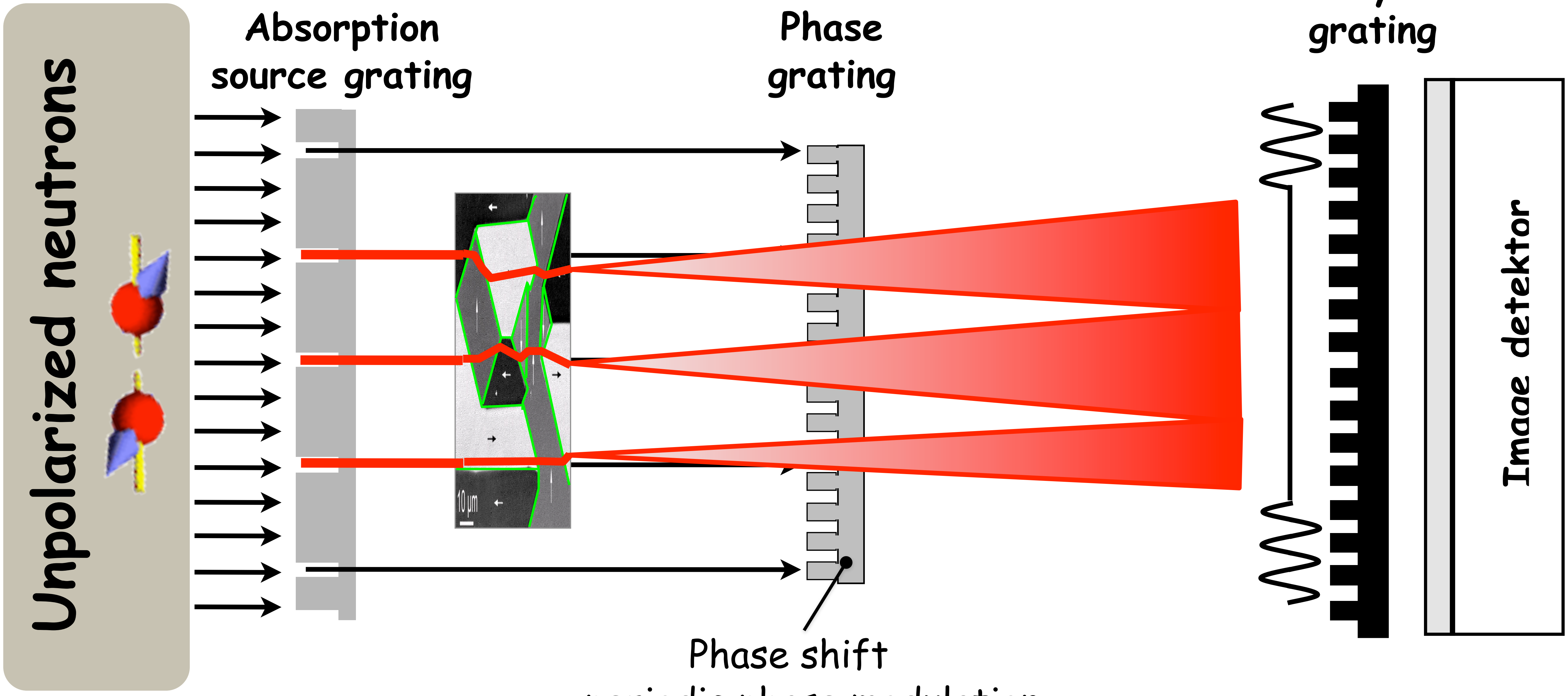


Unpolarized neutrons

Absorption source grating

Phase grating

Absorption analyser grating

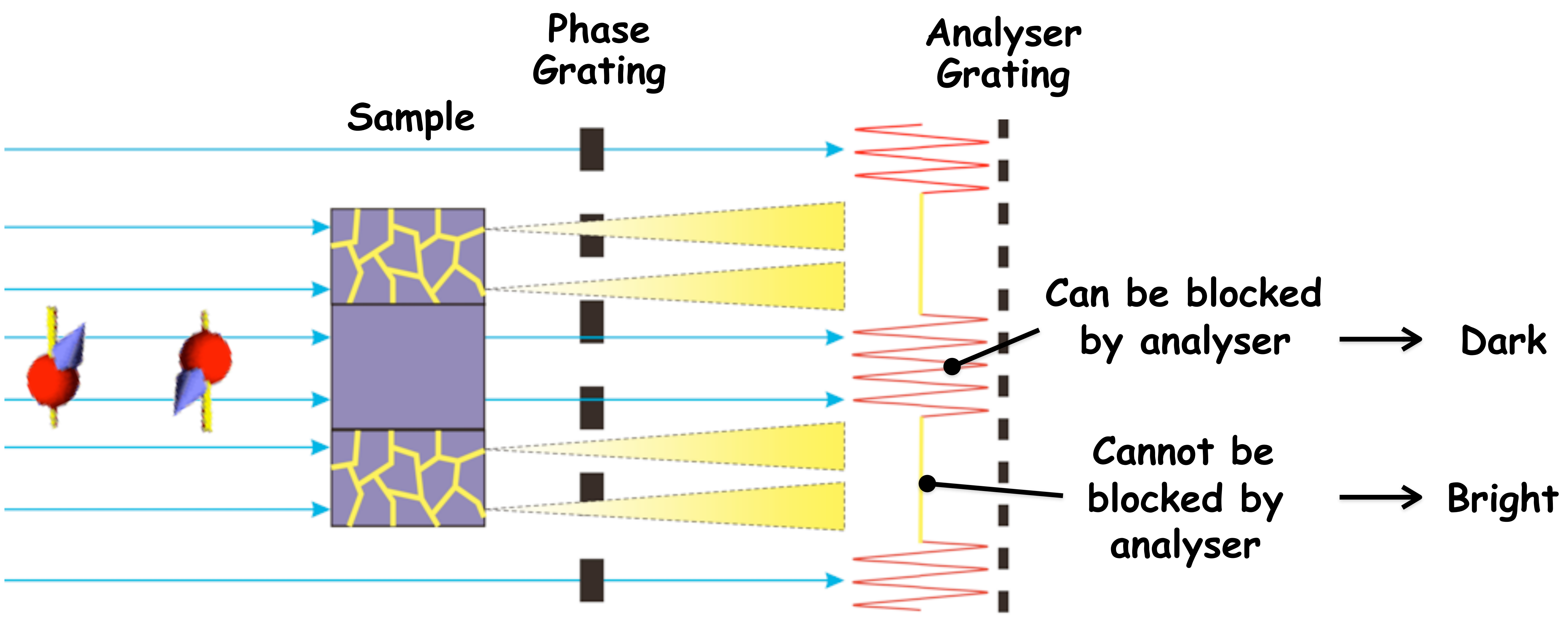
Image detektor

Phase shift
→ periodic phase modulation
→ Talbot effect

8.2 Neutron Dark-Field Microscopy

Principle:

Unpolarized neutrons, refracted at domain walls, locally destroy the interference pattern



→ Bright contrast is caused by refraction

→ Dark field imaging

8.2 Neutron Dark-Field Microscopy

Unpolar … walls,

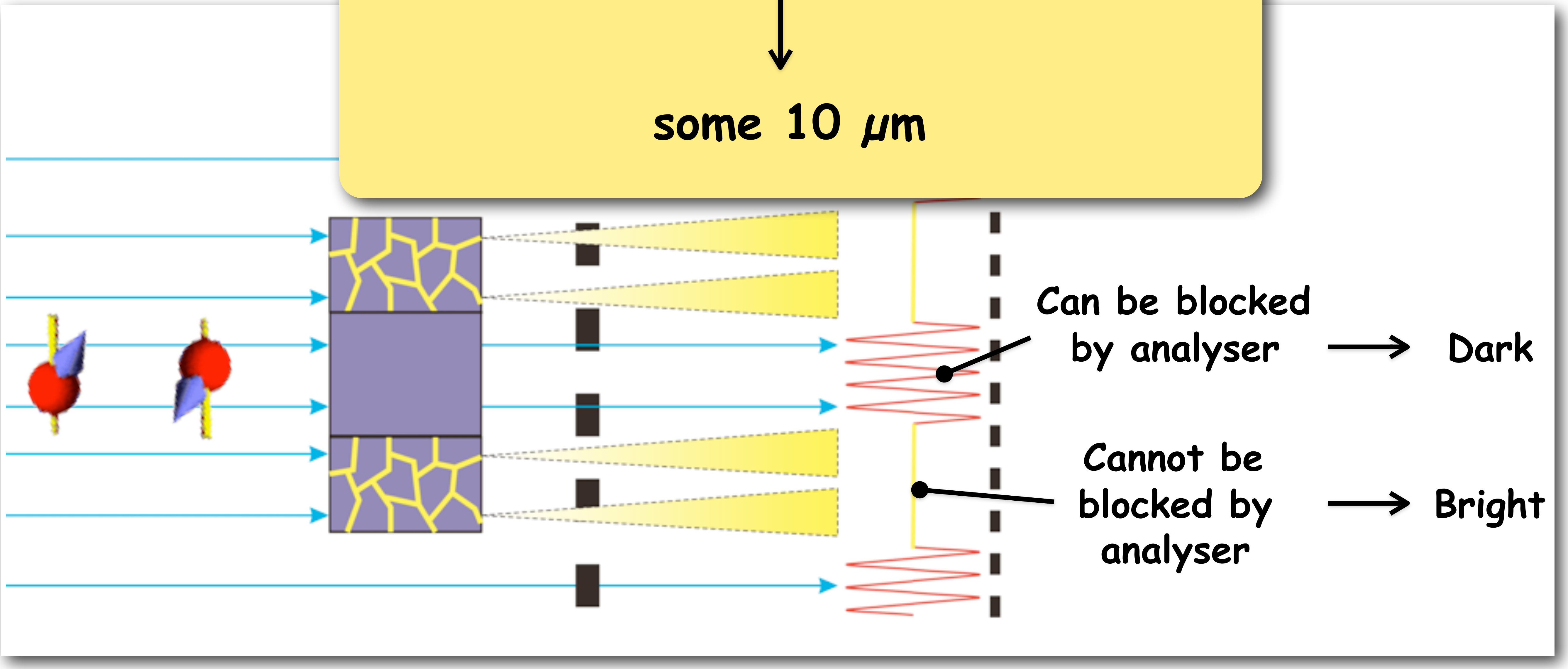
loc … rn

Resolution:

Determined by pixel size of detector and scattering angle

↓

some 10 μm



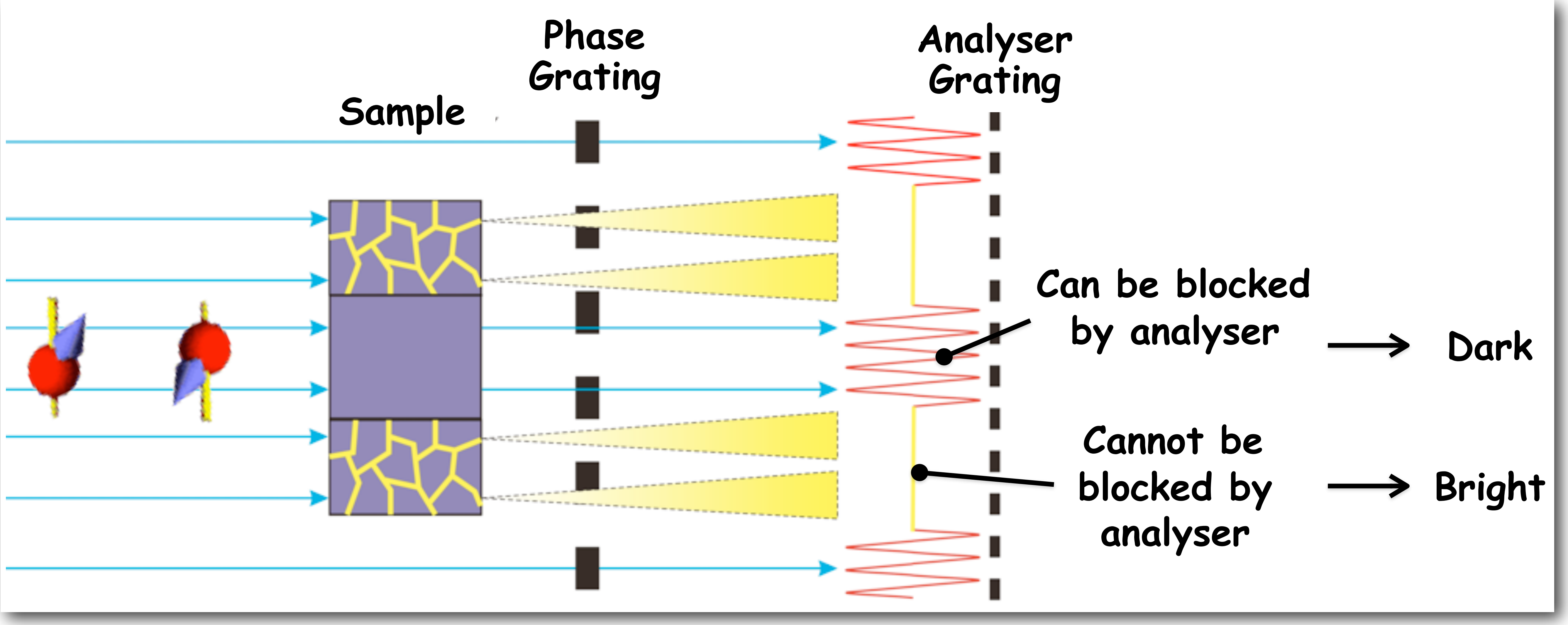
Can be blocked by analyser → Dark

Cannot be blocked by analyser → Bright

→ Bright contrast is caused by refraction

→ Dark field imaging

8.2 Neutron Dark-Field Microscopy



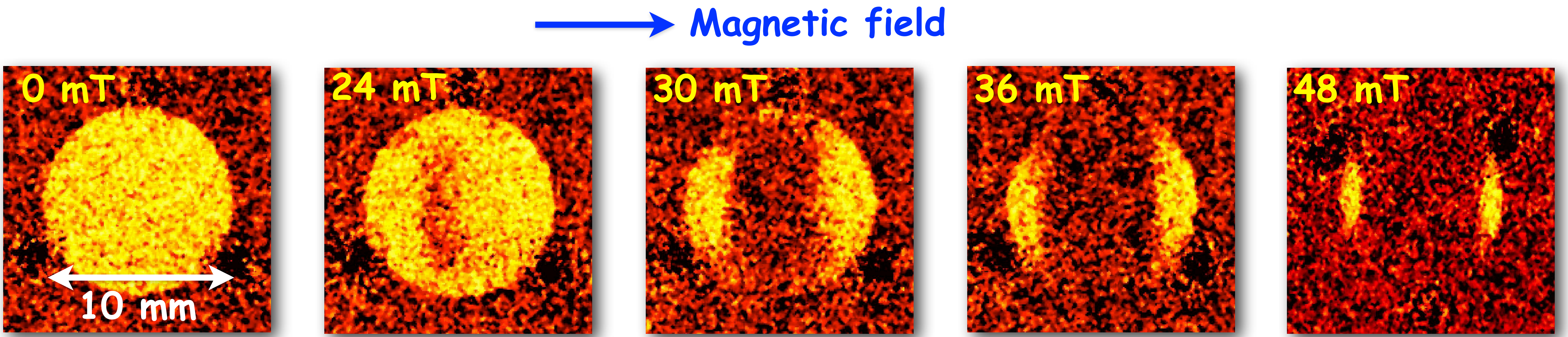
Sample

Phase Grating

Analyser Grating

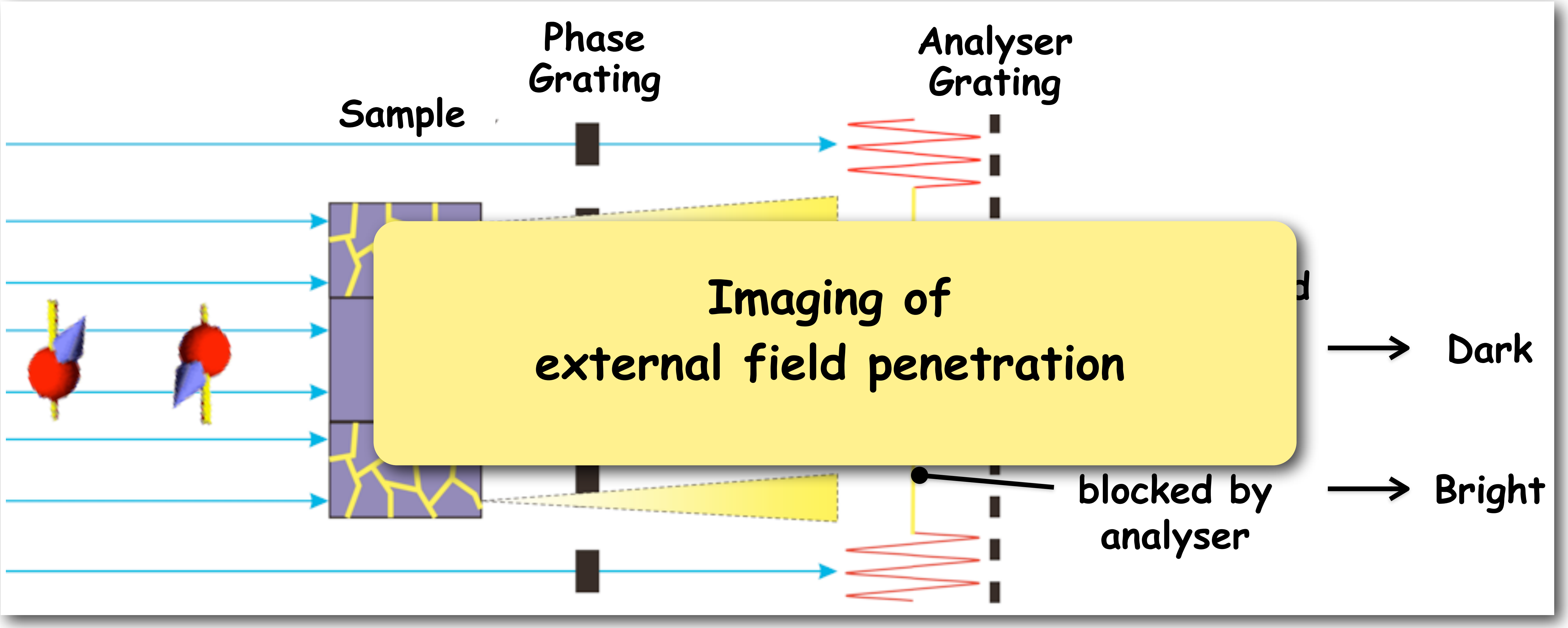
Can be blocked by analyser → Dark

Cannot be blocked by analyser → Bright

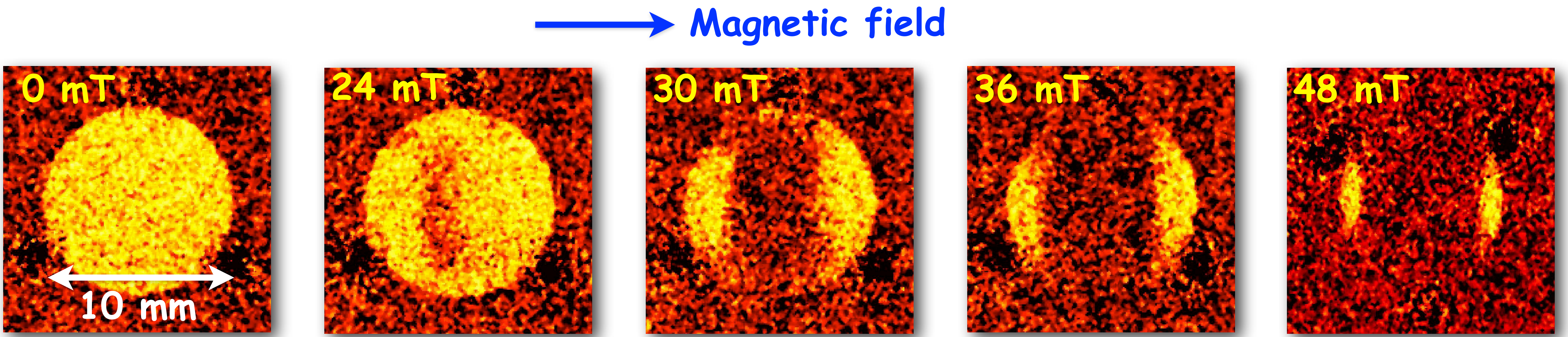


Steel plate,
2 mm thick

8.2 Neutron Dark-Field Microscopy



Imaging of
external field penetration

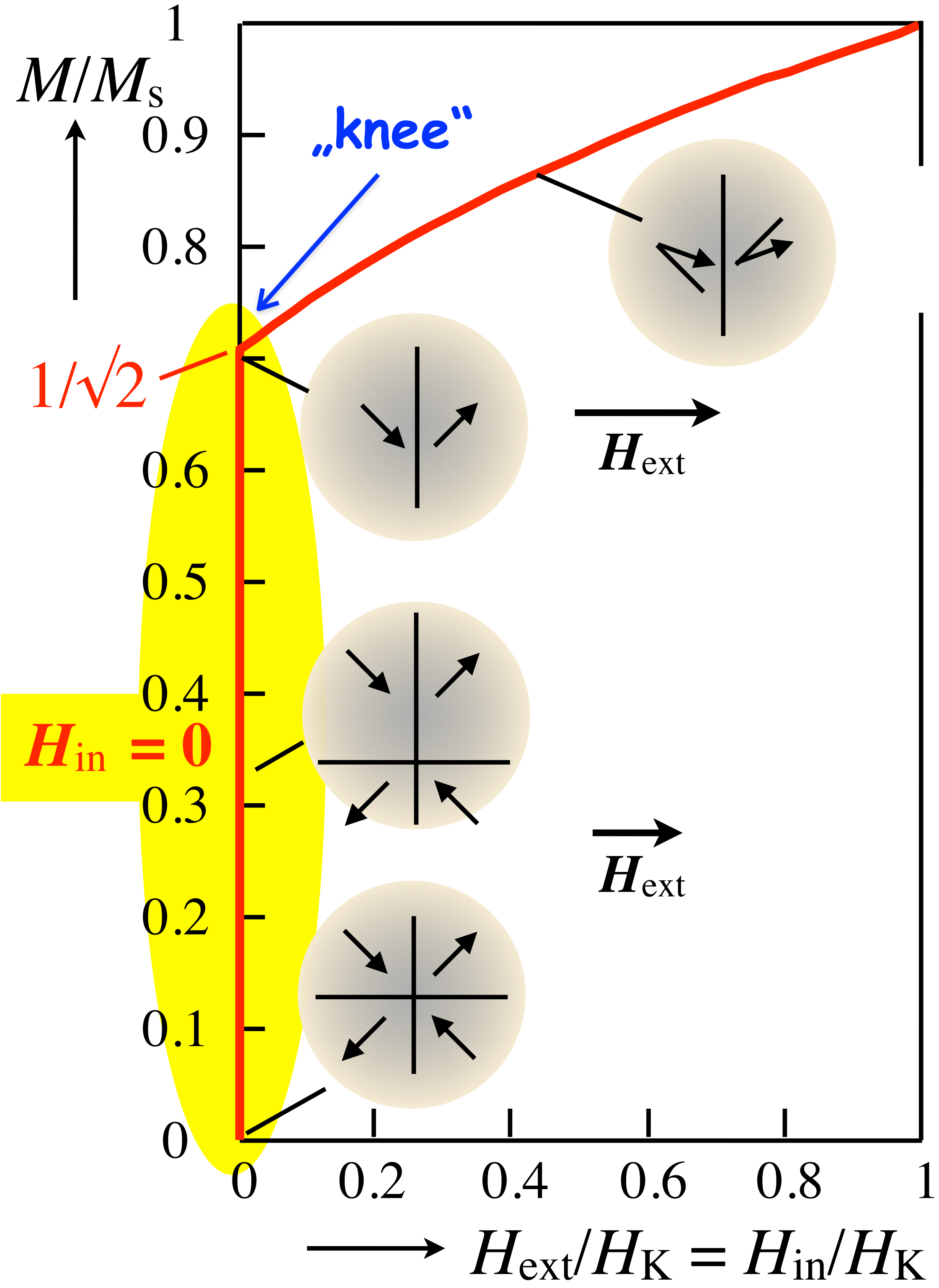


Steel plate,
2 mm thick

Phase Theory and M(H) curve

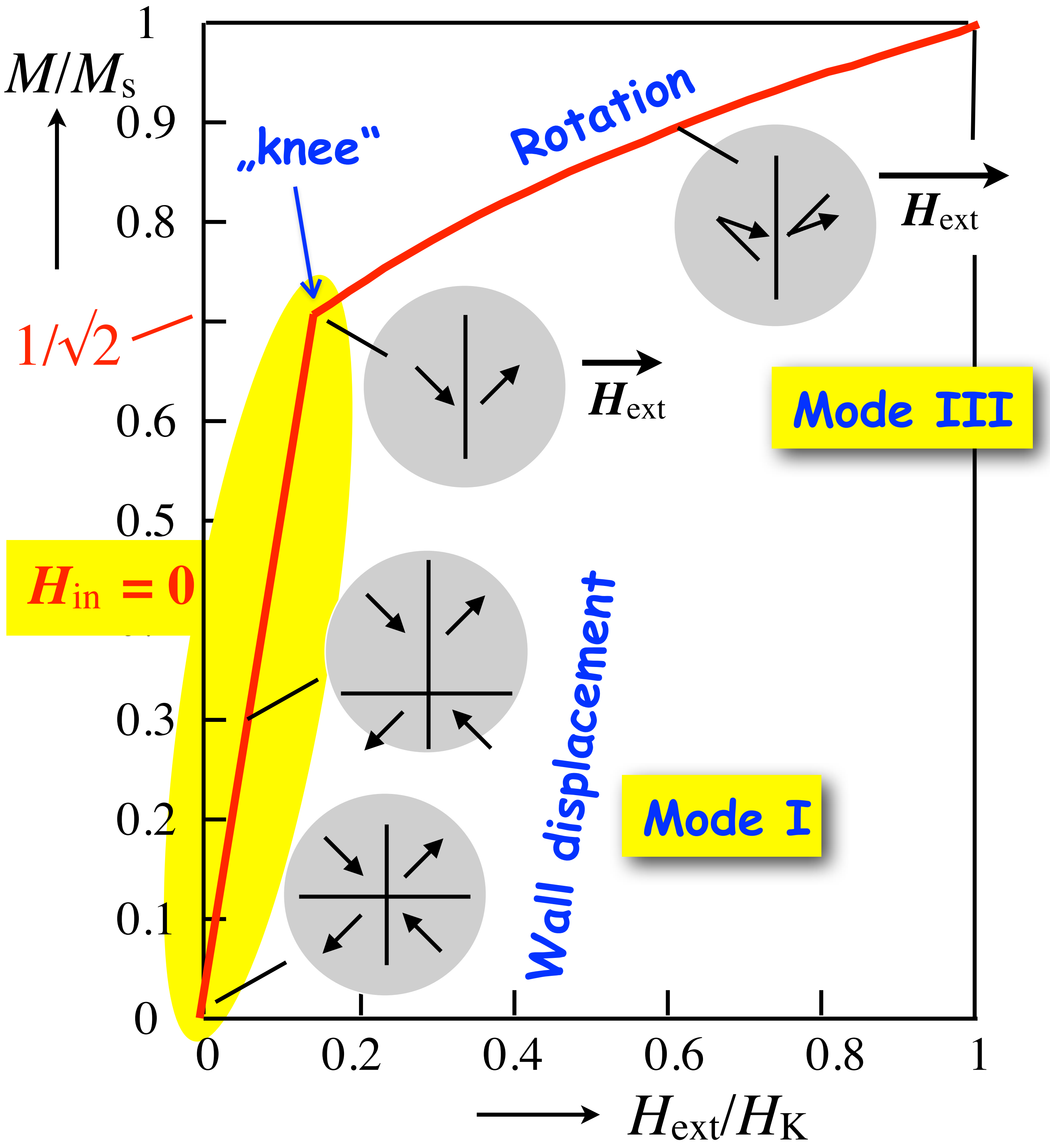
Infinite or closed sample

$N = 0 \rightarrow H_{\text{in}} = H_{\text{ext}}$



Finite sample

$N \neq 0 \rightarrow H_{\text{ext}} = H_{\text{in}} + N \cdot \overline{M} \rightarrow$ shearing



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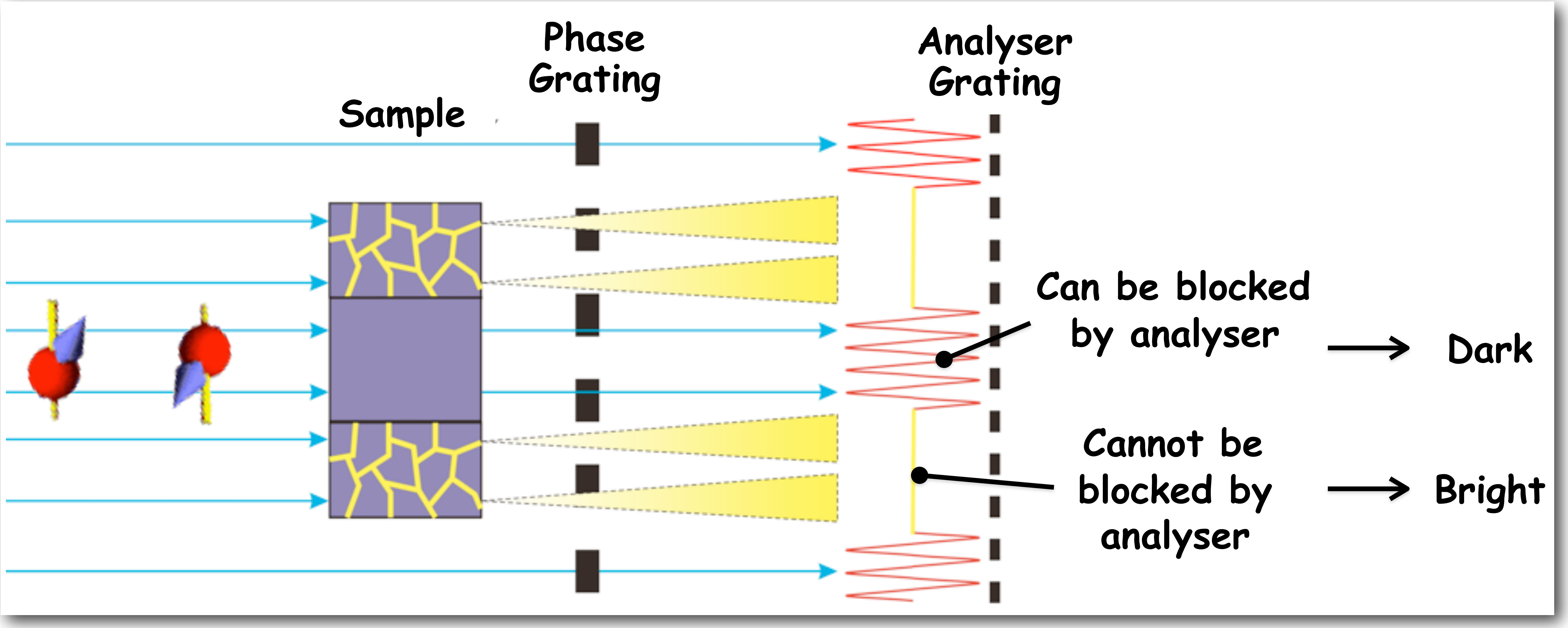


Phase Grating

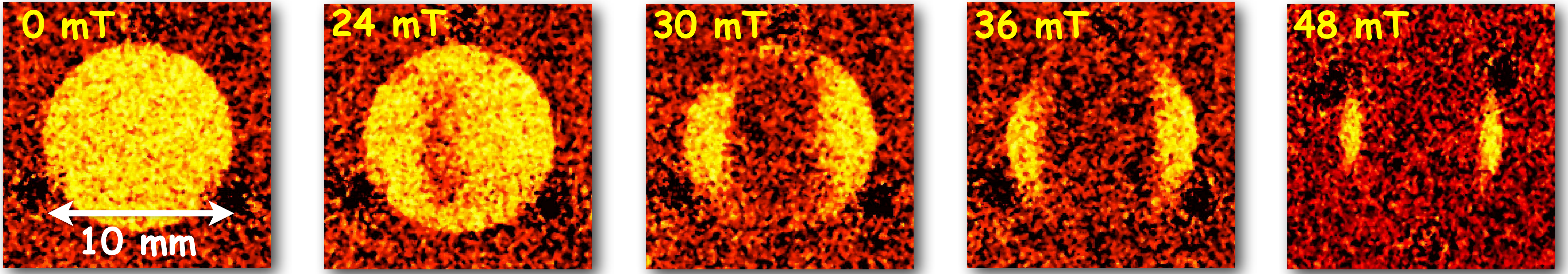
Analyser Grating

Sample

Can be blocked by analyser → Dark

Cannot be blocked by analyser → Bright

→ Magnetic field



Steel plate,
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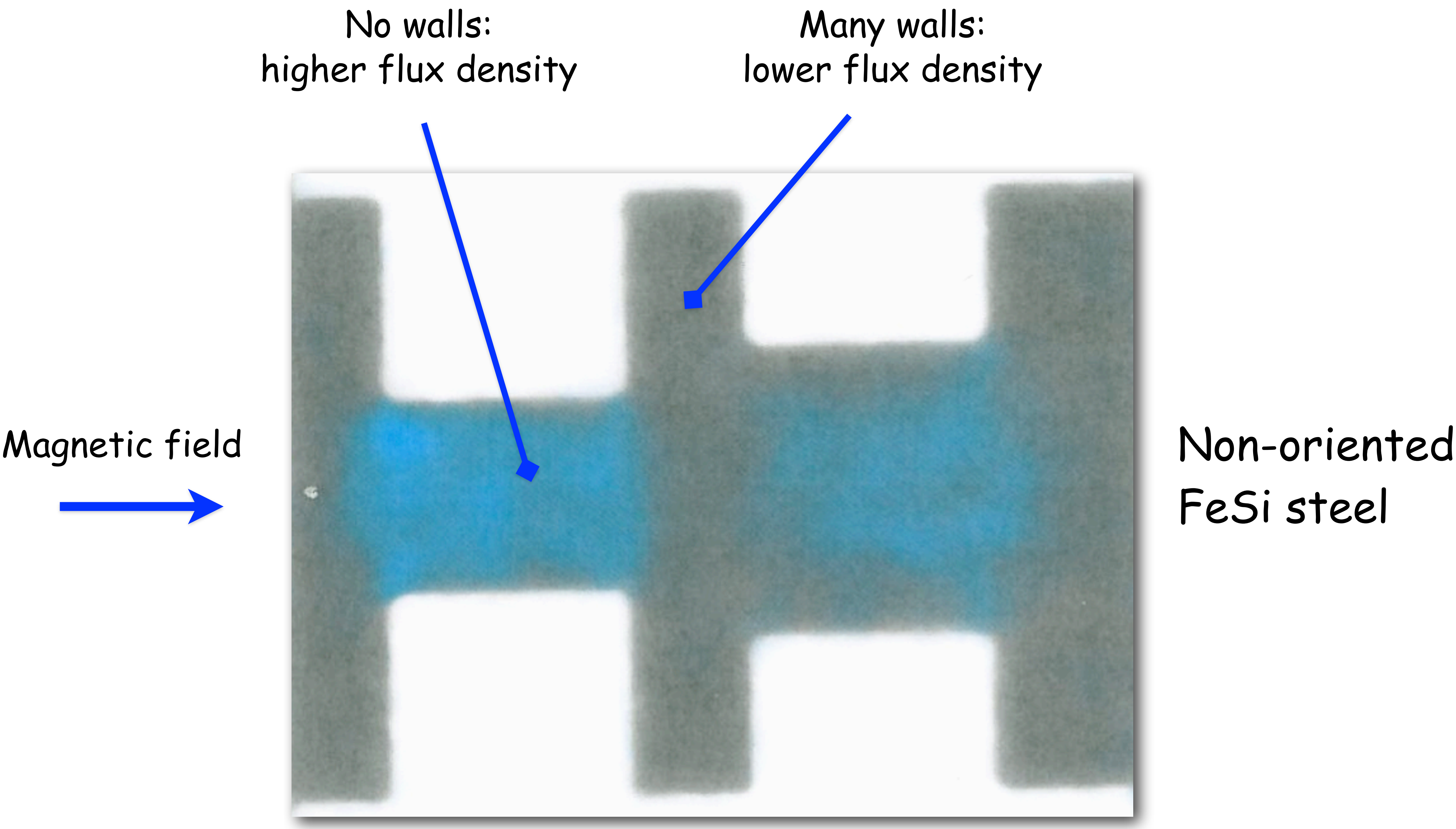
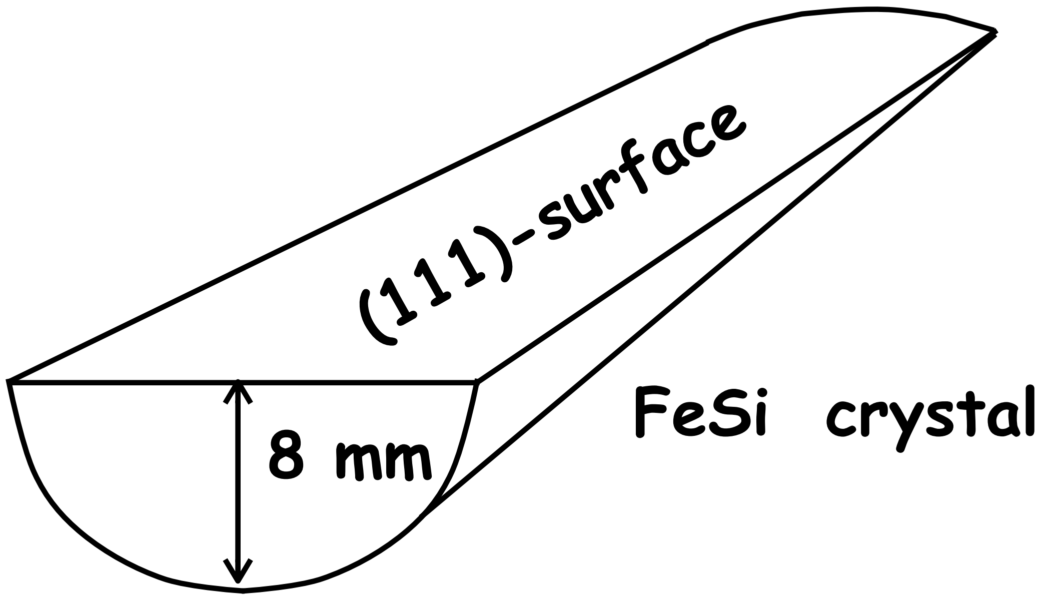
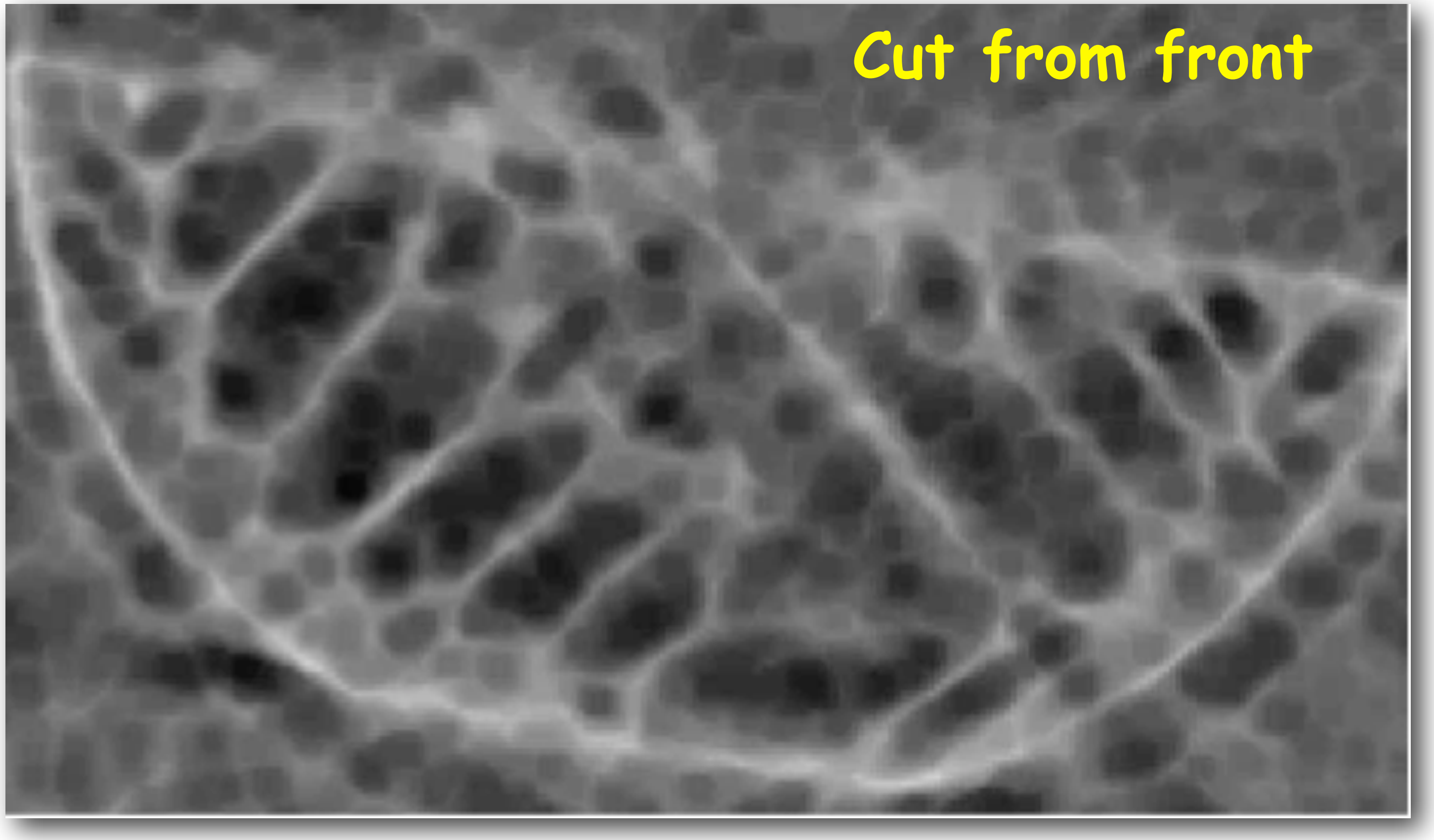
8.2 Neutron Dark-Field Microscopy

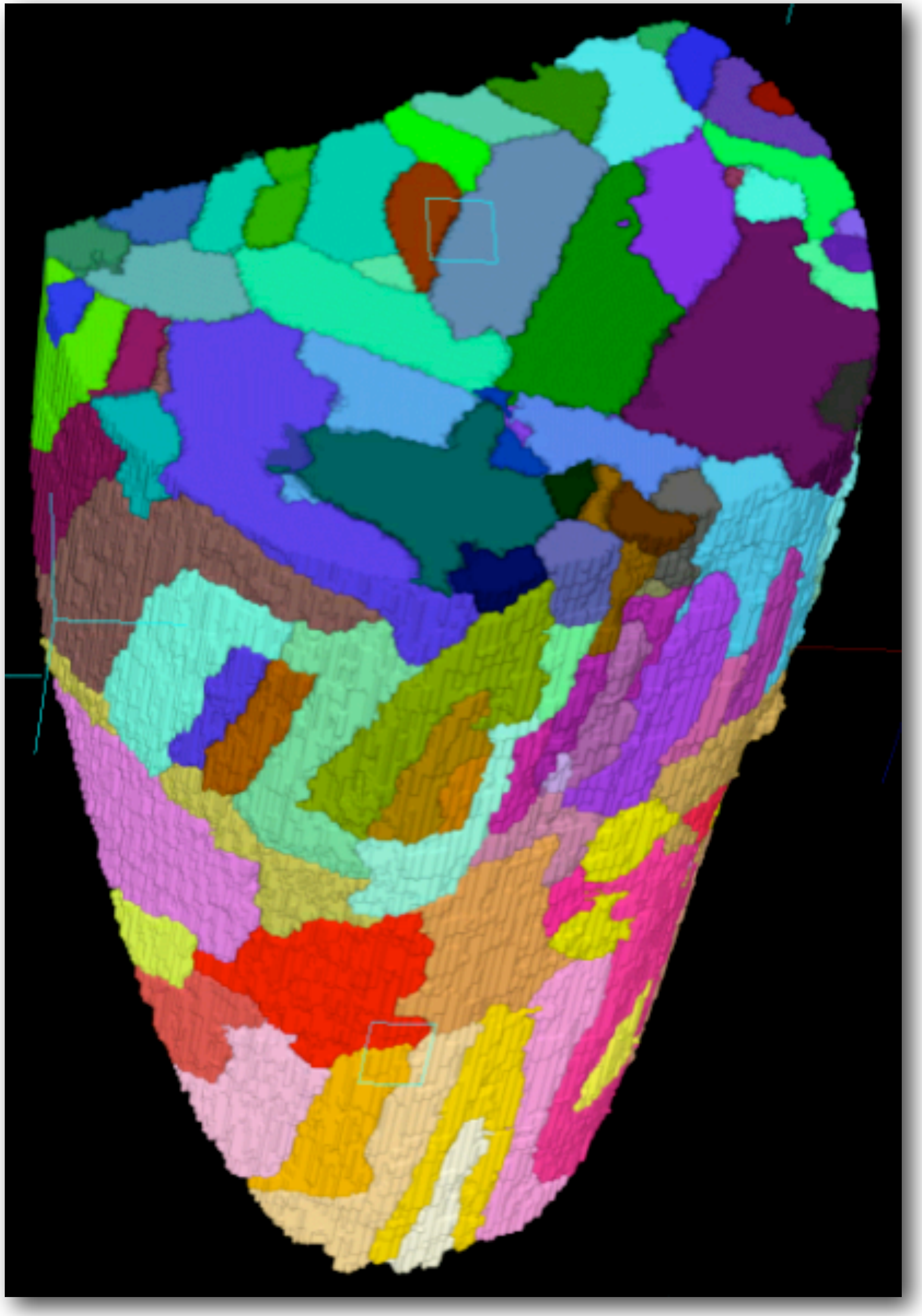


Together with B. Betz and C. Grünzweig (PSI Villigen),
R. Siebert (Fraunhofer IWS Dresden)
unpublished

8.2 Neutron Dark-Field Microscopy

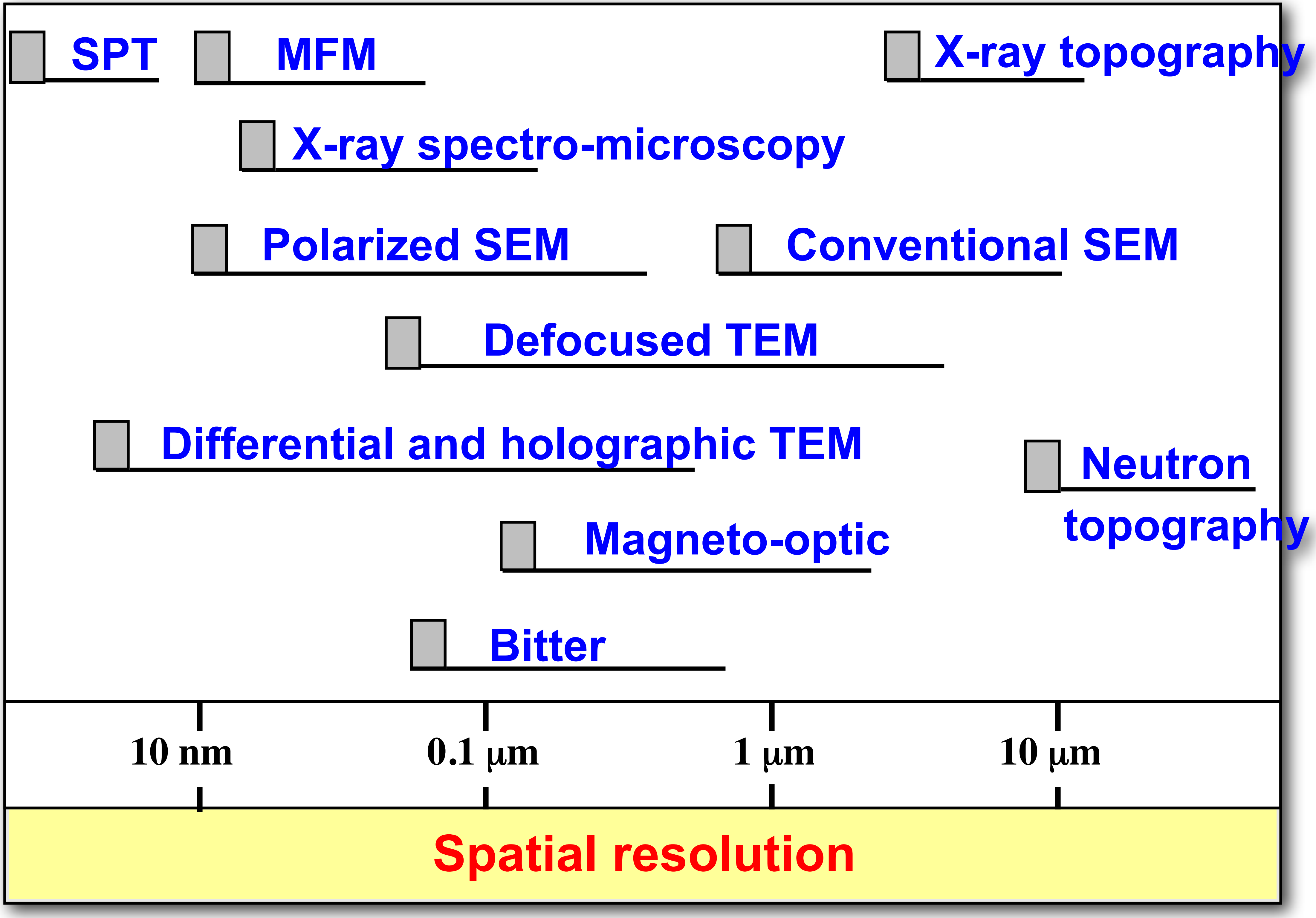






I. Manke, et al.: Three-dimensional imaging of magnetic domains.
Nature Communications, 1:125 doi: 10.1038/ncomms1125 (2010)

Comparison of Domain Obervation Techniques



MFM: Magnetic Force Microscopy

SPT: Spin-Polarized Tunneling

SEM: Scanning (reflection) Electron Microscopy

TEM: Transmission Electron Microscopy